

A Thermodynamic Formulation of the STZ Theory of Deformation in Amorphous Solids

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Thermodynamics and STZs



- Briefly review the shear transformation zone theory of amorphous plasticity.
- Apply the non-equilibrium thermodynamics framework just presented to an STZ system in order to
 - Constrain the theory
 - Relate the microscopic degrees of freedom the relevant "effective temperature" associated with these degrees of freedom
- See Falk, Langer, Annual Reviews of Condensed Matter Physics (in press) arXiv:1004.4684

Micro-mechanical Observations 1



Simulation by T.K. Haxton and A.J. Liu, PRL 99, 195701 (2007)

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Shear Transformation Zones





Postulates:

- STZs have a particular orientation. They are susceptible to shear to the extent that the shear is along this direction.
- STZs are reversible until their environment rearranges. They behave as 2-state systems.
- STZs are transient. They can be created and destroyed by neighboring plastic activity.

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Deformation by STZs



Next consider the shear response by assuming plastic strain rate to be proportional to STZ Flips



Deformation by STZs



Plastic Strain Rate Proportional to Flips

$$\dot{\epsilon}^{pl} = v_0 [R(s)n_- - R(-s)n_+]$$

Master Equation for Densities

$$\dot{n}_{\pm} = R(\pm s) n_{\mp} - R(\mp s) n_{\pm} + \Gamma(s, n_{\pm}) \left[\frac{n_{eq}}{2} - n_{\pm}\right]$$
Rate of
Mechanical
Mixing
Mixing

Re-expressing the Master Eq

Plastic Strain Rate Proportional to Flips

$$\dot{\epsilon}^{pl} = v_0 [R(s)n_- - R(-s)n_+]$$

 Master Equation for Densities $\dot{n}_{+} = +R(s) n_{-} - R(-s) n_{+} + \Gamma \left[\frac{n_{eq}}{2} - n_{+} \right]$ $\dot{n}_{-} = -R(s) n_{-} + R(-s) n_{+} + \Gamma \left[\frac{n_{eq}}{2} - n_{-} \right]$ $\Lambda = \frac{n_{+} + n_{-}}{(N/V)} \quad m = \frac{n_{+} - n_{-}}{n_{+} + n_{-}} \quad \text{would be a scalar, but } m \text{ would be a scalar, but }$

In a full tensorial model Λ the second moment of the STZ orientational distribution.

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Re-expressing the Master Eq



Plastic Strain Rate Proportional to Flips

$$\dot{\epsilon}^{pl} = \epsilon_0 \Lambda \mathcal{C}(s) \left[\mathcal{T}(s) - m \right]$$

- Master Equation for Densities $\dot{\Lambda} = \dot{n}_{+} + \dot{n}_{-} = \Gamma(s, \Lambda, m) [\Lambda_{eq} - \Lambda]$ $\dot{m} = 2\mathcal{C}(s) [\mathcal{T}(s) - m] - \Gamma(s, \Lambda, m) \frac{\Lambda_{eq}}{\Lambda} m$
- need to come up with way to obtain Λ_{eq} and Γ
- original guess for Γ was $s\,\dot{\epsilon}^{pl}$

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Thermodynamics of a Deforming System Bouchbinder, Langer PRE, 80 031131,031132 (2009)

- Consider a thermally isolated system consisting of a deforming subsystem with energy U_C and entropy S_C in contact with a thermal reservoir of energy U_R and entropy S_R .
- Define

$$\chi \equiv \left(\frac{\partial U_C}{\partial S_C}\right)_{\sigma,\{\Lambda\}} \neq \theta \equiv \left(\frac{\partial U_R}{\partial S_R}\right)$$



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• And $U^{tot} = U_C(S_C, s, \{\Lambda\}) + U_R(S_R)$

 $\{\Lambda\}$ represents all the internal state variables needed to describe the subsystem.

• By 1st Law $2 V s \dot{\epsilon}^{tot} = \dot{U}^{tot}$

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The First Law of Thermodynamics



$$2Vs\left(\dot{\epsilon^{el}} + \dot{\epsilon^{pl}}\right) = U^{\dot{t}ot} = \left(\frac{\partial U_C}{\partial \epsilon^{el}}\right)\dot{\epsilon^{el}} + \dots$$

• If $2Vs = (\partial U_C / \partial \epsilon^{el})$ then elastic terms cancel and we find

$$2 V s \dot{\epsilon}^{pl} = \chi \dot{S}_C + \sum_{\alpha} \left(\frac{\partial U_C}{\partial \Lambda_{\alpha}} \right)_{S_C} \dot{\Lambda}_{\alpha} + \theta \dot{S}_R$$

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How to Apply the Second Law?

- Clausius-Duhem Inequality states $S \ge 0$
- Coleman-Noll (1963)
 - Axiomatic approach takes the Clausius-Duhem statement to be the definition of entropy and temperature.
 - We can apply this to a system with two temperatures.
- See Bouchbinder, Langer (2009) for details.

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Enforcing the Second Law



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$$\mathcal{W}(s, \{\Lambda\}) \equiv 2 V s \dot{\epsilon}^{pl} - \sum_{\alpha} \left(\frac{\partial U_C}{\partial \Lambda_{\alpha}}\right)_{S_C}$$

And the first law becomes

$$\chi \dot{S}_C = \mathcal{W}(s, \{\Lambda\}) + Q$$

but this is simply energy conservation, and when we define the specific heat as

 $V c_{eff} \approx \chi (\partial S_C / \partial \chi)$ the first law reduces to

$$V c_{eff} \dot{\chi} = \mathcal{W}(s, \{\Lambda\}) + Q$$

 $\Lambda_{\alpha} > 0$

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- We can write down an expression for configurational entropy $S_C = N\Lambda - N\Lambda \ln \Lambda + N\Lambda \psi(m) + S_1(U_1)$
- and the configurational energy $U_C = N\Lambda e_Z + U_1(S_1)$
- from these and the equations of motion we derive an expression for the rate of plastic work

$$\mathcal{W}(s, \{\Lambda\}) \equiv 2 \, V \, s \, \dot{\epsilon}^{pl} - \sum_{\alpha} \left(\frac{\partial U_C}{\partial \Lambda_{\alpha}}\right)_{S_C} \dot{\Lambda}_{\alpha} \ge 0$$



$$\frac{\mathcal{W}}{V} = 2\Lambda \mathcal{C}(s)[\mathcal{T}(s) - m] \left[sv_0 + \chi \frac{\partial \psi}{\partial m} \right]$$
$$- \left[e_z + \chi \ln \Lambda - \chi \psi(m) + \chi m \frac{\partial \psi}{\partial m} \right] \dot{\Lambda}$$
$$-\Gamma \chi \Lambda m \frac{\partial \psi}{\partial m} \ge 0$$

• We can assure the work is non-negative if each term is non-negative.

ψ_{-} must be an even function peaked at 0



 $\partial \eta$

$$\frac{\mathcal{W}}{V} = 2\Lambda \mathcal{C}(s)[\mathcal{T}(s)]$$



$$2\Lambda C(s)[T(s) - m] \left[sv_0 + \chi \frac{\partial \psi}{\partial m} \right]$$
$$- \left[e_z + \chi \ln \Lambda - \chi \psi(m) + \chi m \frac{\partial \psi}{\partial m} \right] \dot{\Lambda}$$
$$-\Gamma \chi \Lambda m \frac{\partial \psi}{\partial m} \ge 0$$

• The second term will be even if the term in brackets goes to zero at $\Lambda = \Lambda_{eq}$ implying $\Lambda_{eq} = \nu(m) \exp(-e_z/\chi)$ $\nu(m) = \exp\left[\psi(m) - m\frac{\partial\psi}{\partial m}\right]$

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$$\frac{\mathcal{W}}{V} = 2\Lambda \mathcal{C}(s)[\mathcal{T}(s) - m] \left[sv_0 + \chi \frac{\partial \psi}{\partial m} \right]$$
$$- \left[e_z + \chi \ln \Lambda - \chi \psi(m) + \chi m \frac{\partial \psi}{\partial m} \right] \dot{\Lambda}$$
$$-\Gamma \chi \Lambda m \frac{\partial \psi}{\partial m} \ge 0$$

• Similarly the first term will be even if the unknown term goes to zero when $\mathcal{T}(s) = m$ $\frac{\partial \psi}{\partial m} = -\frac{v_0}{\chi} \xi(m) \qquad \mathcal{T}[\xi(m)] = m$

"Athermal" Limit

- We will discuss the theory in the "athermal" limit; when annealing (aging) is negligible on the time scale of the experiment.
- In this limit there is only one relevant physical rate, the rate of plastic dissipation, which must be positive definite.
- The heat flow out of the system must also be proportional to the rate of plastic work, and we can write an equation for the evolution of χ .
- If we also assume we are in the low strain rate limit, there must be a well defined lower bound for χ_{ss} , more than likely this value $\chi_0 = k_B T_g$.

$$c_{eff}\dot{\chi} \propto \mathcal{W} \times \left[1 - \frac{\chi}{\chi_0}\right]$$

• The rate of mixing must also be proportional to the rate of work per STZ

$$= 2\Lambda \mathcal{C}(s)[\mathcal{T}(s) - m]sv_0 = \Gamma \Lambda v_0 s_0$$

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STZ Equations of Motion



$$\begin{split} \dot{\epsilon}^{pl} &= \epsilon_0 \Lambda \mathcal{C}(s) [\operatorname{sign}(s) - m] \\ \dot{m} &= 2\mathcal{C}(s) [\operatorname{sign}(s) - m] \left(1 - \frac{s \, m}{s_0} \right) \\ \dot{\Lambda} &= \frac{2\mathcal{C}(s) \, s}{s_0} [\operatorname{sign}(s) - m] [\Lambda - e^{\psi(0) - e_Z/\chi}] \\ \dot{\chi} &= \frac{2 \, s}{c_{eff}} \epsilon_0 \Lambda \mathcal{C}(s) [\operatorname{sign}(s) - m] \left[1 - \frac{\chi}{\chi_0} \right] \end{split}$$

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Yield in the "Athermal" Limit



$$\dot{\epsilon}^{pl} = \epsilon_0 \nu(0) e^{-e_Z/\chi} \mathcal{C}(s) [\operatorname{sign}(s) - m]$$

$$\dot{m} = 2\mathcal{C}(s) [\operatorname{sign}(s) - m] \left(1 - \frac{s m}{s_0}\right)$$





Development of a Shear Band

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Relating χ to the microstructure



- Consider a linear relation between the χ parameter and the local internal energy

$$C_1 \chi = U - U_0$$
$$\dot{\epsilon}^{pl} = \exp(-e_Z/\chi) f(s)$$

• Is there an underlying scaling?

$$\frac{\dot{\epsilon}^{pl}(y)}{\dot{\epsilon}_b} = \exp\left(\frac{e_Z}{\chi_b} - \frac{e_Z}{\chi(y)}\right)$$
$$\ln\left(\frac{\dot{\epsilon}^{pl}(y)}{\dot{\epsilon}_b}\right) = \frac{e_Z}{\chi_0(\dot{\epsilon}^{pl})} - \frac{C_1 e_Z}{U(y) - U_0}$$

Scaling verifies the hypothesis



Y Shi, MB Katz, H Li, MLF, PRL, 98, 185505 (2007) Emerging Concepts in Glass Physics, Kavli Institute for Theoretical Physics, UCSB

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Numerical Results

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> 0-100% 100-2009 300-400% 500-600%

700-800%

50%

(M Lisa Manning and JS Langer, PRE, 76, 056106(2007)

 These equations closely reproduce the details of the strain rate and structural profiles during band formation



Strain Rate (L/V₀)

6

12

10

(a)

(b)



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Shear Induced Anisotropy in Granular Media

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(experiments by W. Losert and M. Toyia)

- Taylor-Couette cell
- 102mm inner cylinder
- 44mm gap
- 1mm beads or 2mm beads
- Inner cylinder rotated 4-8 mm/s
- Top surface monitored with high speed camera
- Torque measured at inner cylinder



MLF, M. Toiya, W. Losert, arxiv:0802.1752 (2008)

Comparison to Granular Flow Data

- The blue dots represent experimental measurements of displacement at a specified radial position, plotted as a function of the inner cylinder displacement subsequent to shear reversal.
- The red lines are the STZ predictions.





MLF, M. Toiya, W. Losert, arxiv:0802.1752 (2008) Emerging Concepts in Glass Physics,

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STZ Comparison to Shear of a-Si



Emerging Concepts in Glass Physics, Kavli Institute for Theoretical Physics, UCSB **Amorphous** Silicon forms 5fold coordinated liquid-like regions that facilitate shear. **Requires** χ **dynamics Demkowicz and** Argon, PRB 72, 245205 (2005). **Bouchbinder, Langer** and Procaccia, PRE 75, 036108 (2007).

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Summing Up



- We need constitutive theories of plastic response in order to predict mechanical response past the elastic regime.
- Shear Transformation Zone Theory is an attempt to build a thermodynamically based phenomenological theory with a connection to the microscopic physics of deformation.
- The theory exhibits the following behaviors that are seen in simualtion and experiment
 - A range of behavior from perfectly plastic to shear softening
 - Plastic hysteresis (Bauschinger effects)
 - Existence of a dynamically emerging yield stress (transition from creep to superplastic flow ⇒ nonlinear rheology)
 - Diverging time scale for deformation near the yield stress
- Are STZ's ephemeral or persistent?
- How can we more precisely connect atomic scale to parameters?
- What is the physically correct form for R(s)?

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