

A microscopic view of physical aging and plastic deformation in amorphous solids

Jörg Rottler

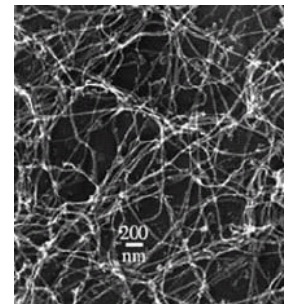
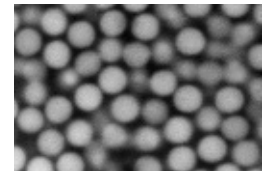
Department of Physics and Astronomy
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KITP workshop “Glasses 2010”

Collaborators:

Mya Warren (now UCSD) and Amy Liu (now Lumerical Solutions)



Nonequilibrium relaxations (physical aging)

- Below T_g , glasses **do not equilibrate**, material can only slowly explore configuration space, α -relaxation time τ increases
- Material properties depend on **waiting time** t_w since the glass was formed and **time** t at which external conditions (stress) are changed
- Almost all glasses age:

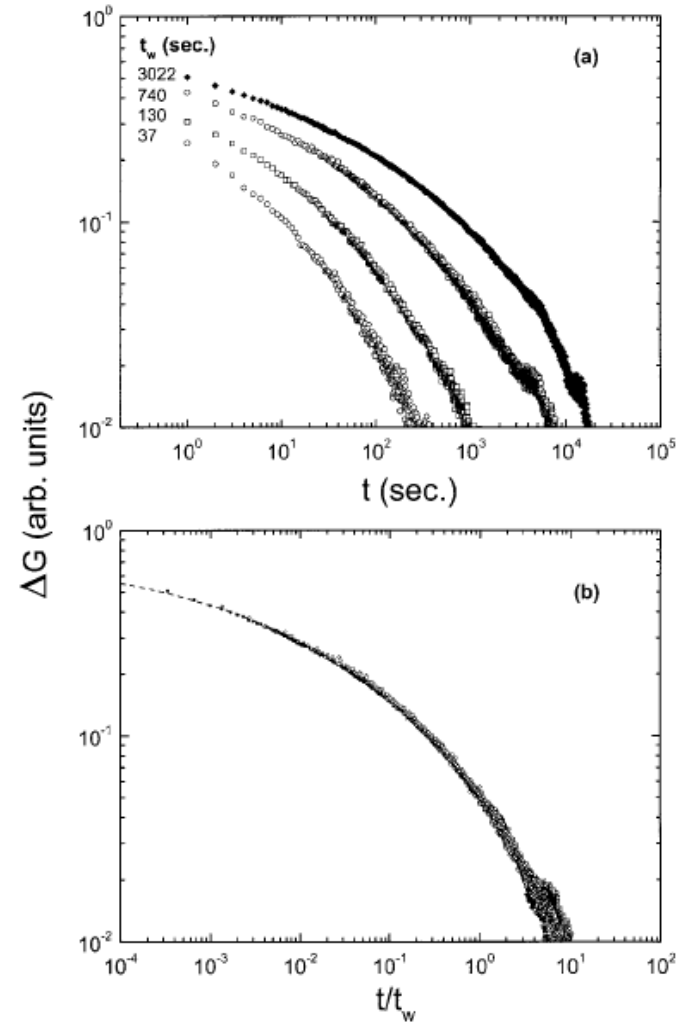
→ thermodynamic quantities (energy, density) $\sim \log t_w$

→ correlation functions $C(t, t_w) : C_{fast}(t) + C_{age}(t / t_w^m)$

- In structural glasses, aging changes the **mechanical properties** such as creep compliance and yield stress.

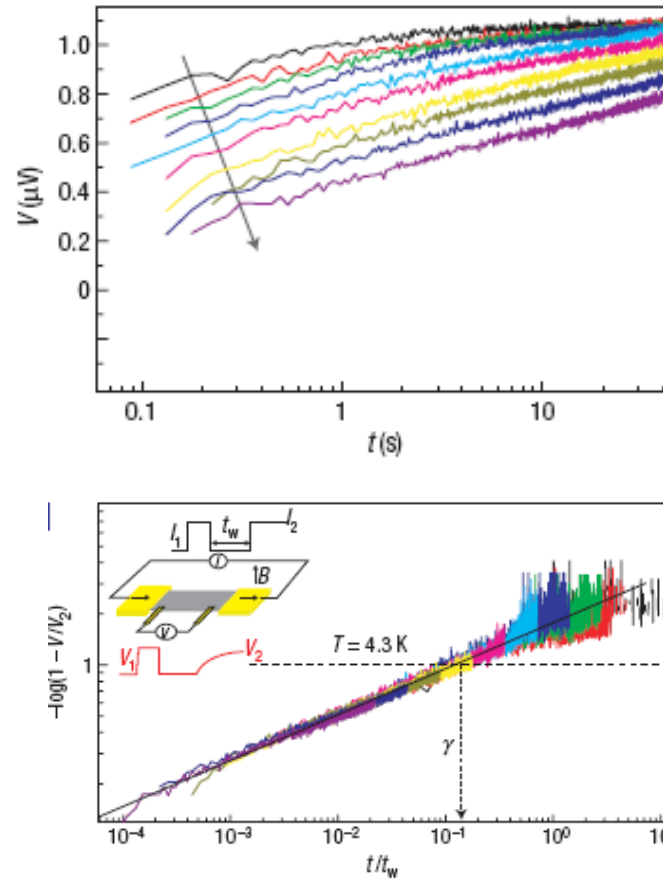
Aging is ubiquitous...

Electron glass



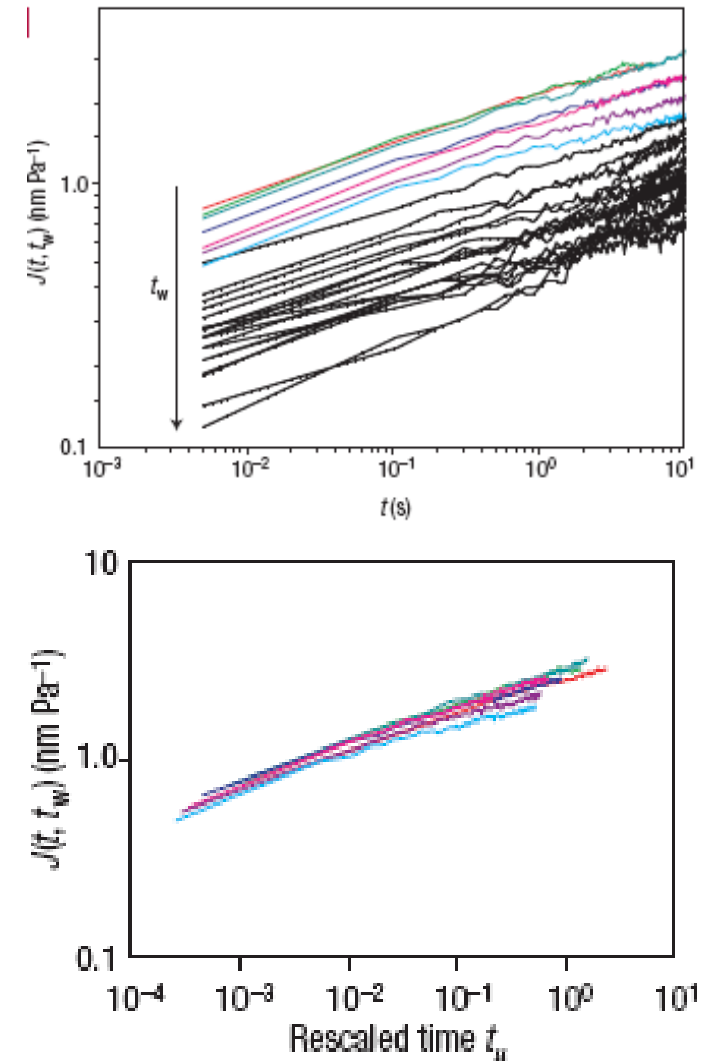
PRL **84**, 3402 (2000)

Vortex glass



Nature Physics **3**, 111 (2007)

Cytoskeleton



Nature Materials **4**, 557 (2005)

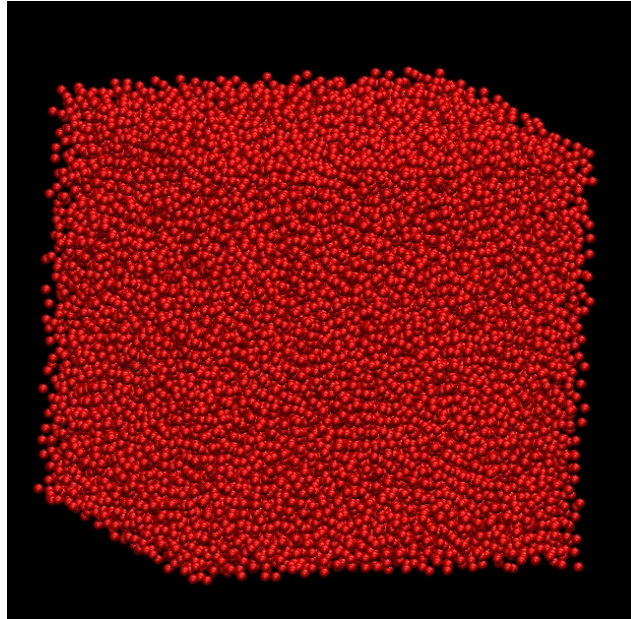
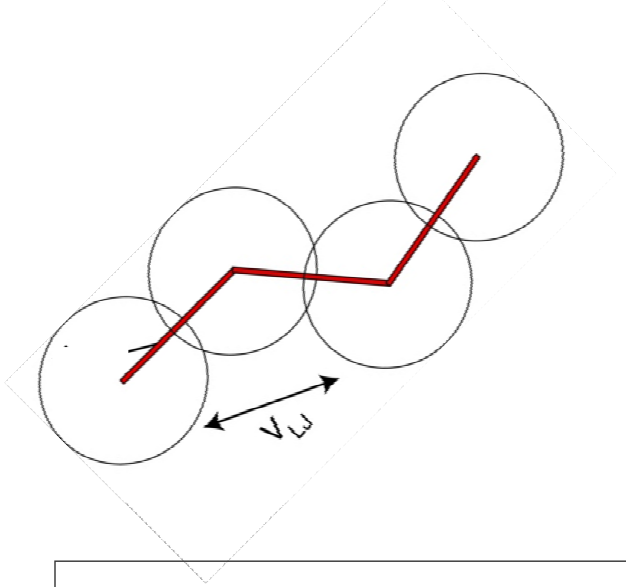
Questions:

- How are the macroscopic physical aging effects related to molecular level processes?
 - How does plastic deformation interact with physical aging?
 - What is the molecular level origin of physical aging in structural glasses?
 - How does active deformation accelerate segmental dynamics?
-

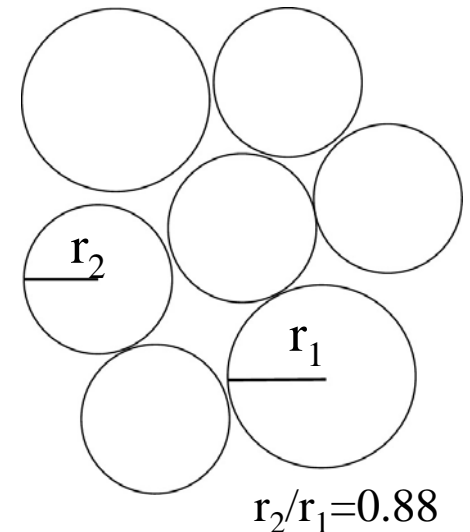
Molecular dynamics

- **Our approach:** minimalistic molecular models that capture generic physics of structural glasses

Bead-spring polymer:



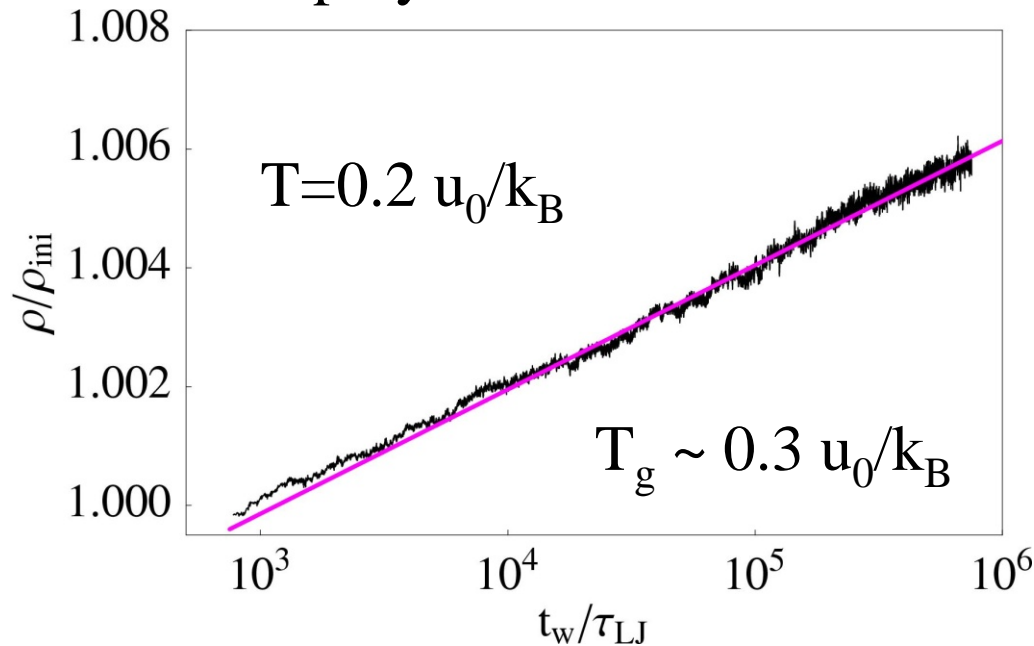
Binary LJ mixture:



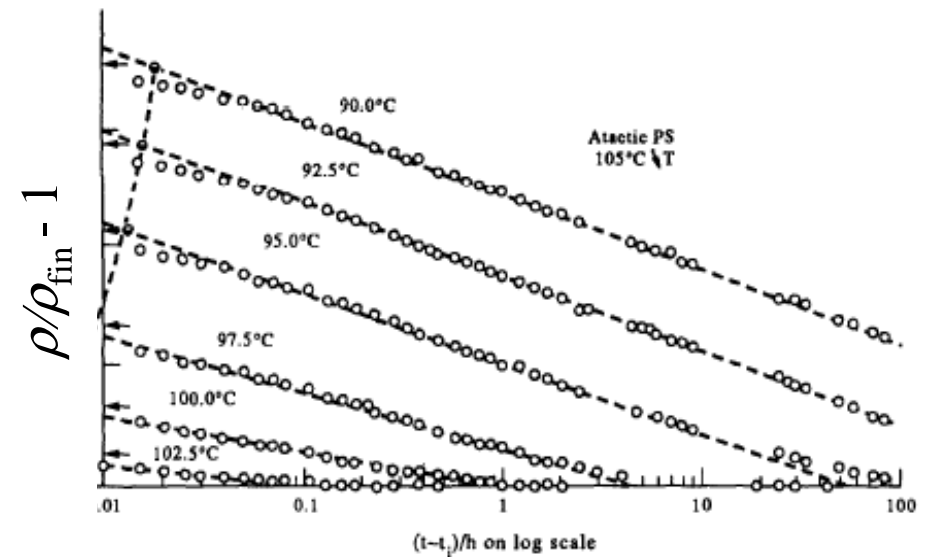
- **Lennard-Jones** (LJ) potential $V_{LJ} \rightarrow$ van der Waals interaction, energy $u_0 \sim \text{meV}$, length $d \sim \text{nm}$, time $\sim \text{ps}$
- **Covalent** bonds, 2 different sphere sizes prevent crystallization \rightarrow computer glass transition to **amorphous solid**

Aging of thermodynamic variables

polymer simulation



atactic polystyrene



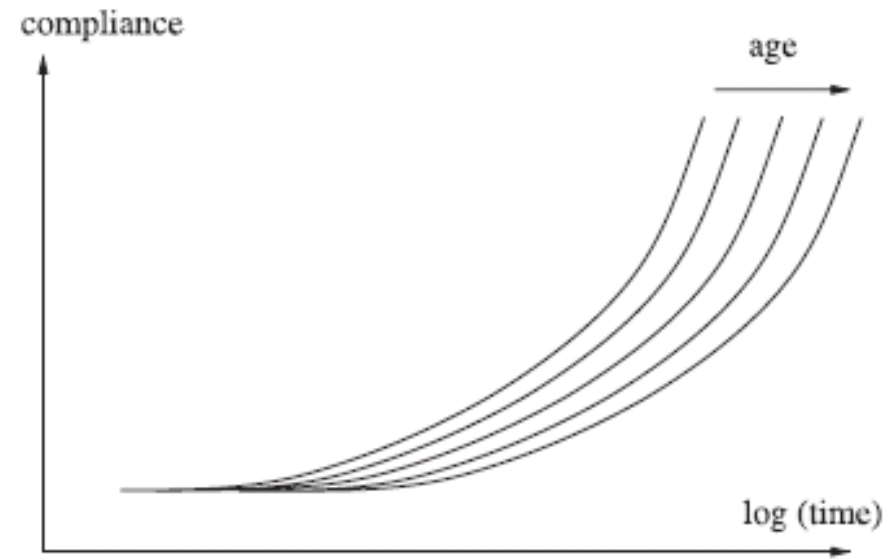
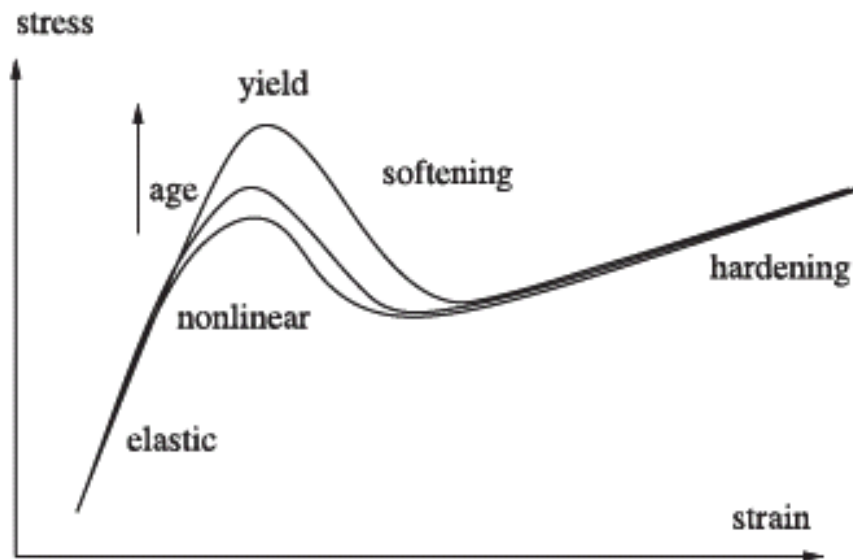
J. Hutchinson, Prog. Poly. Sci. (1995)

- Glass compactifies **logarithmically** in absence of deformation when maintaining zero hydrostatic pressure

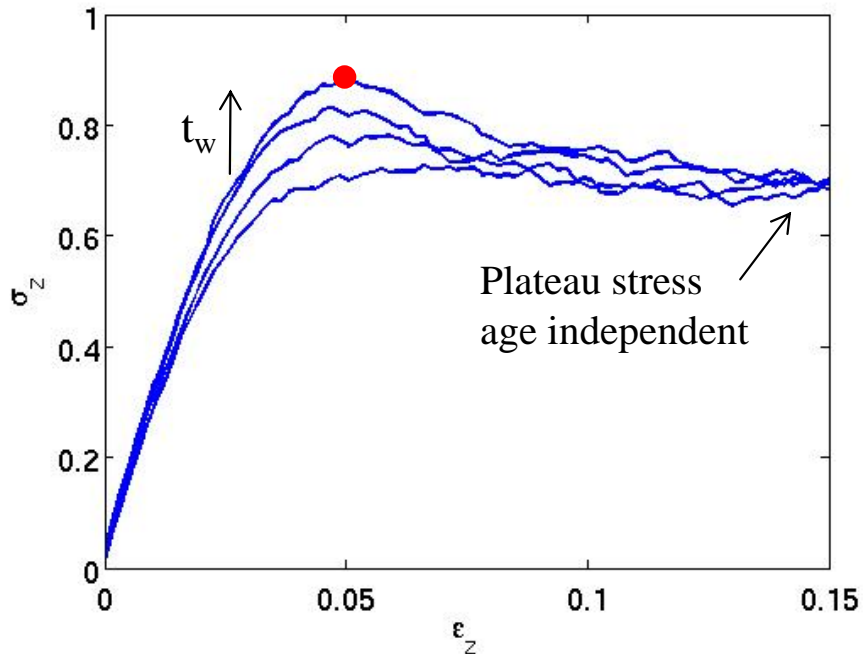
- physical aging rate $r_v = -\frac{1}{V_0} \frac{dV}{d \log t}$

Physical aging and mechanical properties

- Almost all polymer glasses undergo slow **structural relaxations** in the glassy state
- density, enthalpy change with **waiting time** elapsed since the glass was formed
- mechanical properties change with age: **compliance** decreases, **yield stress** increases



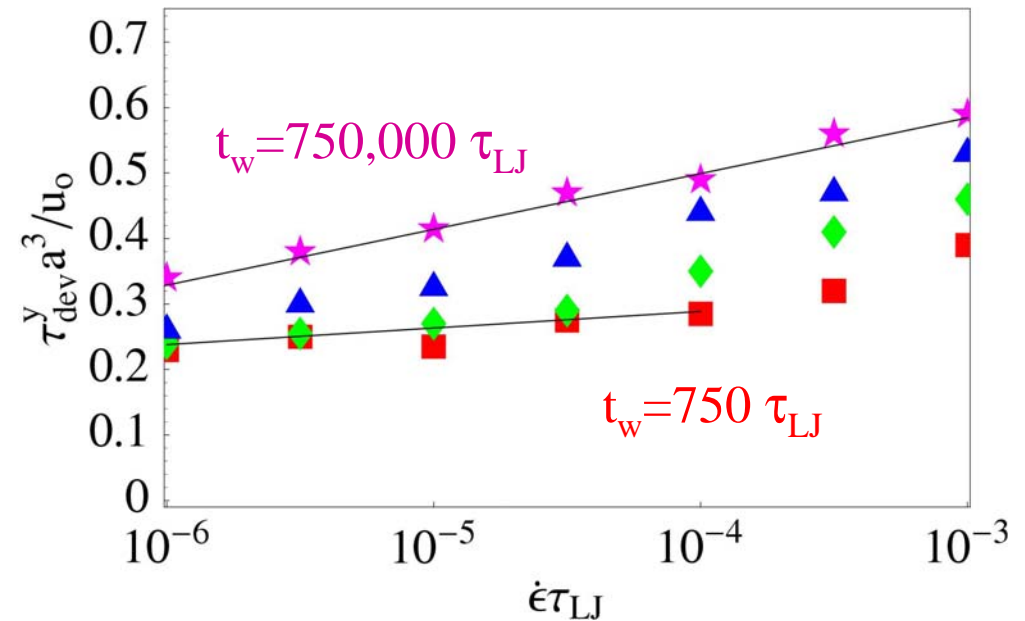
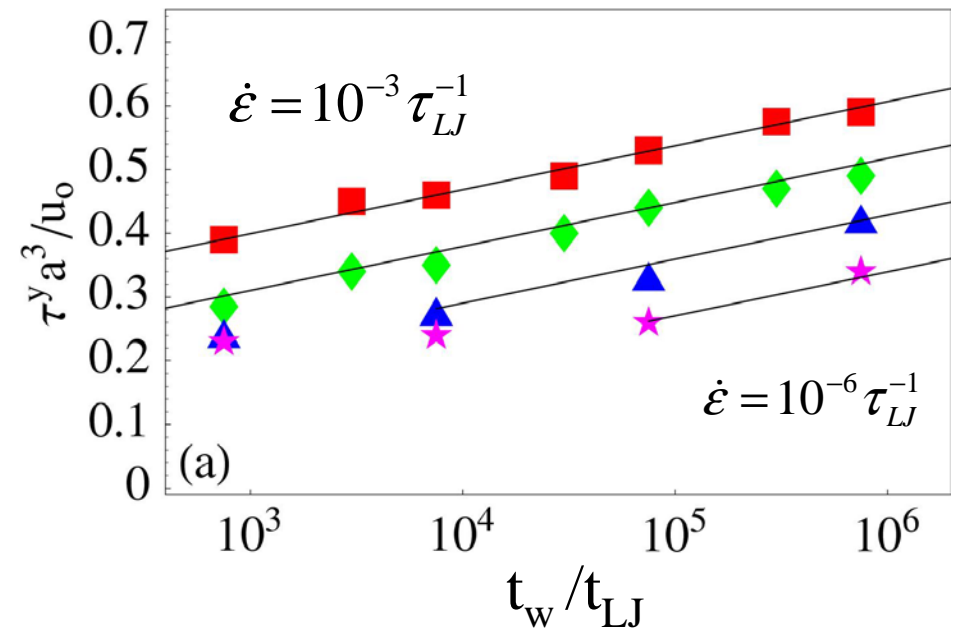
Yield (overshoot) stress



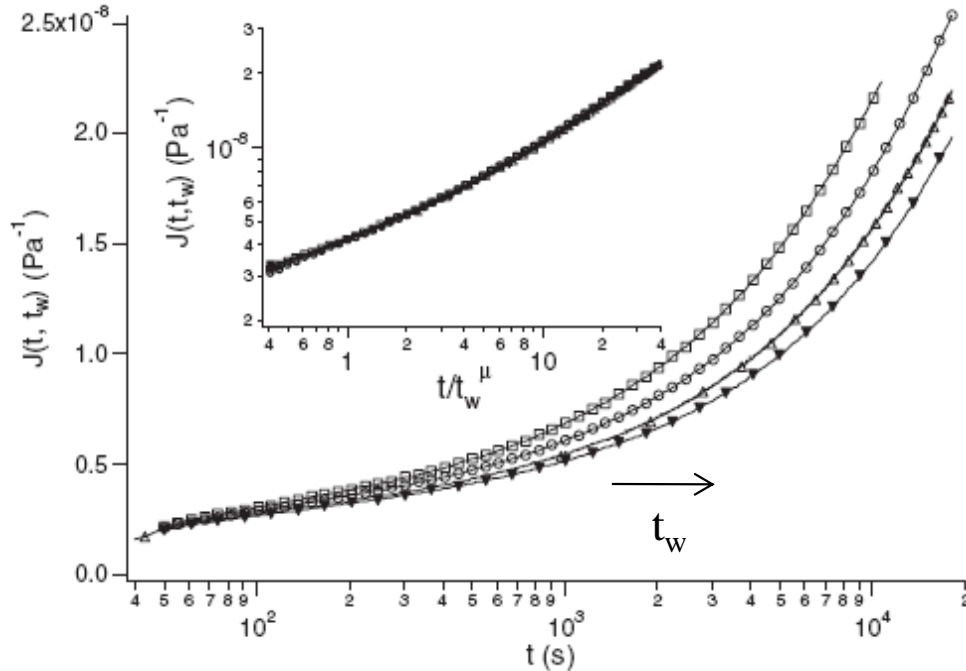
- Age for t_w after down-quench, deform at constant strain rate

Overshoot stress

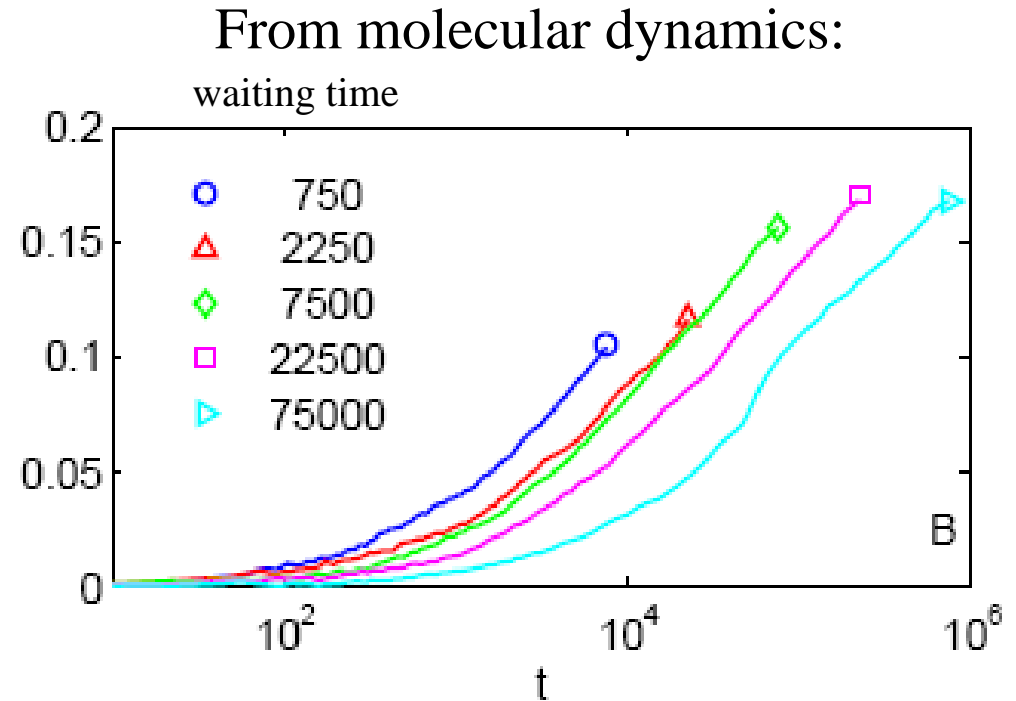
- grows $\sim \log(t_w)$
- grows $\sim \log(\text{strain rate})$



Creep compliance



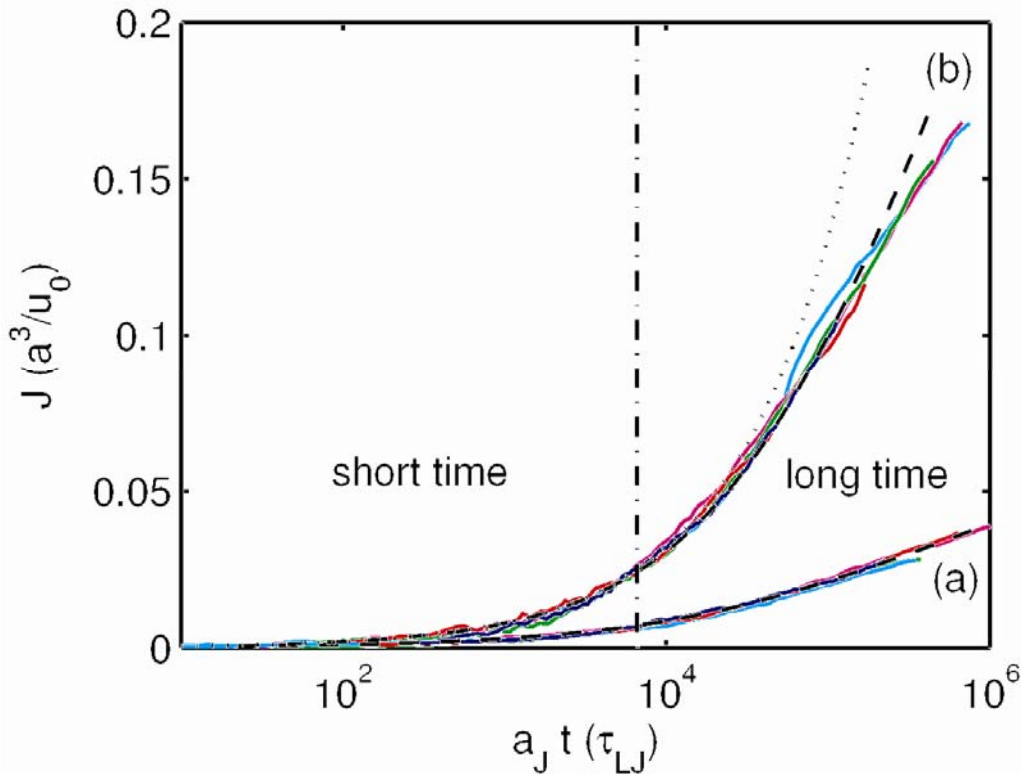
Montes et al, J. Stat. Mech., P03003 (2006)



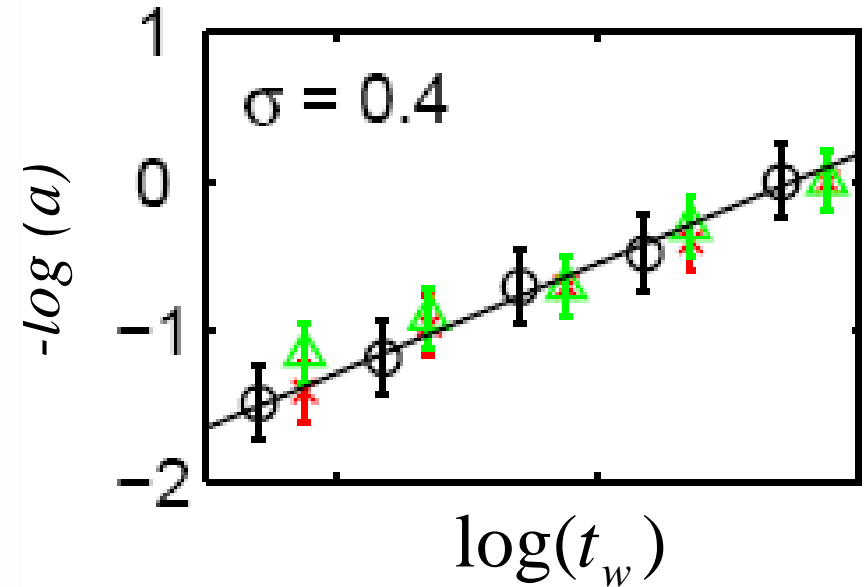
Warren and JR, PR E 76, 031802 (2007)

- Age for t_w after simple down quench.
- Apply **constant load**, measure time-dependent strain $\rightarrow J(t, t_w) = \epsilon(t, t_w) / \sigma$
- Compliance shifts to longer times with increasing t_w , \rightarrow polymer stiffens

Time-waiting time superposition



Shift factor vs waiting time:



$$a_J : t_w^{-m}$$

aging exponent $\mu = \mu(T, \sigma) < 1$:
 \rightarrow subaging

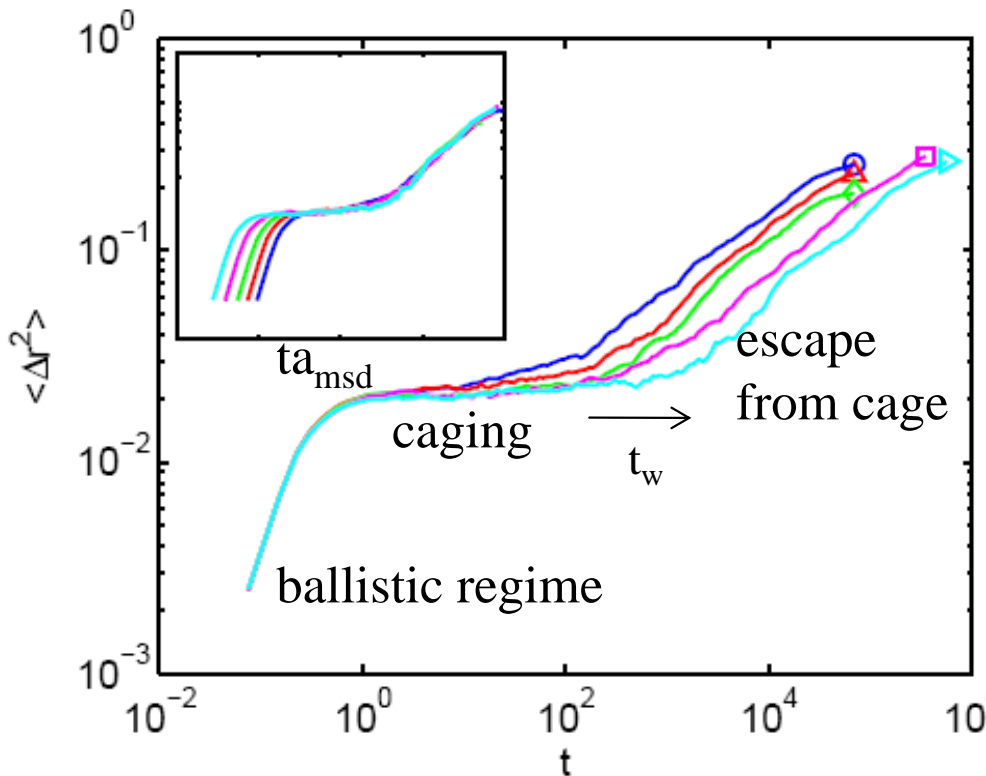
- $\mu(T)$ from microscopic theory (segmental hopping):

Chen & Schweizer PRL **98**, 167802 (2007)

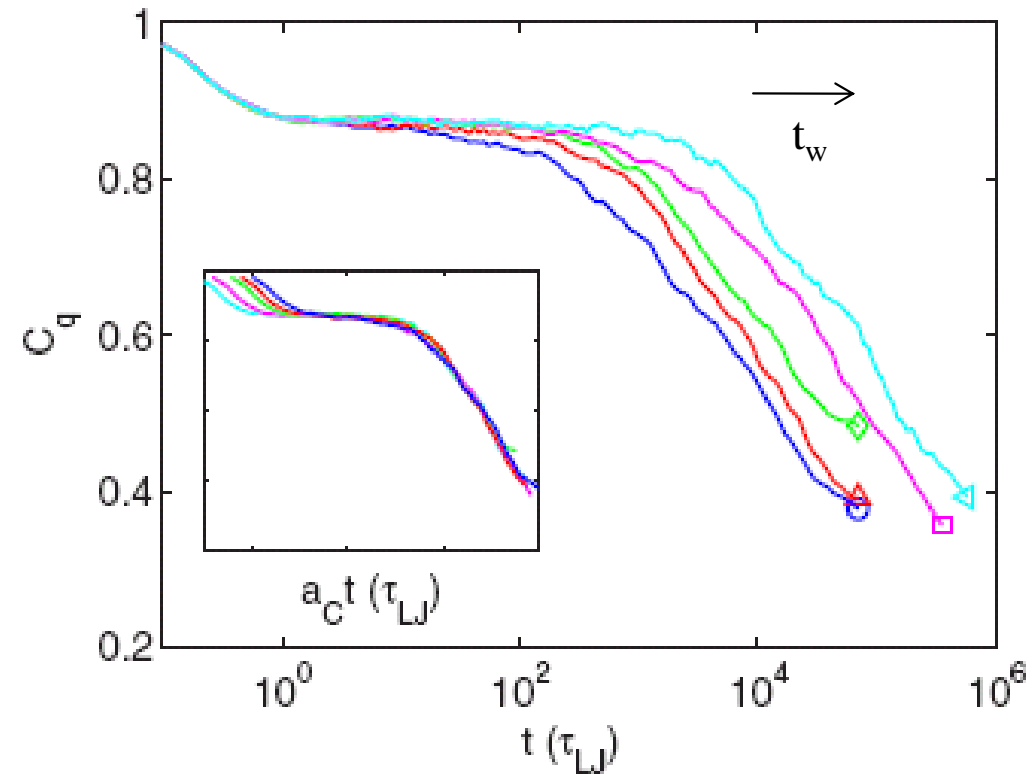
Molecular mobility controls mechanical response

- the mechanical shift factor $a_J(t_w)$ is related to atomic mobility

mean-squared displacement



intermediate scattering function

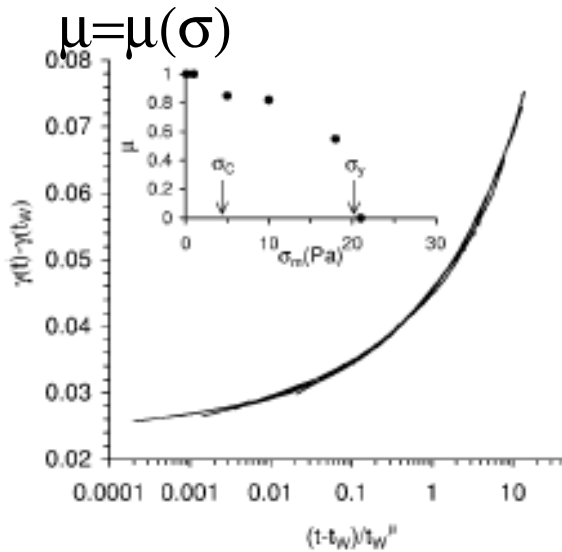


- time to escape from local cages increases with t_w ;
superposition by shifting with factor a_{msd} We find:

$$a_{msd} \sim a_C \sim a_J$$

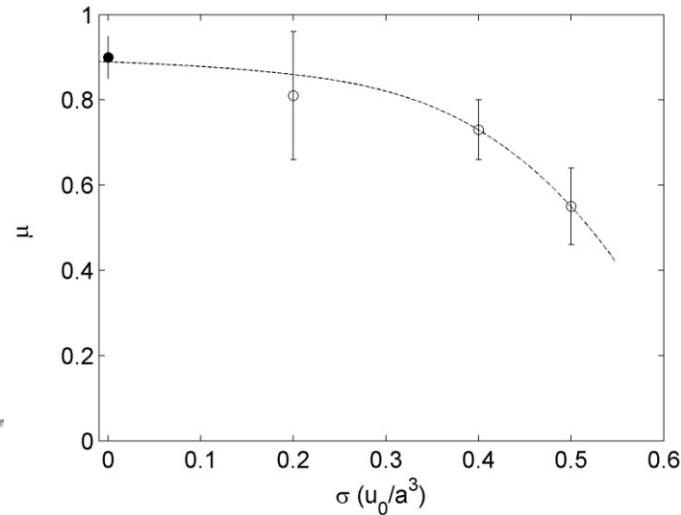
Aging and rejuvenation

- Age for t_w , then apply stress σ and measure aging exponent



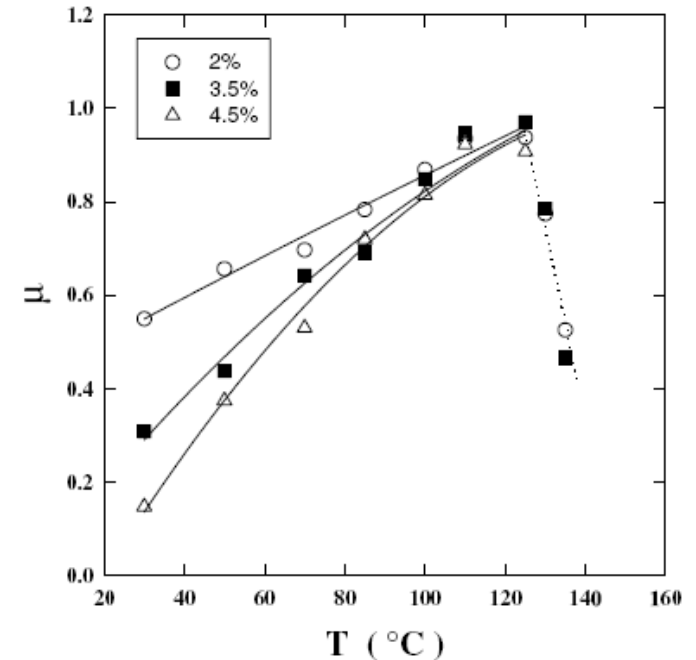
microgel paste

Cloitre et al., PRL (2000)



polymer simulation

Warren and JR, PR E **76**, 031802 (2007)



polycarbonate

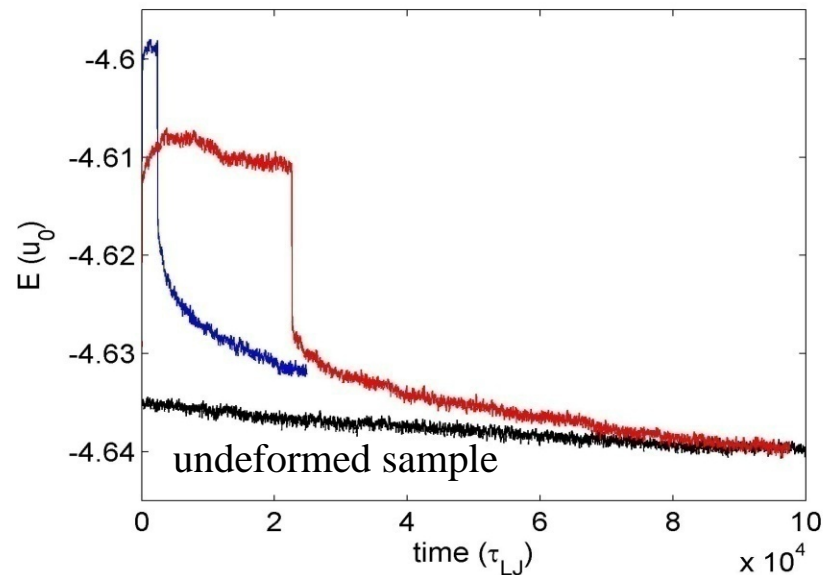
McKenna, J. Phys Condens Mat. (2003)

- μ decreases, relaxation times increase more slowly:
 → system "looks younger" at larger stresses: **mechanical rejuvenation?**

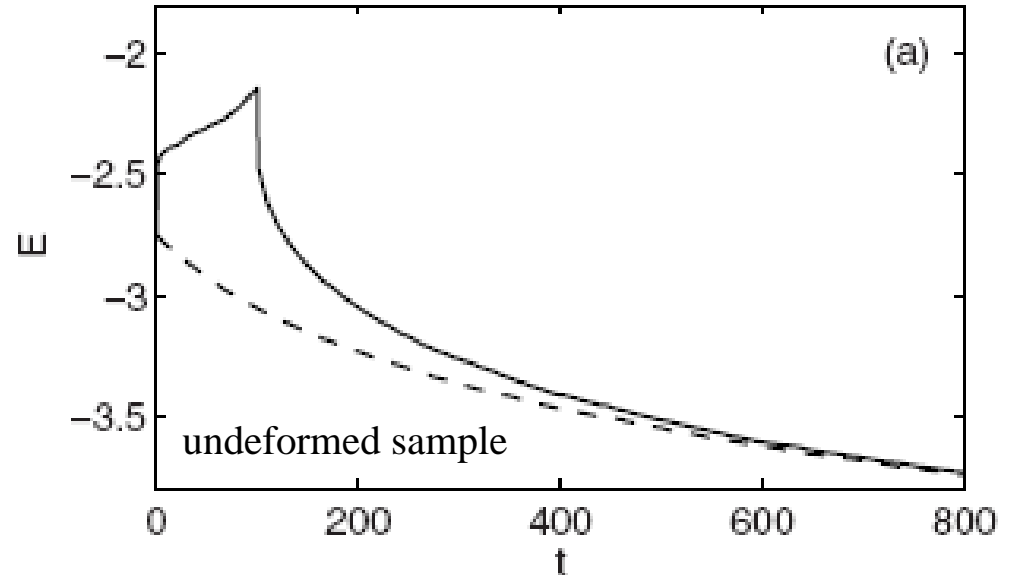
Aging and rejuvenation

- Compare relaxation of unstressed and rejuvenated sample
- Energy of rejuvenated sample decreases faster after stress release

Molecular dynamics

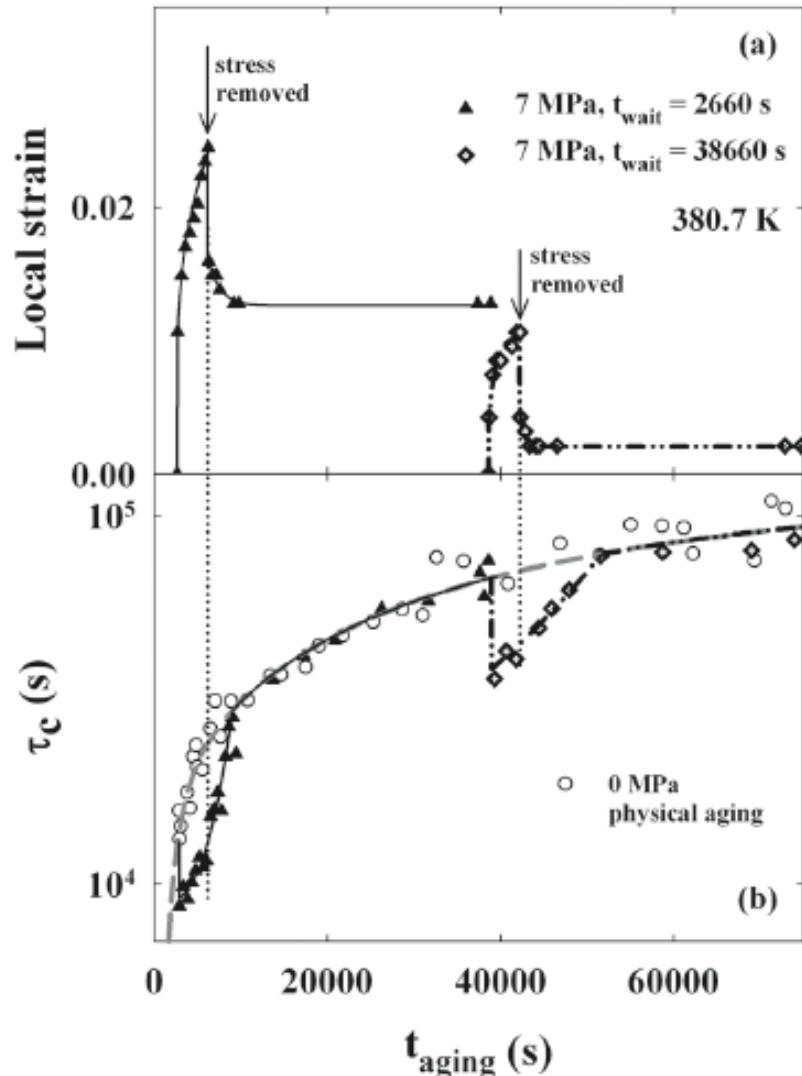


Soft glassy rheology (SGR)



- But the time to reach equilibrium is unchanged!
Is the system really “rejuvenated”?
- Emerging picture: small (subyield) stress only lead to transient acceleration of dynamics, full erasure only after plastic flow.

Experiments on PMMA

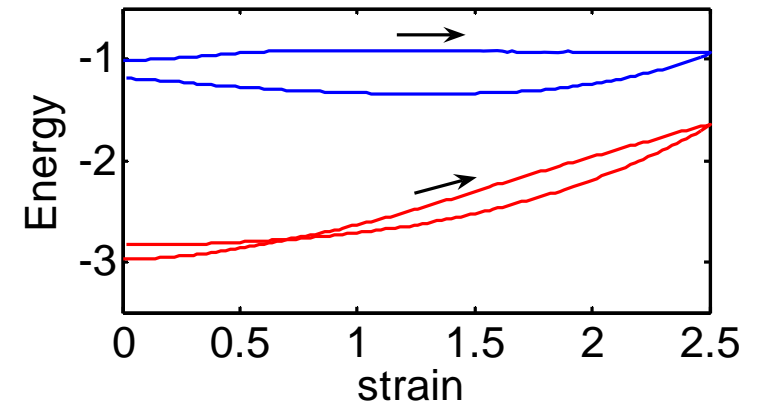


Lee and Ediger (submitted)

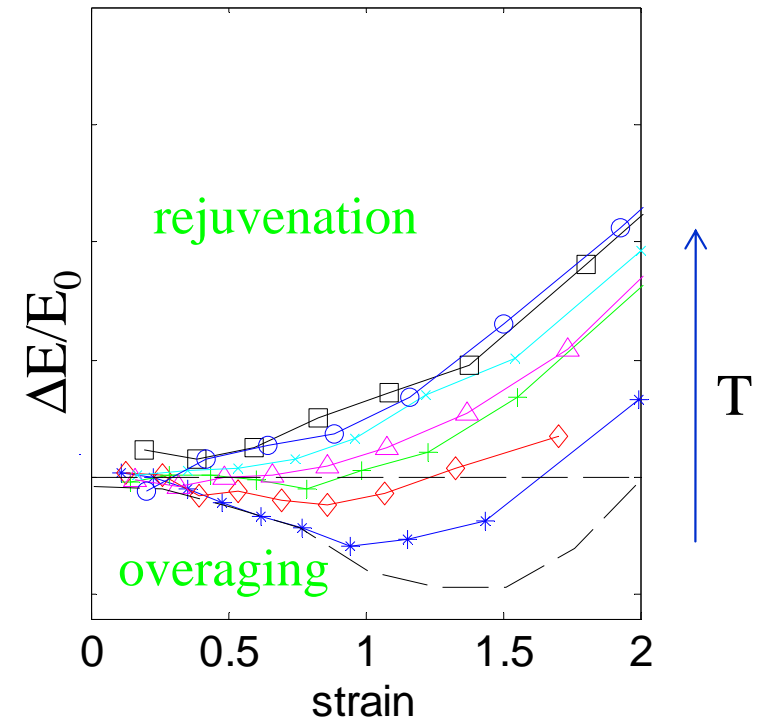
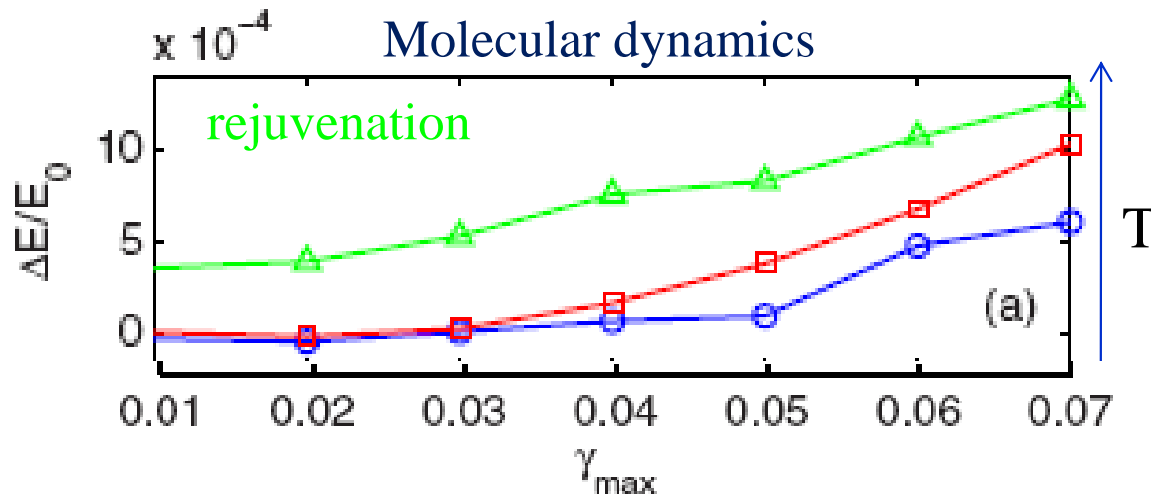
- direct measurement of segmental relaxation via local probe molecules
- extract relaxation times from decay of autocorrelation function
- stresses in the pre-flow regime perturb the aging dynamics only transiently

Rejuvenation in a strain cycle

- Cycle strain, record energy after unloading
- Find regions **of increased** (rejuvenated) and **decreased** (overaged) energy states
- Within SGR model, find regime of overaging for low noise temperature and small strains

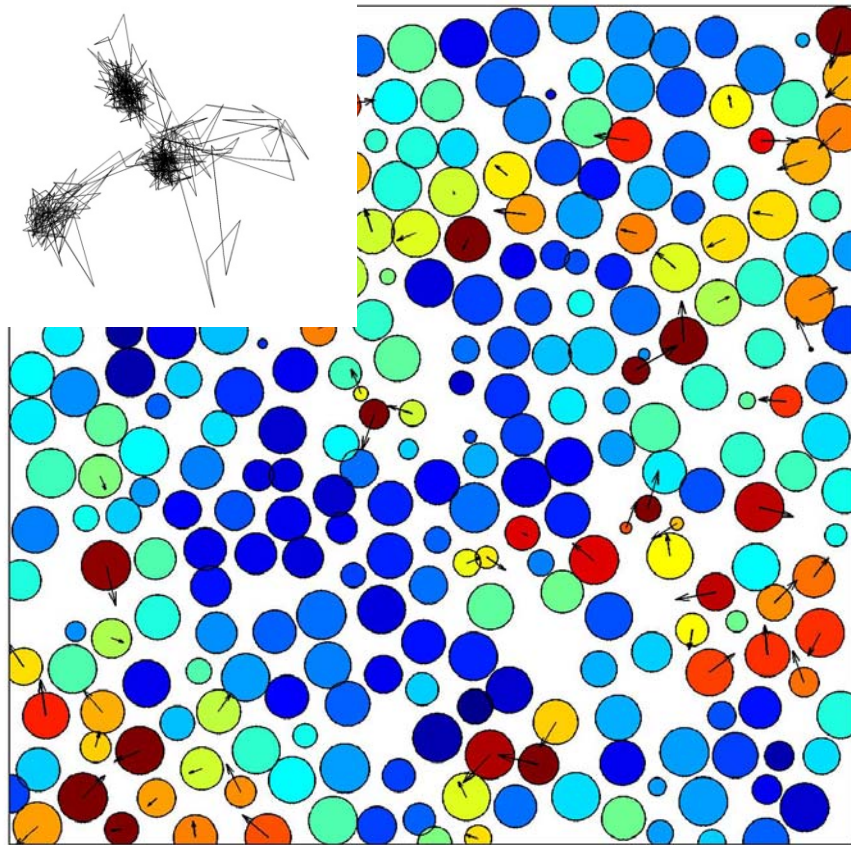


Soft Glassy Rheology (SGR)



Warren and JR, PR E **78**, 041502 (2008)

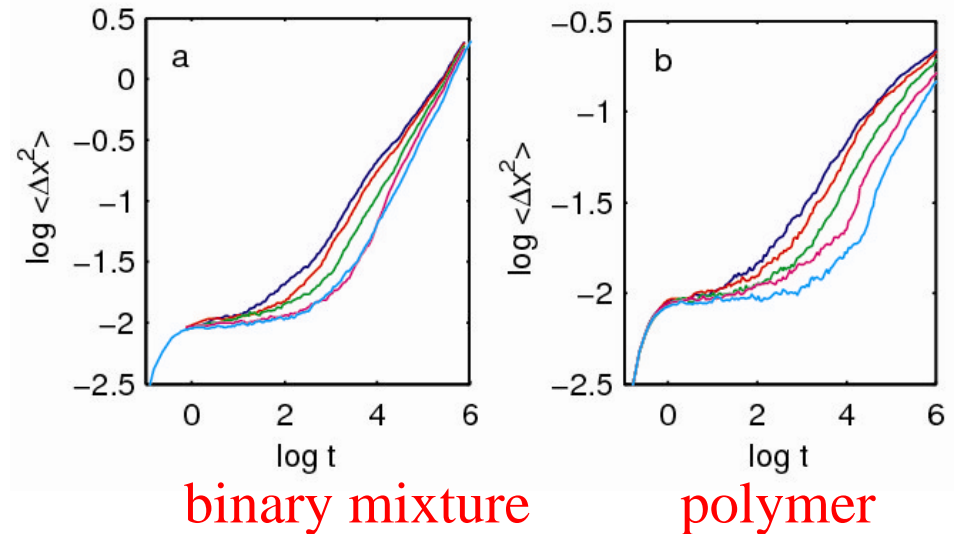
Atomistic features of glassy dynamics



2D slice through a polymer glass

Red = highly mobile

Blue = immobile



binary mixture

polymer

- Structural relaxations are
 - Collective
 - Spatially heterogeneous
 - Temporally intermittent
- Trajectories are well described as a series of hops between long lived caged states

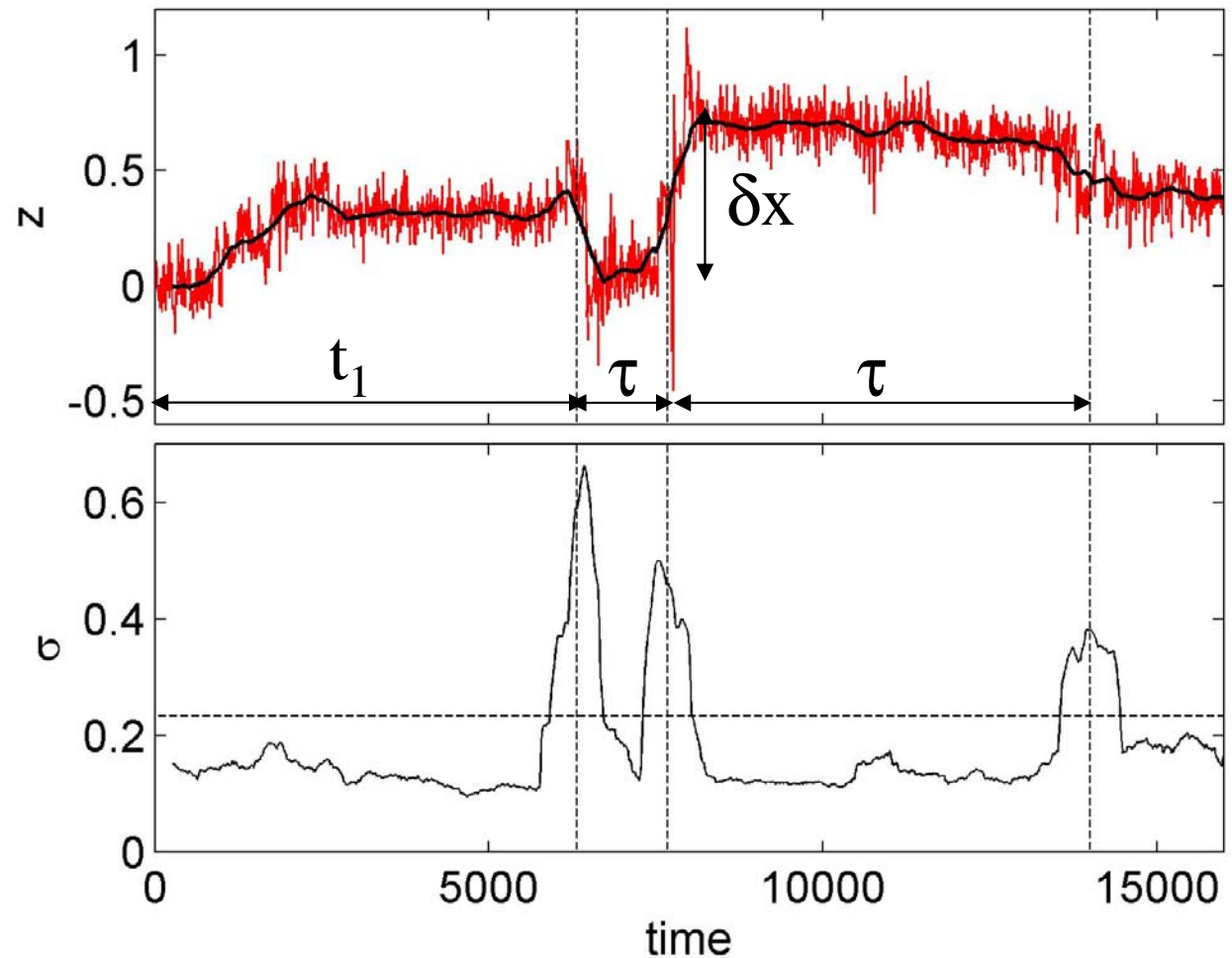
Analysis of particle (segmental) trajectories

Record particle trajectories

Calculate running average and standard deviation σ

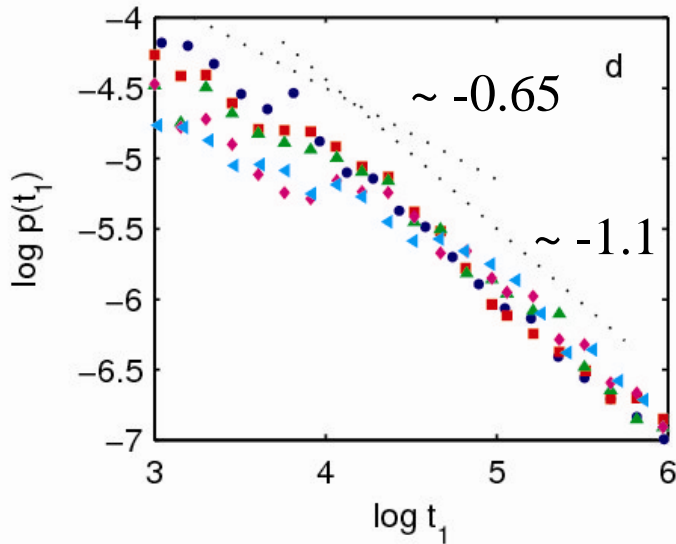
Hops identified through a threshold in σ , i.e. through their activity

Find hop times and displacements



Hop times and displacements – polymer glass

First hop time

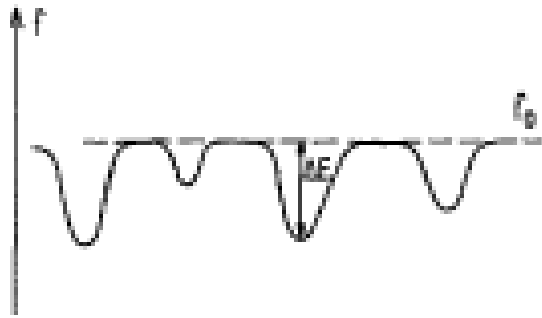


- Age-dependent
- Two power laws, tail moves to longer times with increasing age, likelihood of small t_1 decreases
- Persistent times and displacements **age-independent**
- Pure power law for persistent time
- As in **trap model** of glassy dynamics: trap energies redrawn from stationary potential energy landscape

Warren and JR, EPL (2009)

Aging in an energy landscape picture

- Consider **activated hops in a random energy landscape**:
- Escape rate: $w : \exp[-bE]$
- Distribution of trap depths: $r(E) : \exp[-b_g E]$
- Energy drawn anew after every jump (annealed disorder)

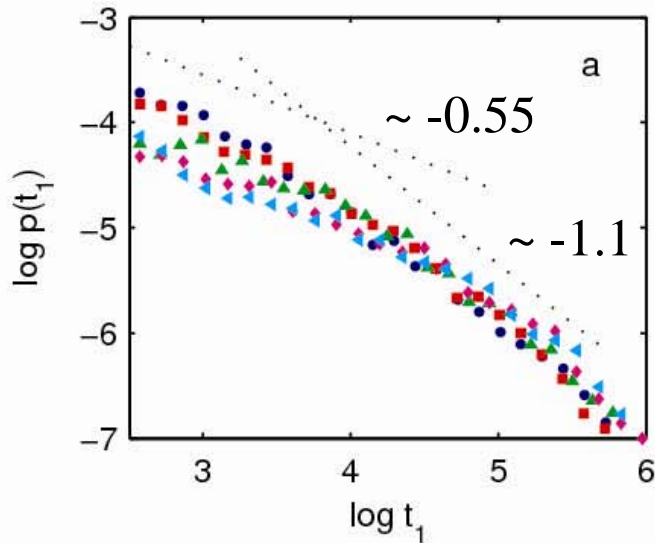


(Bouchaud 1992, Monthus and Bouchaud 1996)

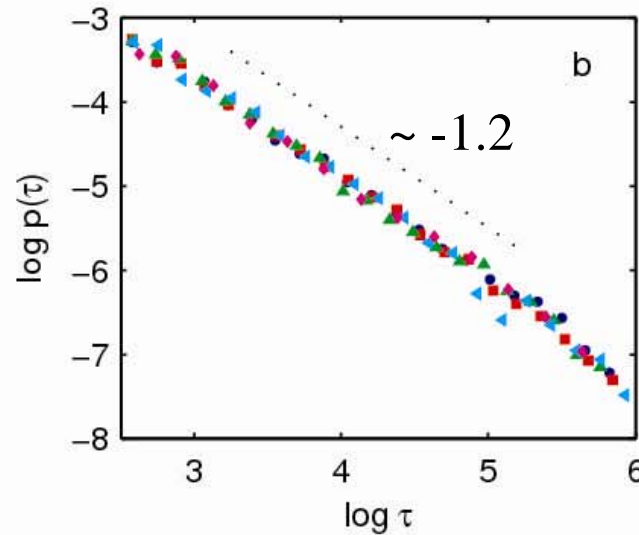
- For $T < T_g$, the dynamics becomes nonstationary and "ages"
- System **does not equilibrate due** to the presence of very deep traps
- predicts distribution of trapping times $P(t) \propto t^{-(1+T/T_g)}$
- mean trapping time infinite \rightarrow aging

Hop times and displacements – binary mixture

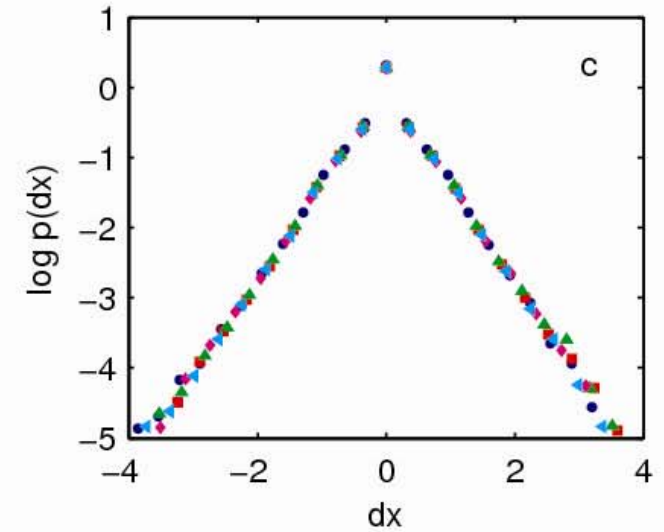
First hop time



Persistent time

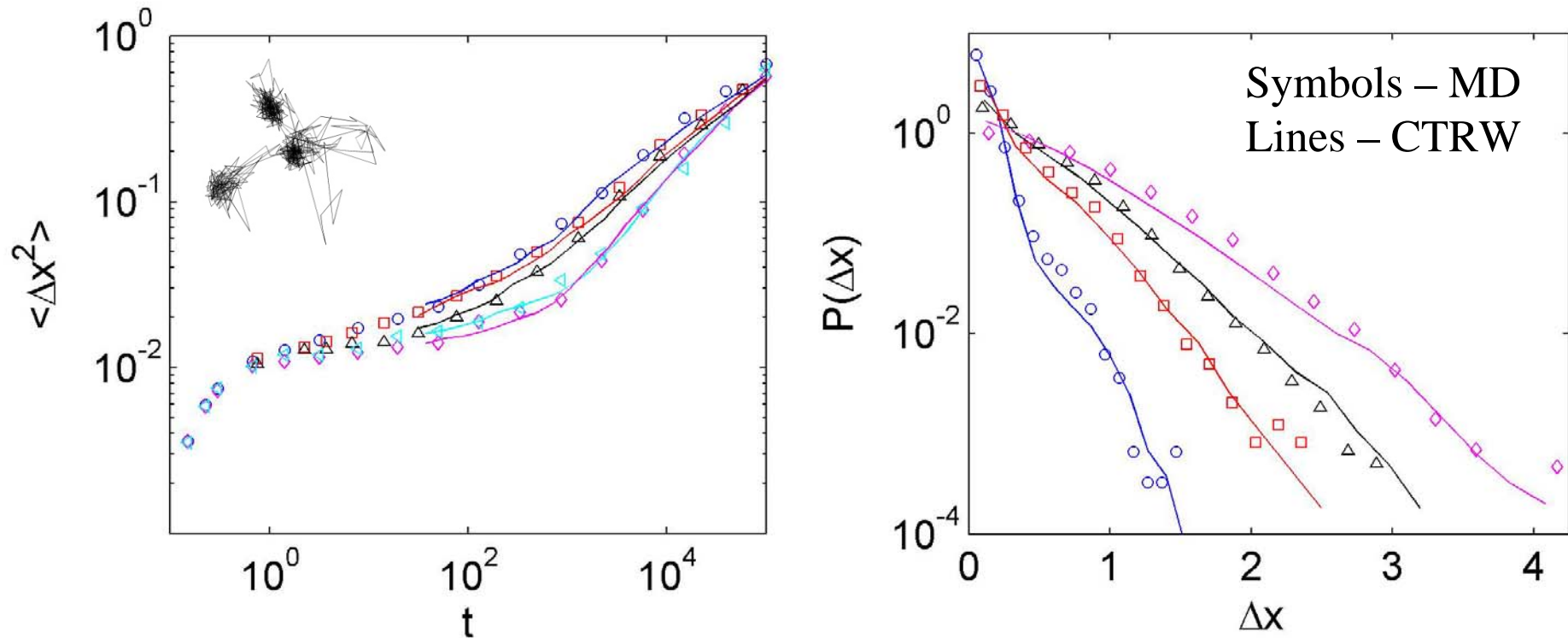


Displacement



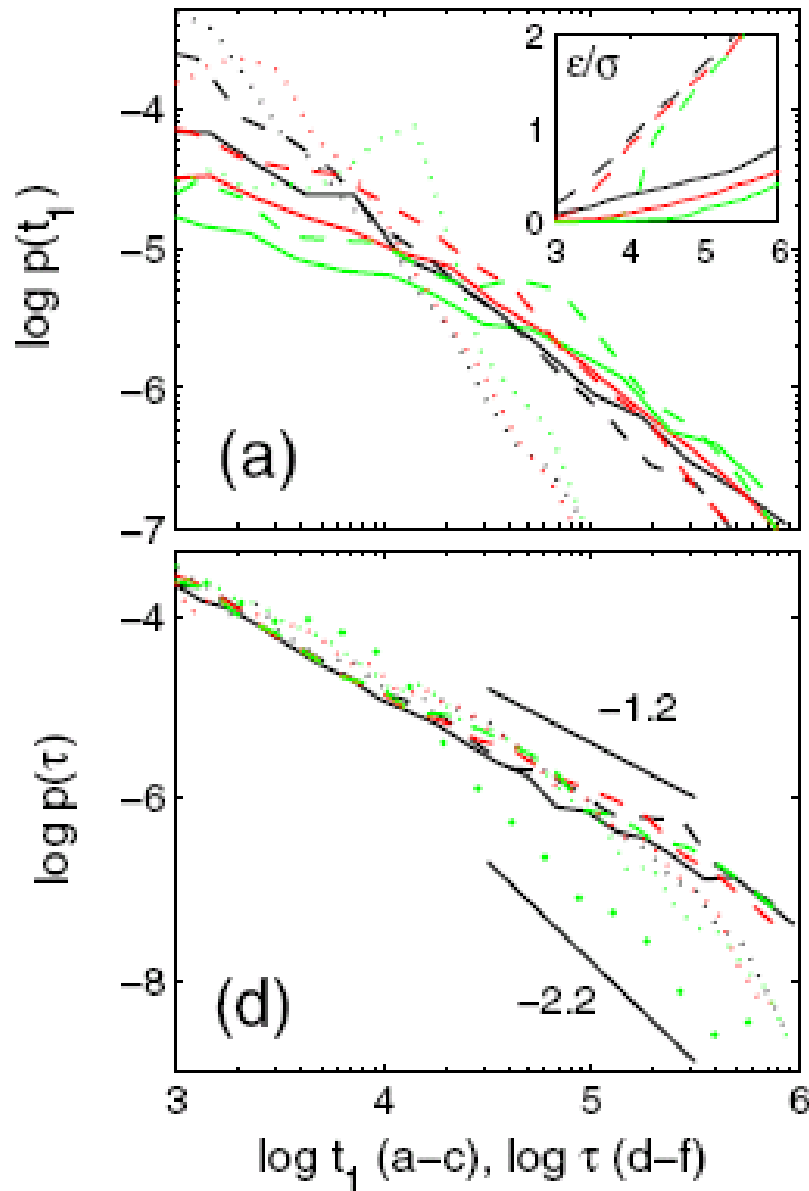
- Hop time and persistence time distributions unchanged
 - Displacement distribution **purely exponential**, no Rouse regime
- Findings are **general** for structural glasses, not polymer specific.

Continuous time random walk



- Particle dynamics are modeled using a continuous time random walk (CTRW) with measured hop statistics (no adjustable parameters).
- **Aging is self-generating!**
- All of the important physics has been captured

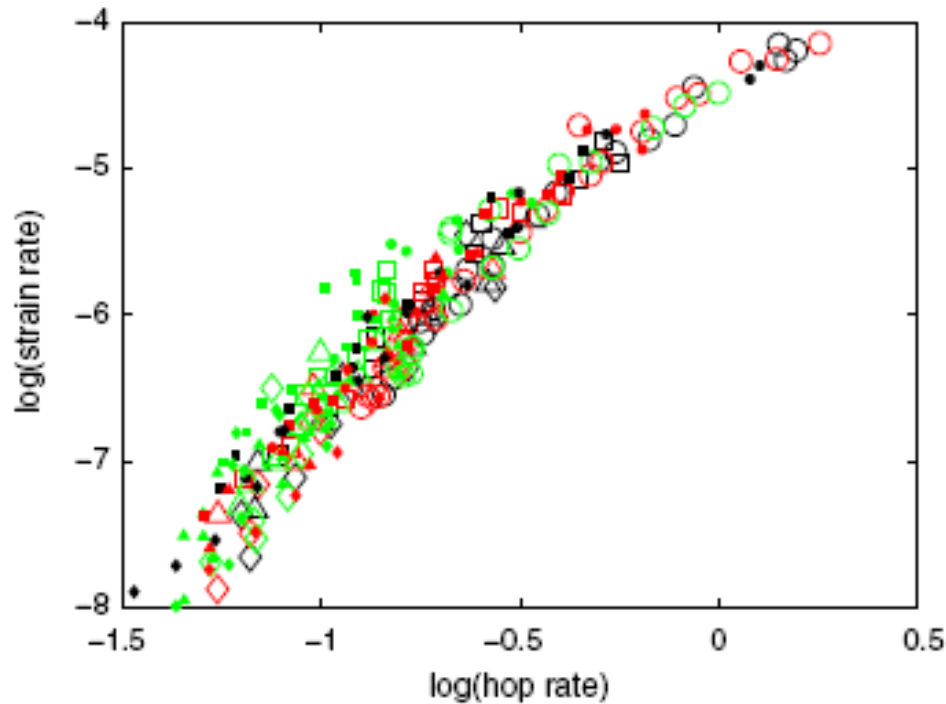
Accelerated dynamics: constant stress (creep)



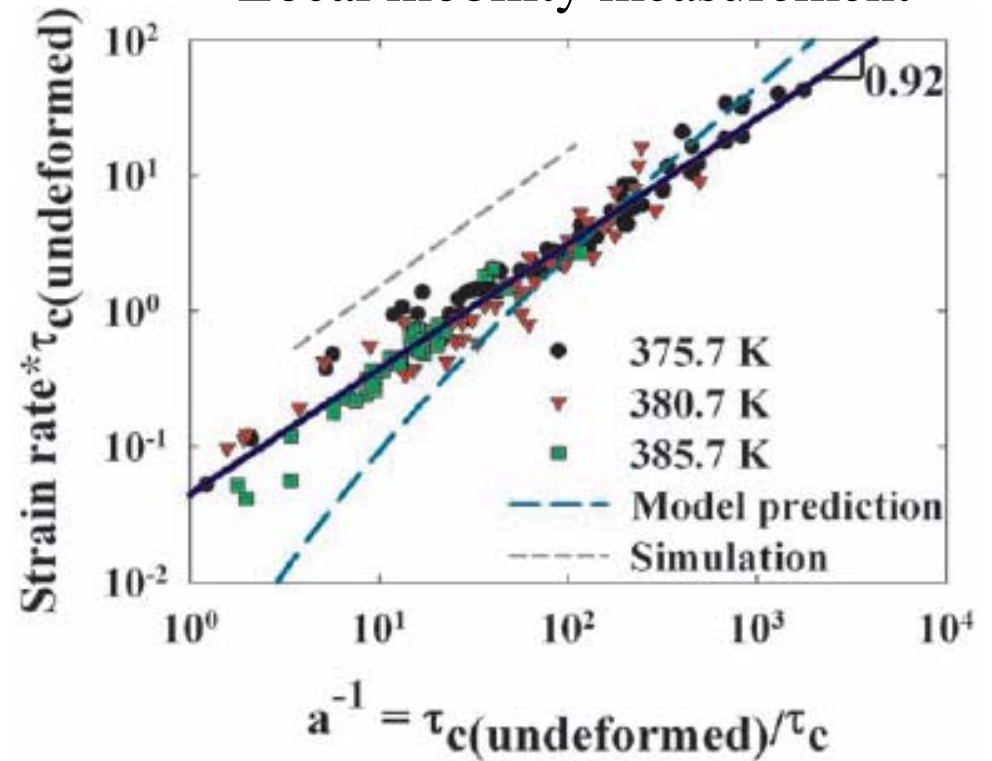
- three different t_w and three stresses $\sigma=0$ (solid), $\sigma=0.4$ (dashed) and $\sigma=0.5$ (dotted)
- first hop time dist. **narrows** with increasing stress, power law tail steepens (see expts. Ediger group)
- persistence time distribution modified for large times
- power law exponent decreases below -2 \rightarrow aging is stopped

Average hop rate and mobility

Simulation



Local mobility measurement

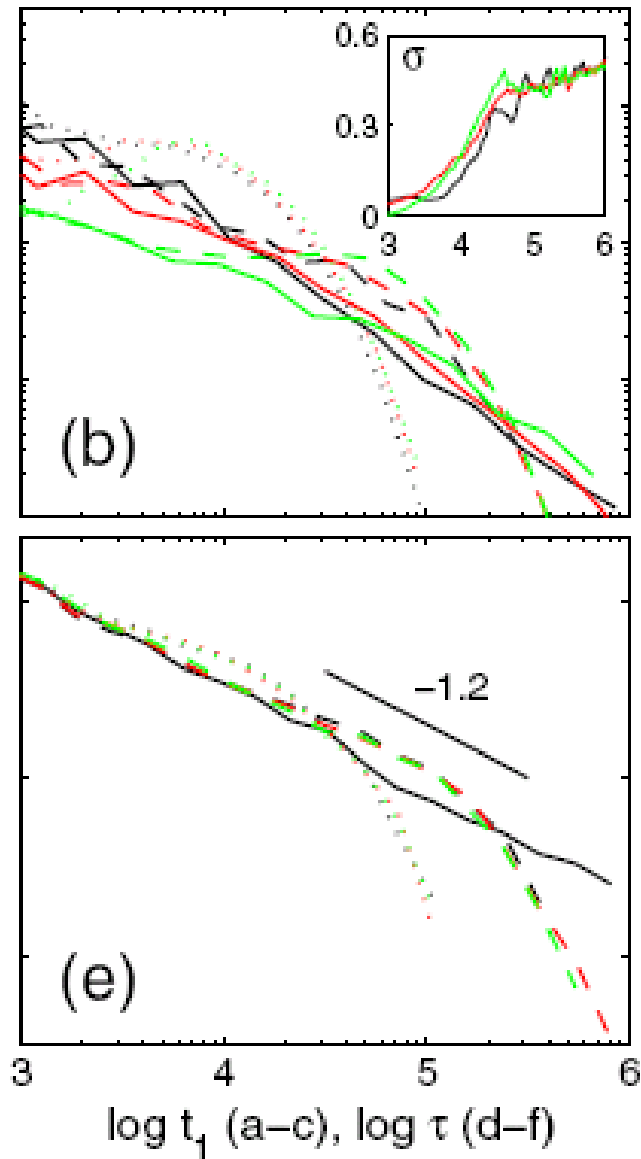


- strain rate is a universal function of hop rate
- data collapse for different stresses and ages

- qualitatively similar relationship observed in recent experiments

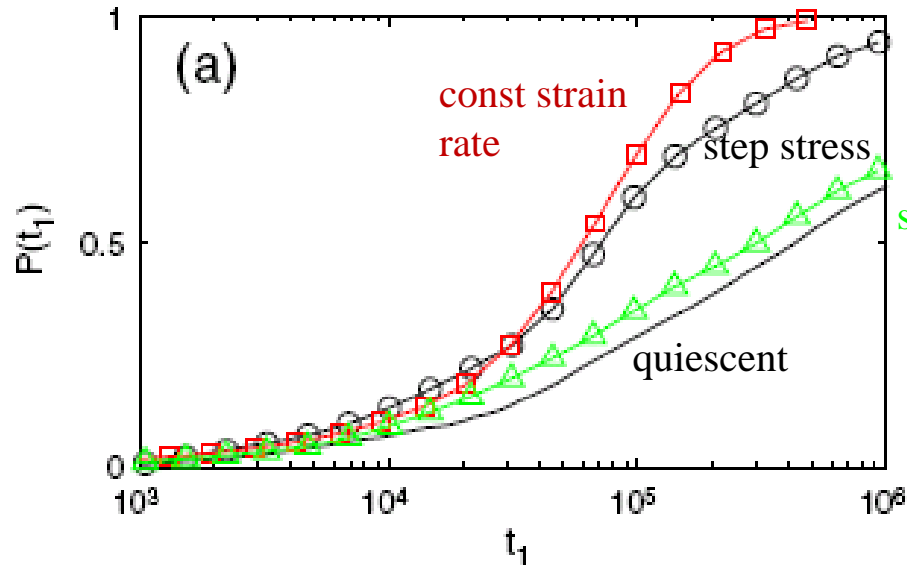
(Lee et al., Science (2009))

Accelerated dynamics: constant strain rate



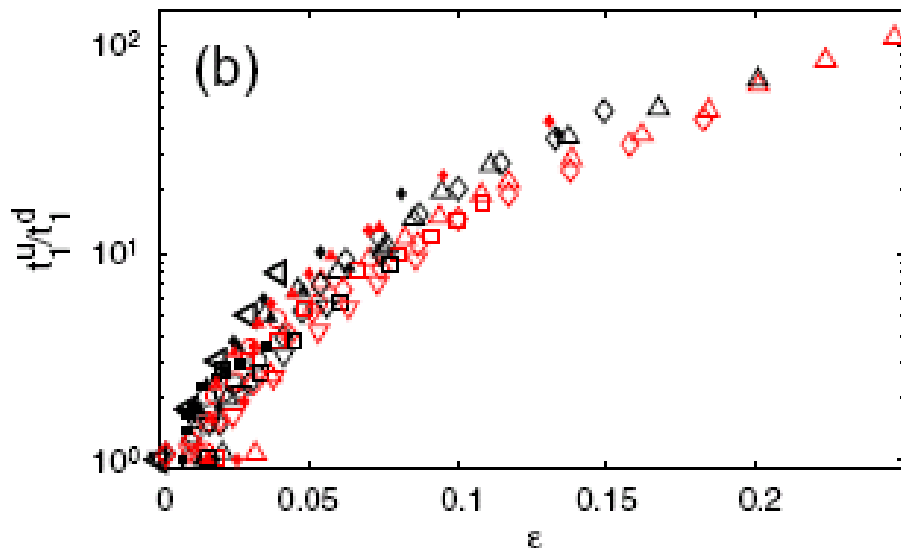
- three different t_w and three strain rates $\dot{\epsilon}$
- distributions “cut off” at times $\sim 0.1/\dot{\epsilon}$ but no change at small times
- aging stopped due to truncation of persistence time distribution

Acceleration ratio



step strain

$$P(t) = \int_0^t p(t_1) dt_1$$



- describe transformation of hop time dists through cumulatives

- define an **acceleration ratio** through times when $P_u(t_1^u) = P_d(t_1^d)$

- acceleration ratio collapses onto universal curve when plotted against total strain

- (local) strain is good variable to describe accelerated dynamics (see also SGR model)

Summary

- Simulations reproduce mechanical behavior typical of glasses. Slow relaxation (aging) changes yield stress and creep compliance.
- Molecular mobility controls mechanical response
- **Reduction of aging exponent** under subyield stress, but glass returns to original aging trajectory; erasure of aging only through plastic flow
- Robust picture of activated hopping dynamics with broad distribution of relaxation times
 - **only first hop time waiting time dependent**
 - particle/segment 'forgets' its age after one hop
 - supports picture of **annealed disorder** as assumed in trap model
- Accelerated dynamics: narrowing of relaxation time spectrum
universal dependence of acceleration on strain