## EXPERIMENTALLY TUNING THE GROUND STATE OF BaFe<sub>2</sub>As<sub>2</sub> BY ORBITAL DIFFERENTIATION

Priscila F. S. Rosa<sup>1,2</sup>, C. Adriano<sup>1</sup>, T. M. Garitezi<sup>1</sup>, T. M. Garitezi<sup>1</sup>, T. Grant<sup>2</sup>, Z. Fisk<sup>2</sup> and P. G. Pagliuso<sup>1</sup>

Institute of Physics "Gleb Wataghin", UNICAMP, Campinas SP, 13083-970, Brazil
 University of California, Irvine, California 92697, USA



Magnetism, Bad Metals and Superconductivity: Iron Pnictides and Beyond

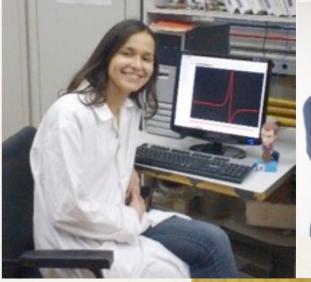
October 7, 2014

## FeAs Team













- Prof. Cris Adriano
- Prof. Eduardo Bittar
- Prof. Eduardo Granado
- \* Dr. Priscila Rosa
- \* Dr. Ted Grant

Prof. Pascoal Pagliuso

Prof. Ricardo Urbano

Dr. Thales Garitezi

Prof. Zachary Fisk





## FeAs Team









- \* Prof. Cris Adriano
- Prof. Eduardo Bittar
- Prof. Eduardo Granado
- \* Dr. Priscila Rosa
- \* Dr. Ted Grant

Prof. Pascoal Pagliuso

Prof. Ricardo Urbano

Dr. Thales Garitezi

Prof. Zachary Fisk





\* Introduction and Motivation: Fe-based Superconductors

- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

 Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

\* Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

Conclusions

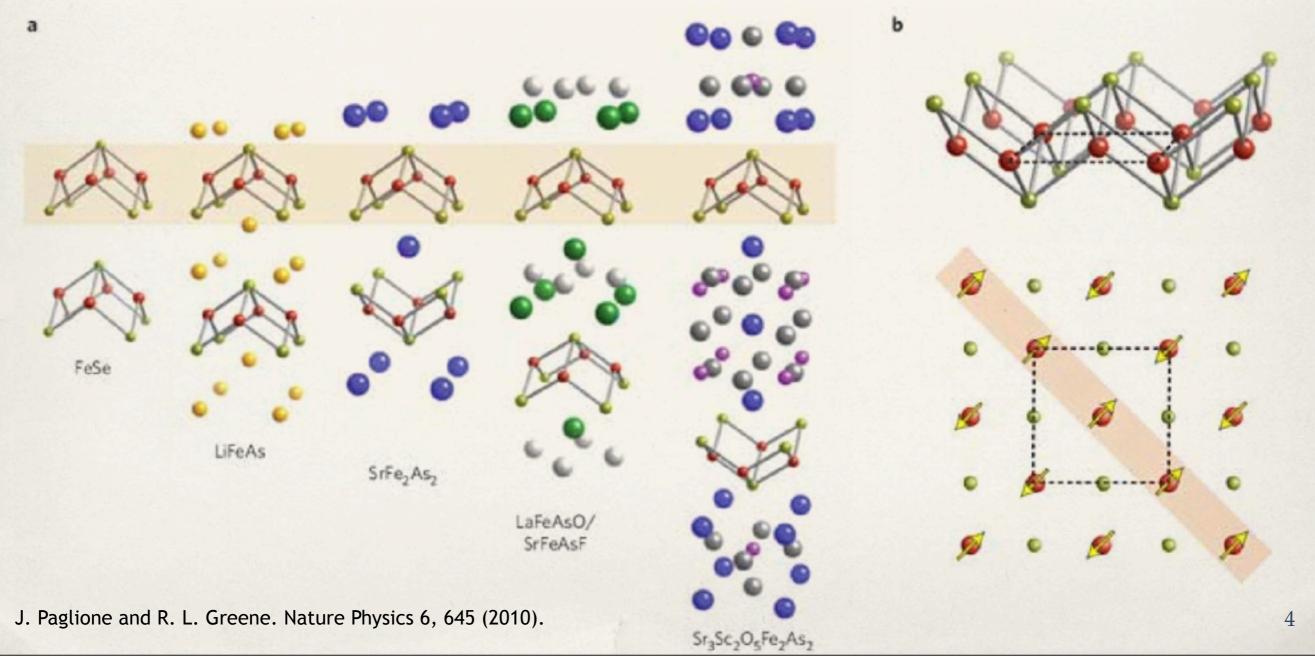
- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

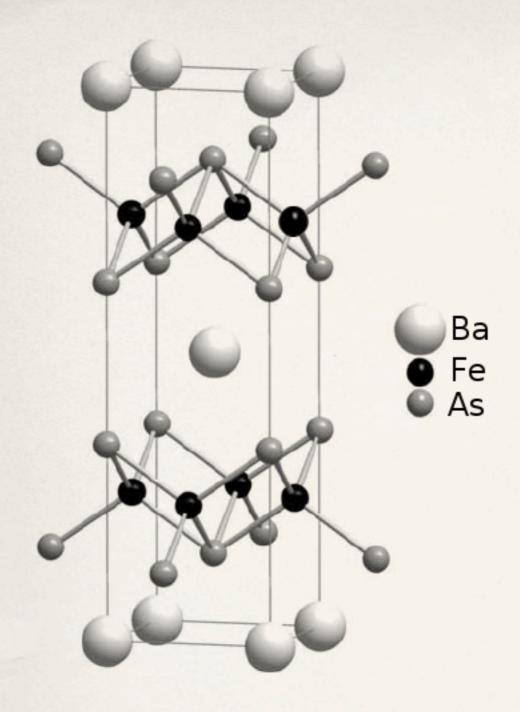
\* Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

\* Conclusions

## Fe-based superconductors

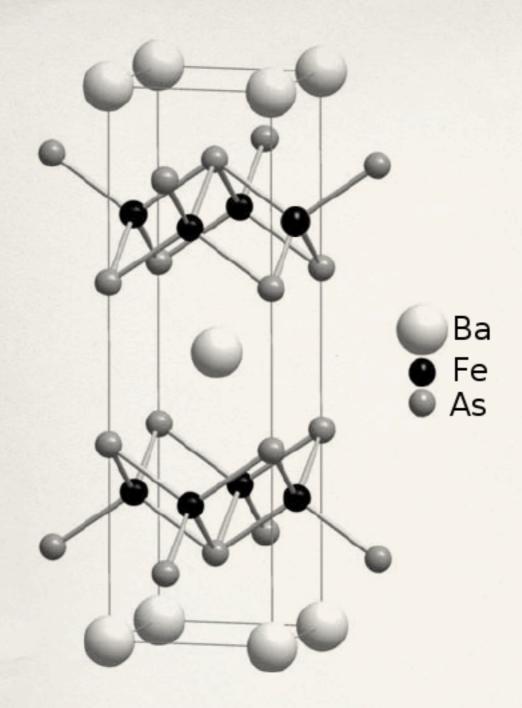
- Common Structural Parameter: FeAs<sub>4</sub> tetrahedra
- \* Common Electronic Parameter: Spin-Density Wave (SDW) magnetic instability.





Good sample quality;

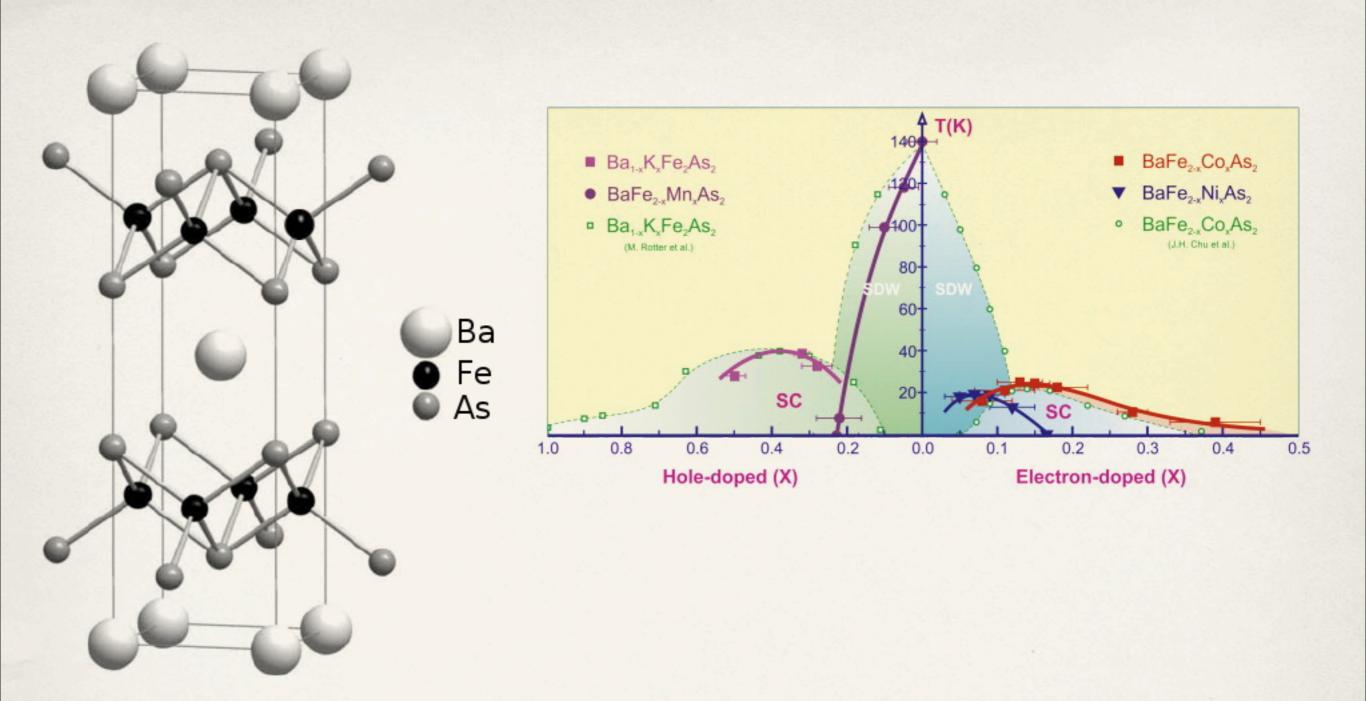
Variety of compounds.



#### Good sample quality;

#### Variety of compounds.

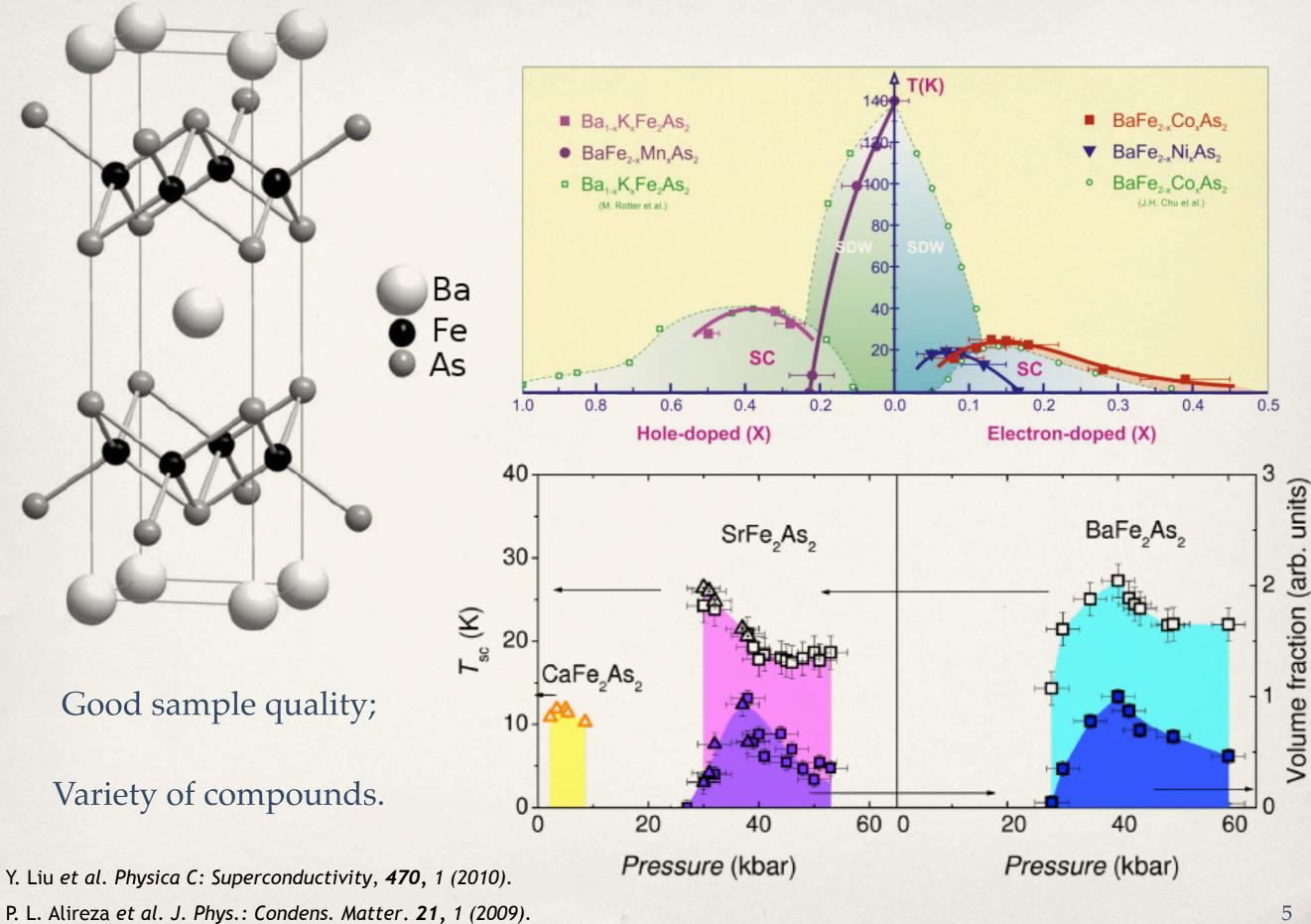
Y. Liu et al. Physica C: Superconductivity, 470, 1 (2010).P. L. Alireza et al. J. Phys.: Condens. Matter. 21, 1 (2009).



Good sample quality;

#### Variety of compounds.

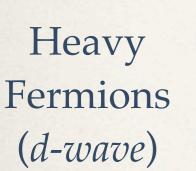
Y. Liu et al. Physica C: Superconductivity, 470, 1 (2010).P. L. Alireza et al. J. Phys.: Condens. Matter. 21, 1 (2009).

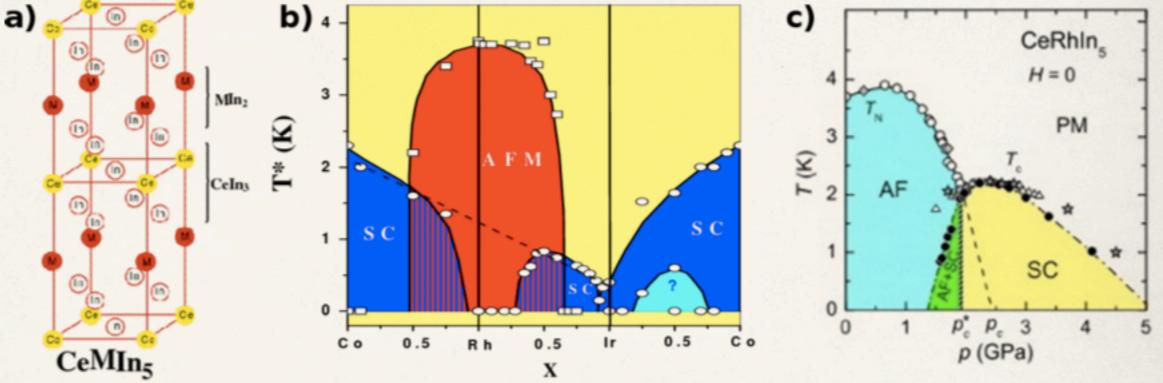


How one can relate Fe-based superconductors to other unconventional superconductors?

P. G. Pagliuso *et al.* Physica B 312-313 (2002); G. Knebel *et al. J. Phys. Soc. Japan* **77** 114704 (2008) N. Yamada and M. Ido. Physica C 203, 240 (1992).

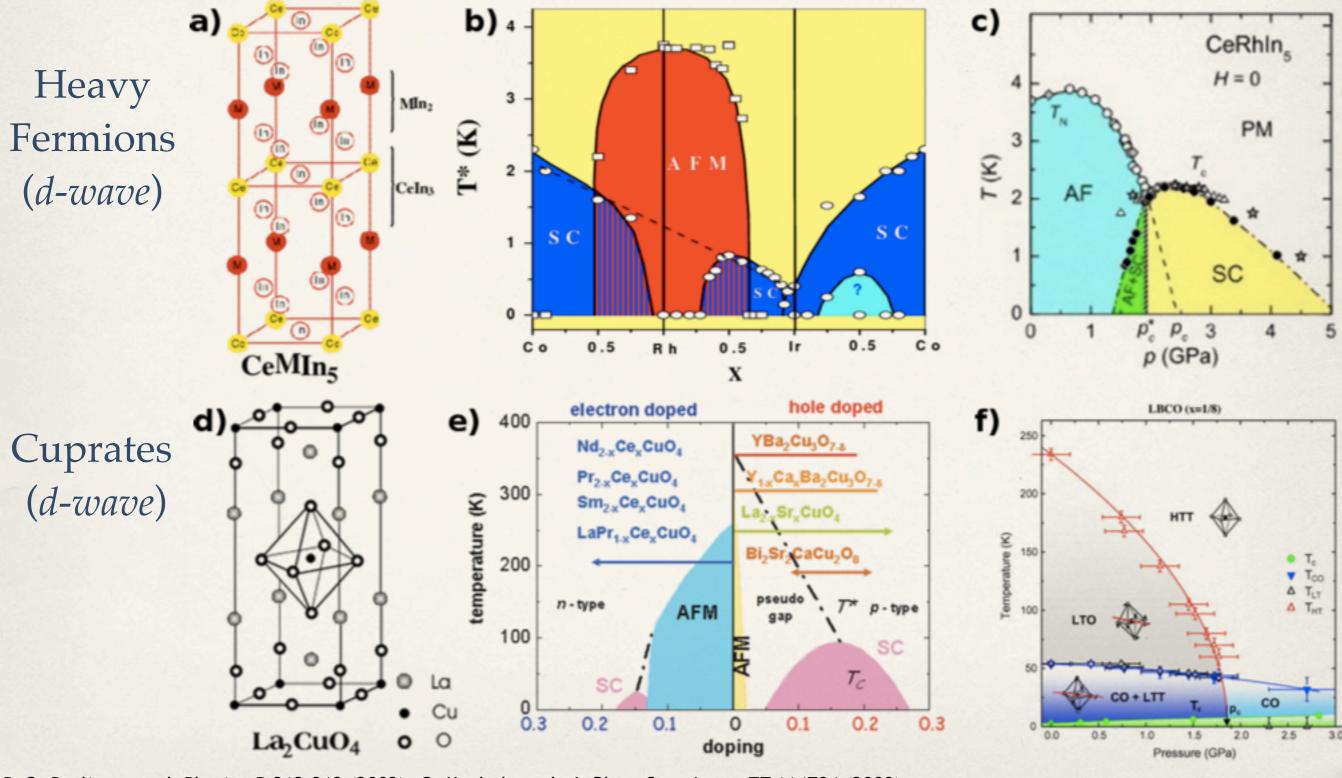
How one can relate Fe-based superconductors to other unconventional superconductors?





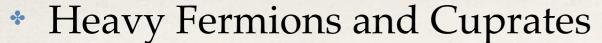
P. G. Pagliuso *et al.* Physica B 312-313 (2002); G. Knebel *et al. J. Phys. Soc. Japan* **77** 114704 (2008) N. Yamada and M. Ido. Physica C 203, 240 (1992).

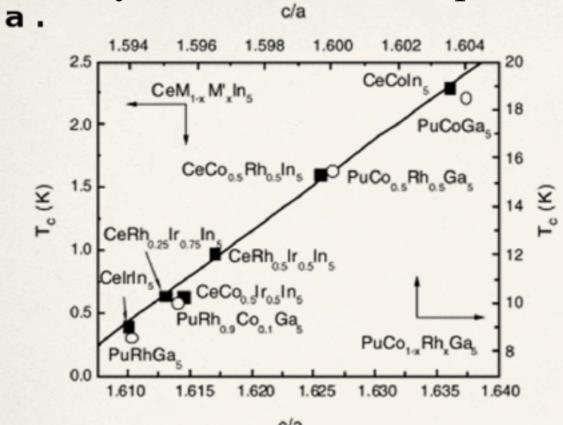
How one can relate Fe-based superconductors to other unconventional superconductors?



P. G. Pagliuso *et al.* Physica B 312-313 (2002); G. Knebel *et al. J. Phys. Soc. Japan* **77** 114704 (2008) N. Yamada and M. Ido. Physica C 203, 240 (1992).

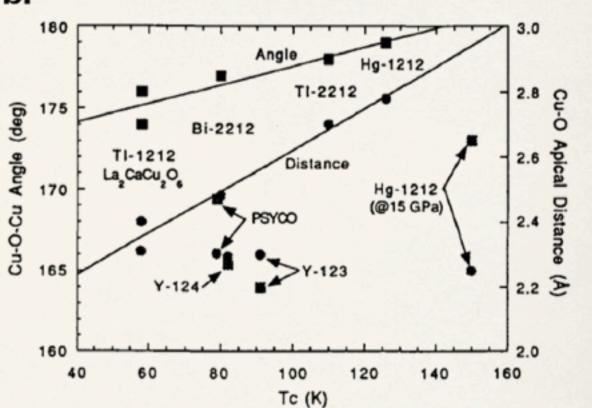
### Structural Similarities





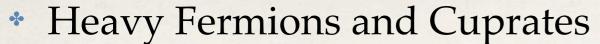


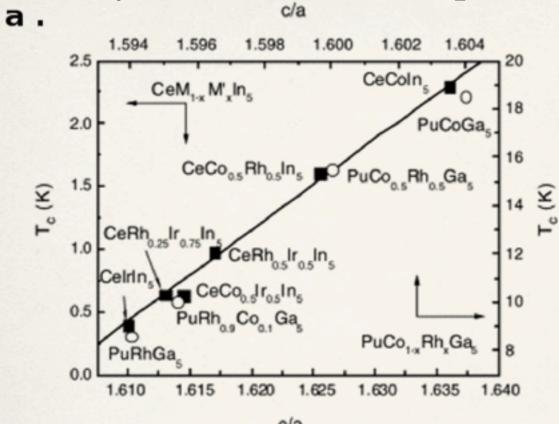
b.



J. D. Jorgensen *et al.* Invited paper for the Conference on High Temperature SC, 1995, Poland.

### **Structural Similarities**

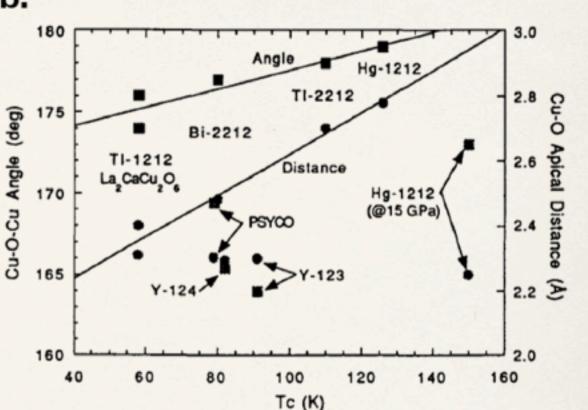






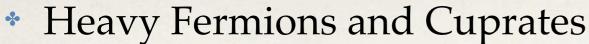
#### What about the FeAs?

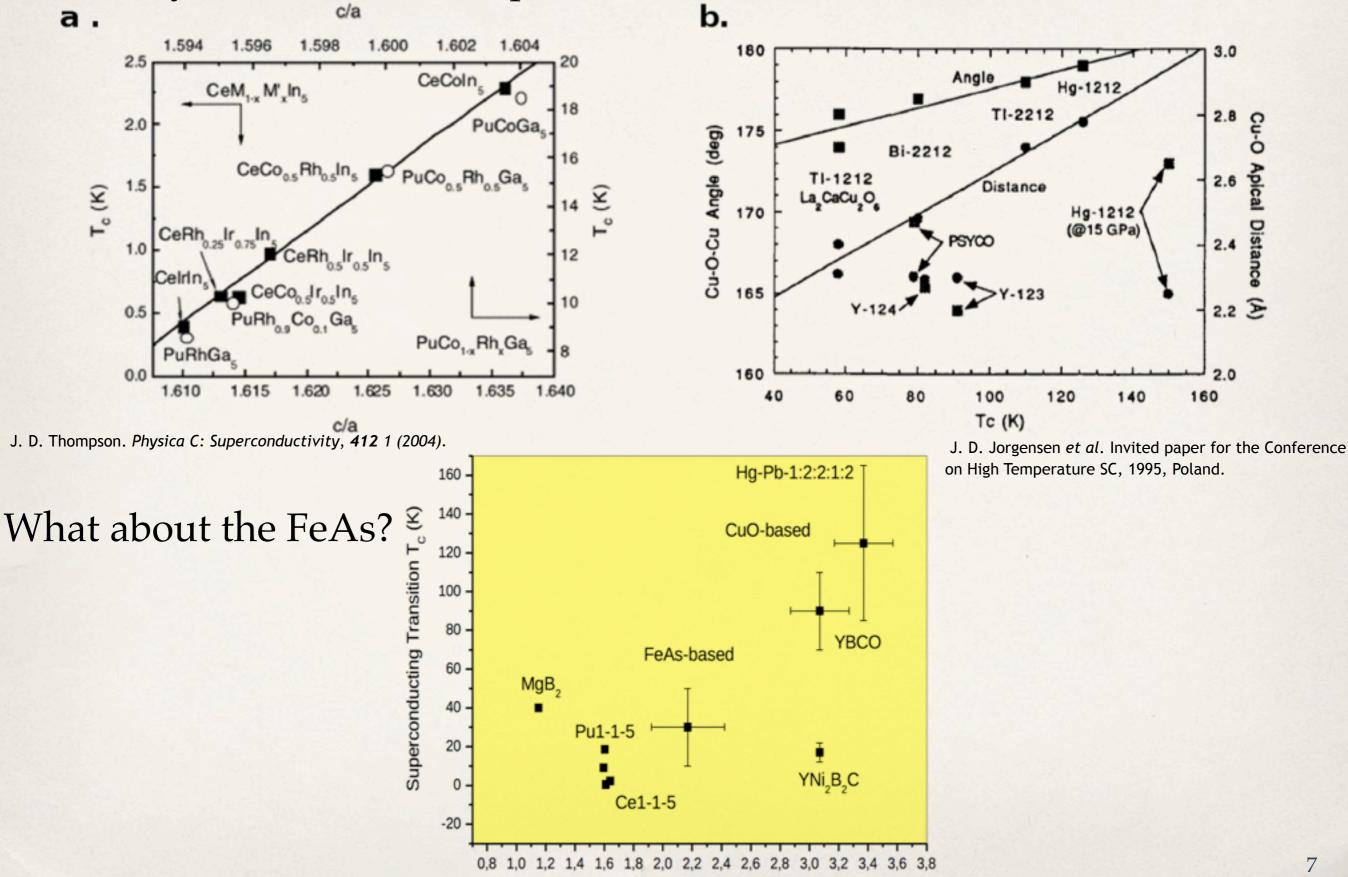
b.



J. D. Jorgensen *et al.* Invited paper for the Conference on High Temperature SC, 1995, Poland.

### Structural Similarities





\* Introduction and Motivation: Fe-based Superconductors

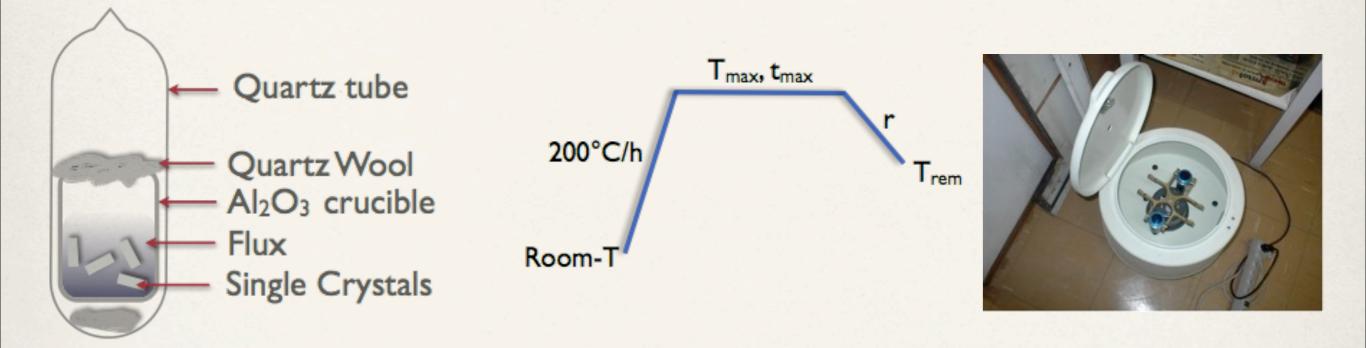
- \* Alternative Single Crystal Growth and Macroscopic Properties
- Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

\* Conclusions and Perspectives

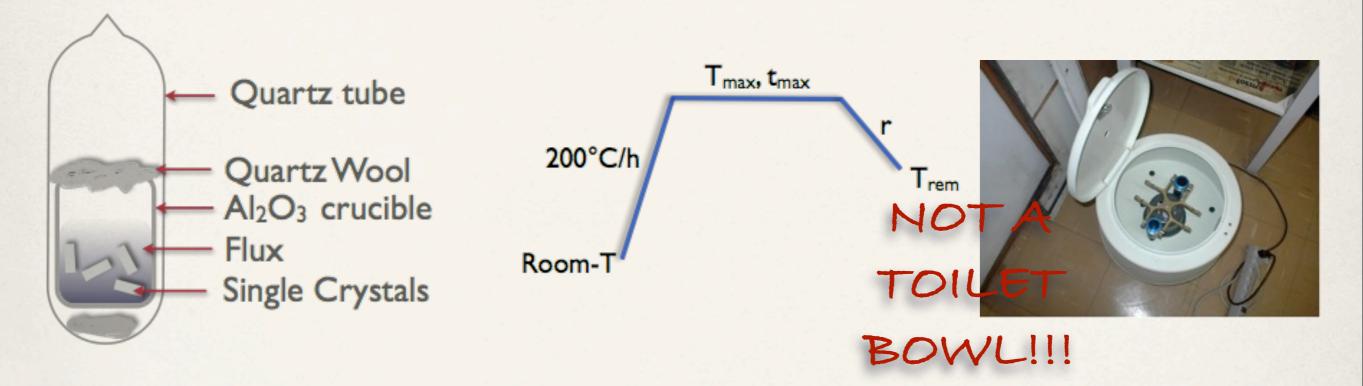
\* Introduction and Motivation: Fe-based Superconductors

- \* Alternative Single Crystal Growth and Macroscopic Properties
- \* Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

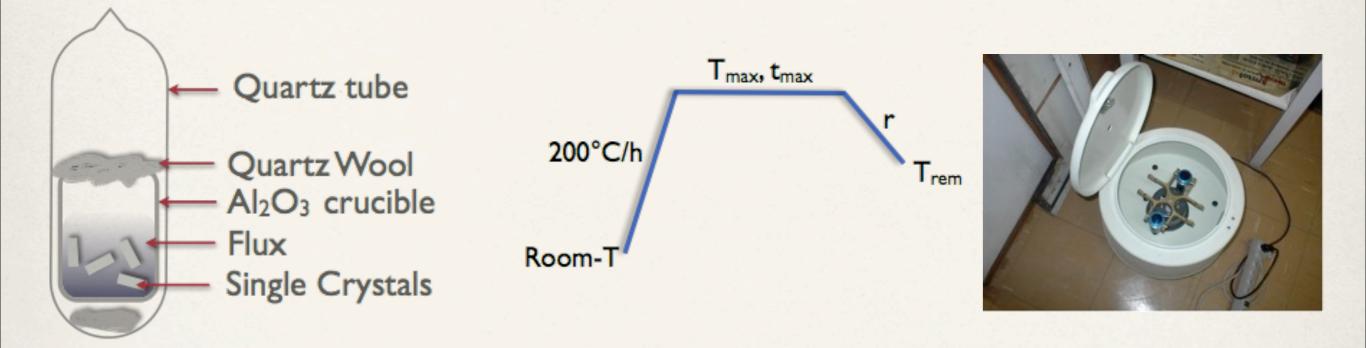
\* Conclusions and Perspectives



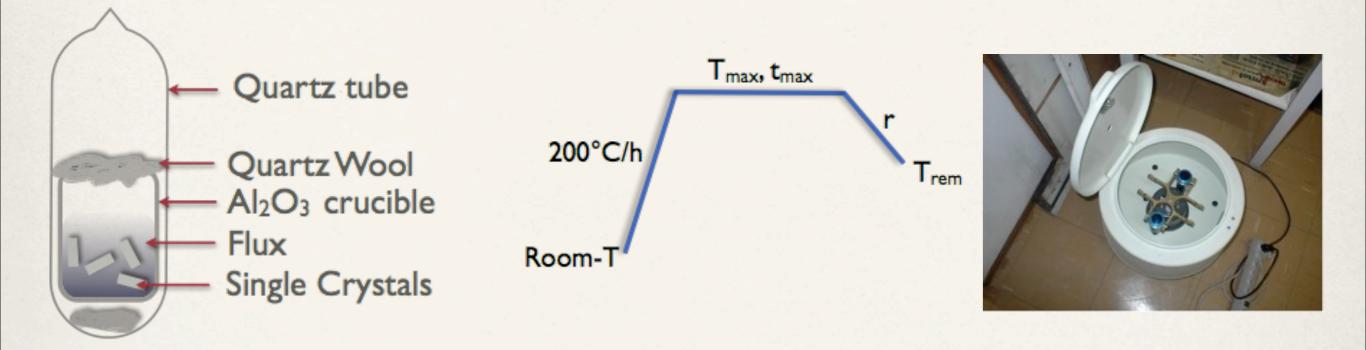
 Low melting point elements (Al, Ga, In, Sn, Pb, Sb e Bi): low temperature of synthesis, well-defined morphology.



 Low melting point elements (Al, Ga, In, Sn, Pb, Sb e Bi): low temperature of synthesis, well-defined morphology.



 Low melting point elements (Al, Ga, In, Sn, Pb, Sb e Bi): low temperature of synthesis, well-defined morphology.



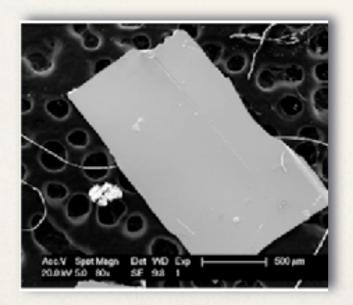
- Low melting point elements (Al, Ga, In, Sn, Pb, Sb e Bi): low temperature of synthesis, well-defined morphology.
- Current disadvantages: Sn-incorporation and possible non-stoichiometric 122 compounds using self-(FeAs-) flux.

## Indium flux $\rightarrow$ BaFe<sub>2</sub>As<sub>2</sub>

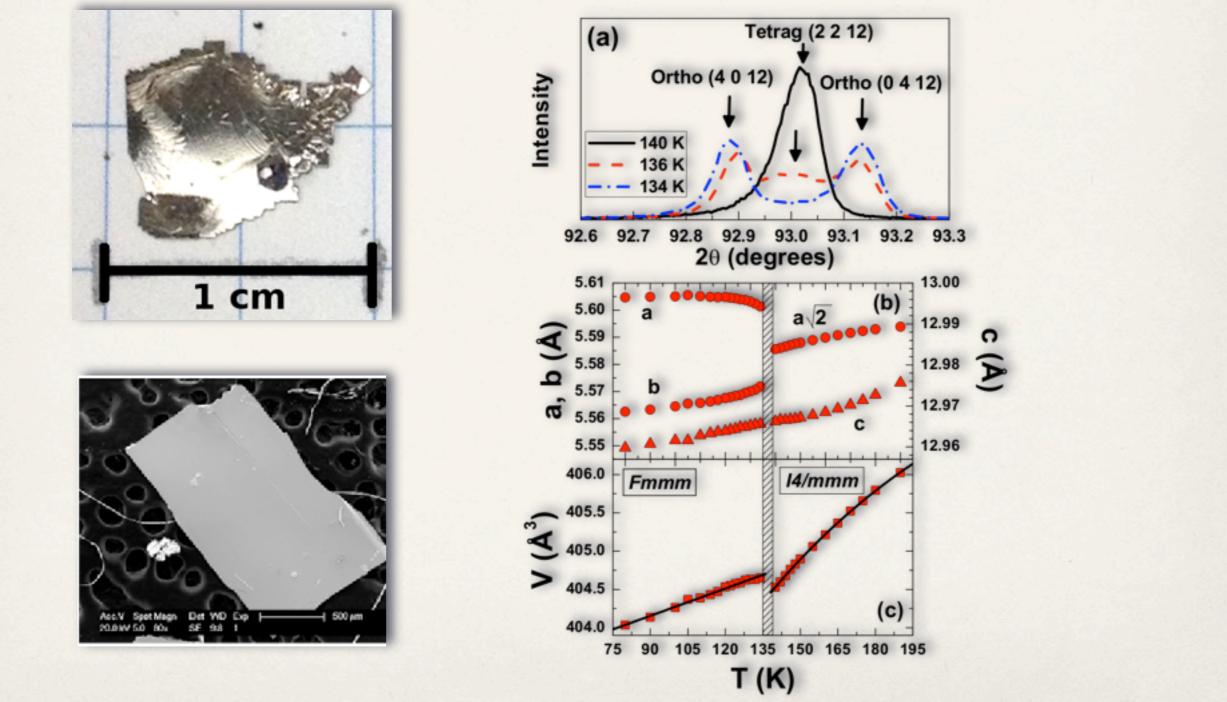


## Indium flux $\rightarrow$ BaFe<sub>2</sub>As<sub>2</sub>

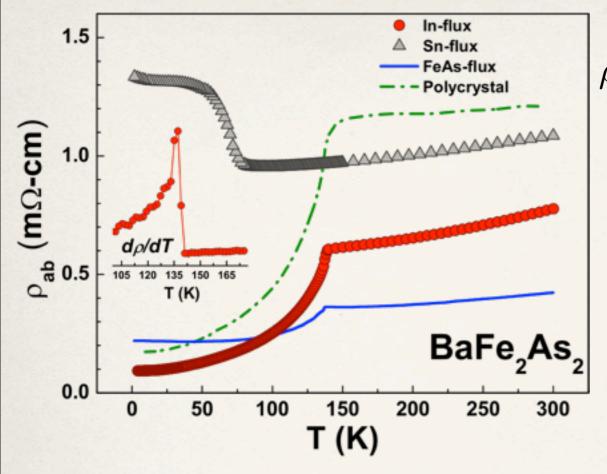




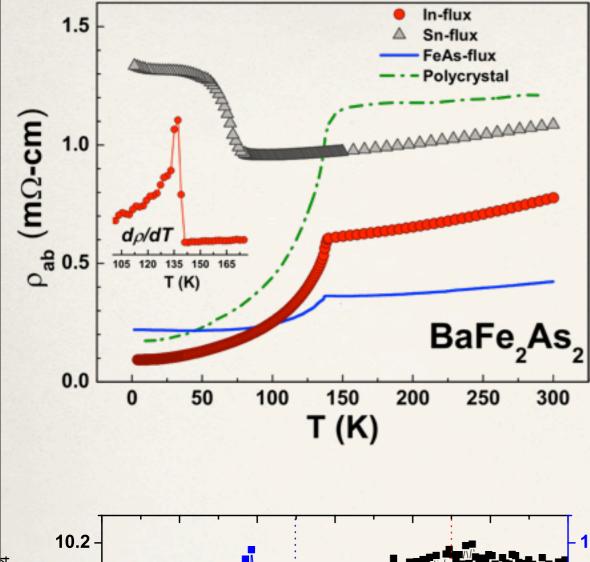
#### Indium flux $\rightarrow$ BaFe<sub>2</sub>As<sub>2</sub>

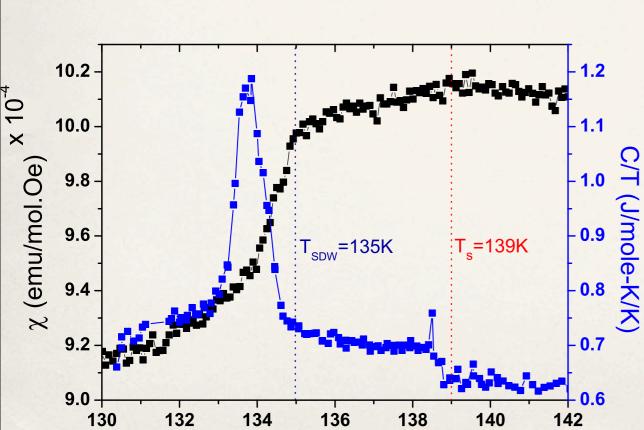


T. M. Garitezi, C. Adriano, P. F. S. Rosa et al. Brazilian Journal of Physics 43 (2013).



 $ho_0 \approx 0.1 m \Omega.cm$  $RRR \sim 8$ 





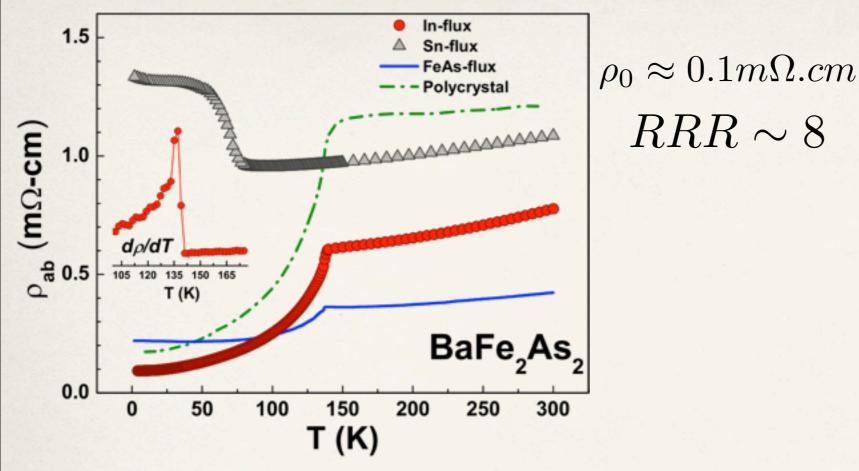
T (K)

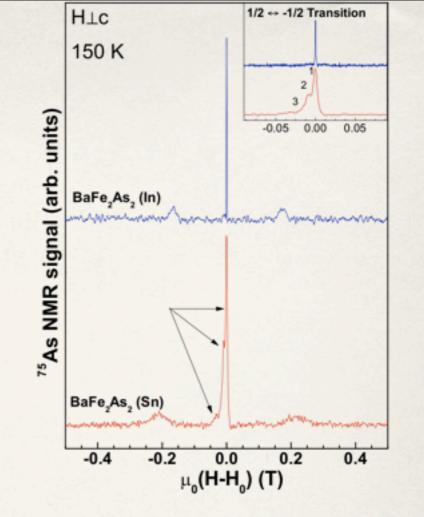
 $ho_0 \approx 0.1 m \Omega.cm$  $RRR \sim 8$ 

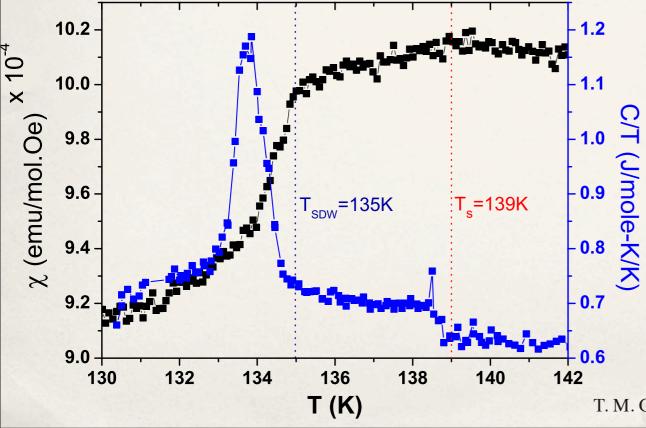
Thursday, October 9, 2014

T. M. Garitezi, G. G. Lesseux et al. JOURNAL OF APPLIED PHYSICS 115, 17D711 (2014) 11

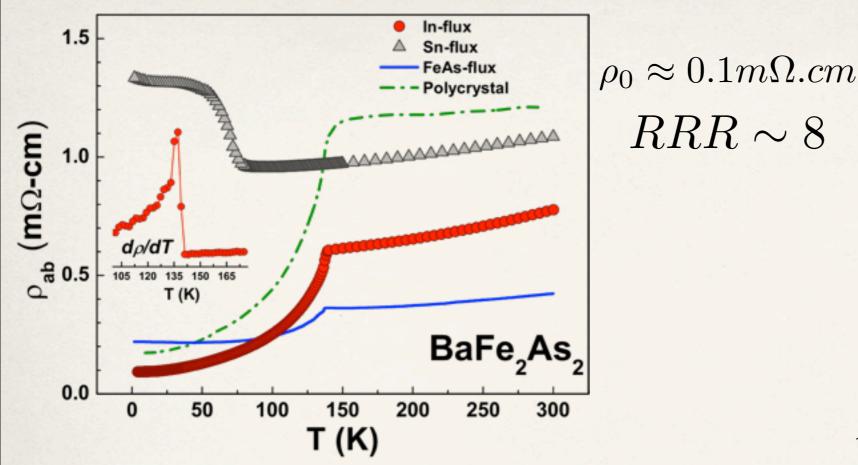
 $RRR \sim 8$ 



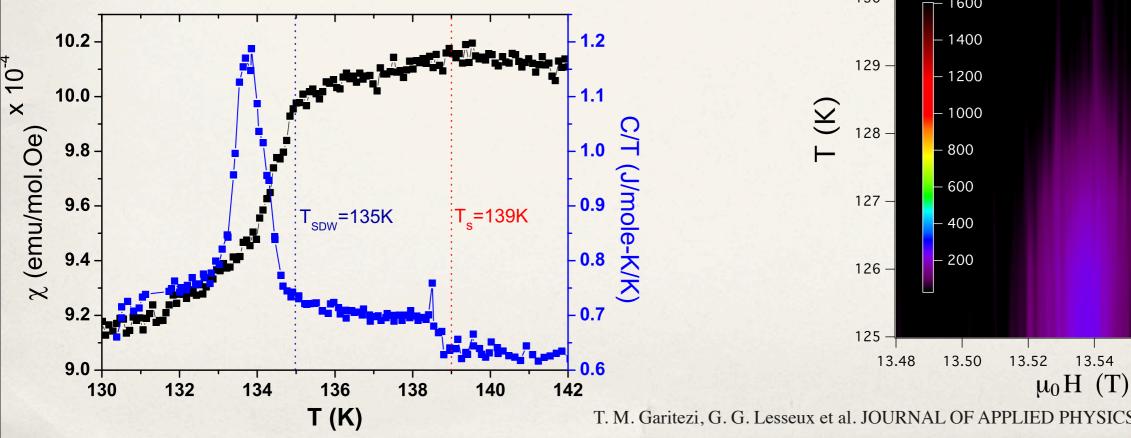




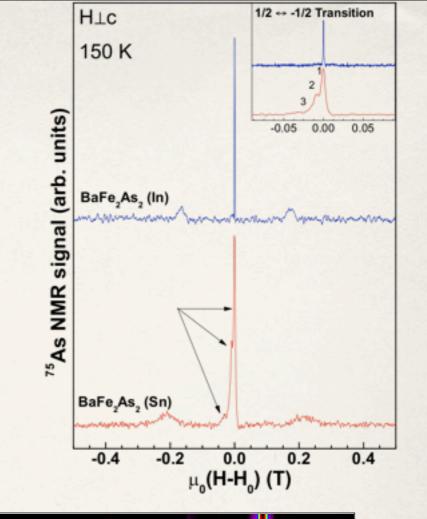
11 T. M. Garitezi, G. G. Lesseux et al. JOURNAL OF APPLIED PHYSICS 115, 17D711 (2014)

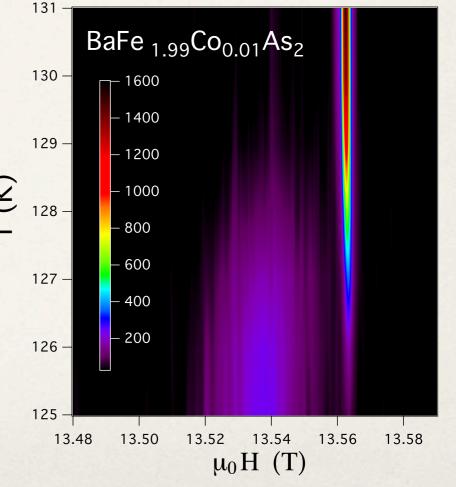


T. M. Garitezi, C. Adriano, P. F. S. Rosa et al. Brazilian Journal of Physics 43 (2013).



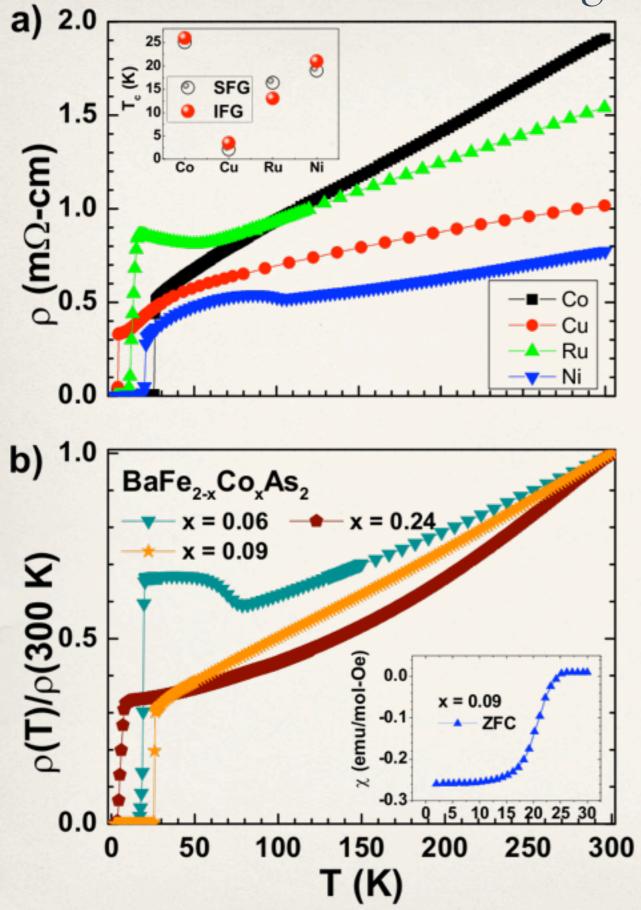
 $RRR \sim 8$ 



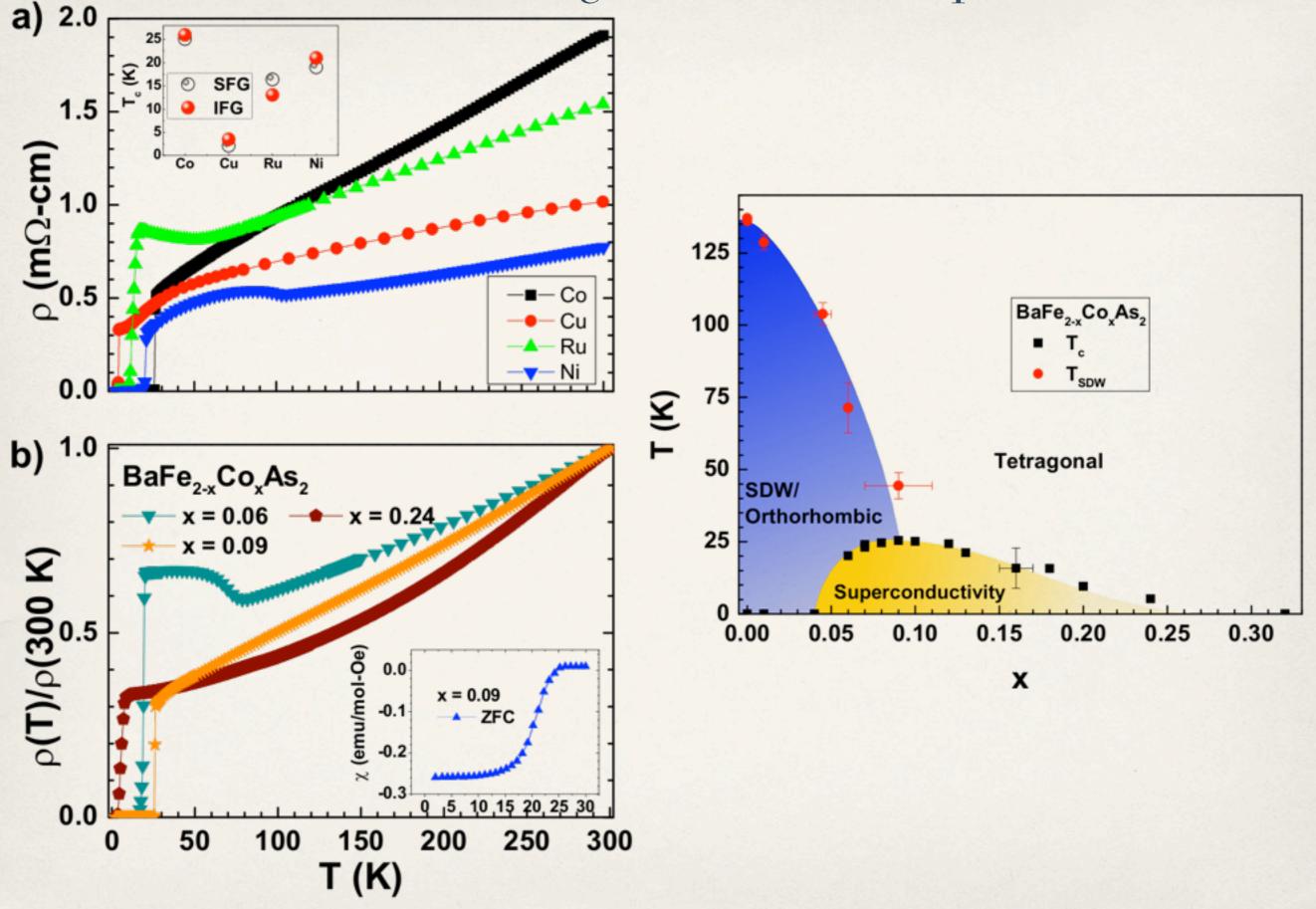


11 T. M. Garitezi, G. G. Lesseux et al. JOURNAL OF APPLIED PHYSICS 115, 17D711 (2014)

#### $BaFe_{2-x}M_xAs_2$ : higher critical temperatures

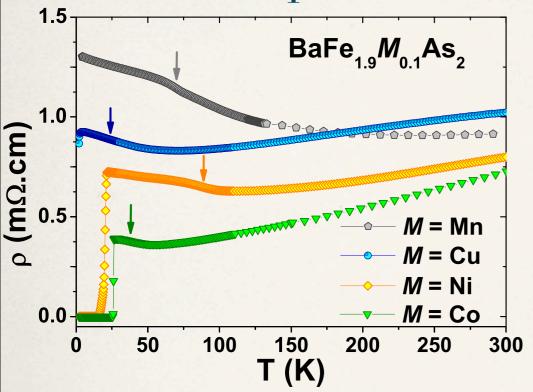


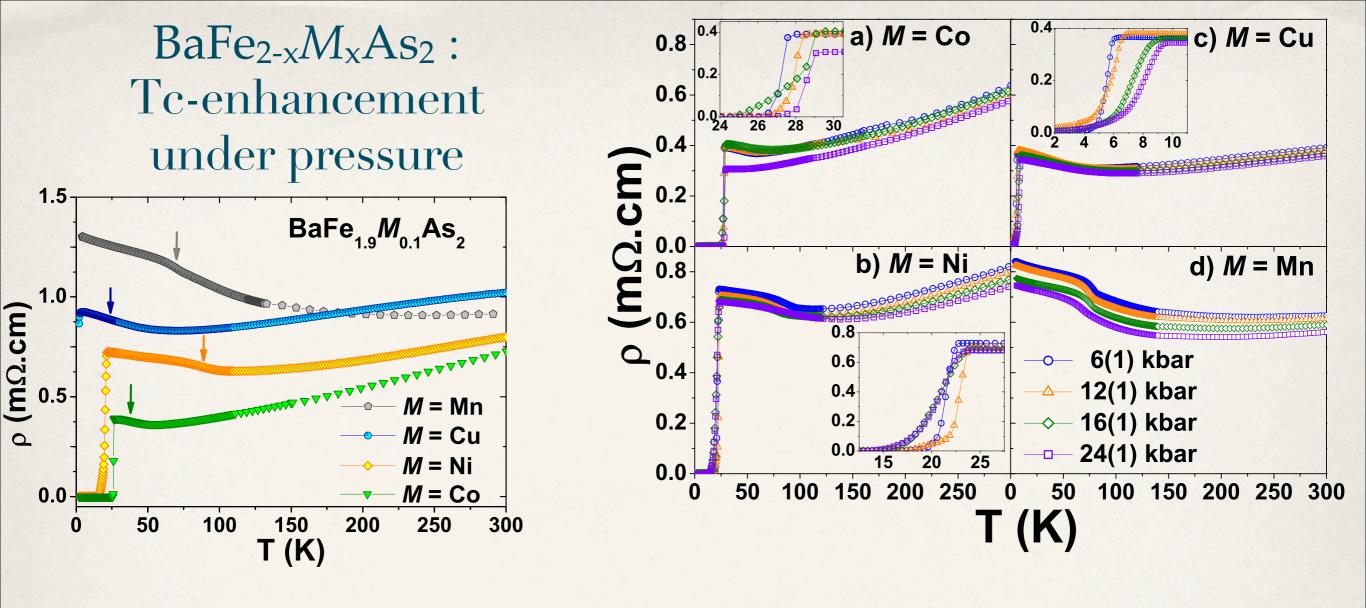
#### $BaFe_{2-x}M_xAs_2$ : higher critical temperatures

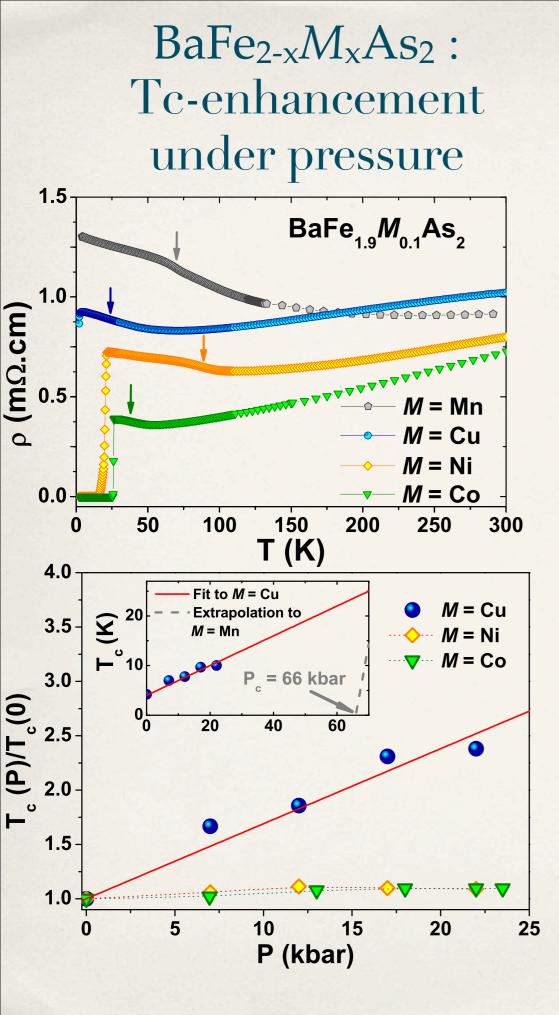


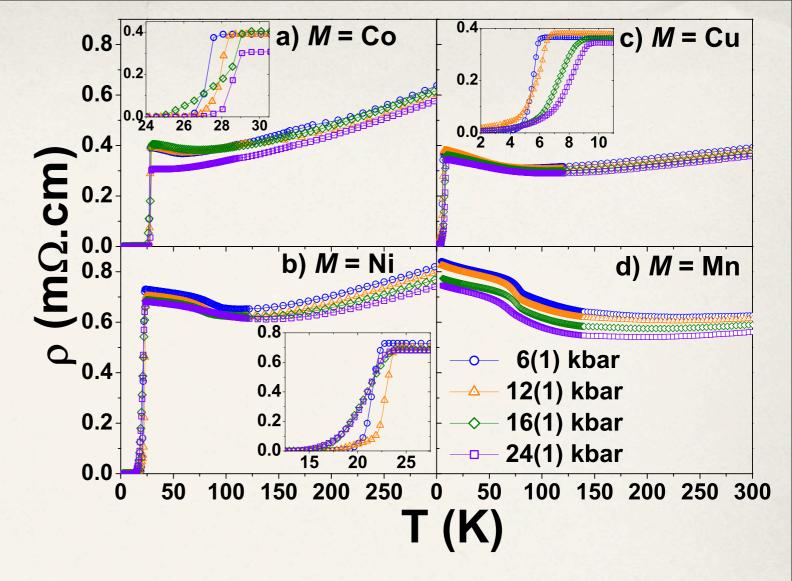
T. M. Garitezi, C. Adriano, P. F. S. Rosa et al. Brazilian Journal of Physics 43 (2013).

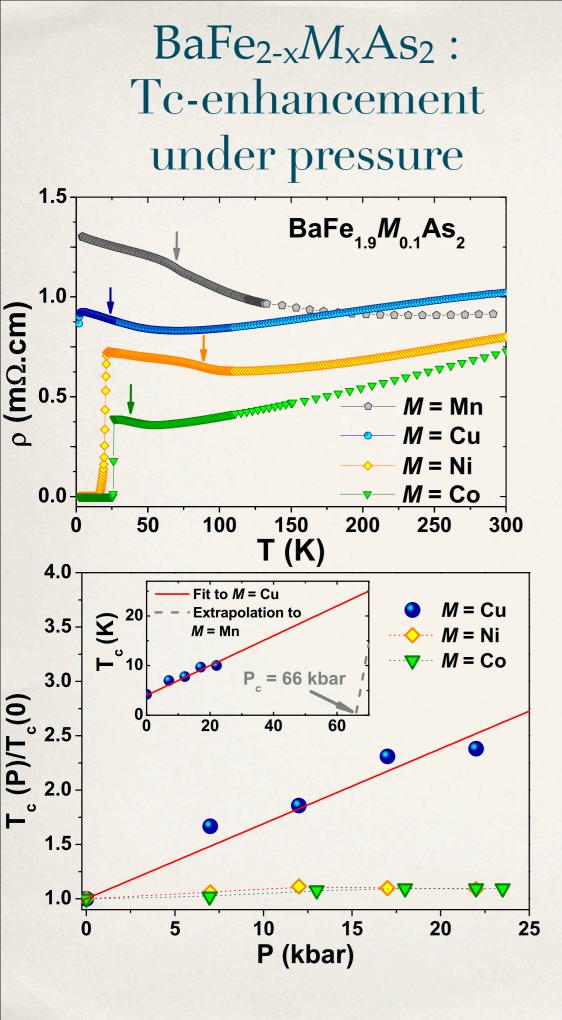
 $BaFe_{2-x}M_xAs_2$ : Tc-enhancement under pressure

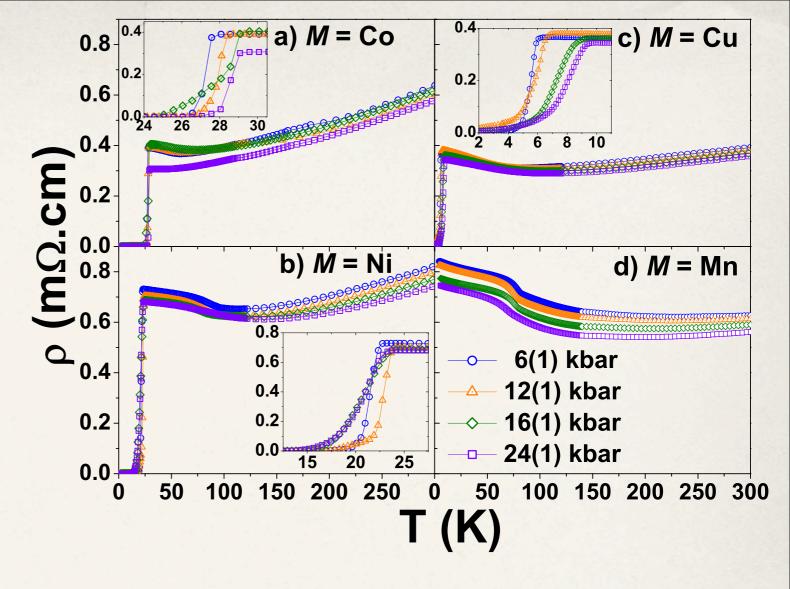


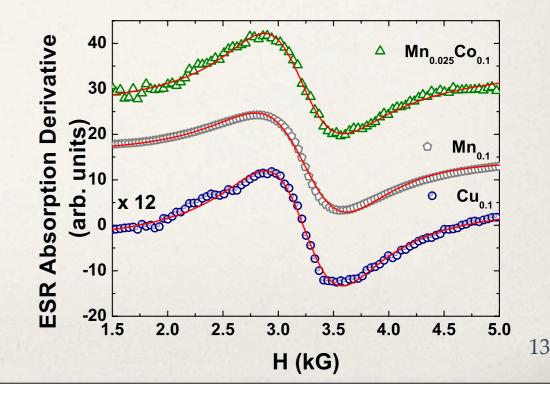


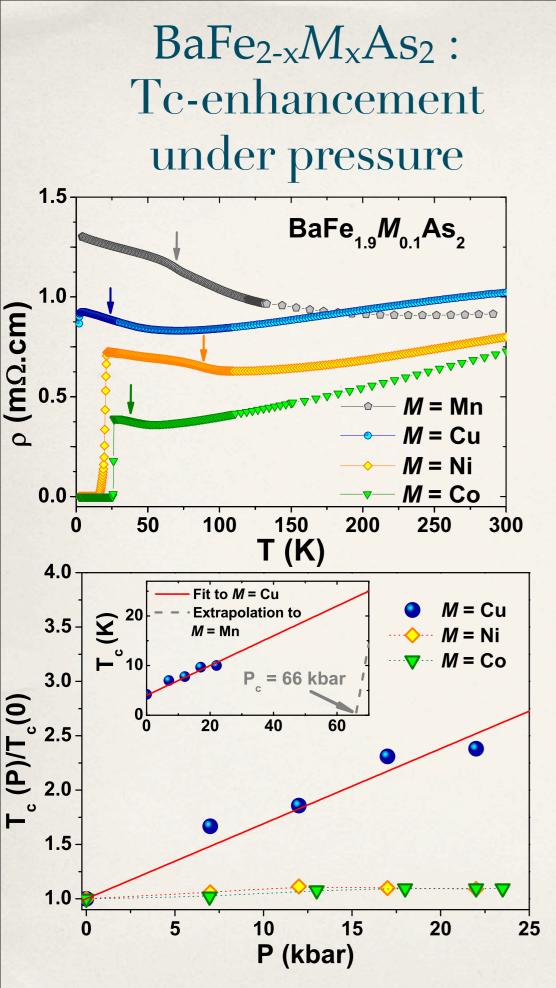




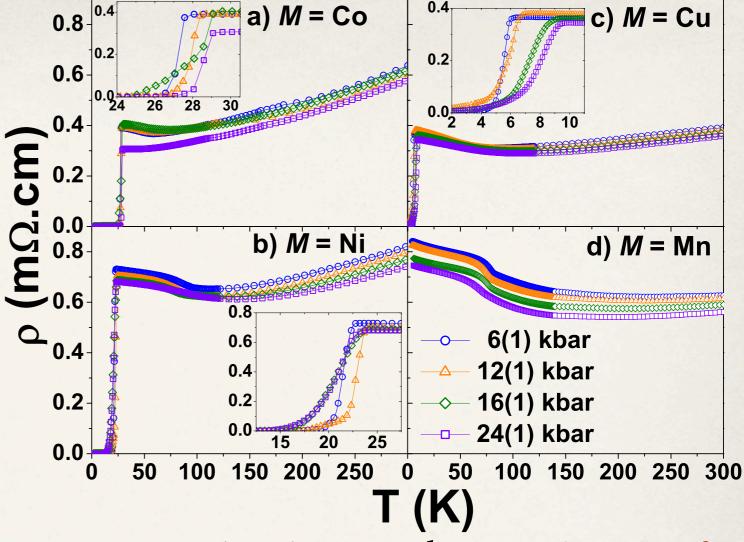




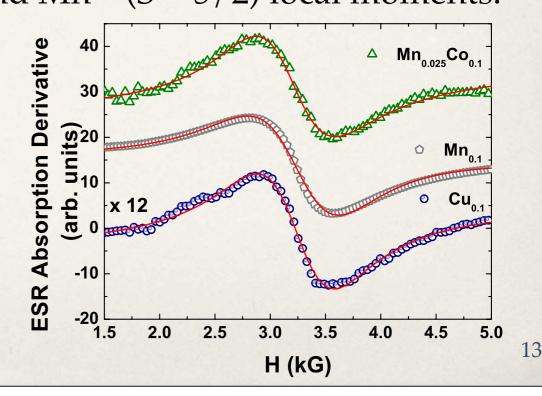




**P. F. S. Rosa** et al.Sci. Rep. 4, 6252; DOI:10.1038/srep06252 (2014).



• Magnetic pair-breaking mechanism due to  $Cu^{2+}$ (S = 1/2) and Mn<sup>2+</sup> (S = 5/2) local moments.



# Outline

- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

 Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

\* Conclusions and Perspectives

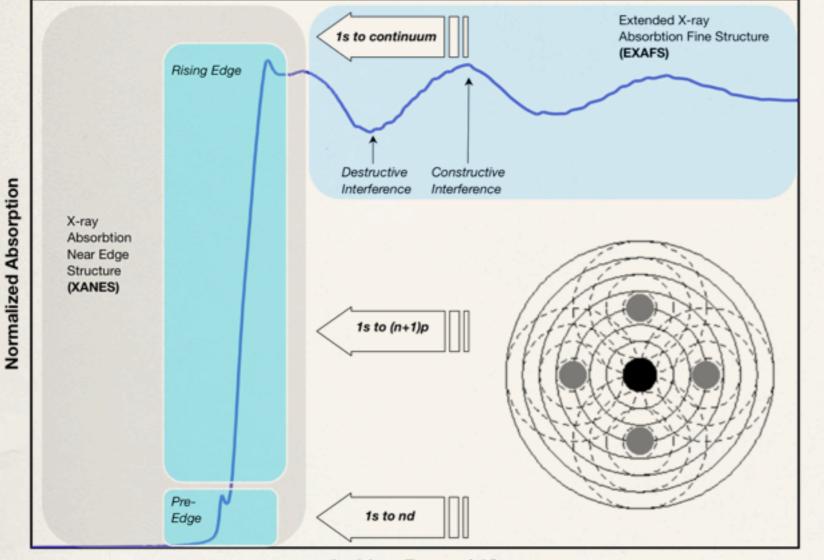
# Outline

- \* Introduction and Motivation: Fe-based Superconductors
- \* Alternative Single Crystal Growth and Macroscopic Properties

 Microscopic Investigation: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

\* Conclusions and Perspectives

X-Ray Absorption Spectroscopy



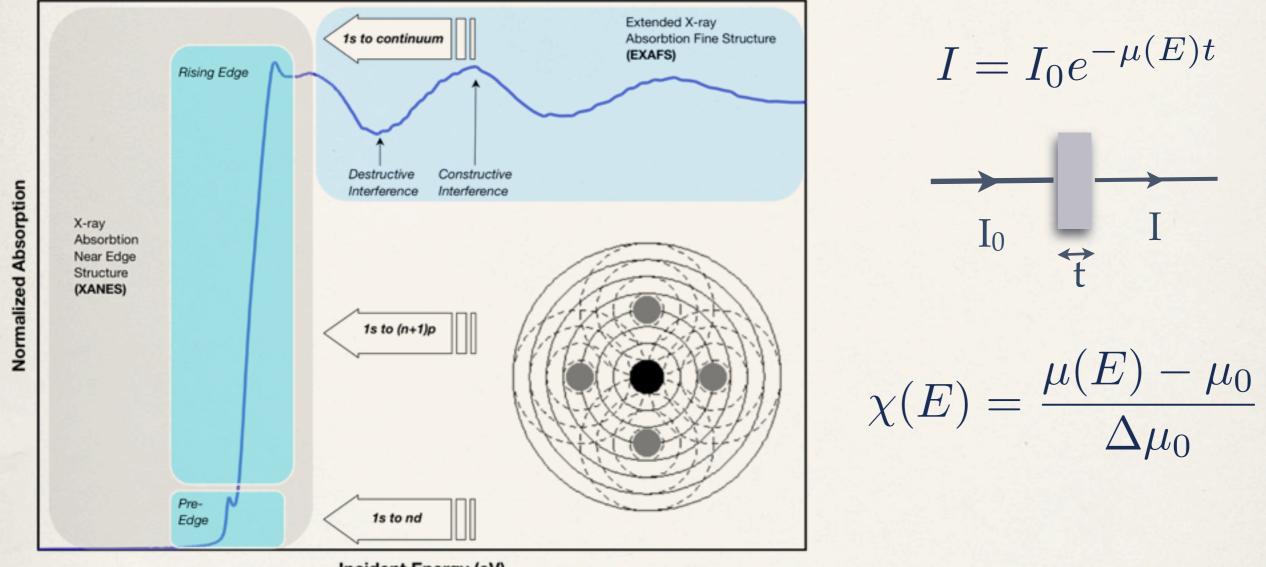
Incident Energy (eV)

 $I = I_0 e^{-\mu(E)t}$ 

 $I_0$ 

For hv < 40 keV — photoelectron emission

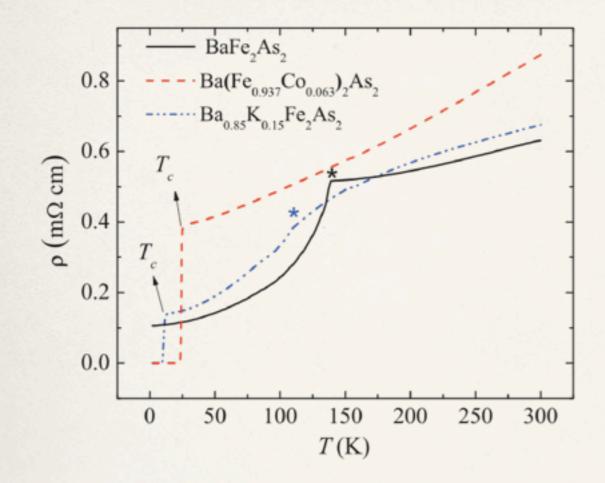
X-Ray Absorption Spectroscopy



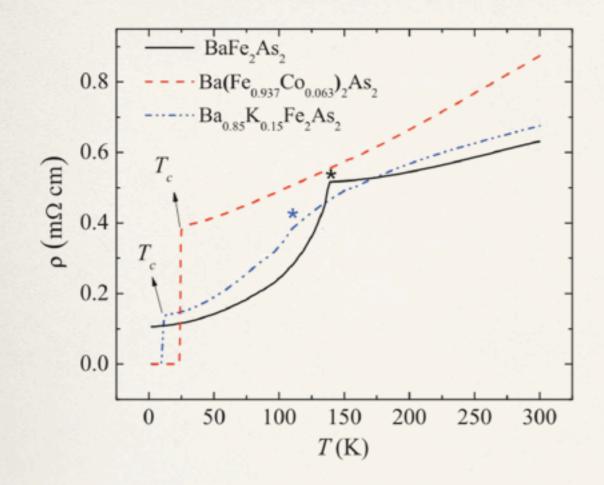
Incident Energy (eV)

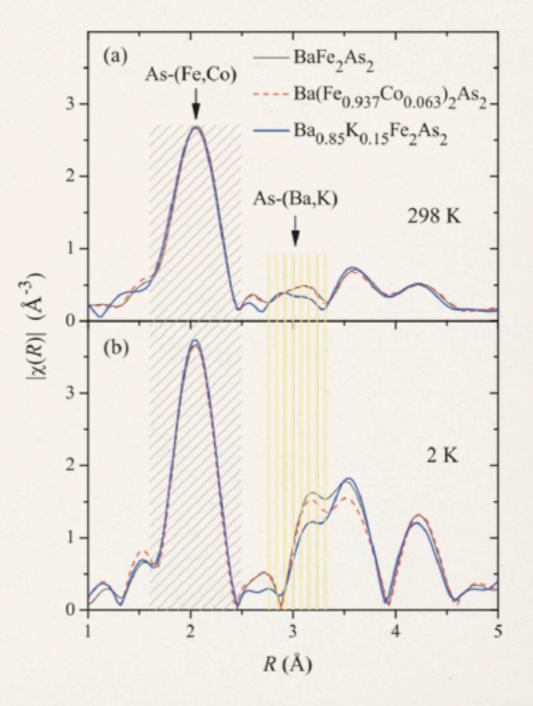


\* EXAFS in the As *K* edge in transmission mode:

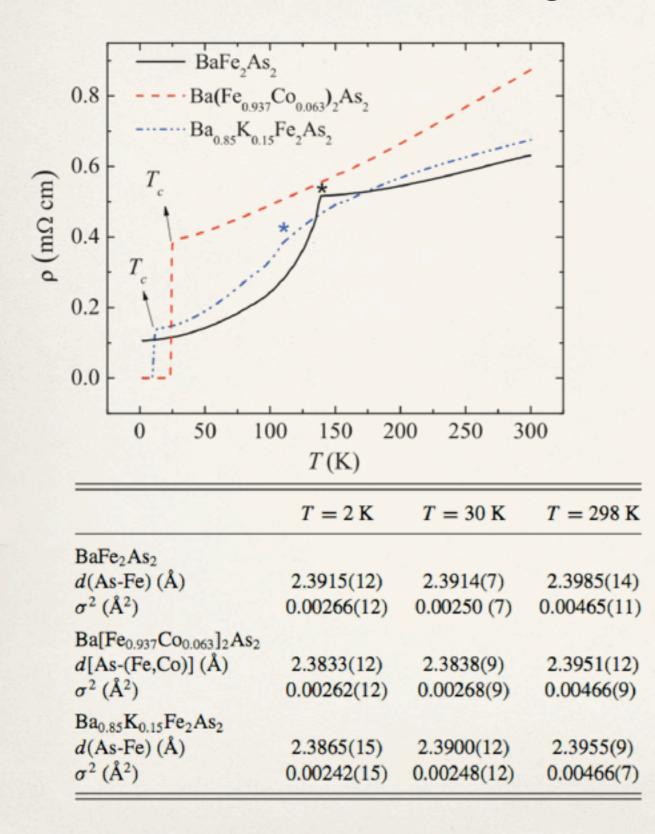


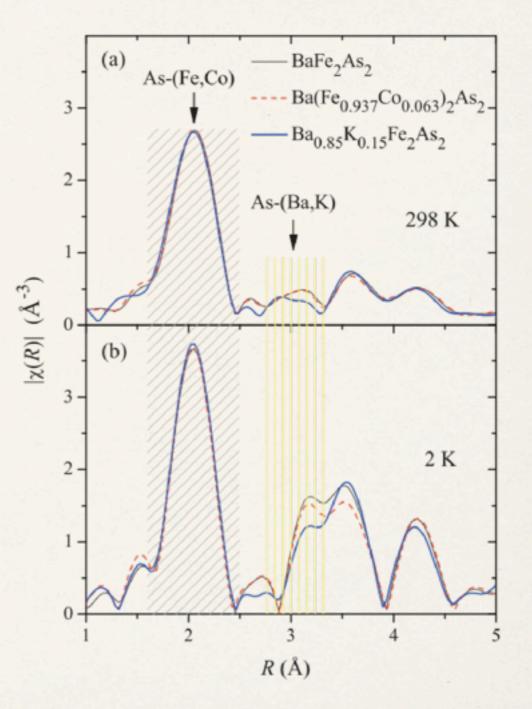
Microscopic Properties: Evolution of the FeAs distance\* EXAFS in the As *K* edge in transmission mode:



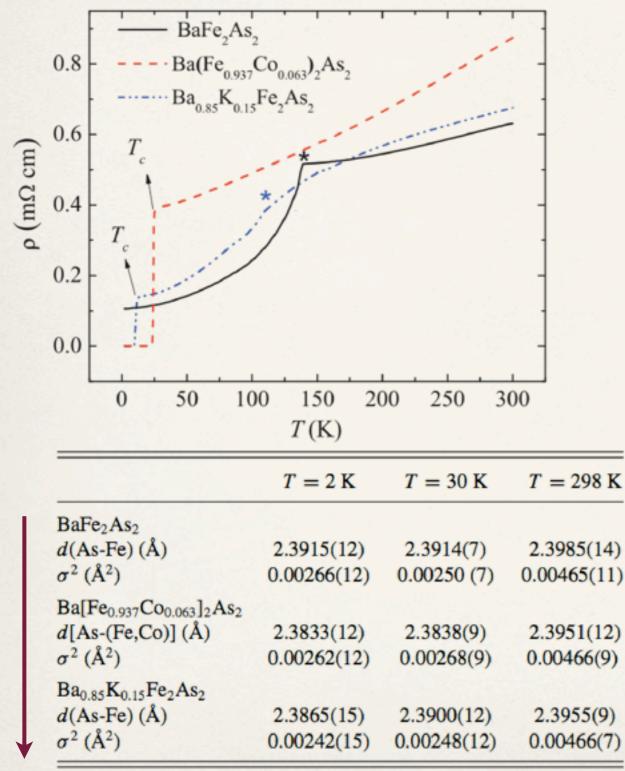


Experiments performed at XAFS2 beamline of the Brazilian Synchrotron Light Laboratory (LNLS). Microscopic Properties: Evolution of the FeAs distance\* EXAFS in the As *K* edge in transmission mode:



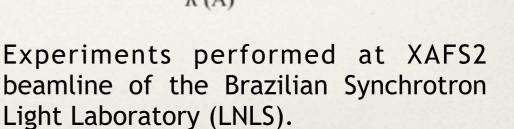


Experiments performed at XAFS2 beamline of the Brazilian Synchrotron Light Laboratory (LNLS). Microscopic Properties: Evolution of the FeAs distance\* EXAFS in the As *K* edge in transmission mode:



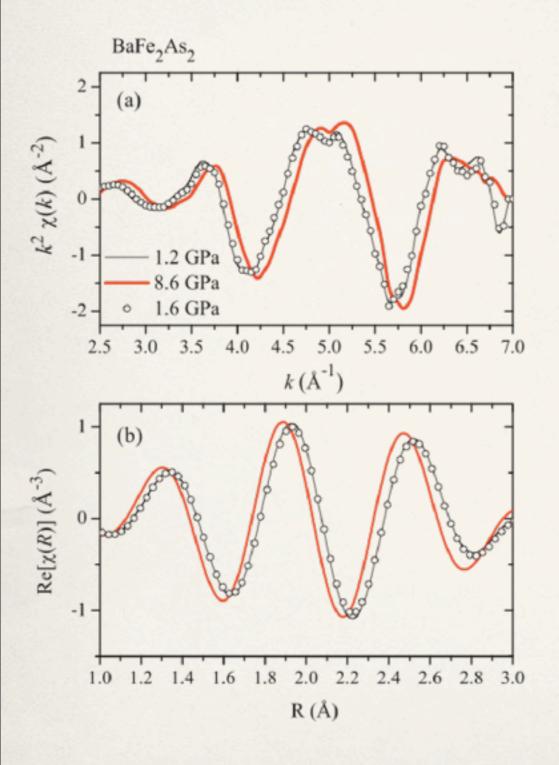
Decrease of FeAs distance with both K and

Co substitution

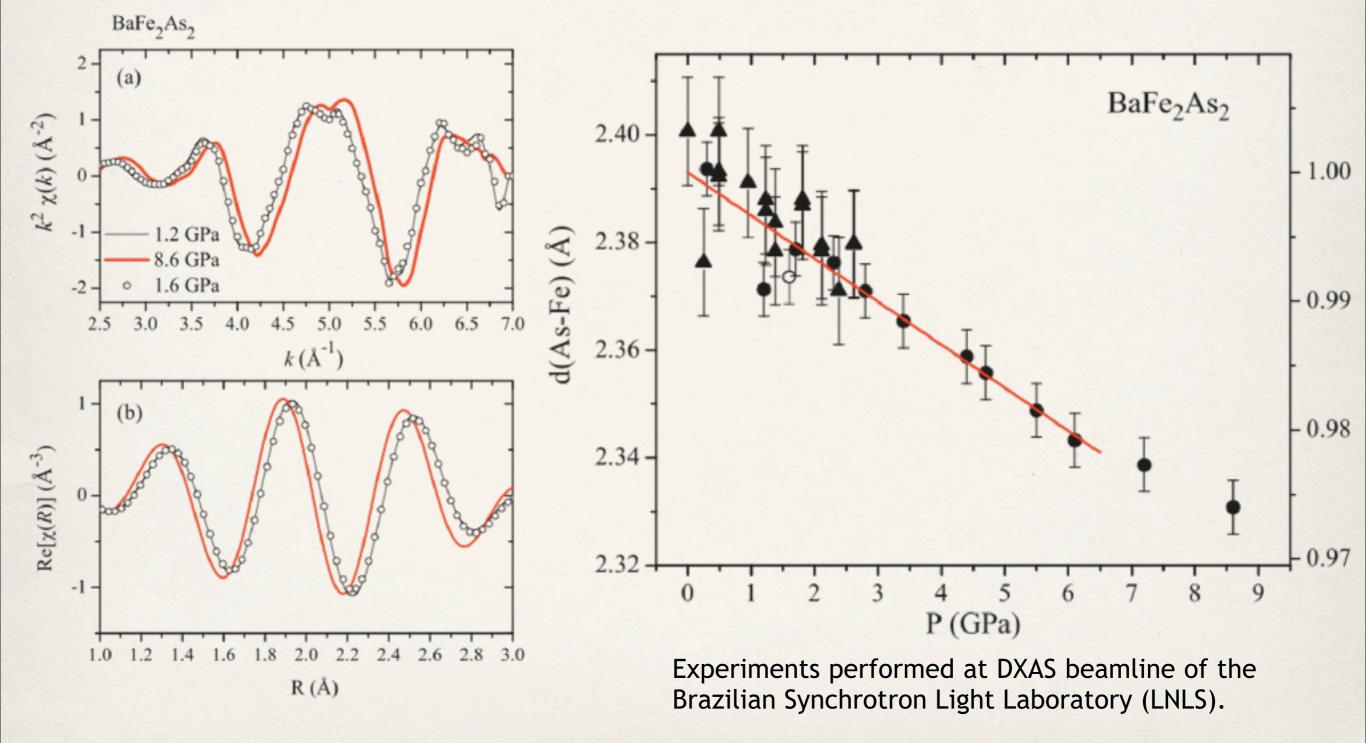


BaFe, As, (a) As-(Fe,Co) Ba(Fe0.937Co0.063)2As2 3 Ba0.85K0.15Fe2As2 2 As-(Ba,K) 298 K  $|\chi(R)| (\text{Å}^{-3})$ (b) 3 2 K 2 R (Å)

\* Decrease of d(Fe-As) also with applied pressure.



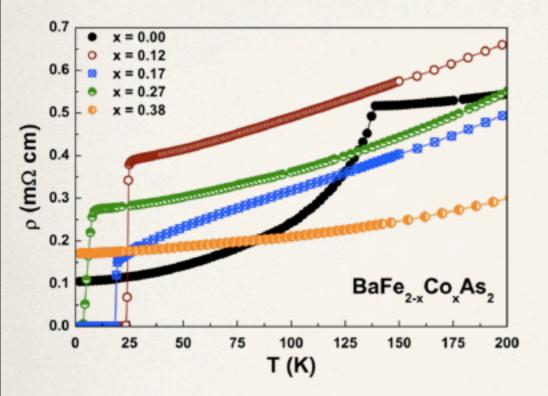
\* Decrease of d(Fe-As) also with applied pressure.



E. Granado, L. Mendonça-Ferreira, F. Garcia, G. de M. Azevedo, G. Fabbris, E. M. Bittar, C. Adriano, T. M. Garitezi, P. F. S. Rosa *et al.* Phys. Rev. B **83**, 184508 (2011).

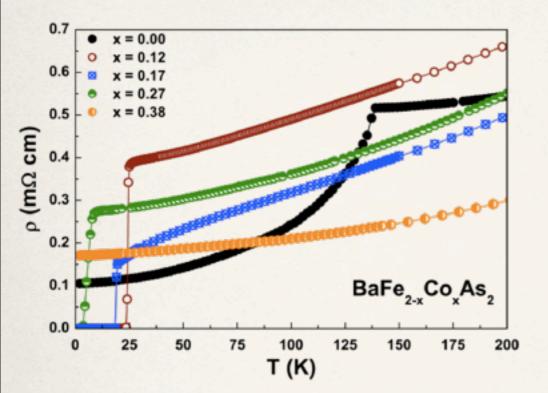
#### Microscopic Properties: Fe Valence (XANES)

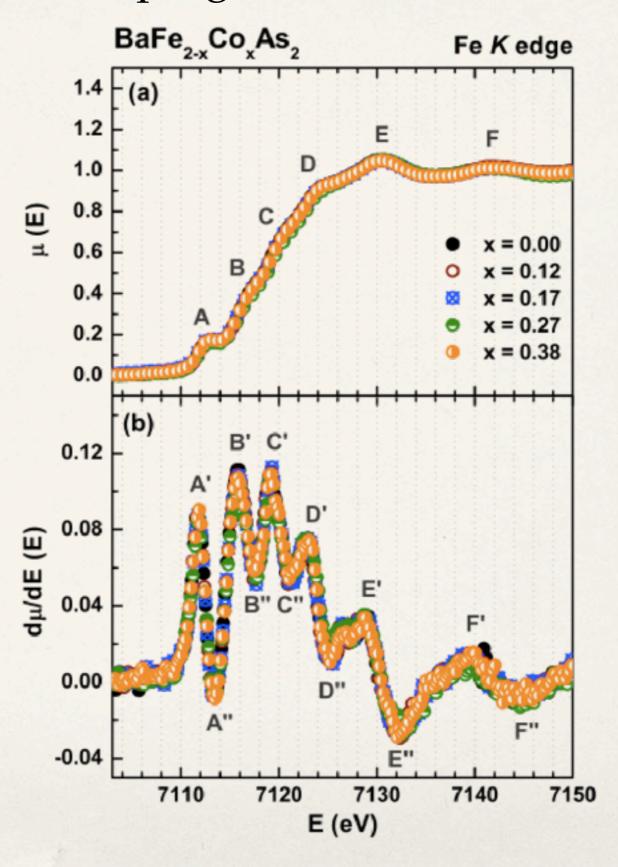
\* Is Cobalt an effective carrier doping?



#### Microscopic Properties: Fe Valence (XANES)

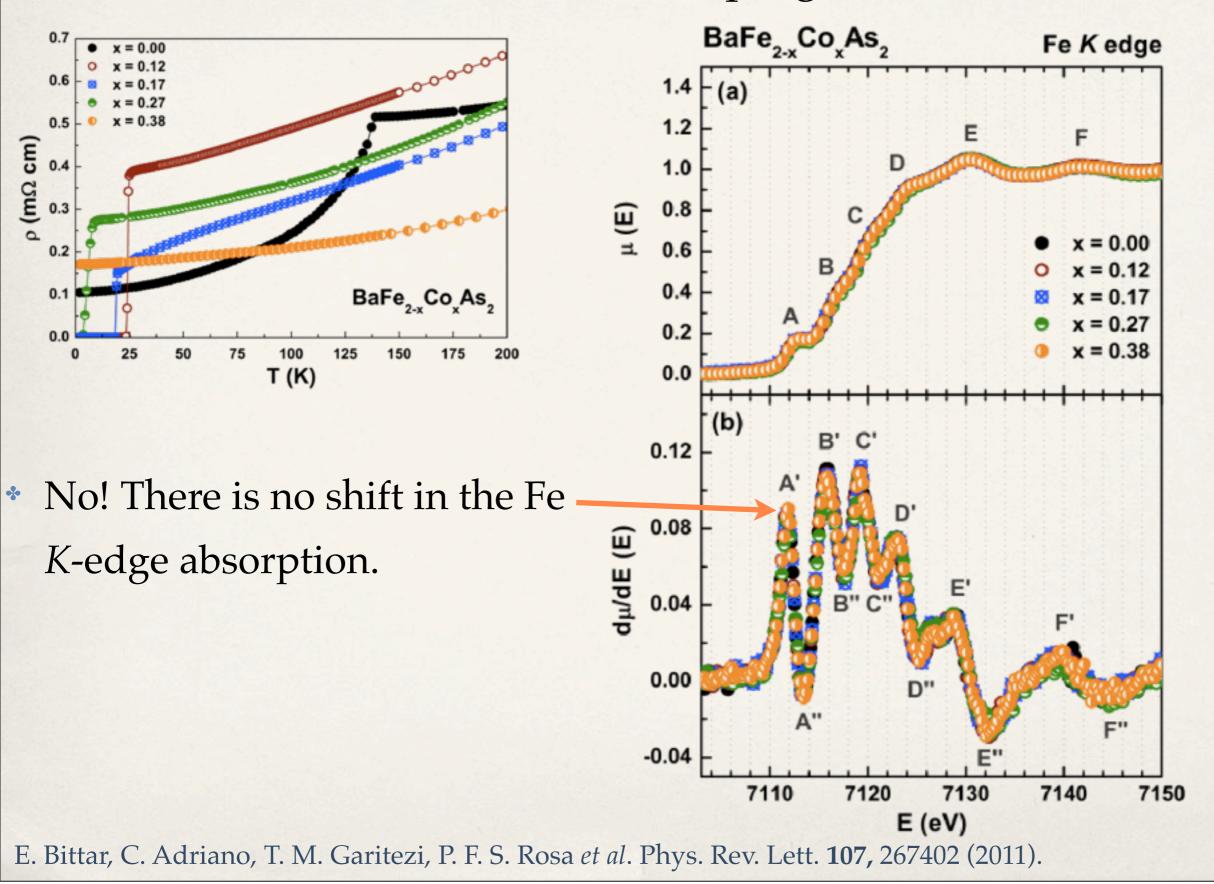
\* Is Cobalt an effective carrier doping?

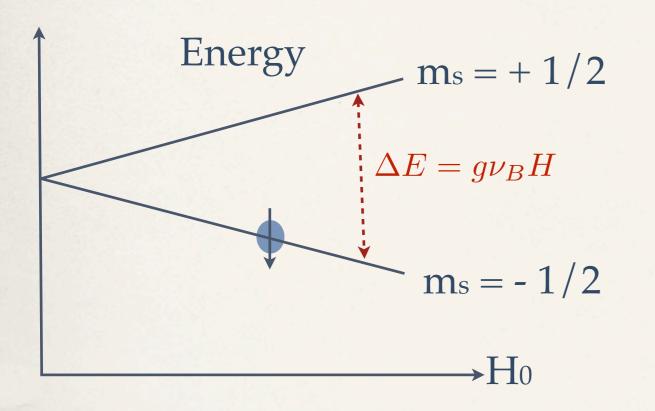


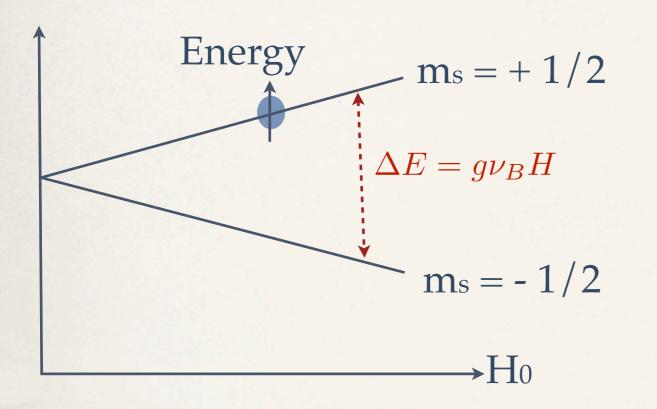


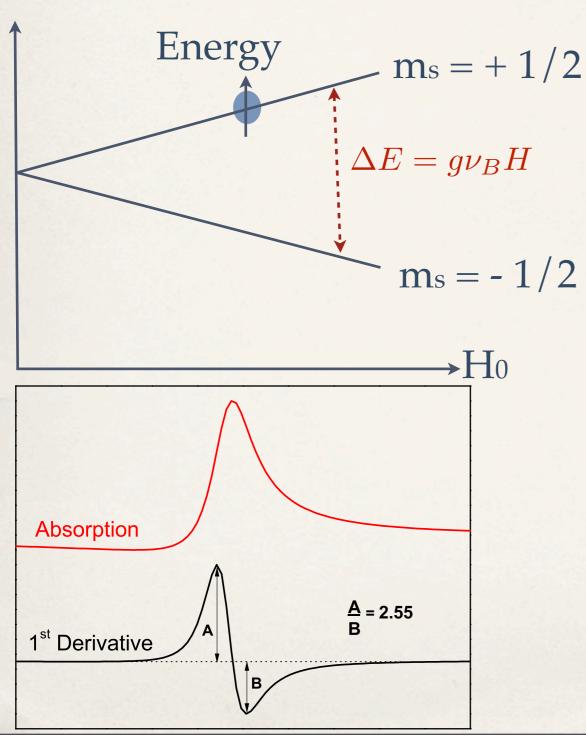
#### Microscopic Properties: Fe Valence (XANES)

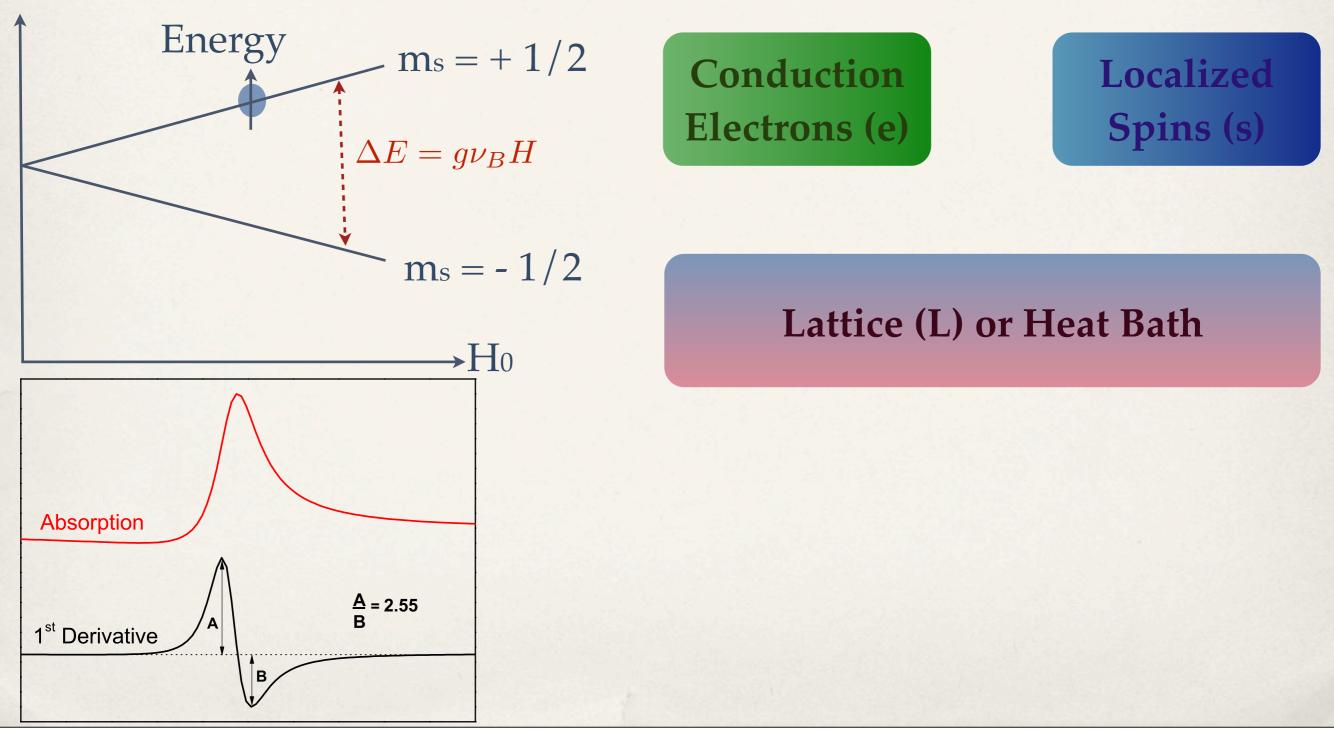
\* Is Cobalt an effective carrier doping?

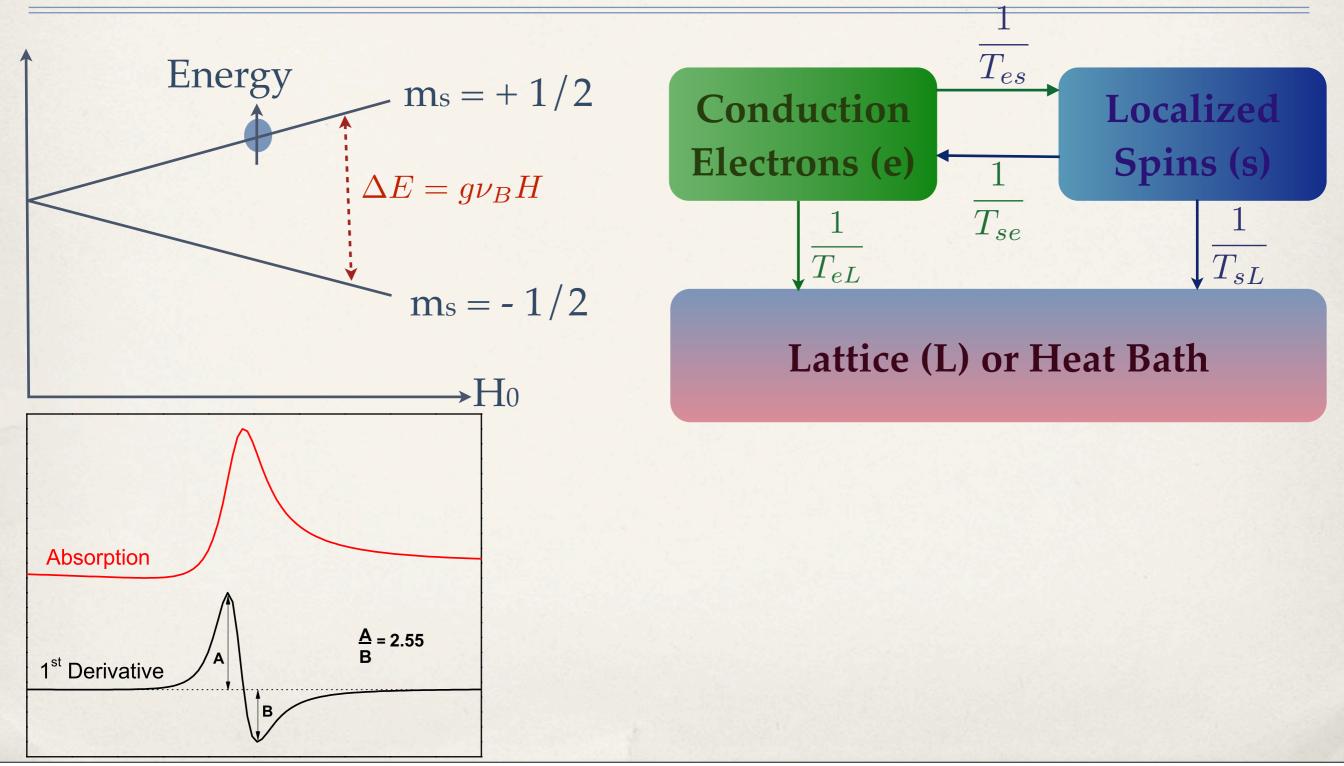


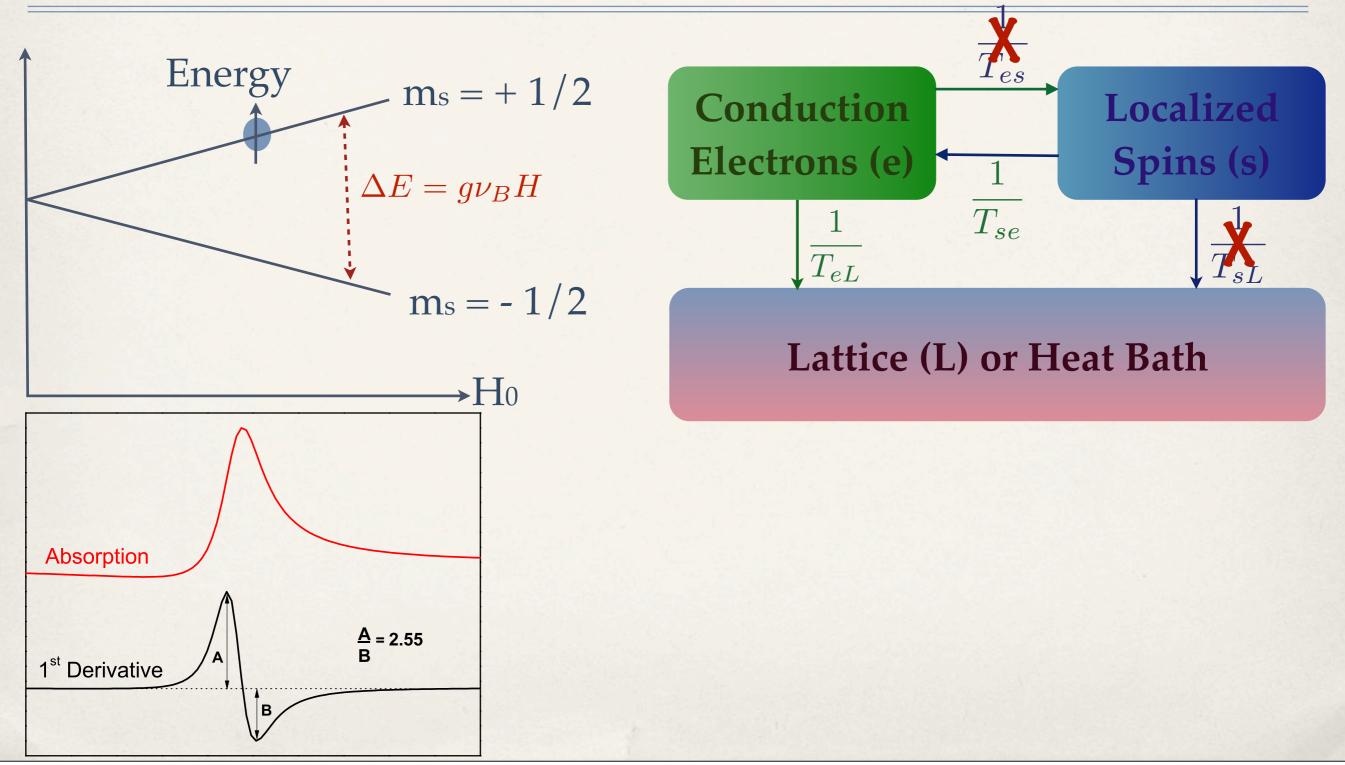


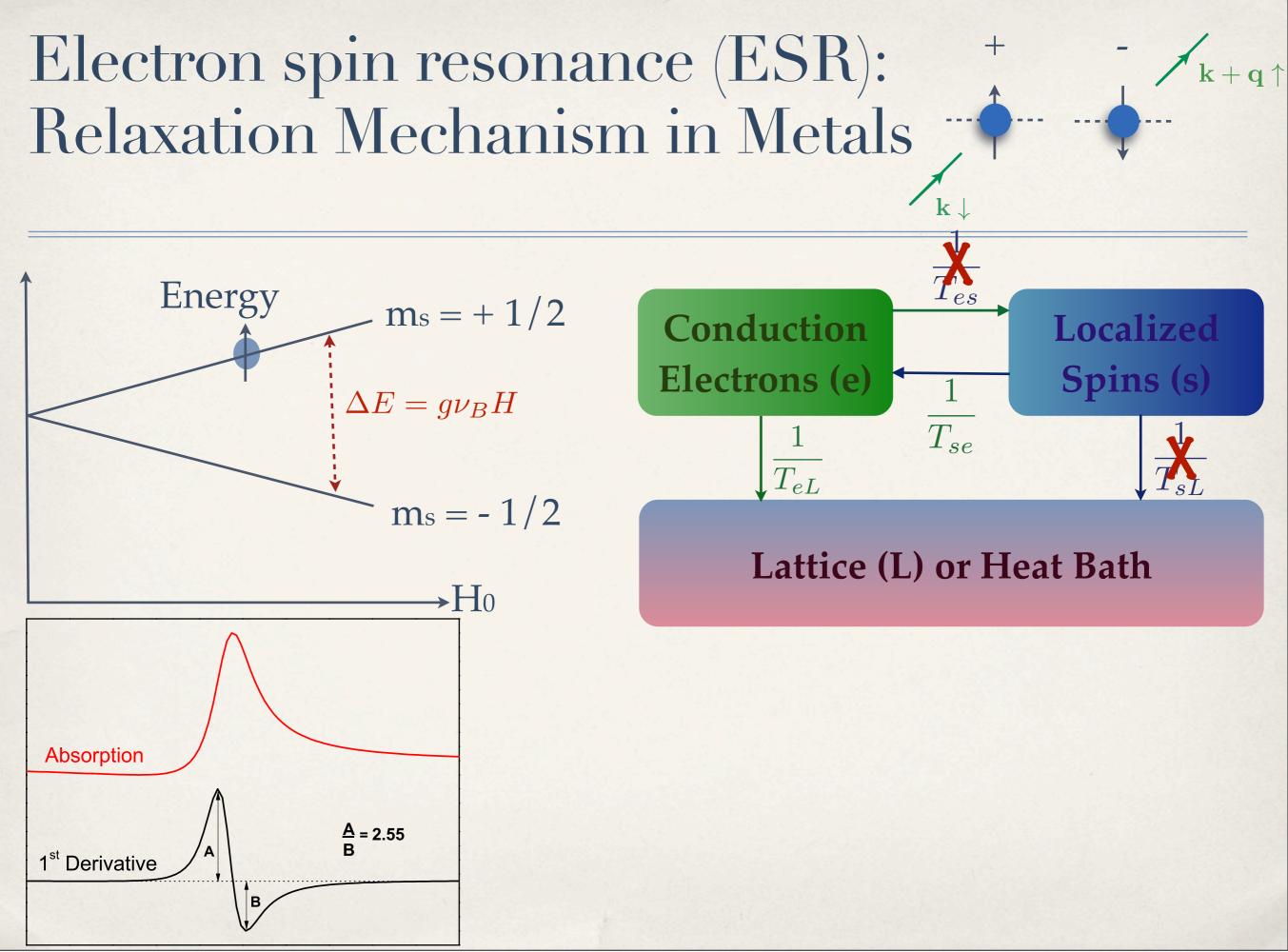


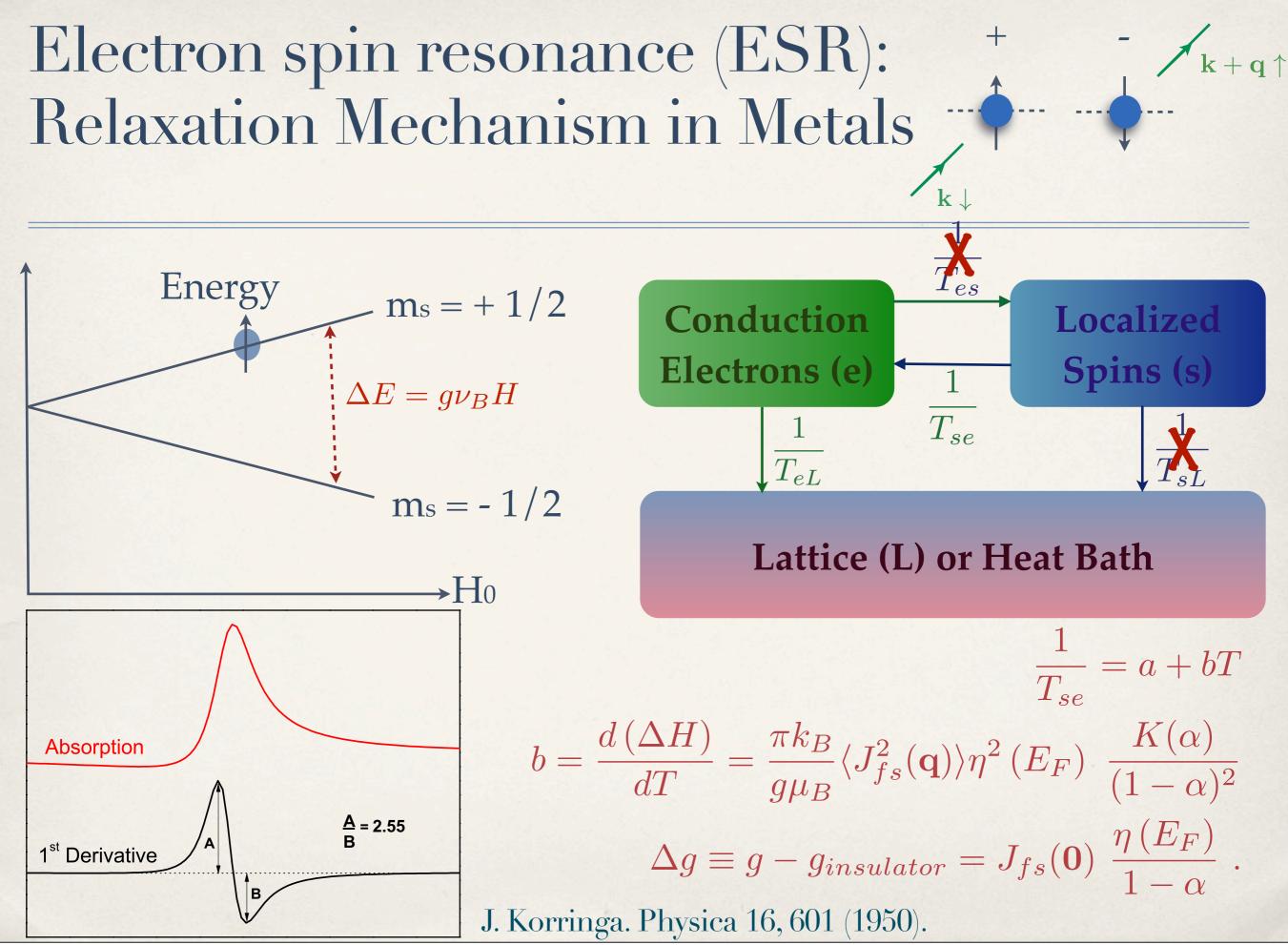




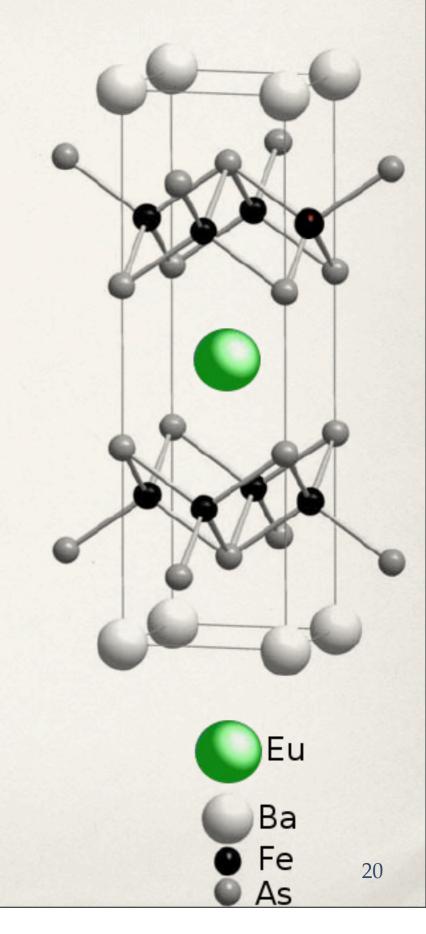


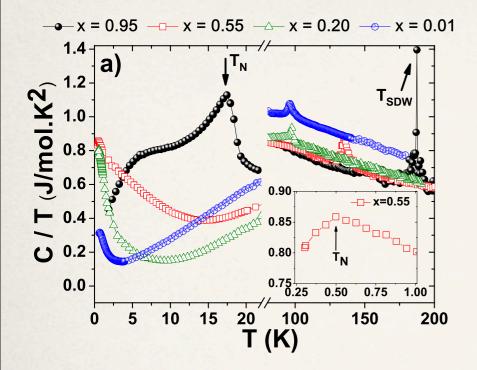


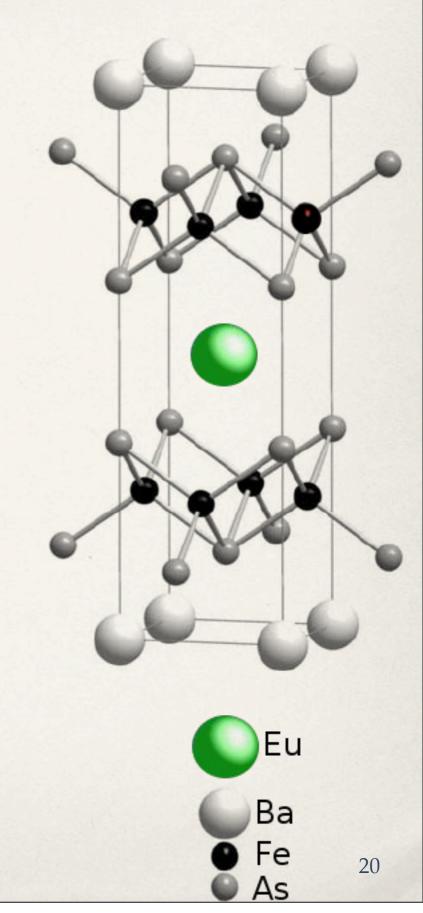


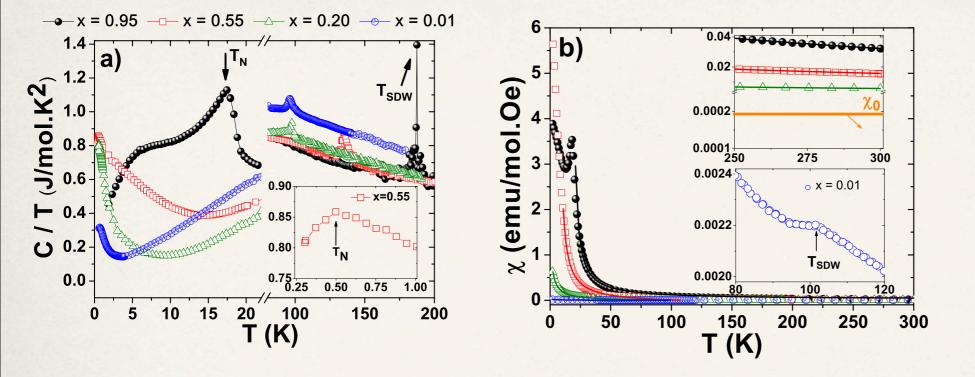


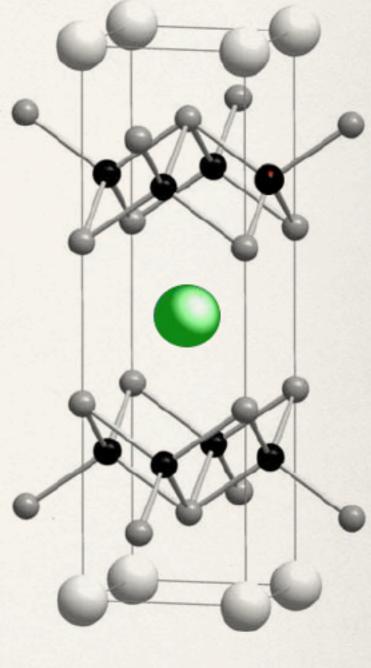
Thursday, October 9, 2014

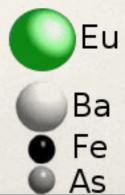


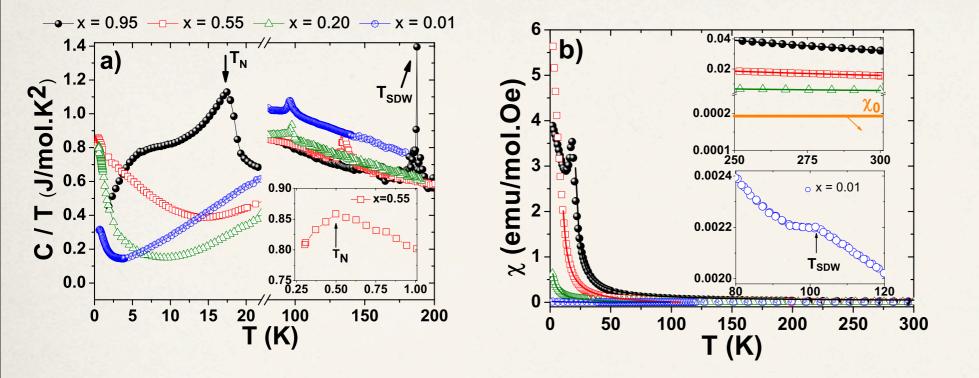


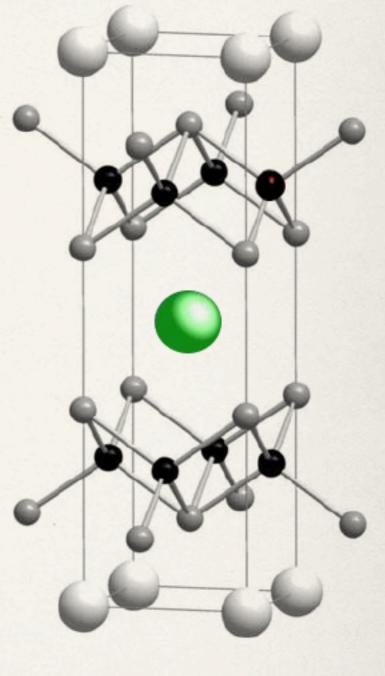












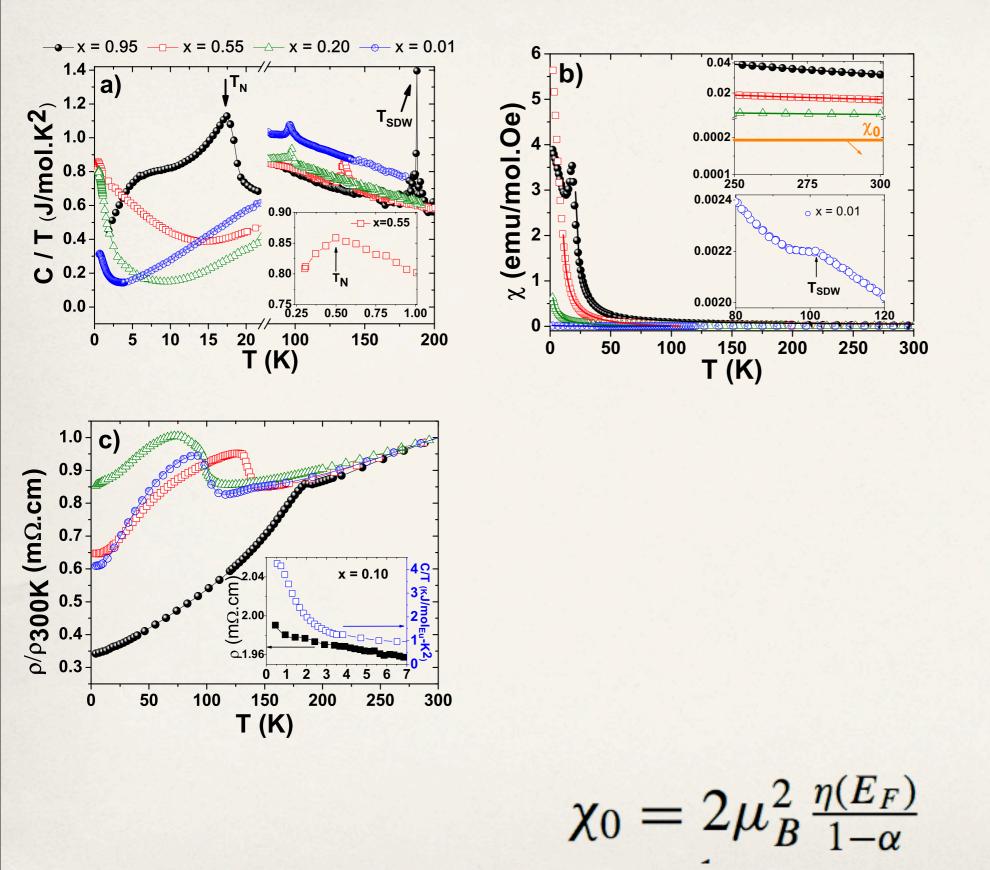
 $\chi_0 = 2\mu_B^2 \frac{\eta(E_F)}{1-\alpha}$ 

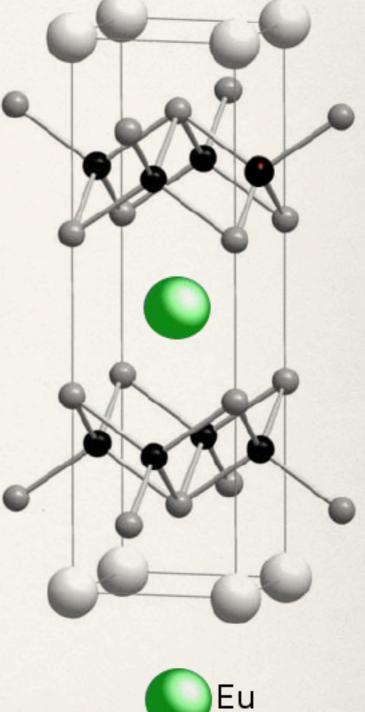
Eu

Ba

Fe

As

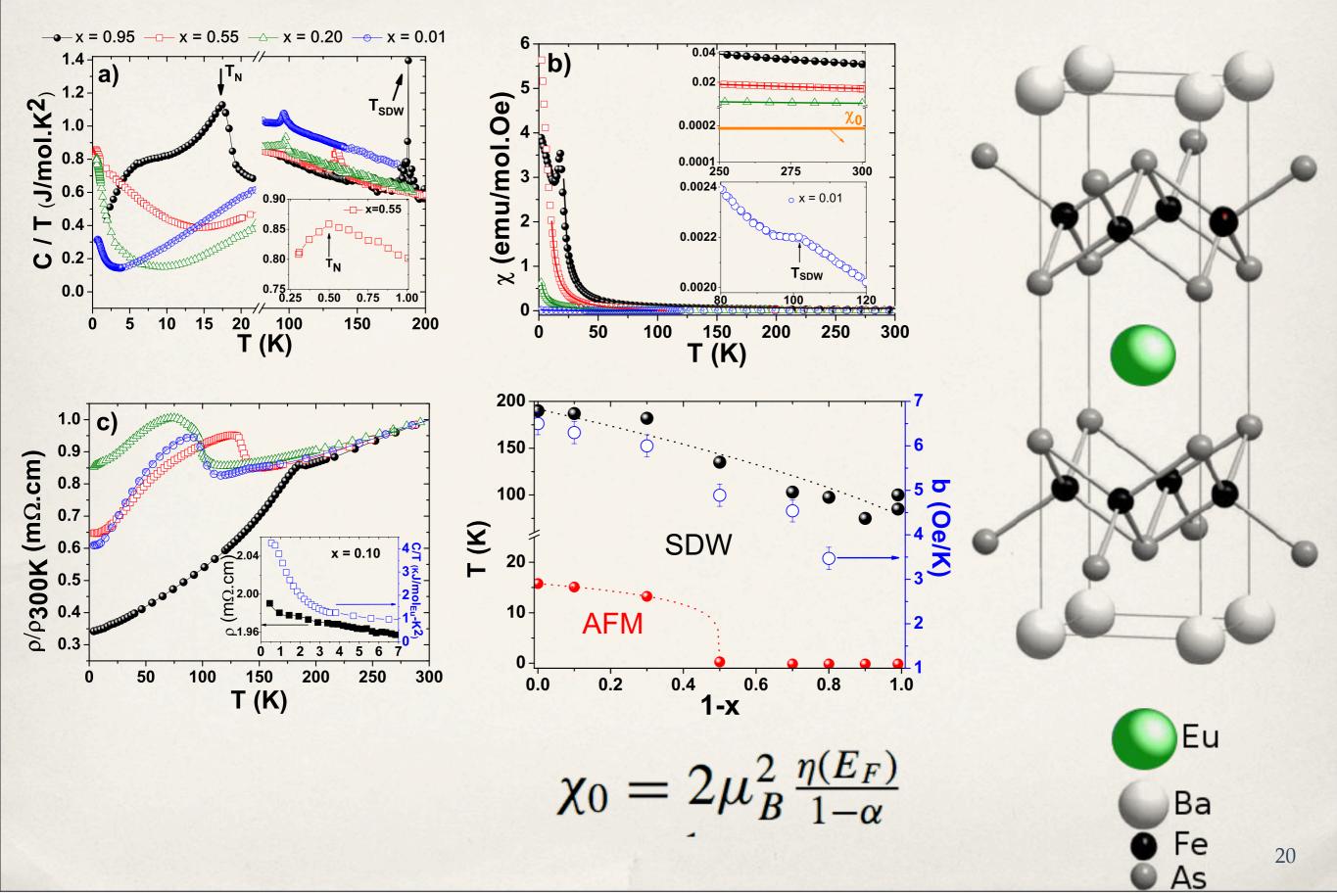




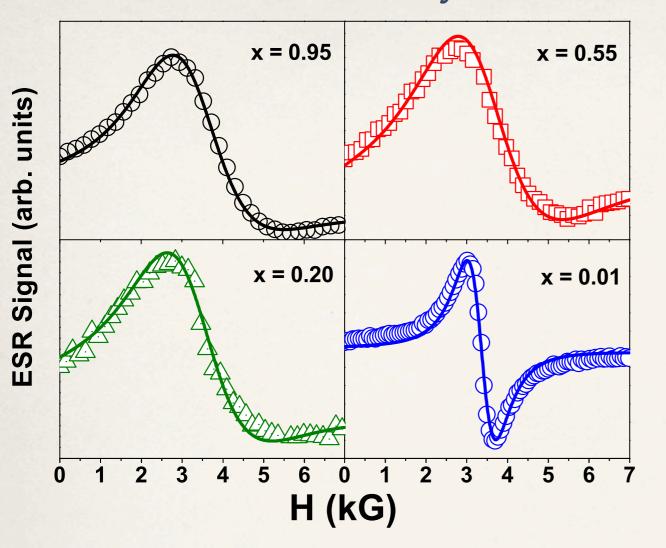
Ba

Fe

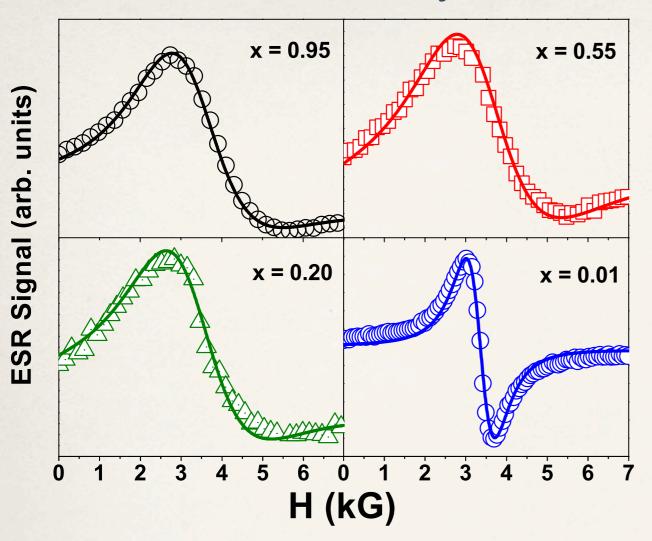
As

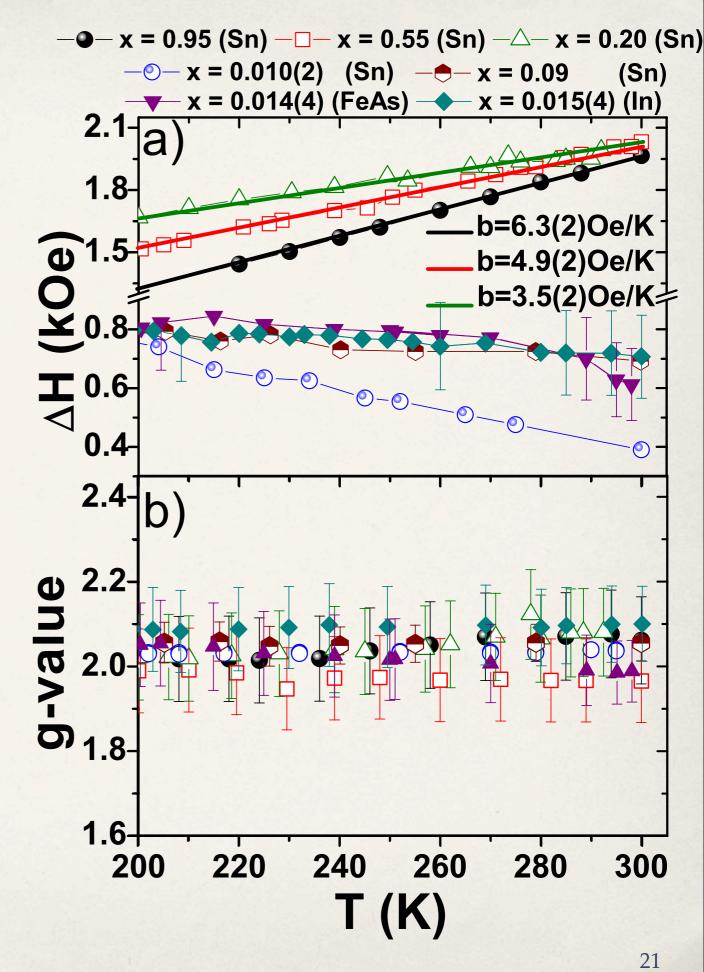


#### Eu<sup>2+</sup> ESR Analysis

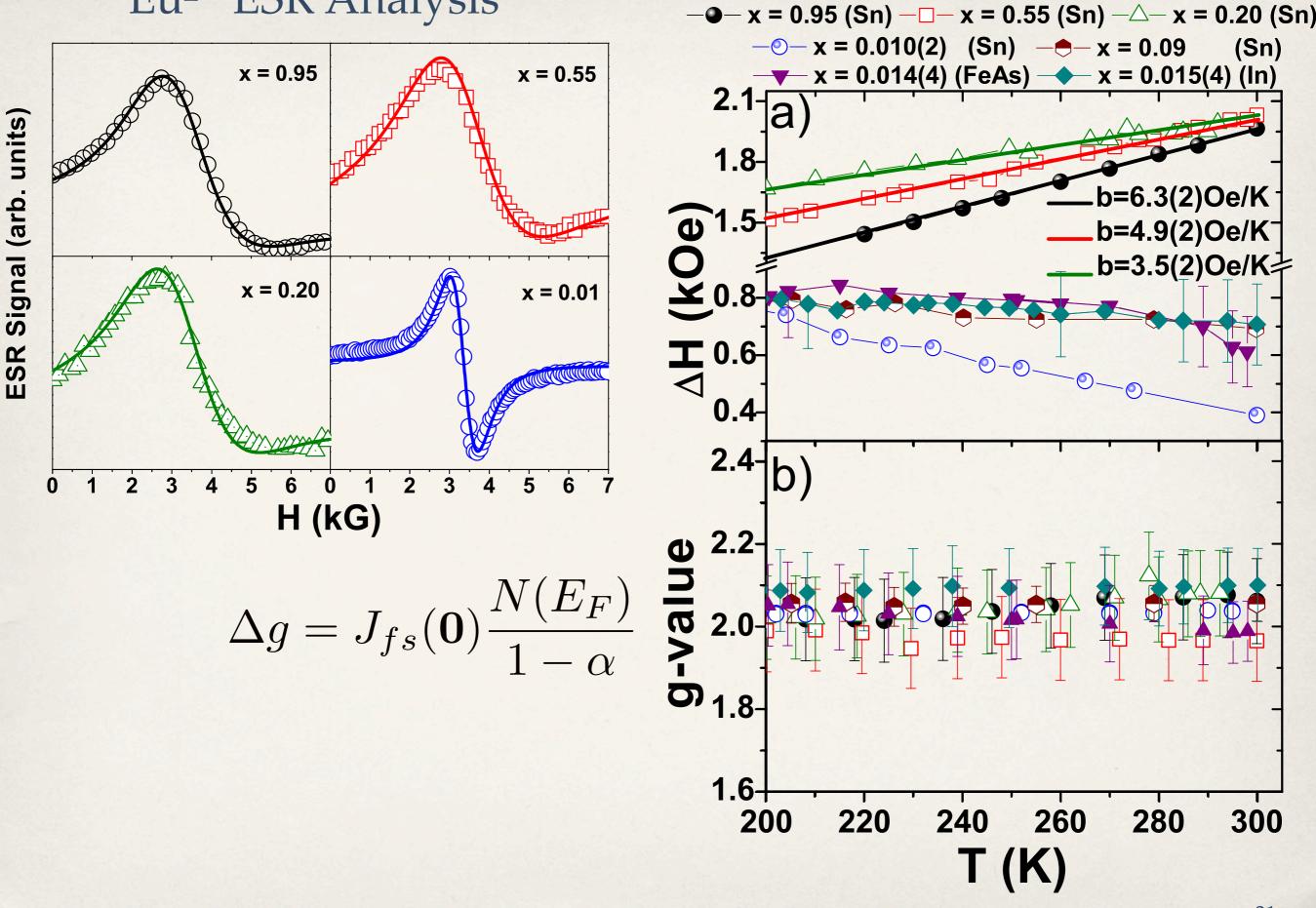




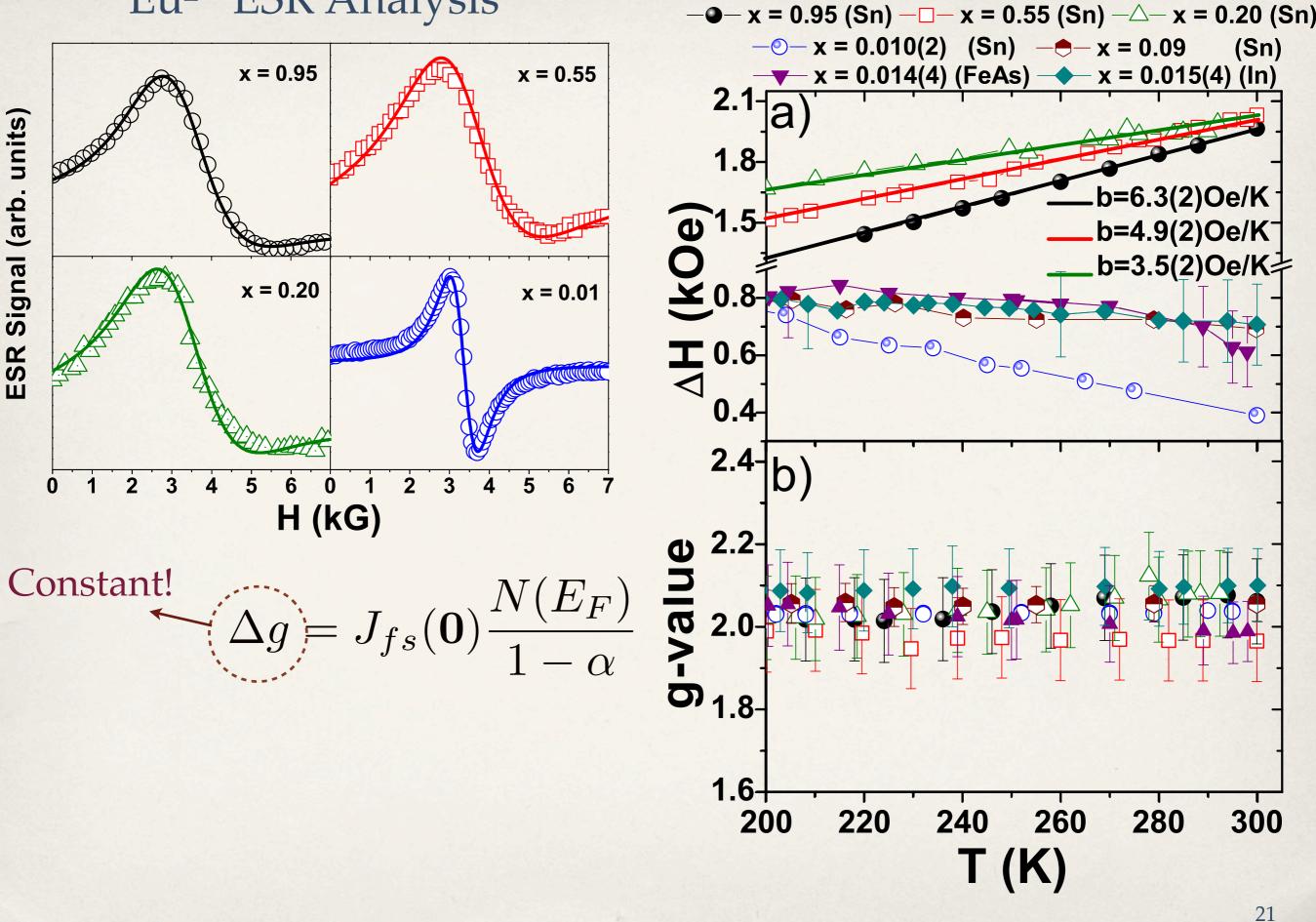


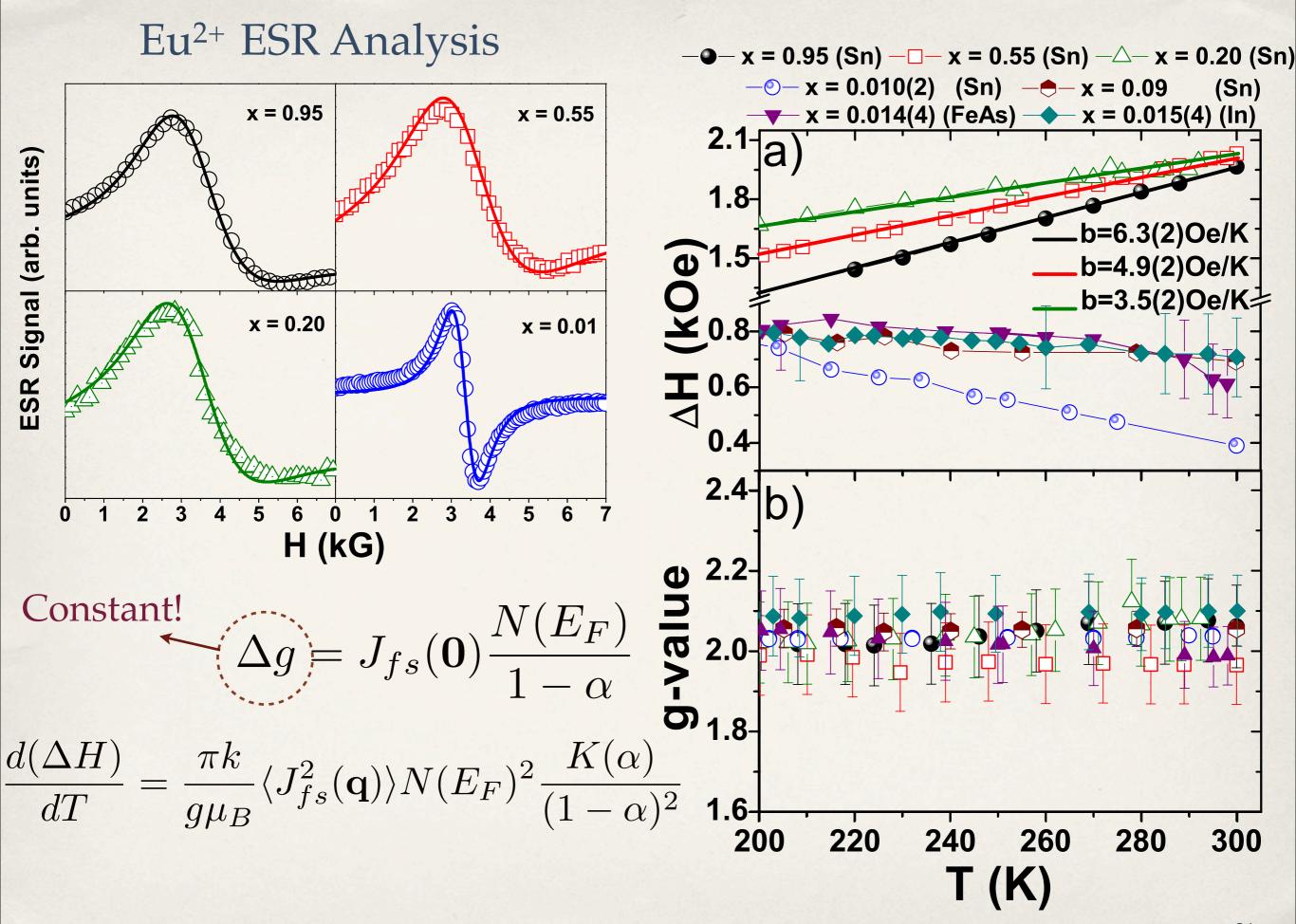


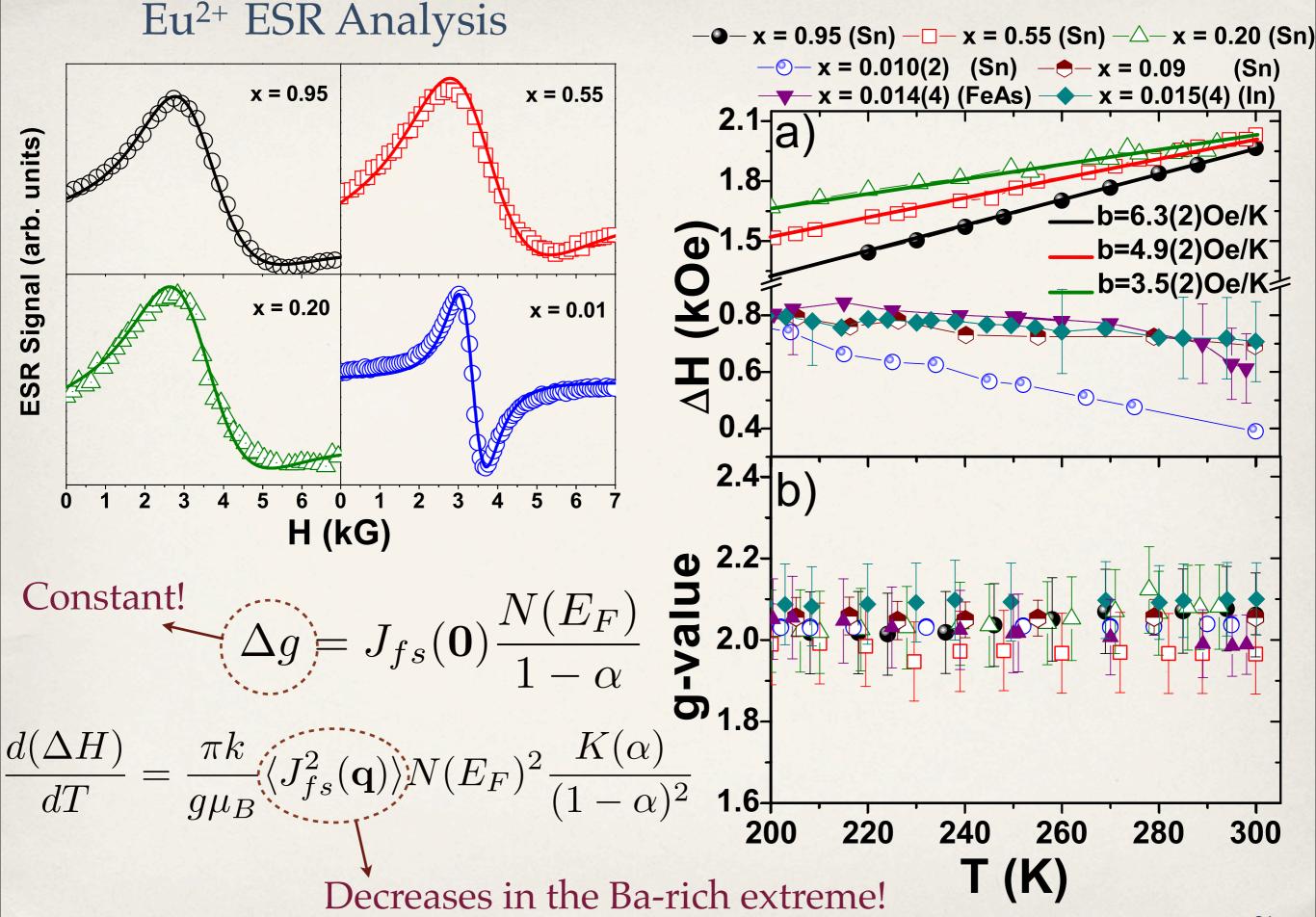


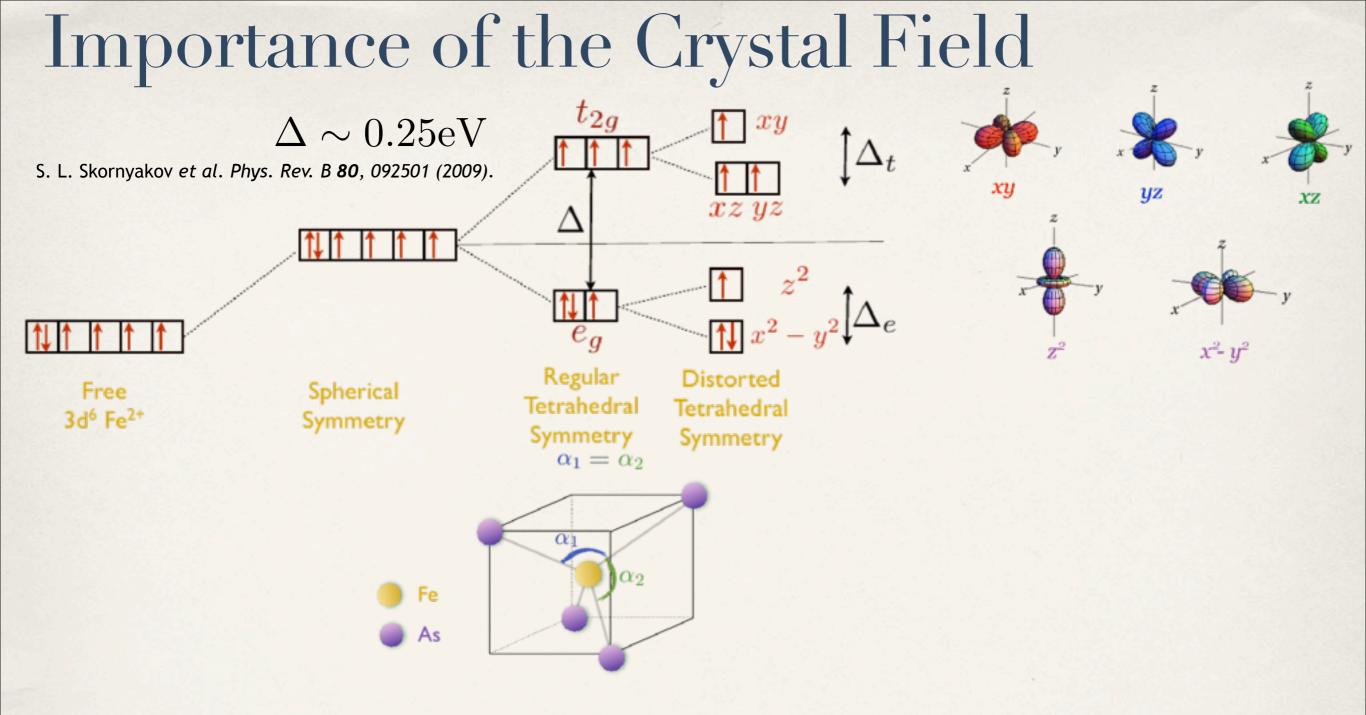


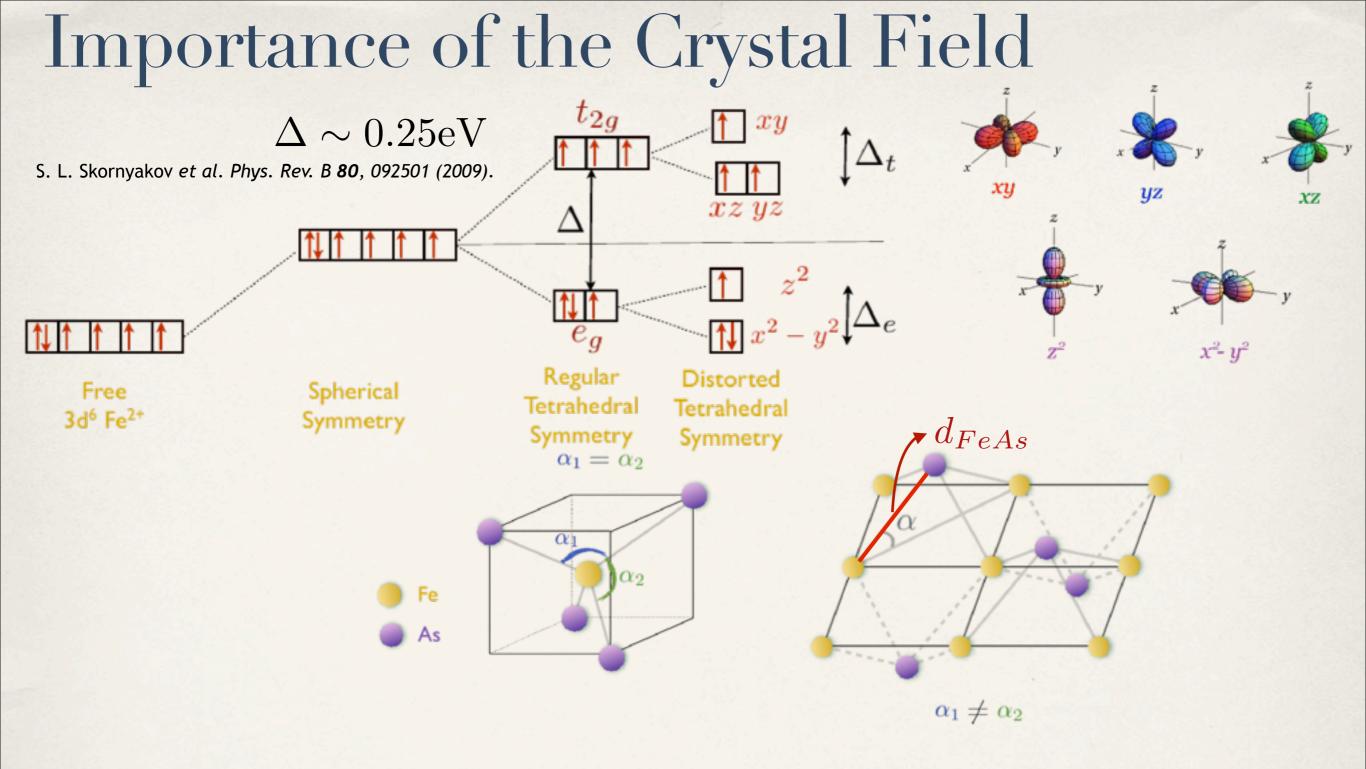


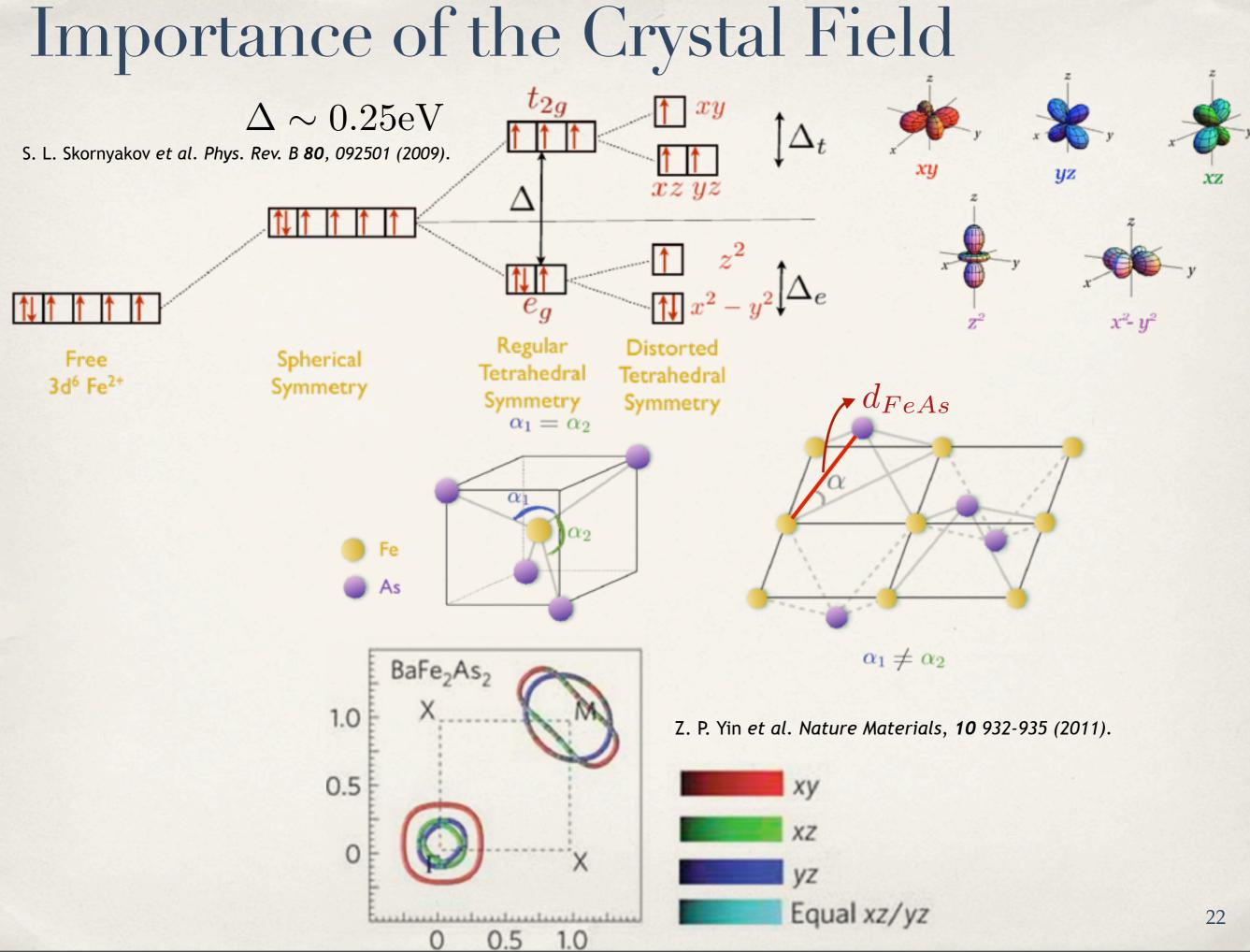


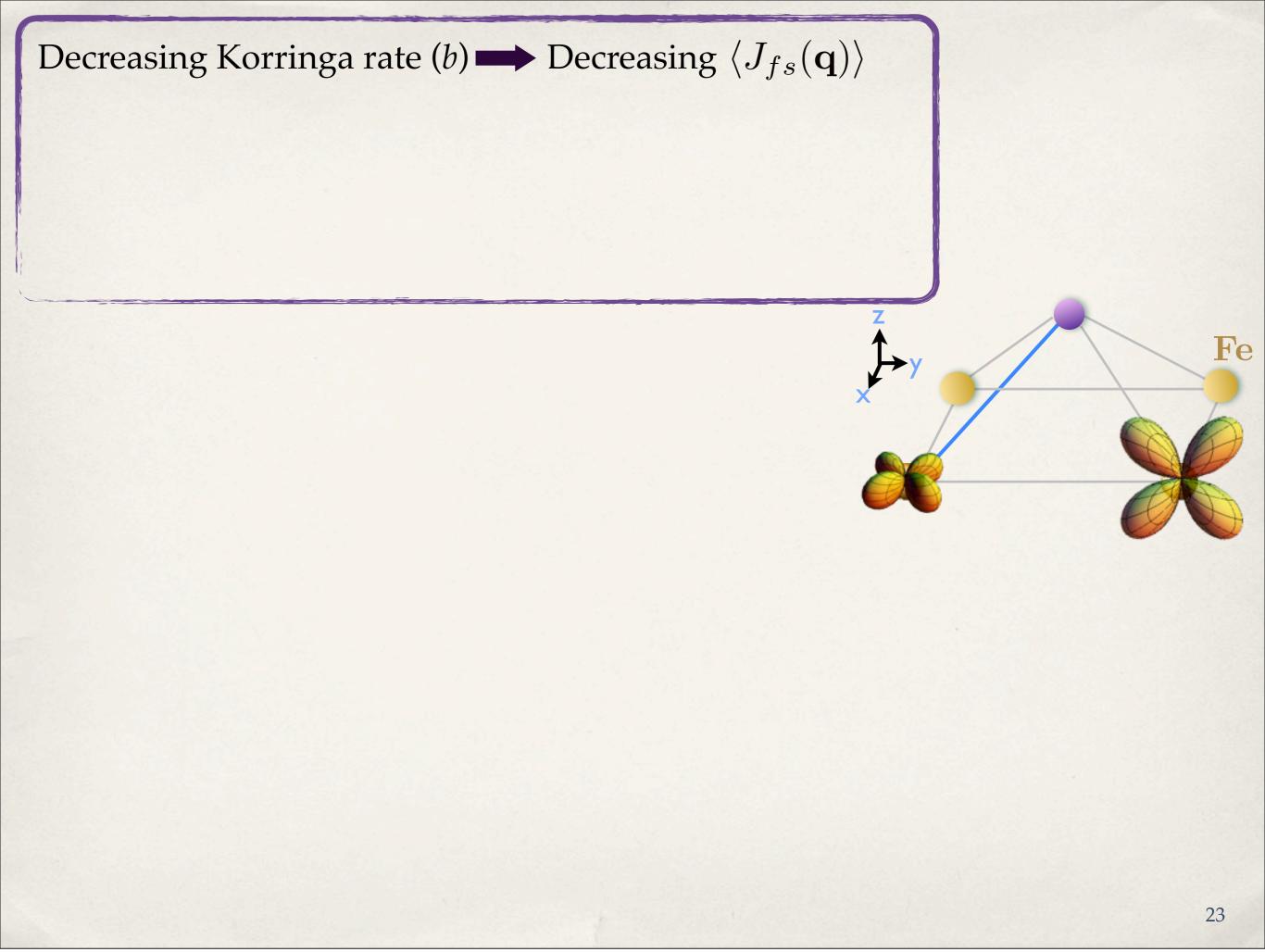


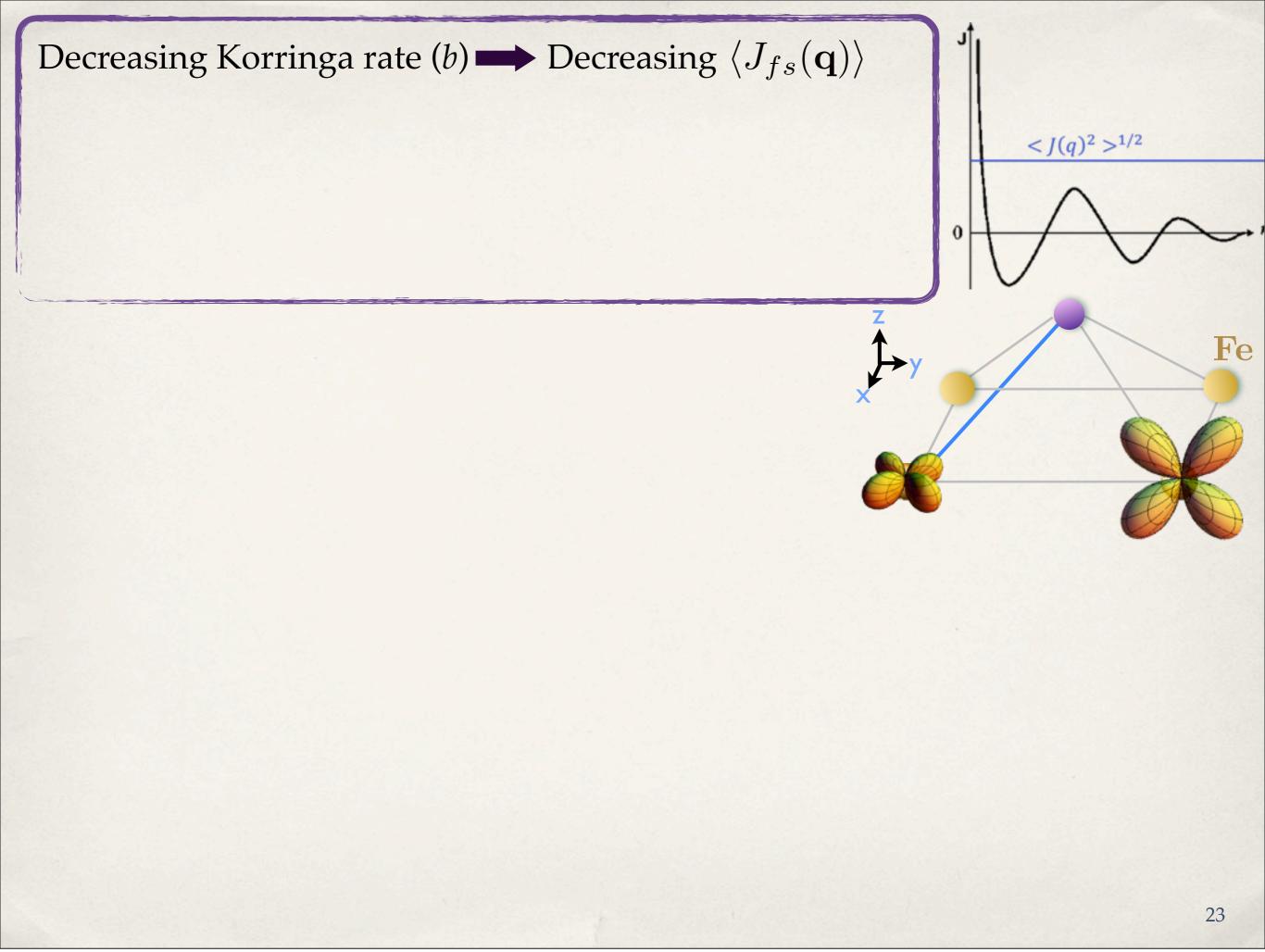








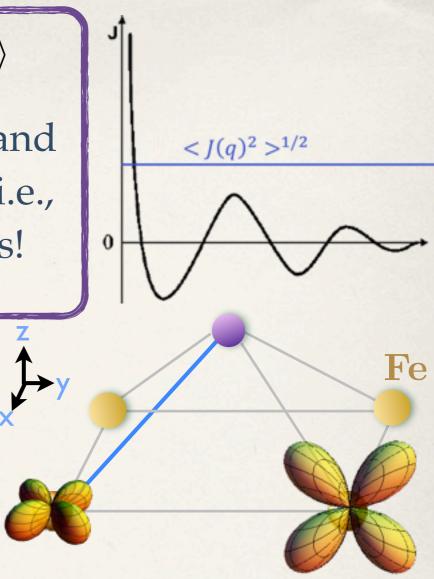




#### Decreasing Korringa rate (b) $\blacksquare$ Decreasing $\langle J_{fs}(\mathbf{q}) \rangle$

3d Fe bands are more anisotropic (less s-like) and
are, in average, further away from the Eu site, i.e., the planar orbital character xy/x<sup>2</sup>-y<sup>2</sup> increases!

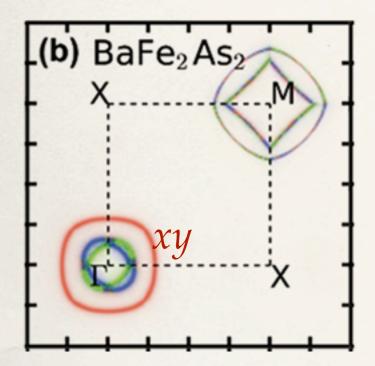
P. F. S. Rosa et al. Phys. Rev. B 86 165131 (2012).

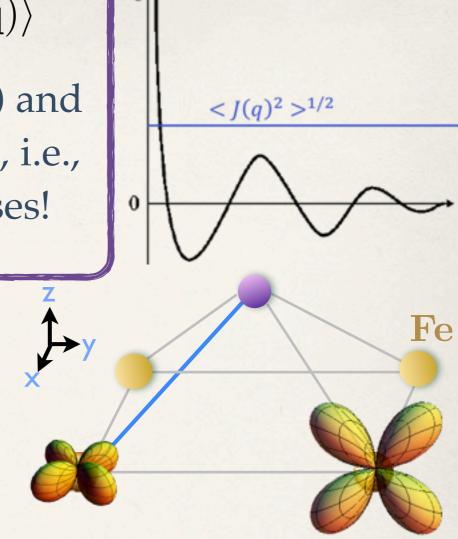


Decreasing Korringa rate (b)  $\longrightarrow$  Decreasing  $\langle J_{fs}(\mathbf{q}) \rangle$ 

*3d* Fe bands are more anisotropic (less *s*-like) and
are, in average, further away from the Eu site, i.e., the planar orbital character xy/x<sup>2</sup>-y<sup>2</sup> increases!

P. F. S. Rosa et al. Phys. Rev. B 86 165131 (2012).

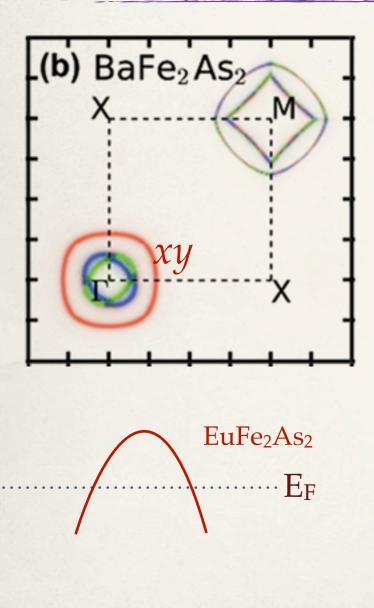


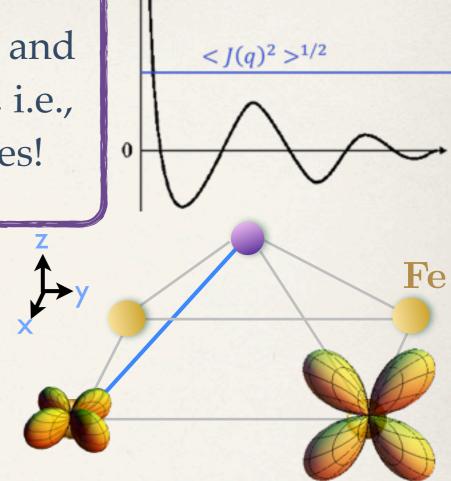


Decreasing Korringa rate (b)  $\longrightarrow$  Decreasing  $\langle J_{fs}(\mathbf{q}) \rangle$ 

*3d* Fe bands are more anisotropic (less *s*-like) and
are, in average, further away from the Eu site, i.e., the planar orbital character xy/x<sup>2</sup>-y<sup>2</sup> increases!

P. F. S. Rosa et al. Phys. Rev. B 86 165131 (2012).

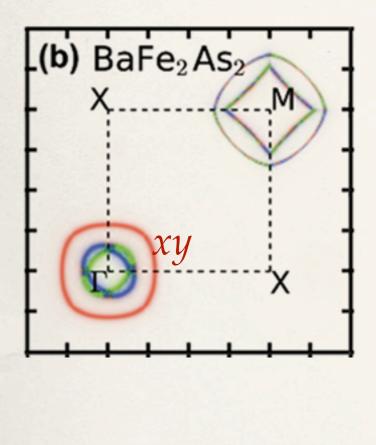


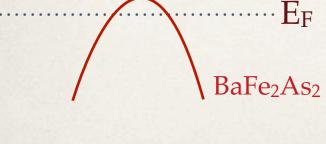


Decreasing Korringa rate (b)  $\blacksquare$  Decreasing  $\langle J_{fs}(\mathbf{q}) \rangle$ 

*3d* Fe bands are more anisotropic (less *s*-like) and
are, in average, further away from the Eu site, i.e., the planar orbital character xy/x<sup>2</sup>-y<sup>2</sup> increases!

P. F. S. Rosa et al. Phys. Rev. B 86 165131 (2012).

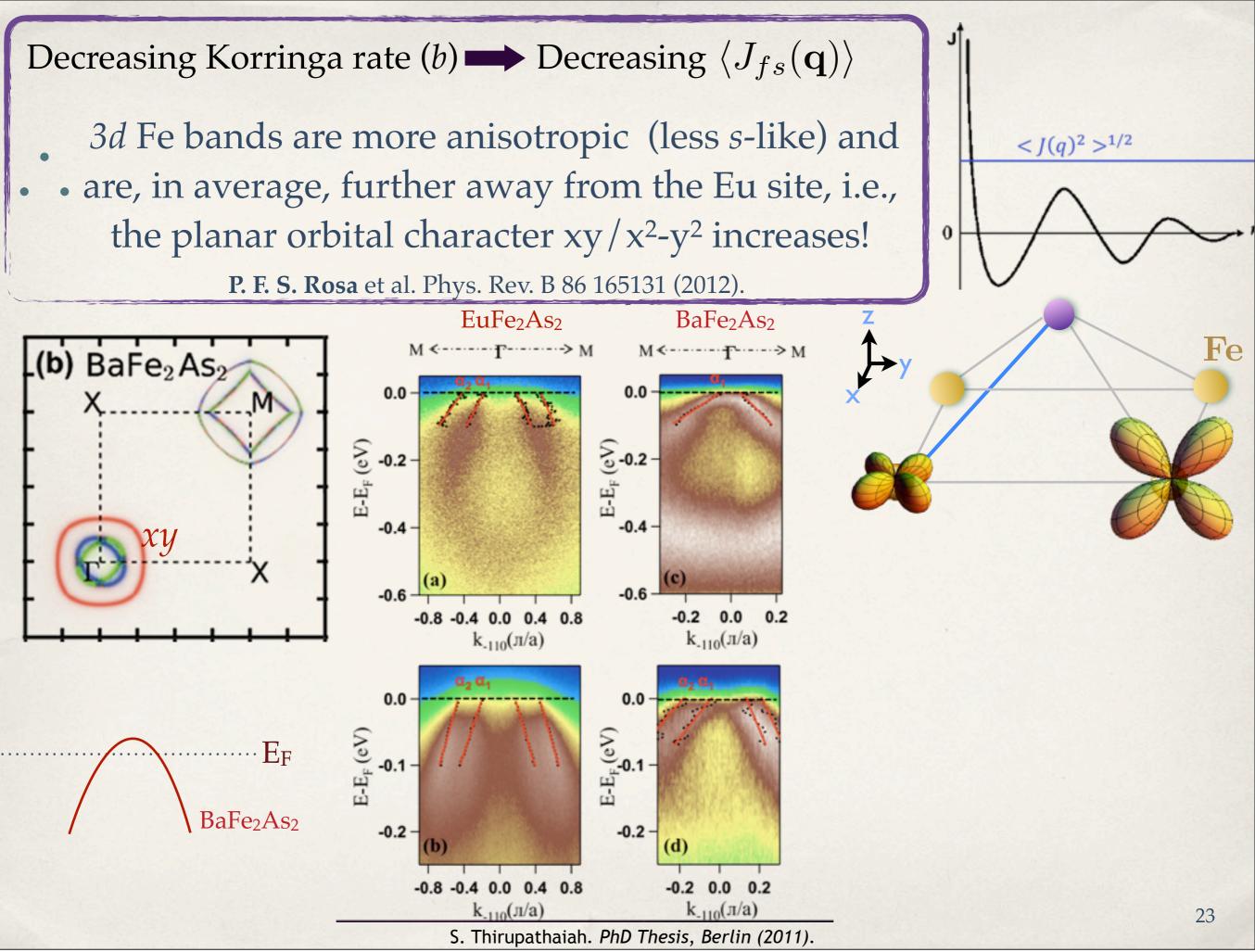


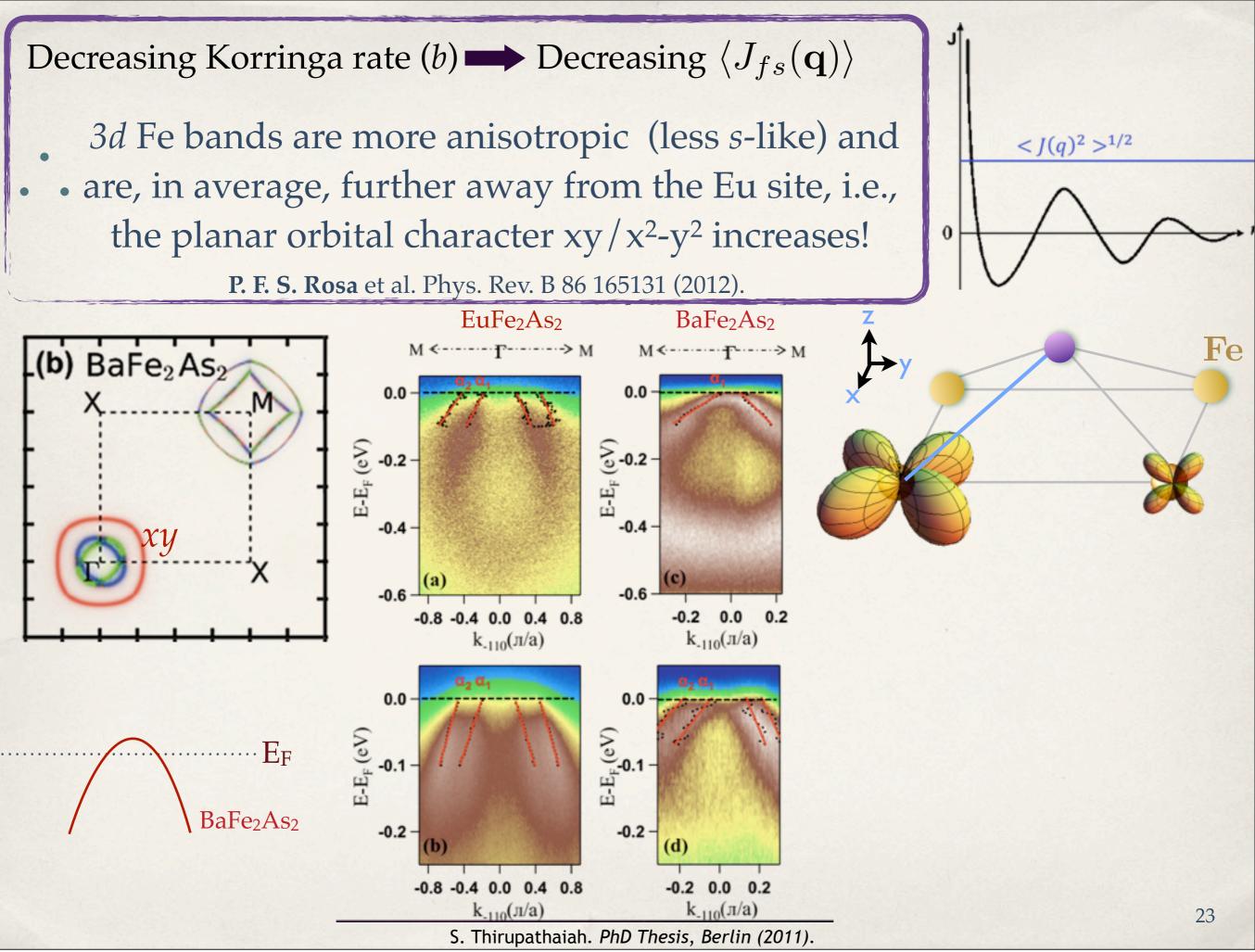


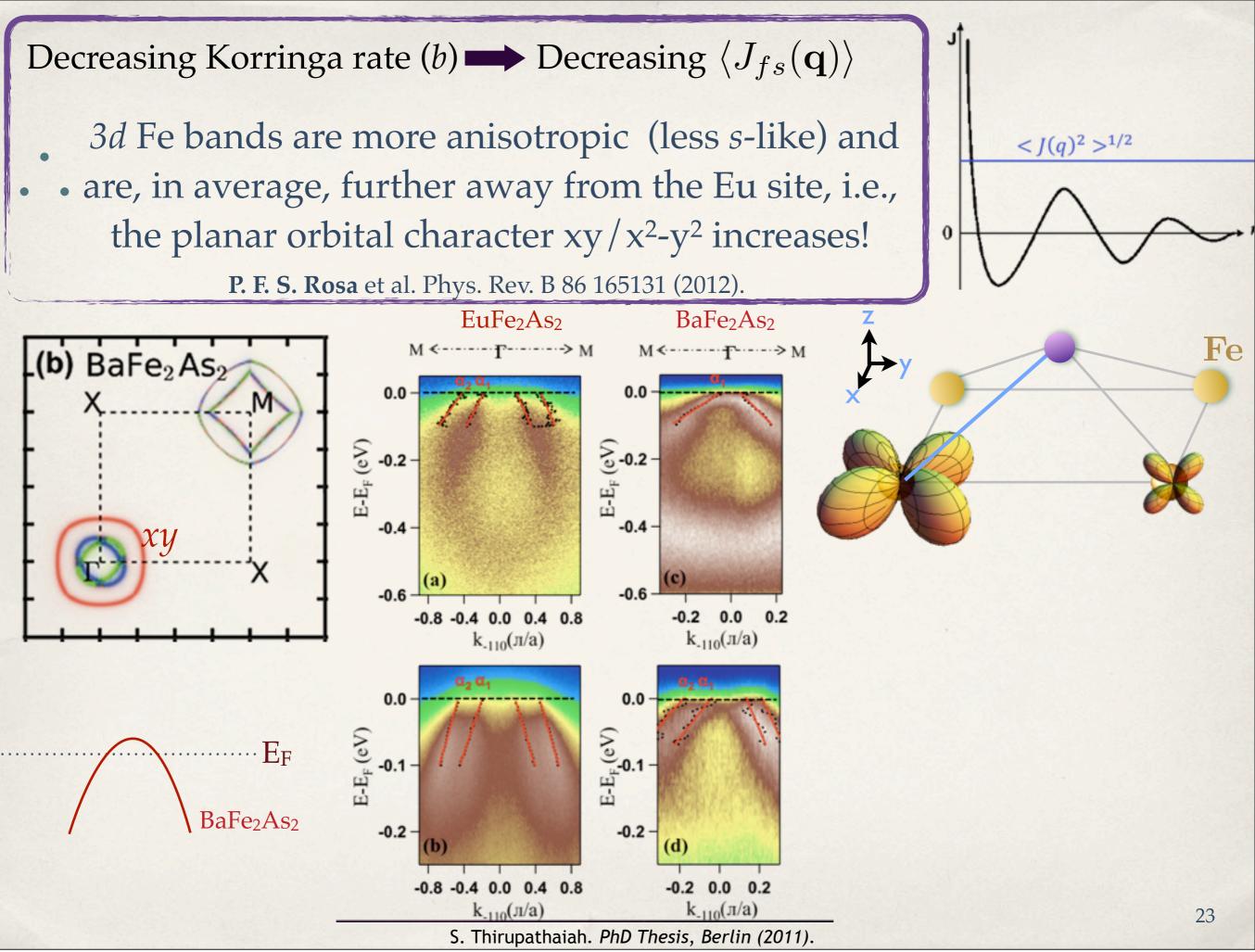
Fe

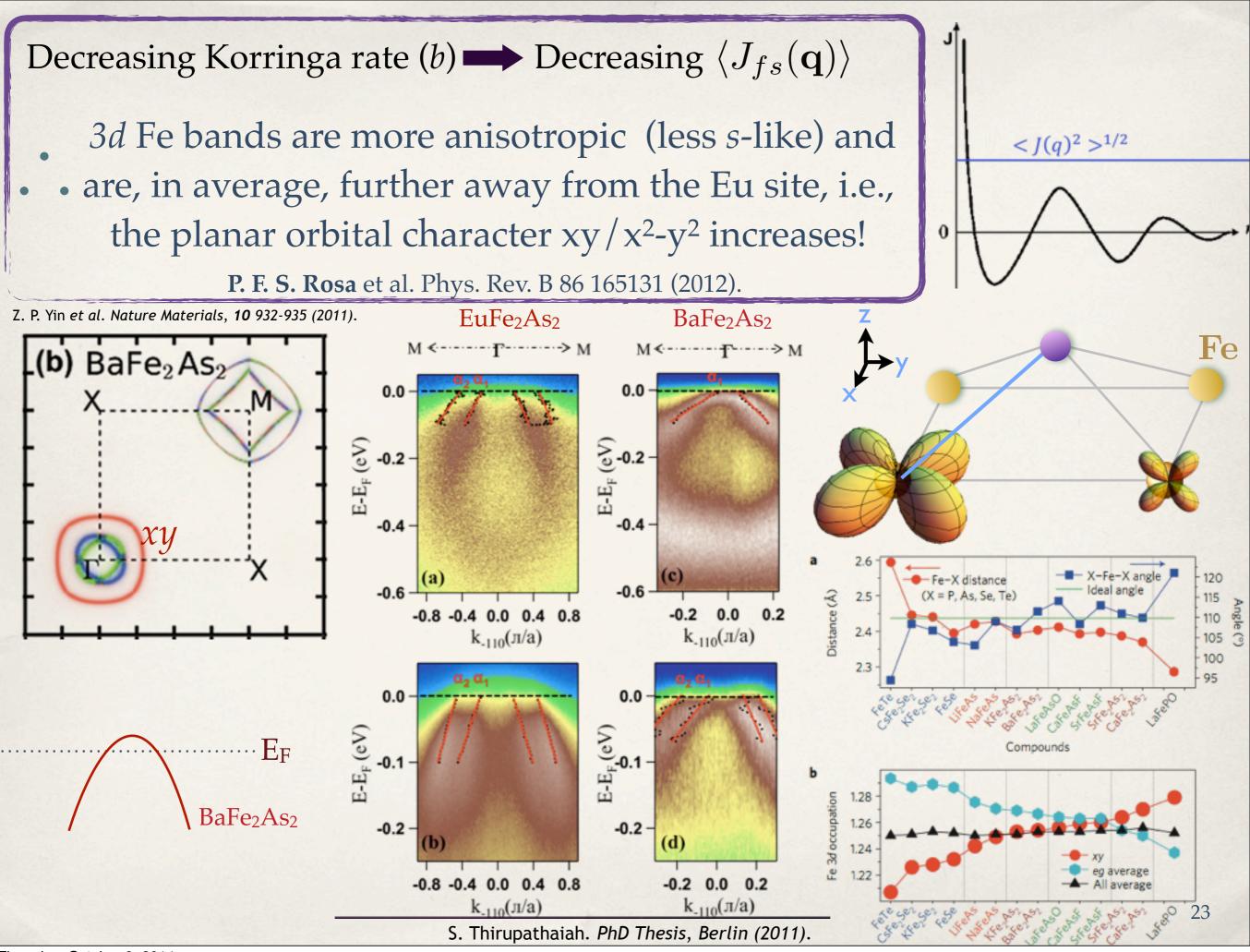
 $< J(q)^2 >^{1/2}$ 

0

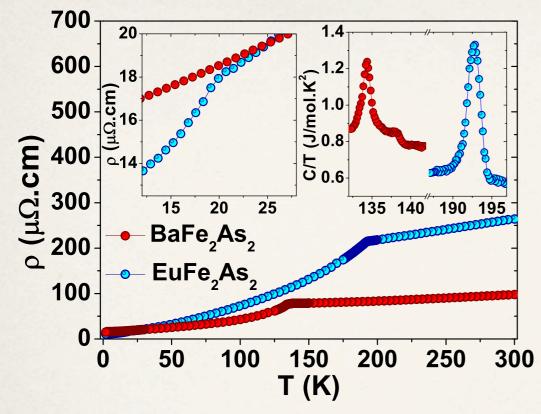




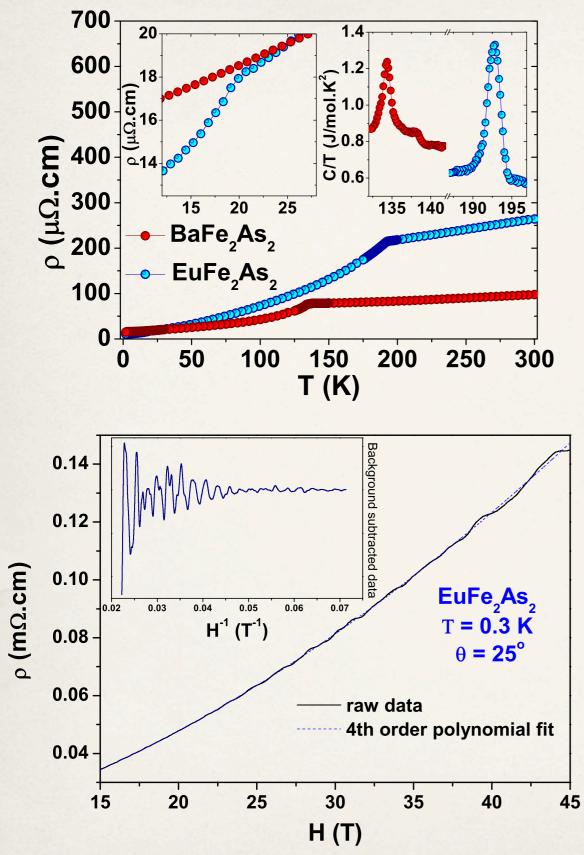




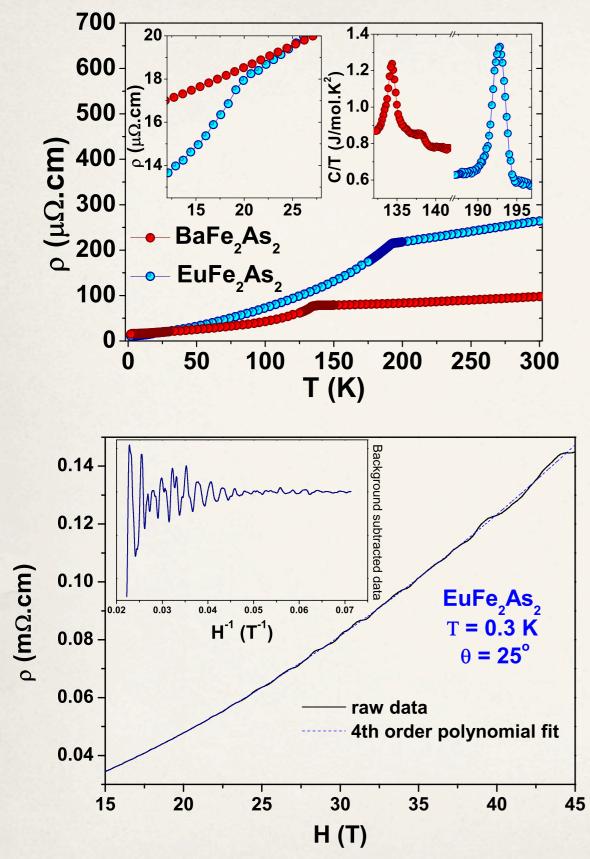
### Quantum Oscillations in EuFe<sub>2</sub>As<sub>2</sub>

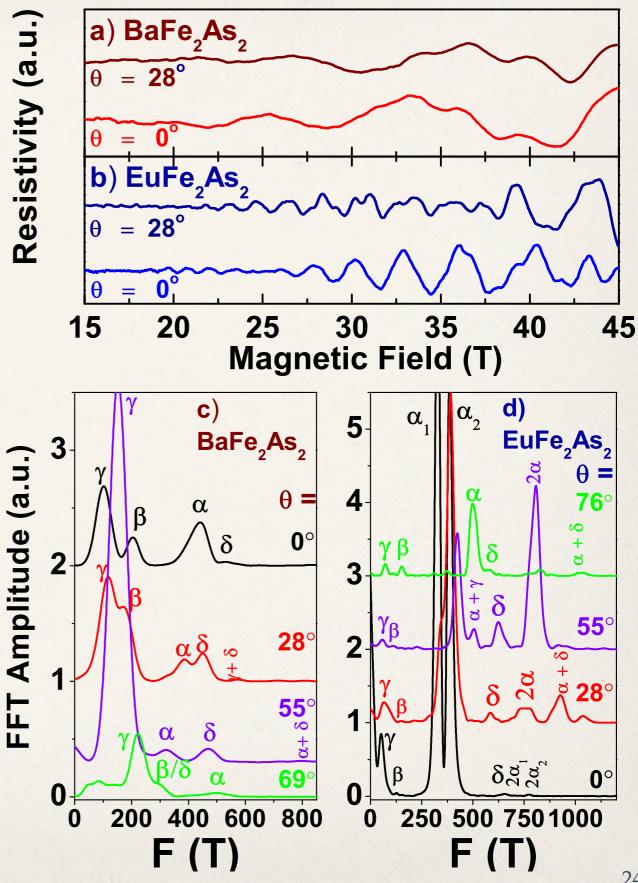


### Quantum Oscillations in EuFe<sub>2</sub>As<sub>2</sub>



### Quantum Oscillations in EuFe<sub>2</sub>As<sub>2</sub>





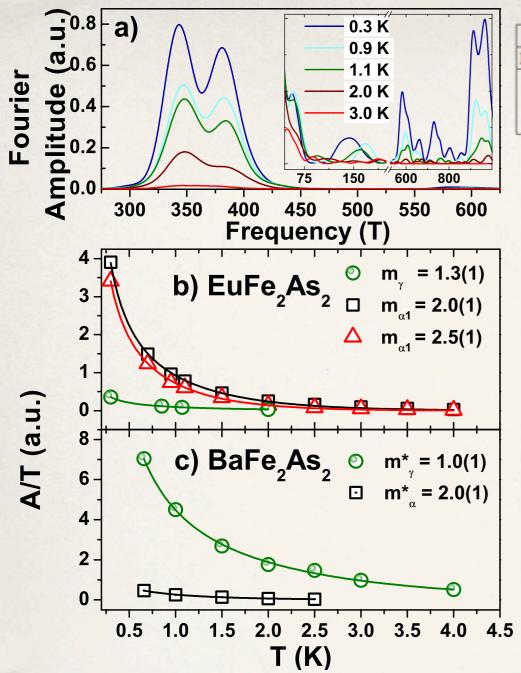


TABLE I: Comparison between the SdH frequencies of Ba122 and Eu122 obtained experimentally in this work.

	BaFe <sub>2</sub> As <sub>2</sub>						EuFe <sub>2</sub> As <sub>2</sub>					
Pocket	F(T)	$A/A_{BZ}$ (%)	$m^*/m_e^*$	$m^*_{\rm DFT}/m^*_e$	R	$T_D$ (K)	F (T)	$A/A_{BZ}$	$m^*/m_e^*$	$m^*_{\rm DFT}/m^*_e$	R	$T_D$
Y	90(10)	0.3	0.7(2)	0.4	2.1	5(2)	60(10)	0.2	1.0(2)	0.4	1.2	-
α	430(10)	1.4	1.5(2)	-0.8	2.1	4(1)	340(10)/380(10)	1.3	1.5(2)/1.9(2)	-0.8	1.4	4(1)
δ	510(10)	1.7	-	1.2	0.8	3(1)	580(10)	1.4	-	1.7	0.5	3(1)

TABLE II: Comparison between the SdH frequencies of Ba122 obtained experimentally in refs. [21, 22]

		BaFe <sub>2</sub> As <sub>2</sub> r	BaFe <sub>2</sub> As <sub>2</sub> ref. [22].					
Pocket	F(T)	$A/A_{BZ}$ (%)	$m^*/m_e^*$	$T_D$ (K)	F (T)	$A/A_{BZ}$	$m^*/m_e^*$	$T_D$
γ	80(10)	0.3	0.7(2)	3(1)	$\sim 90$		0.9(1)	-
α	440(10)	1.7	1.2(3)	4(1)	$\sim 440$	1.3	2.1(1)	4(1)
δ	-	-	-	-	$\sim 500$	1.4	2.4(3)	3(1)

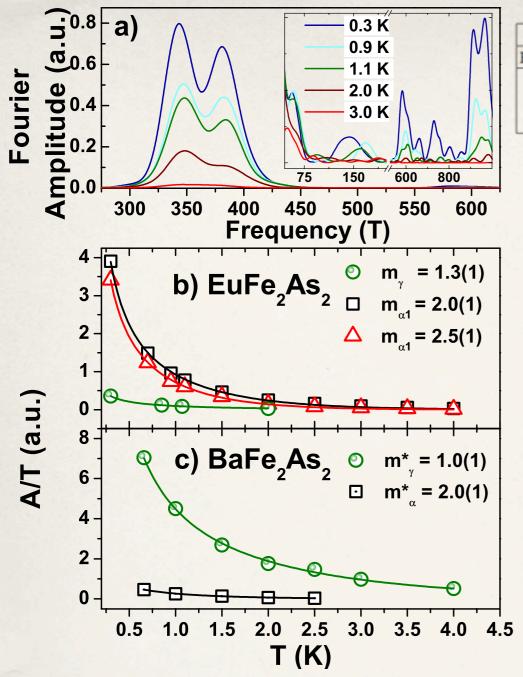
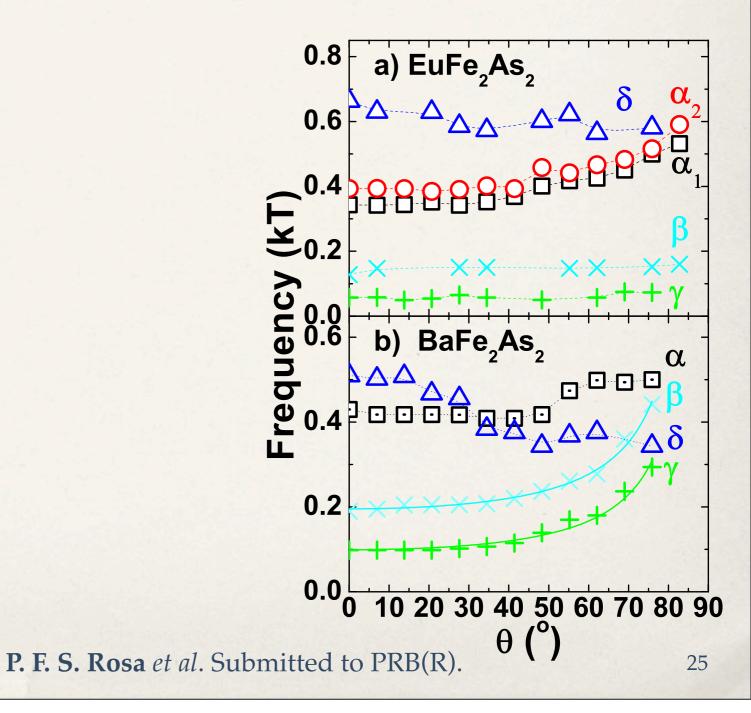


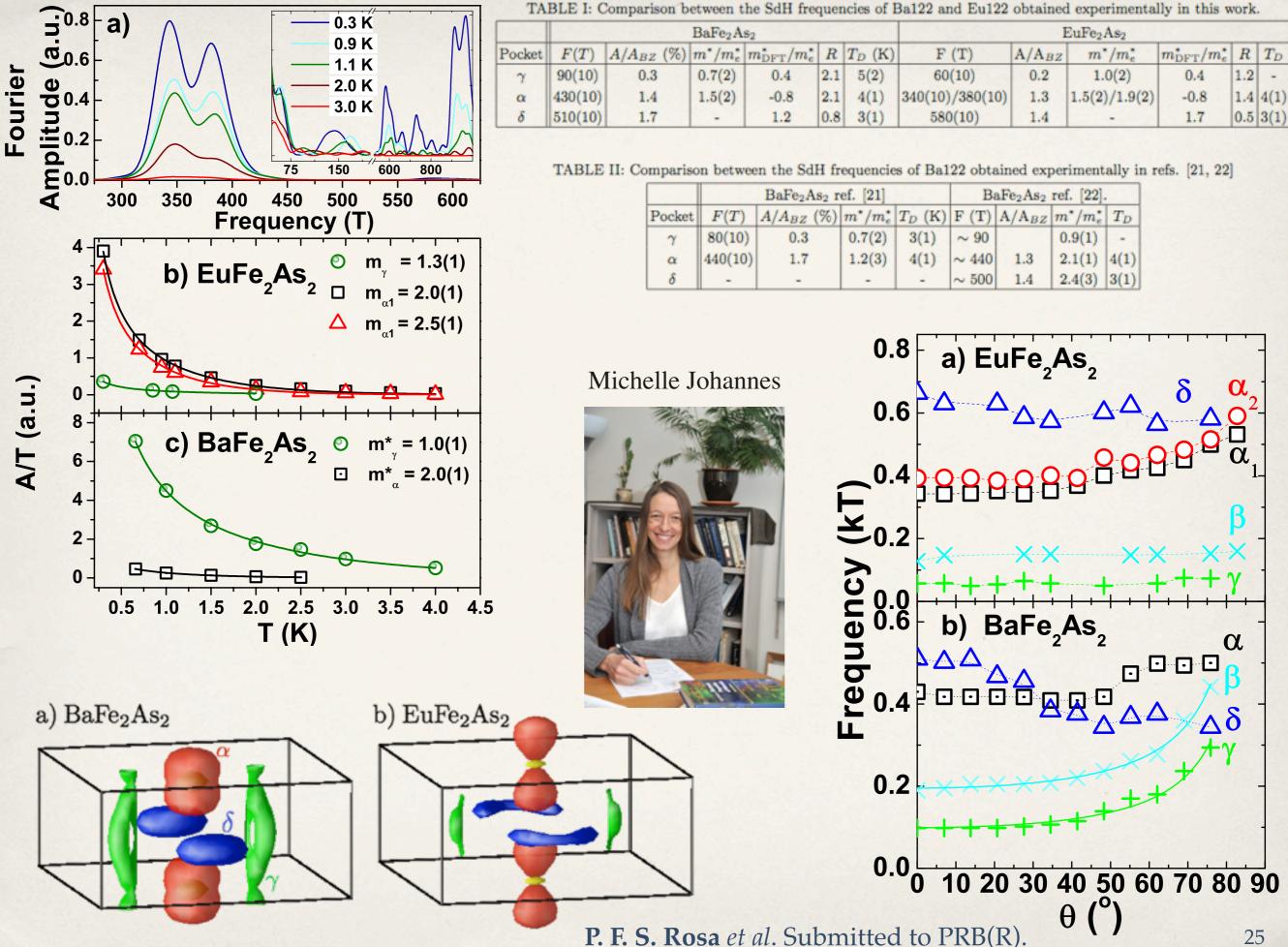
TABLE I: Comparison between the SdH frequencies of Ba122 and Eu122 obtained experimentally in this work.

	BaFe <sub>2</sub> As <sub>2</sub>					EuFe <sub>2</sub> As <sub>2</sub>						
Pocket	F(T)	$A/A_{BZ}$ (%)	$m^*/m_e^*$	$m^*_{\rm DFT}/m^*_e$	R	$T_D$ (K)	F (T)	$A/A_{BZ}$	$m^*/m_e^*$	$m^*_{ m DFT}/m^*_e$	R	$T_D$
Y	90(10)	0.3	0.7(2)	0.4	2.1	5(2)	60(10)	0.2	1.0(2)	0.4	1.2	-
α	430(10)	1.4	1.5(2)	-0.8	2.1	4(1)	340(10)/380(10)	1.3	1.5(2)/1.9(2)	-0.8	1.4	4(1)
δ	510(10)	1.7	-	1.2	0.8	3(1)	580(10)	1.4	-	1.7	0.5	3(1)

TABLE II: Comparison between the SdH frequencies of Ba122 obtained experimentally in refs. [21, 22]

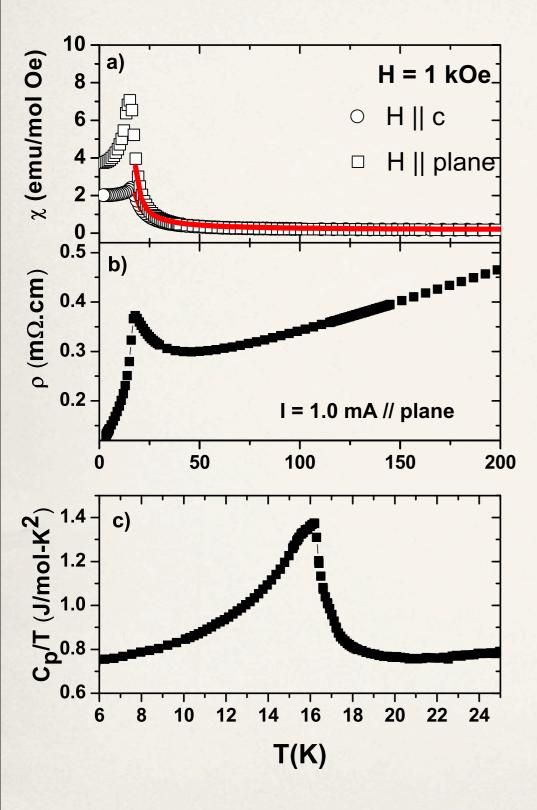
Pocket		BaFe <sub>2</sub> As <sub>2</sub> r	BaFe <sub>2</sub> As <sub>2</sub> ref. [22].					
	F(T)	$A/A_{BZ}$ (%)	$m^*/m_e^*$	$T_D$ (K)	F (T)	$A/A_{BZ}$	$m^*/m_e^*$	$T_D$
γ	80(10)	0.3	0.7(2)	3(1)	$\sim 90$		0.9(1)	-
α	440(10)	1.7	1.2(3)	4(1)	$\sim 440$	1.3	2.1(1)	4(1)
δ	-	-	-	-	$\sim 500$	1.4	2.4(3)	3(1)



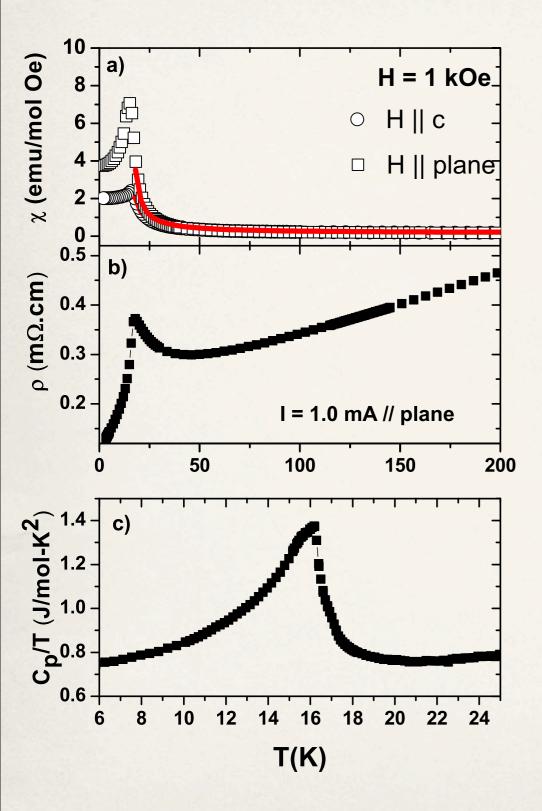


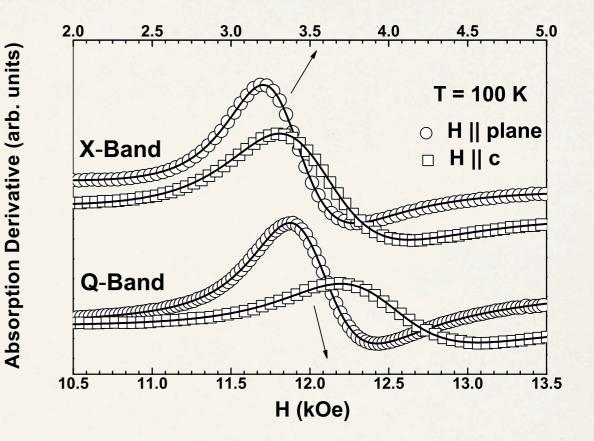
Evidence for the 3d Fe band contribution to the FS \*  $EuIn_2As_2 : T_N = 19 K$ 

\*  $EuIn_2As_2$  :  $T_N = 19 K$ 

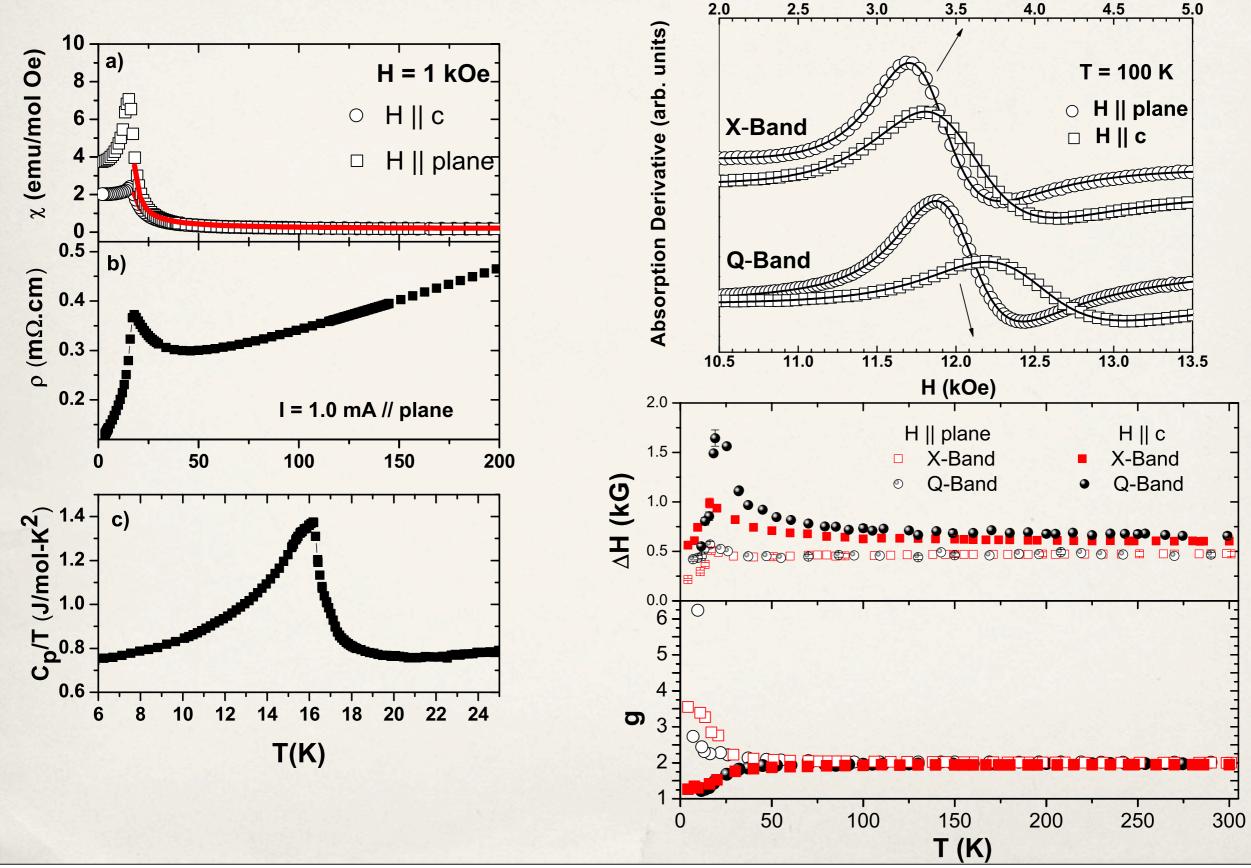


\*  $EuIn_2As_2$  :  $T_N = 19 K$ 

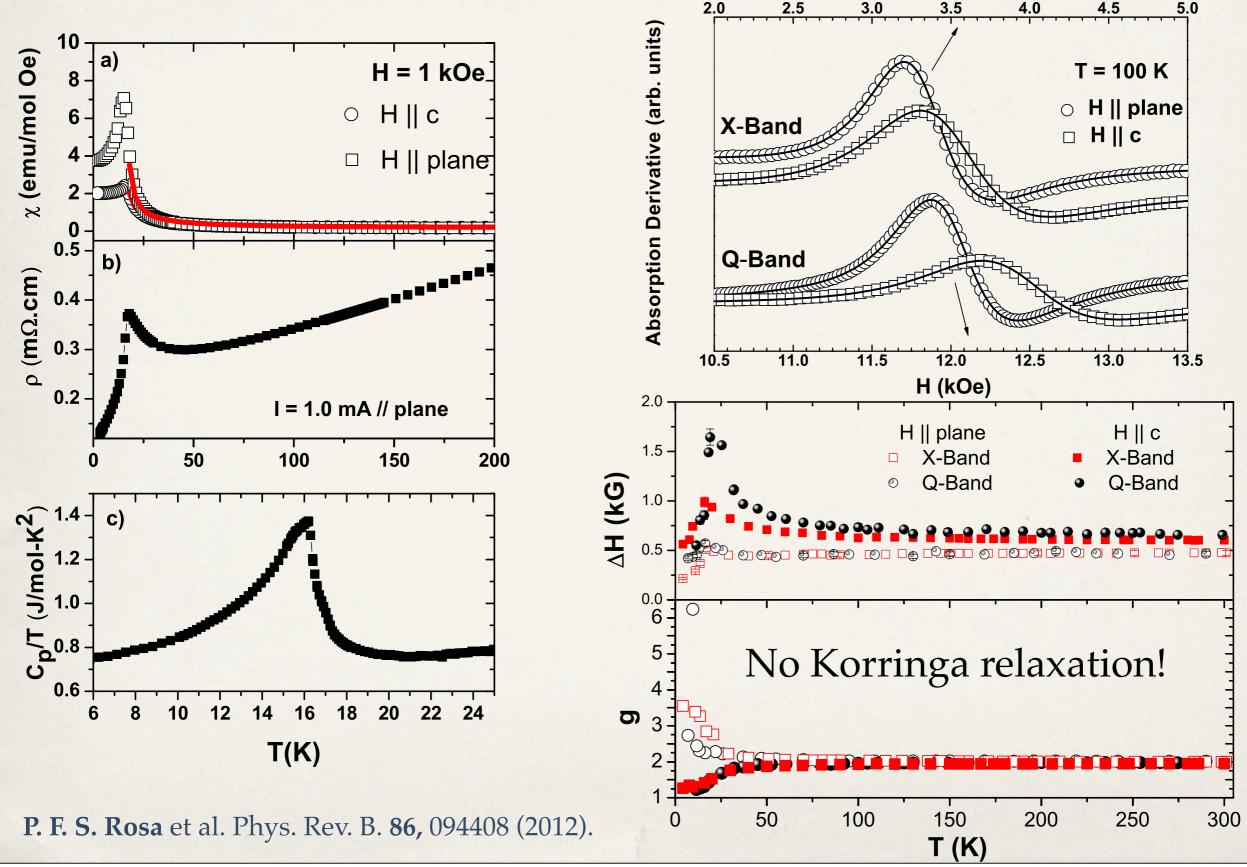




\*  $EuIn_2As_2$  :  $T_N = 19 K$ 

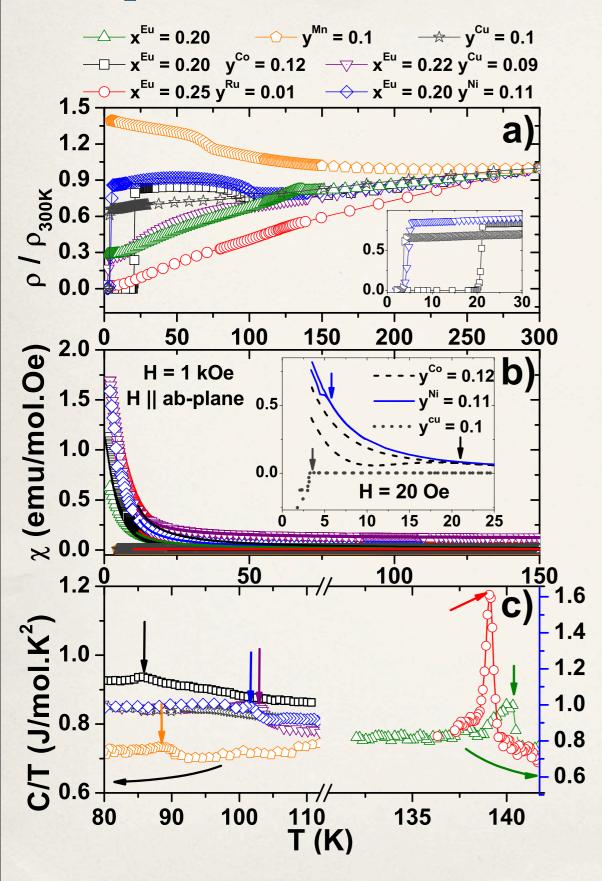


\*  $EuIn_2As_2$  :  $T_N = 19 K$ 

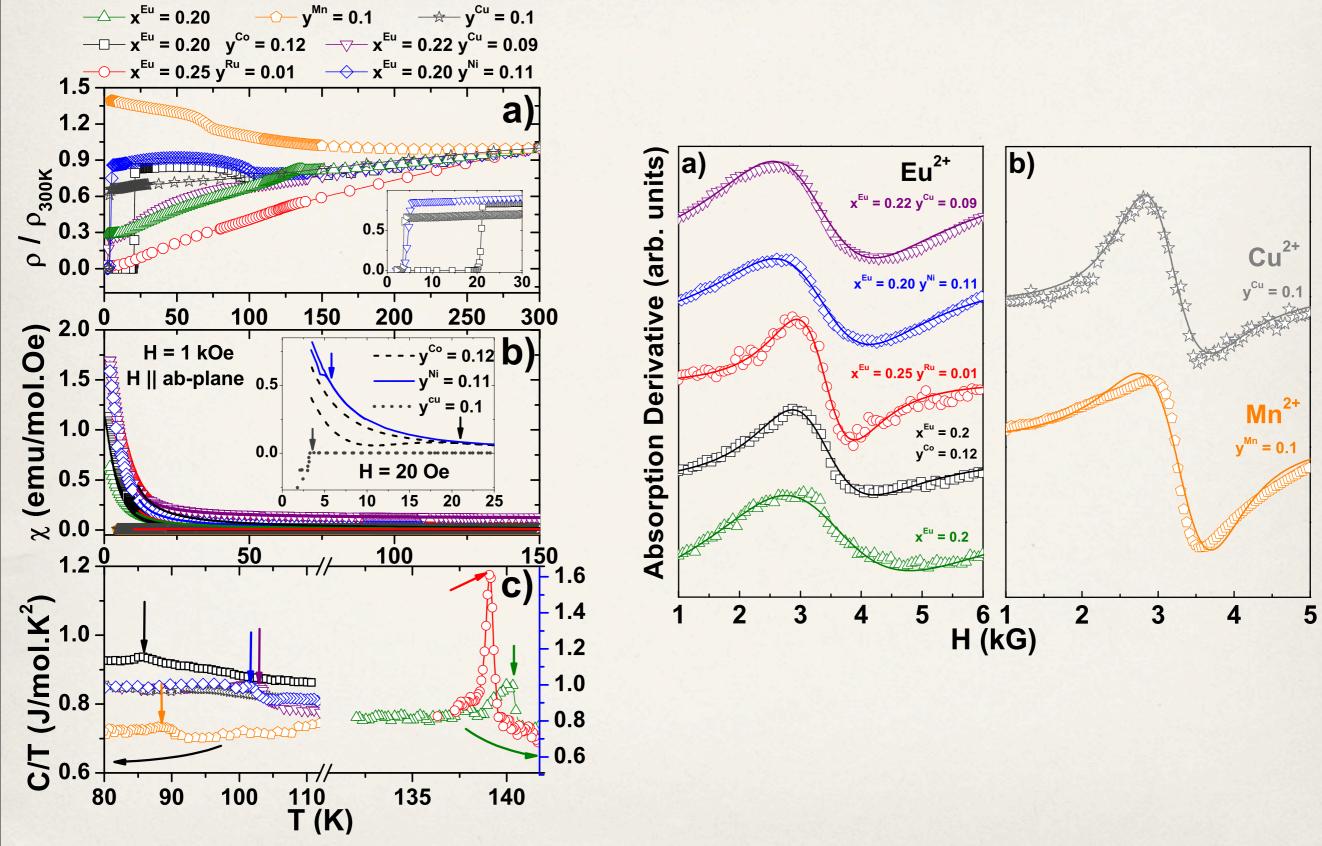


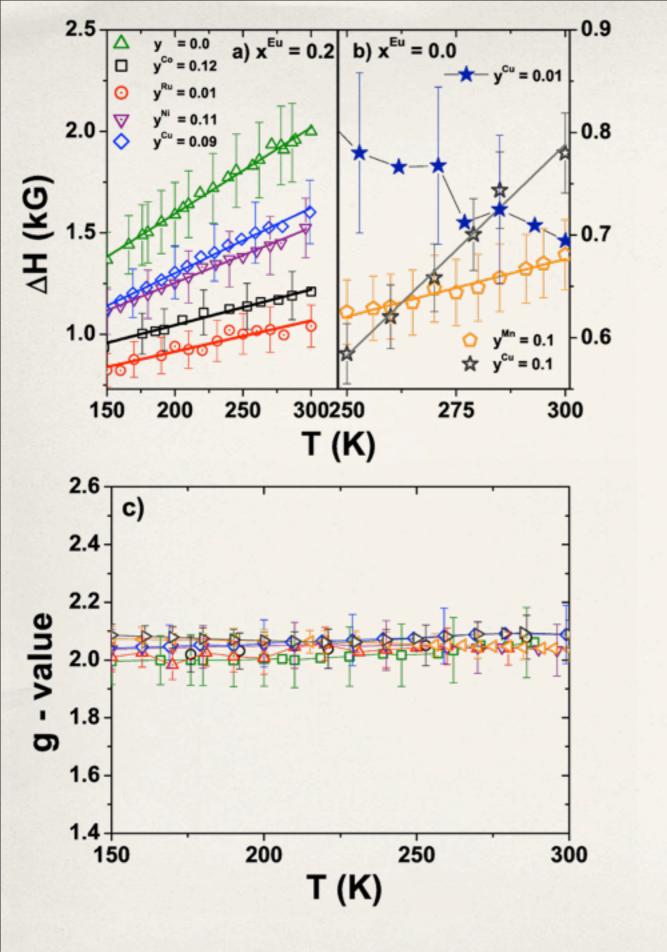
26

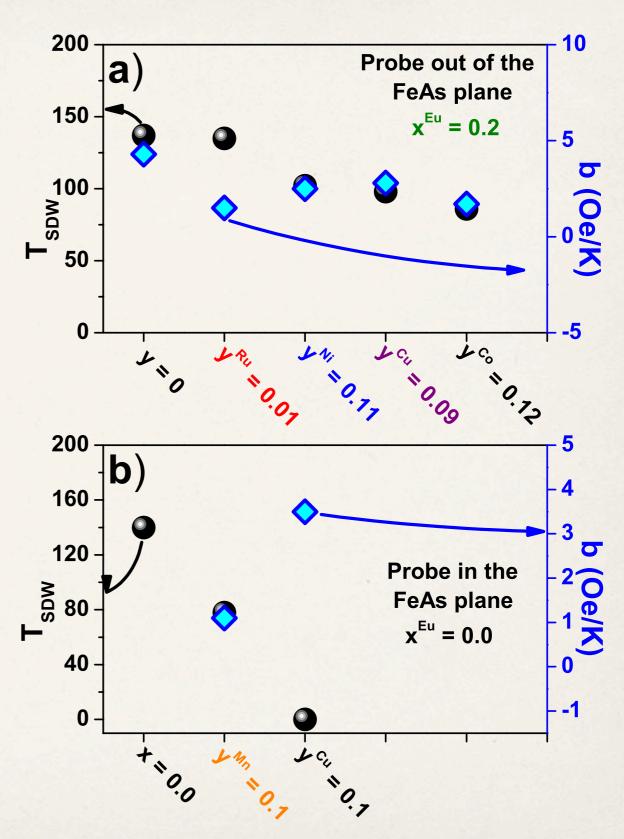
### Site-specific ESR on $Ba_{1-x}Eu_xFe_{2-y}M_yAs_2$ (M = Co, Cu, Ni, and Ru)

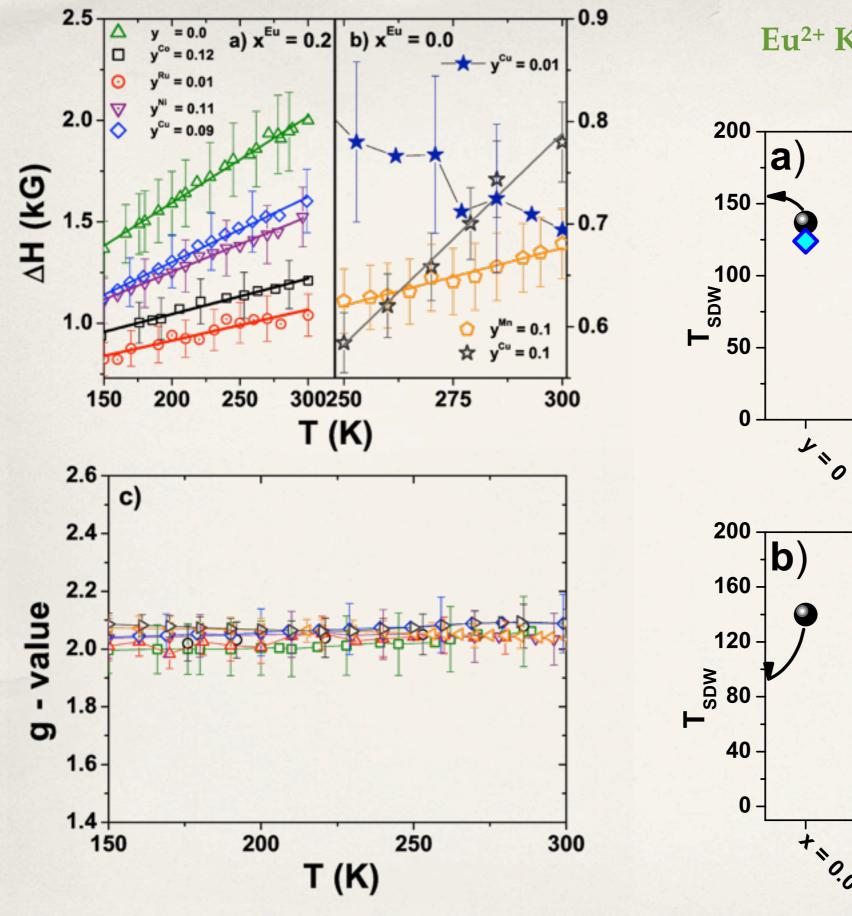


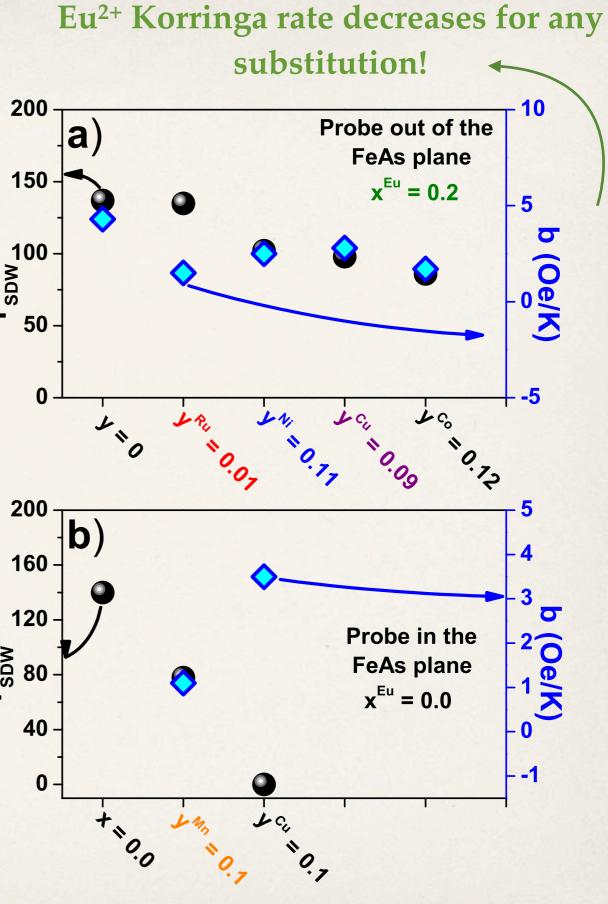
### Site-specific ESR on $Ba_{1-x}Eu_xFe_{2-y}M_yAs_2$ (M = Co, Cu, Ni, and Ru)

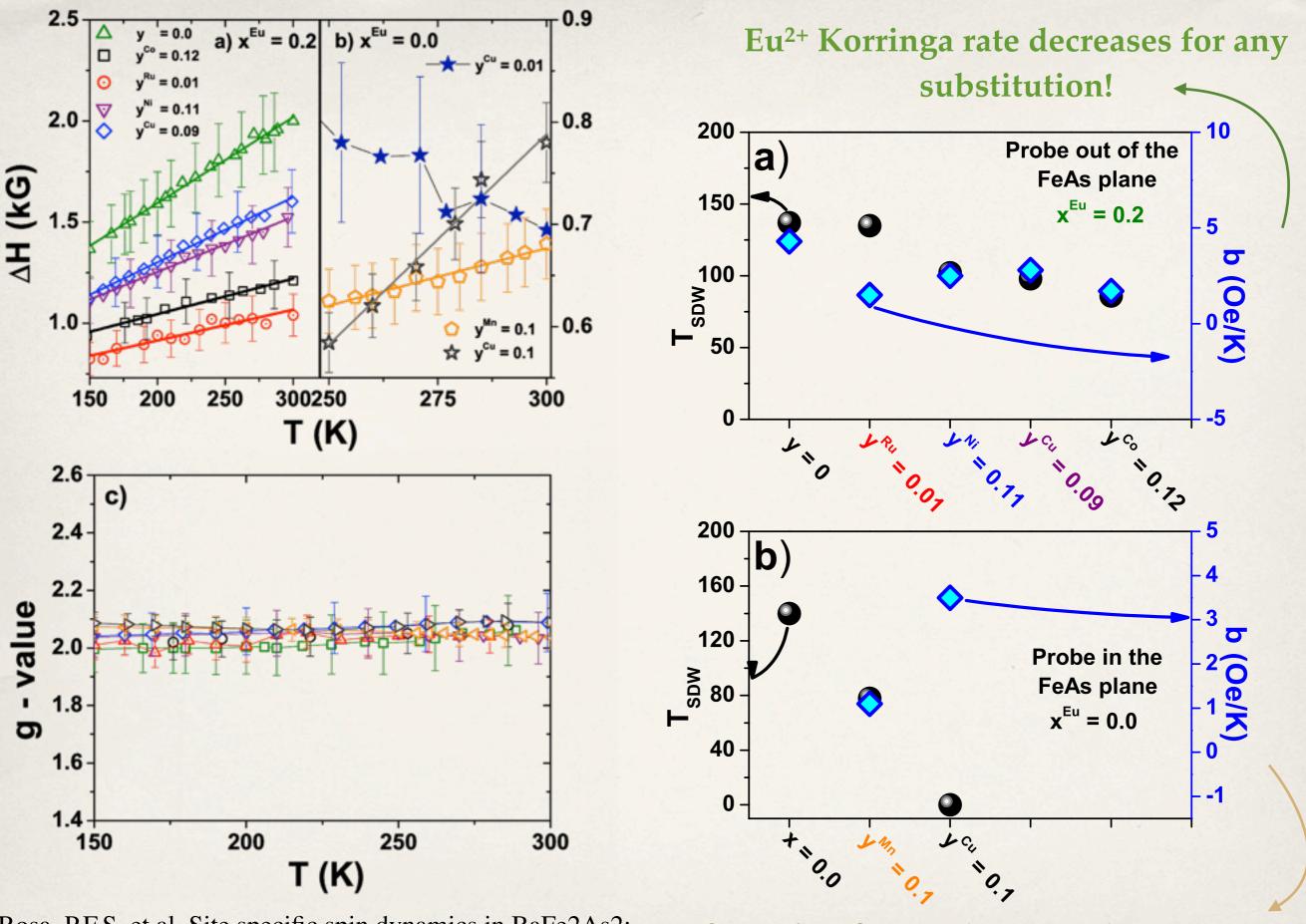












Rosa, P.F.S. et al. Site specific spin dynamics in BaFe2As2: tuning the ground state by orbital differentiation. Sci. Rep. 4, 6543; DOI:10.1038/srep06543 (2014).

Mn<sup>2+</sup> and Cu<sup>2+</sup> Korringa rate increases as the SDW phase is suppressed! 28

# Outline

- Introduction and Motivation: Fe-based Superconductors
- Crystal Synthesis and Characterization

 Microscopic Techniques: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

# Outline

- \* Introduction and Motivation: Fe-based Superconductors
- Crystal Synthesis and Characterization

\* Microscopic Techniques: X-Ray Absorption Spectroscopy (XANES and EXAFS) and Electron Spin Resonance (ESR)

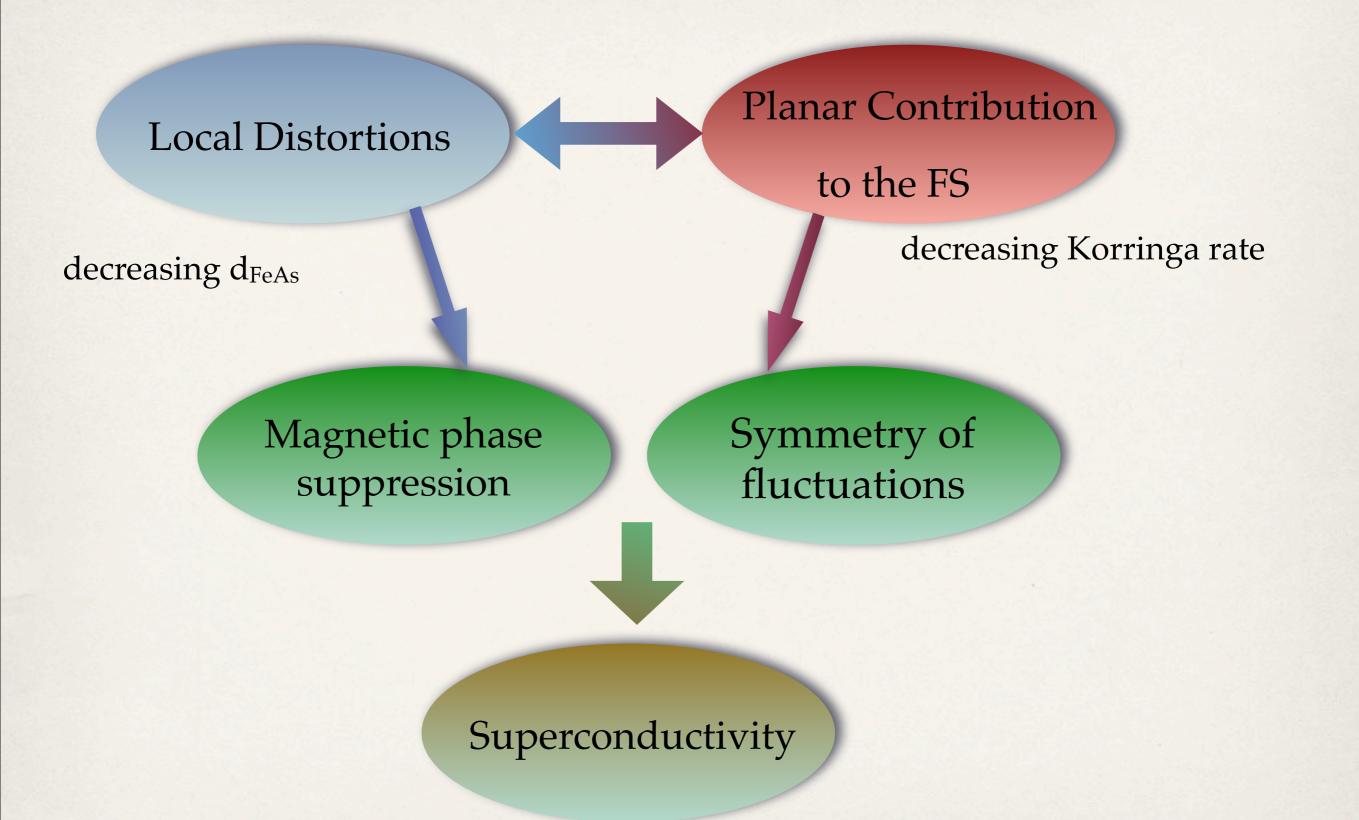
High-quality single crystal growth by the In-flux technique;

- High-quality single crystal growth by the In-flux technique;
- Decrease of the Fe-As distance as the magnetic SDW is suppressed by both chemical substitution (Co e K) and applied pressure;

- High-quality single crystal growth by the In-flux technique;
- Decrease of the Fe-As distance as the magnetic SDW is suppressed by both chemical substitution (Co e K) and applied pressure;
- \* Fe<sup>2+</sup> valence unchanged by Co-substitution;

- High-quality single crystal growth by the In-flux technique;
- Decrease of the Fe-As distance as the magnetic SDW is suppressed by both chemical substitution (Co e K) and applied pressure;
- \* Fe<sup>2+</sup> valence unchanged by Co-substitution;
- Similar critical temperatures independent of chemical substitution (due to the same local distortion), except for Cu<sup>2+</sup> and Mn<sup>2+</sup> that have local magnetic moment (*unconventional magnetic impurity pair breaking*).

- High-quality single crystal growth by the In-flux technique;
- Decrease of the Fe-As distance as the magnetic SDW is suppressed by both chemical substitution (Co e K) and applied pressure;
- \* Fe<sup>2+</sup> valence unchanged by Co-substitution;
- Similar critical temperatures independent of chemical substitution (due to the same local distortion), except for Cu<sup>2+</sup> and Mn<sup>2+</sup> that have local magnetic moment (*unconventional magnetic impurity pair breaking*).
- \* Magnetic phase suppression causes increasing planar  $(xy/x^2-y^2)$  contribution to the 3*d* Fe bands at the Fermi surface.



# Thank you for your attention!