multiscale aspects of thermal transport

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Top-down approach: from Fourier ...



Joseph Fourier 1768–1830 "Analytic theory of Heat" continuum theory, partial differential equations

Steady-state condition:



thermal conductivity

...to Peierls kinetic theory of heat transport

Heat carriers: electrons and lattice vibrations (phonons)

Selectrons. Wiedman-Franz
law: $\frac{\kappa}{\sigma} = LT$

Phonons obey a transport equation analogous to the Boltzmann transport equation, but with quantum statistics.



Rudolf Peierls 1907-1995

> Oxford Classic Texts IN THE PHYSICAL SCIENCES

Electrons and Phonons

J. M. Ziman

Boltzmann-Peierls equation for phonons $\frac{\partial n}{\partial t} + v\frac{\partial n}{\partial r} + \frac{F}{m}\frac{\partial n}{\partial t} = \left(\frac{\partial n}{\partial t}\right)_{collision}$ Transport equation Relaxation $\left(\frac{\partial n}{\partial t}\right)_{collision} = -\frac{n-n^0}{\tau}$ stationary $\vec{v} \cdot \vec{\nabla} n = -\frac{n - n^0}{\tau}$ conditions $n = n^0 + \delta n \longrightarrow \delta n = -\tau \frac{\partial n^0}{\partial T} \vec{v} \cdot \vec{\nabla} T$ linearize

Solution of the linearized Boltzmann-Peierls equation the heat flux (power/area) is linked to δn : $\vec{J} = \sum_{BZ} \int_{BZ} d^3q \frac{\hbar\omega_s(q)}{8\pi^3} \delta n_s(q) \vec{v}_s(q)$ using Fourier's law and $\delta n = -\tau \frac{\partial n^0}{\partial T} \vec{v} \cdot \vec{\nabla} T$ $J_{\alpha} = \sum \int_{BZ} \int_{BZ} d^3q \frac{\hbar\omega_s(q)}{8\pi^3} \frac{\partial n_s^0(q)}{\partial T} \tau v_{s,\alpha}(q) v_{s,\beta}(q) \nabla_{\beta} T$ $\kappa_{lphaeta}$ here we assumed that normal modes populations can be treated separately!

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Solution of the Boltzmann-Peierls equation

In a more human-readable and computerpracticable way:

$$\kappa_{\alpha\beta} = \sum C(s,q) v_{\alpha} v_{\beta} \tau$$

phonon mean free path λ

more approximations on top:
single relaxation time
v: speed of sound

s,q

Collisions: where does λ (or τ) come from?

Anharmonicity: phonon-phonon scattering (3-phonon processes are the lowest order)

Ø Defects

Isotopic mass disorder

Boundaries

 $\tau_b^{-1} = (\langle 1/c \rangle_b lF)^{-1},$

$$\tau_a^{-1} = BT\omega^2 e^{-C/T},$$

$$\tau_i^{-1} = A \omega^4$$
.

Simple models from BP: Callaway's model (1959) Disregard distinct phonon branches Assume linear dispersion: $\omega = \bar{v_q}q$ \oslash Effective lifetime τ^* and λ^* $\kappa(T) = \rho^{-2/3} \int_{0}^{\omega_{D}} d\omega \lambda^{*} \frac{\hbar\omega}{2\pi} \frac{\partial n}{\partial T}$

Refinements followed, but do not work for nanostructures (more parameters)

Thriving research activity in thermal transport



new (and nanostructured) thermoelectric materials

Atomistic methods: equilibrium MD

The thermal conductivity is just another response function.

Second Green-Kubo theory: $\kappa = \frac{1}{kT^2V} \int_0^\infty dt \langle J(o)J(t) \rangle$

or Einstein relation:

 $\kappa_{\alpha\beta} = \frac{1}{k_B T^2} \lim_{t \to \infty} \lim_{V \to \infty} \frac{1}{2V} \frac{d}{dt} \langle [R_{\alpha}(t) - R_{\beta}(0)]^2 \rangle$

Size and time convergence require special care

Size and time convergence: carbon nanotubes

Several long (100 ns) runs are needed

need to sample well low frequency modes, which contribute a lot



Size and time convergence: graphene

- k converges, unlike purely
 2D and 1D models:
 flexural modes
- Convergence occurs from above: in general convergence is nonmonotonic
- phonons are heat carriers but also scatterers



L.F. Pereira, DD (in preparation)

MD, lattice dynamics and the Boltzmann-Peierls equation

 $\kappa_{\alpha\beta} = \sum_{s,q} C(s,q) v_{\alpha} v_{\beta} \tau$

Shopping list:

Normal modes e_i: diagonalization of the force constant matrix

Group velocities: dispersion curves

Phonon lifetimes: projection of a MD trajectory on the normal modes (all orders of anharmonicity)

Phonon lifetimes from MD

Green-Kubo formula for τ :

Energy of mode *i*:

Projection of the atomic displacements over the eigenmode e_i of the dynamical matrix

$$\lambda_{i} = v_{i}\tau_{i} \qquad \text{v: from dispersion curves}$$

$$\tau_{i} = 2\int_{0}^{\infty} dt \langle E_{i}(0)E_{i}(t) \rangle$$

$$E_{i} = \frac{\omega_{i}^{2}S_{i}^{*}S_{i}}{2} + \frac{\dot{S}^{*}\dot{S}}{2}$$

$$S_{i}(q) = N^{-1/2}\sum_{j}^{N} \sqrt{M}e^{-iqr_{j,0}}e_{i}^{*}(q) \cdot u_{j}$$

Ladd, Moran, Hoover Phys. Rev. B 34, 5058 (1986) P. Chantrenne et al. J. Appl. Phys. 97, 104318 (2005) A.J.H McGaughey et al. Phys. Rev. B 69, 094303 (2004)

Resolving contributions to k -> customize heat transport





Suspended CNT: ~1600 W/m K On a Si substrate: ~250 W/m K

DD & G. Galli PRL 99, 255502 (2007)

Non-Equilibrium MD

 $\frac{1}{\kappa_L} = \frac{1}{k} + \frac{A}{L}$

O Use Fourier's law

- Thermostats are applied to hot/cold temperature reservoirs
- Temperature gradients are computed at stationary nonequilibrium conditions

Size scaling. Usually:



Comparing EMD and NEMD

Good news: sometimes they give the same result



amorphous silicon

1/k is not linear with 1/L

Y. He, DD, G. Galli, APL (2011)

Comparing EMD and NEMD

... but some other times they don't! (graphene)



(non)-Equilibrium phonon populations

non-equilibrium phonon
 populations must be
 explicitly considered

this would imply failure of the single-mode relaxation time approximation



(Dis)advantages of MD

- Full details, atomistic treatment
- Relatively large scales (million-atom systems)
- No truncation of anharmonicity
- Possibility to simulate systems out of equilibrium
- No quantum effects (important at room temperature for Si, crucial for C)
- Need for very accurate forcefields: so far the method cannot be implemented in a DFT framework

Other atomistic approaches

ballistic transport by non-equilibrium Green's function or scattering theory (elastic scattering)
very good at low T, but no anharmonicity
self-consistent solution of the Boltzmann-Peierls equation (limited to 3-phonon scattering)
extremely expensive, and fails at high T

Recent and future developments

neural network approximation of DFT potentials
 recently applied to phase-changing materials

ø pushing the limit of the scattering matrix approach to million atom simulations

And-matter/soft-matter interfaces: model systems and realistic composite materials

Chances for multiscale modeling?

- Systematic application of DFT-quality neural network potentials
- How much can we learn from model systems?
- Energy transfer in biological systems (cell membranes)
- Quantum dynamics

Systems and references

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