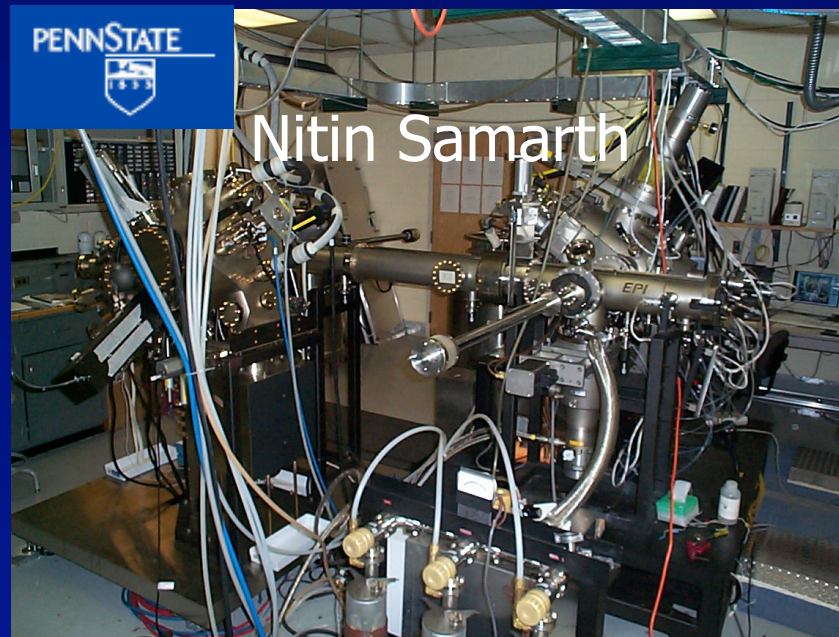


# Spin Transport & Scattering in Ferromagnetic Semiconductor Heterostructures



## Outline

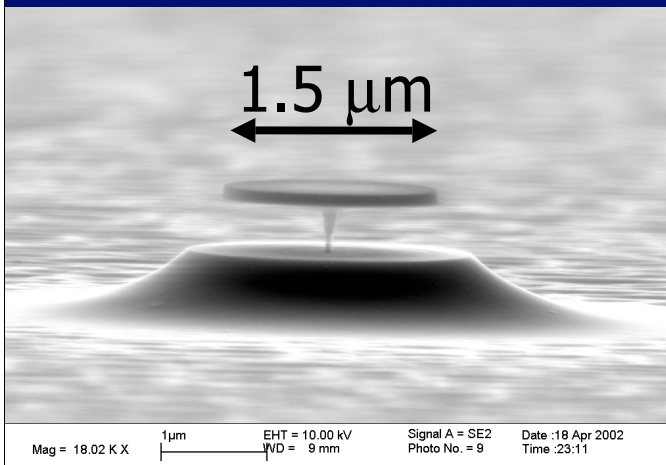
- Controlling spins in semiconductor heterostructures: overview
- Spin transport & scattering in (Ga,Mn)As devices:
  - Non-collinear spin valve effect in trilayer devices (Xiang et al, in preparation)
  - Pinning and controlling domain walls at constrictions & interfaces (Eid et al, in preparation)
- Interfacial control of ferromagnetism in (Ga,Mn)As
  - Exchange biasing of (Ga,Mn)As by MnO (Eid et al., APL **85**, 1556 [2004], Eid et al. J. Appl. Phys. **97**, 10D304 [2005])
  - Nanoengineered  $T_C$  in submicron (Ga,Mn)As wires (Eid et al, APL **86**, 152505 [2005])

Epitaxial integration and patterning of magnetic materials with established growth/processing protocols

- paramagnetic semiconductors
- ferromagnetic semiconductors
- ferromagnetic metals
- antiferromagnets

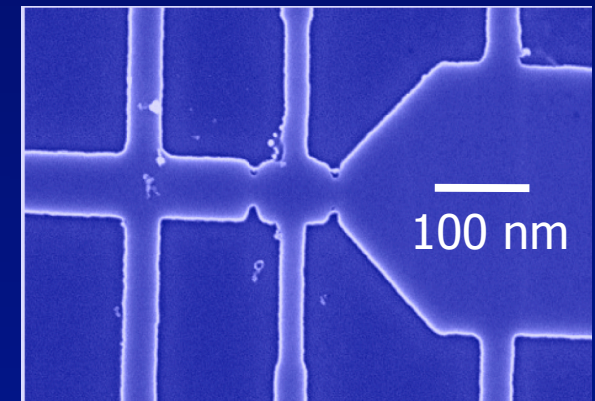
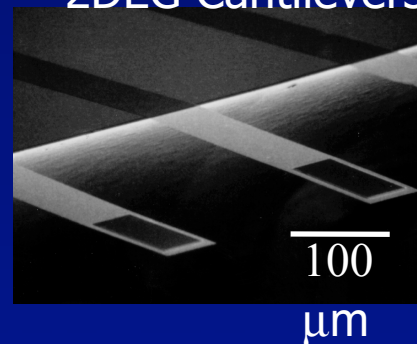


Frustrated magnetic arrays of Py mesas



II-VI microdisk on  
~100nm diameter AlGaAs  
pedestal

GaAlAs/ZnSe/(Zn,Cd,Mn)Se  
2DEG Cantilevers



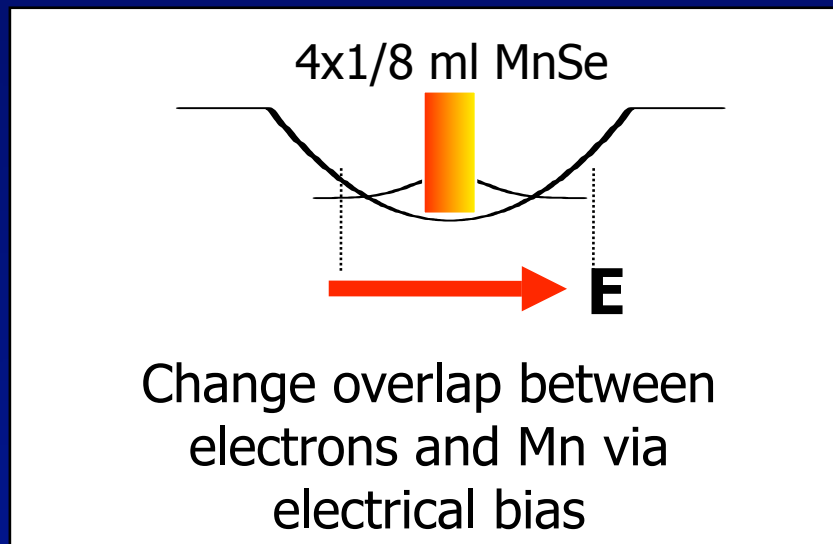
Submicron (Ga,Mn)As device  
For domain wall pinning

# Towards Semiconductor Spintronics - I

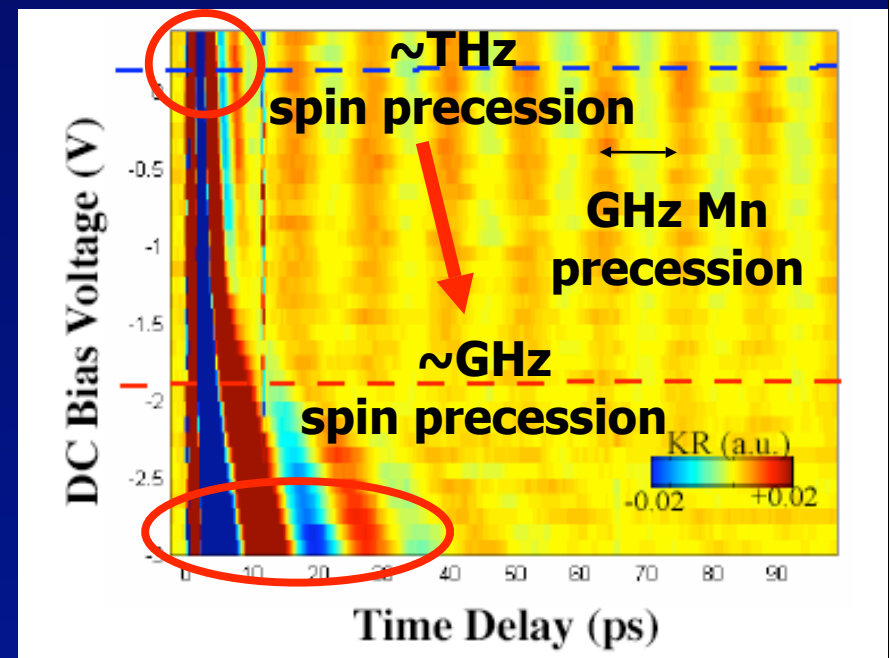
Create, control & detect spin polarization of electrons/holes in semiconductors

- Electric fields
- Exchange interactions

## Electrically controlled spin dynamics



Parabolic potential using (Zn,Cd)Se "digital alloy"; Mn ions in center.



Myers et al, PRB **72**, 041302(R) [2005]

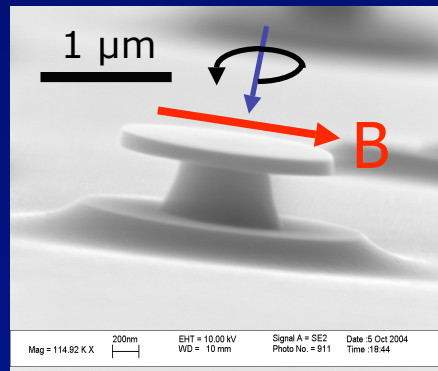


# Towards Semiconductor Spintronics - II

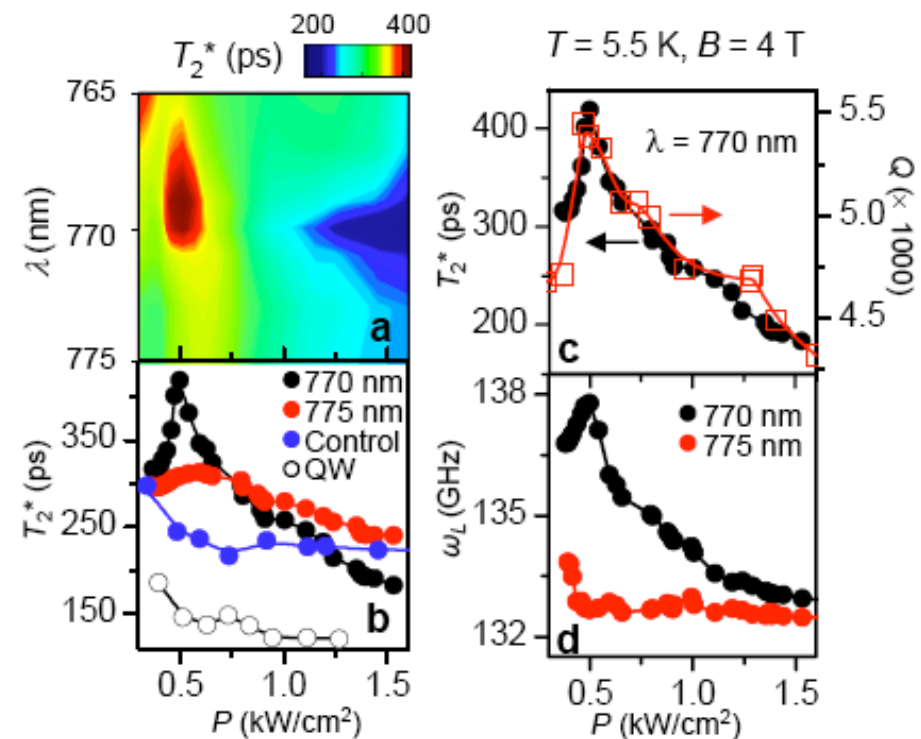
Create, control & detect spin polarization of electrons/holes in semiconductors

- Electric fields
- Exchange interactions
- Circularly polarized photons

Q-factor engineering of electron spin coherence in GaAs/GaAlAs microdisk lasers



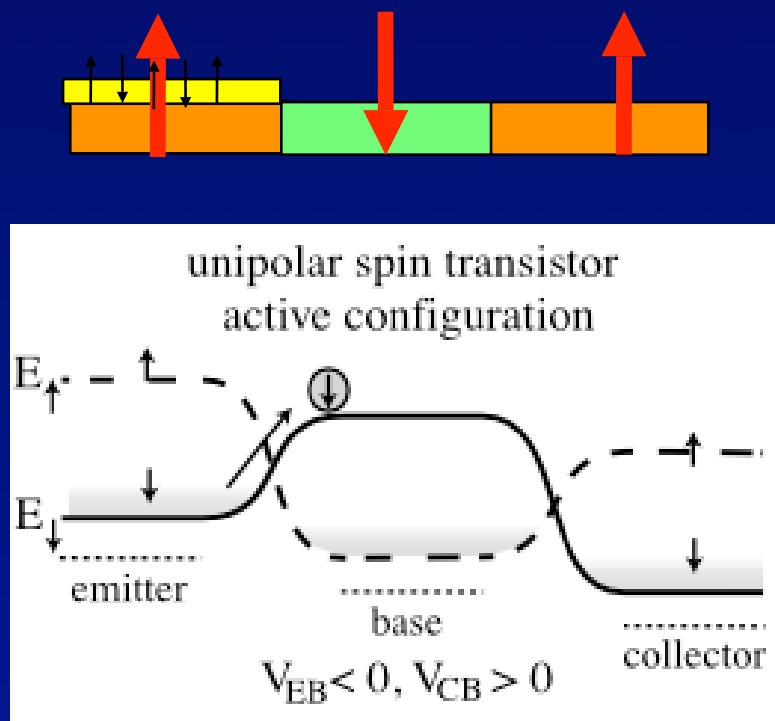
Ghosh, Wang et al, Nature (Materials), **56** (2006)



# Towards Semiconductor Spintronics -- III

Exploit spin transport in both conventional and magnetic semiconductors

- Spin injection & spin polarized transport
- Spin Hall effect
- Unipolar spin diodes & transistors (Flatte & Vignale)
- Magnetic bipolar transistors (Flatte et al, Zutic et al)

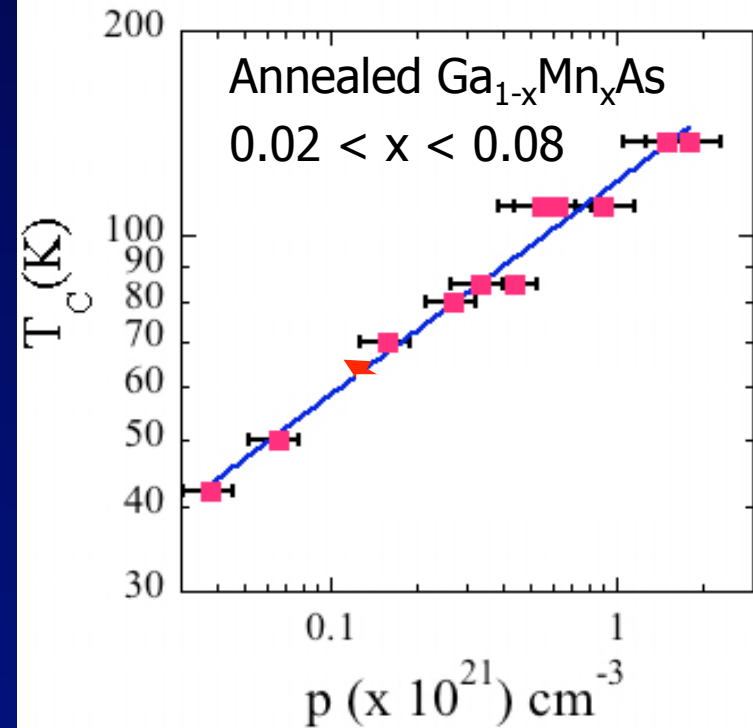
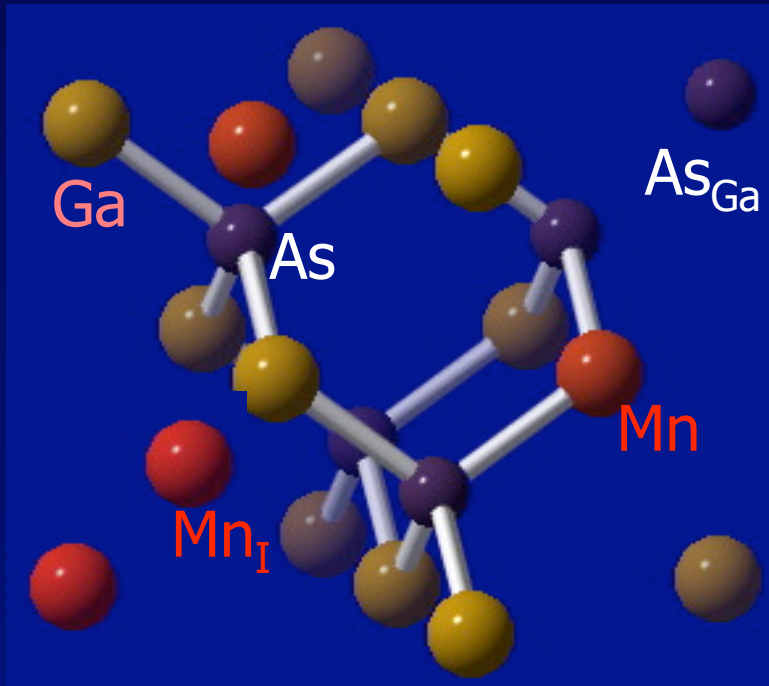


Need to control:

- Switching field of different device elements via
    - Shape anisotropy
    - Magneto-crystalline anisotropy, strain
    - Exchange bias
  - Domain wall locations
    - Pinned: fixed architecture
    - Moveable: reconfigurable
- (Holleitner et al., 2004, Yamounichi et al. 2004)

# $\text{Ga}_{1-x}\text{Mn}_x\text{As}$ : the “canonical” ferromagnetic semiconductor

Ku et al., APL **82**, 2302 (2003)

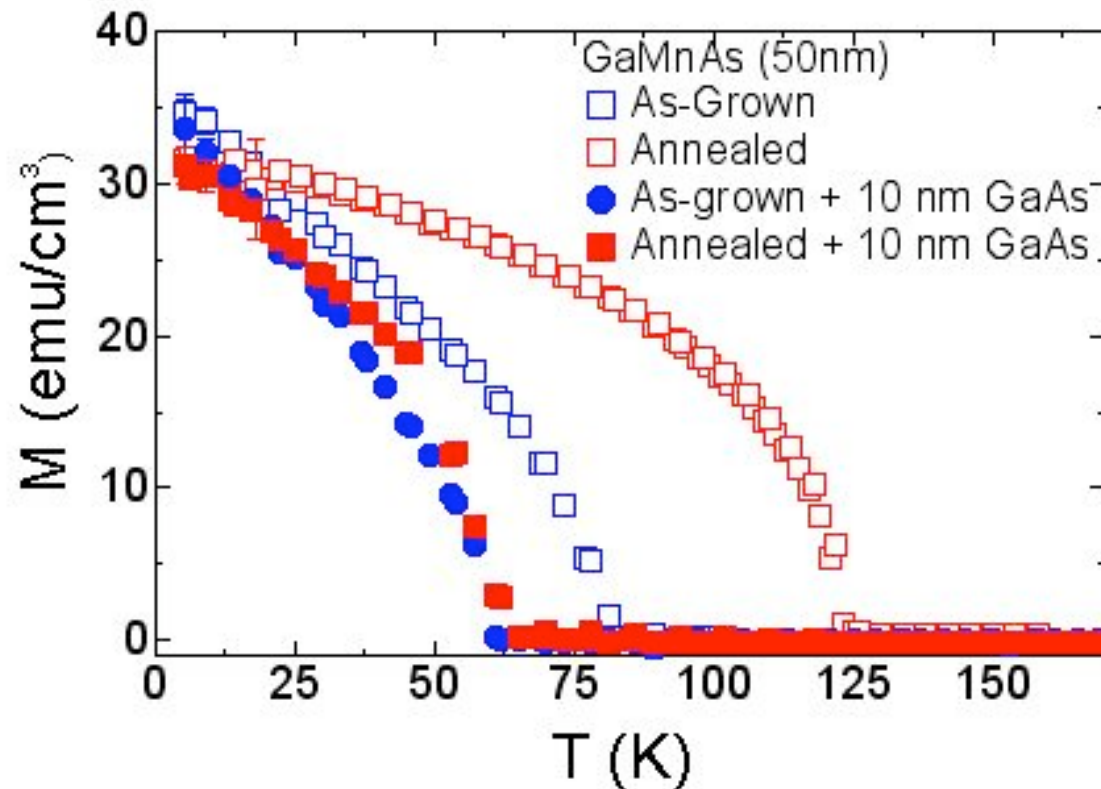


- Hole-mediated ferromagnetism:  $\text{Mn}^{2+}$  ( $S = 5/2$ ) in zinc-blende GaAs lattice
- Low temperature MBE: Mn interstitial & As antisite defects (donors)
- Post-growth annealing: Mn interstitials to free surface of sample [Yu et al., PRB (2002), Edmonds et al, PRL (2004)]
- $T_C$  can be increased up to  $\sim 170$  K. ( $\sim 240$  K? Tanaka)
- Origin of ferromagnetism: impurity band (e.g, Burch et al, 2006)

# Annealing Effects are Suppressed in Heterostructures!

M. B. Stone et al. Appl. Phys. Lett. **83**, 4568 (2003)

See also: Chiba et al, APL (2003)



GaAs



- Capping (Ga,Mn)As with a thin epitaxial layer of GaAs suppresses beneficial aspects of annealing
- Diffusing interstitials (donors) create pn junction at interface?

# I. Non-collinear spin valve effect in ferromagnetic semiconductor trilayer devices

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With **G. Xiang**, M. Zhu, B. L. Sheu, & P. Schiffer (in preparation)



# Searching for the spin valve effect in (Ga,Mn)As devices

VOLUME 61, NUMBER 21

PHYSICAL REVIEW LETTERS

21 NOVEMBER 1988

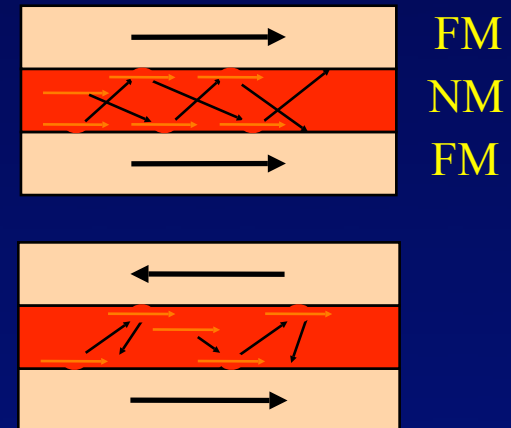
## Giant Magnetoresistance of (001)Fe/(001)Cr Magnetic Superlattices

M. N. Baibich,<sup>(\*)</sup> J. M. Broto, A. Fert, F. Nguyen Van Dau, and F. Petroff  
*Laboratoire de Physique des Solides, Université Paris-Sud, F-91405 Orsay, France*

P. Eitenne, G. Creuzet, A. Friederich, and J. Chazelas  
*Laboratoire Central de Recherches, Thomson CSF, B.P. 10, F-91401 Orsay, France*  
(Received 24 August 1988)

We have studied the magnetoresistance of (001)Fe/(001)Cr superlattices prepared by molecular-beam epitaxy. A huge magnetoresistance is found in superlattices with thin Cr layers: For example, with  $t_{Cr}=9 \text{ \AA}$ , at  $T=4.2 \text{ K}$ , the resistivity is lowered by almost a factor of 2 in a magnetic field of 2 T. We ascribe this giant magnetoresistance to spin-dependent transmission of the conduction electrons between Fe layers through Cr layers.

PACS numbers: 75.50.Rr, 72.15.Gd, 75.70.Cn



In (Ga,Mn)As:

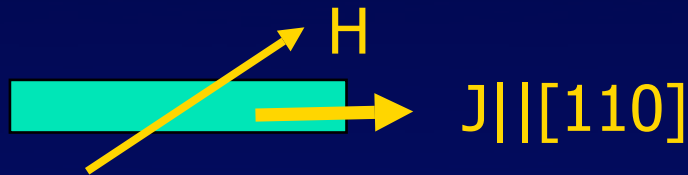
- Short hole spin diffusion length (~few nm)
- Short elastic mean free path (<2 nm)
- Strong magnetoresistance (anisotropic magnetoresistance, anomalous Hall effect, planar Hall effect)

$$\frac{\Delta R}{R} = \frac{(\alpha - 1)^2}{4(\alpha + pd_{NM}/d_{FM})(1 + pd_{NM}/d_{FM})}$$

Key Factors:

- $\alpha = \rho_{\downarrow}/\rho_{\uparrow}$
- $p = \rho_{NM}/\rho_{\uparrow}$
- $d_{NM}/d_{FM}$

# Spin Valve Effect in Ferromagnetic Semiconductor Trilayers

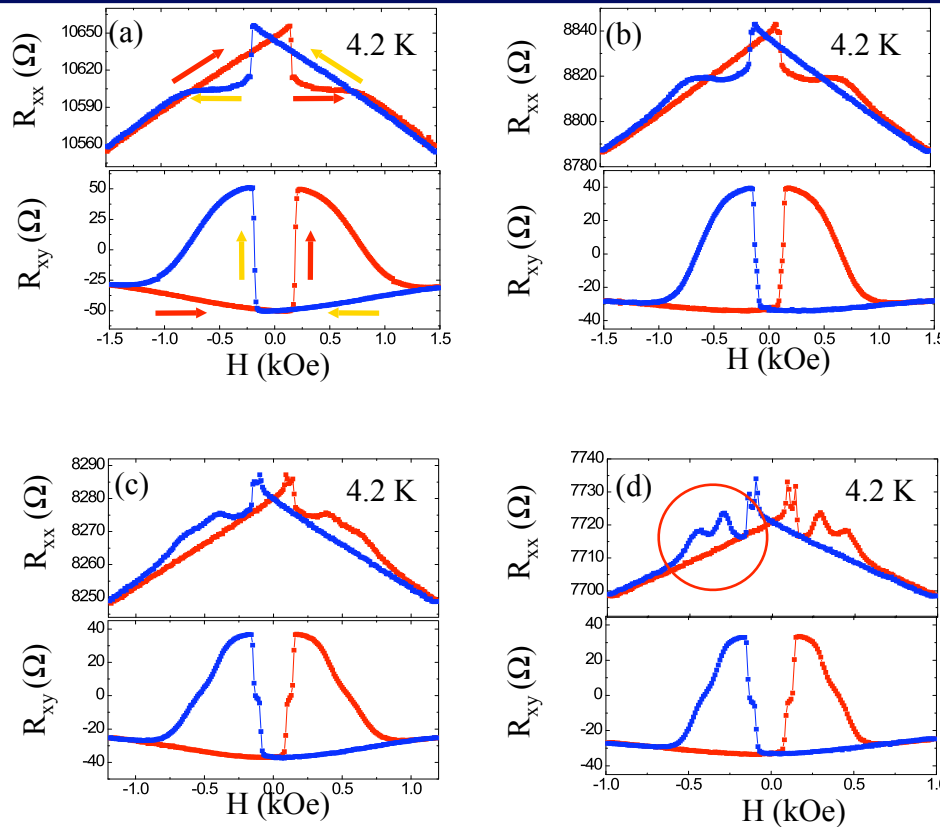


As-grown

30 minute anneal



GaMnAs  
GaAs:Be  
GaMnAs



60 min anneal

120 minute anneal

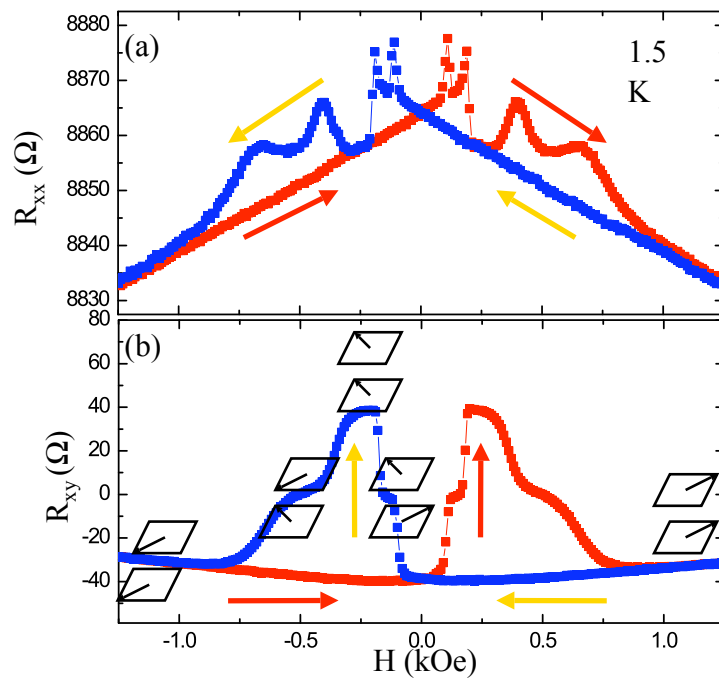
- Trilayer devices wherein ferromagnetic layers & spacer layer have comparable conductivity
- Annealing creates distinct  $H_C$  for upper/lower GaMnAs layers
- Speculate that annealing also produces spin-dependent scattering at interfaces
- Spacer thickness varied: 2 nm, 5 nm & 10 nm
- Annealing time varied for 5 nm spacer

# Spin Valve Effect in Ferromagnetic Semiconductor Trilayers

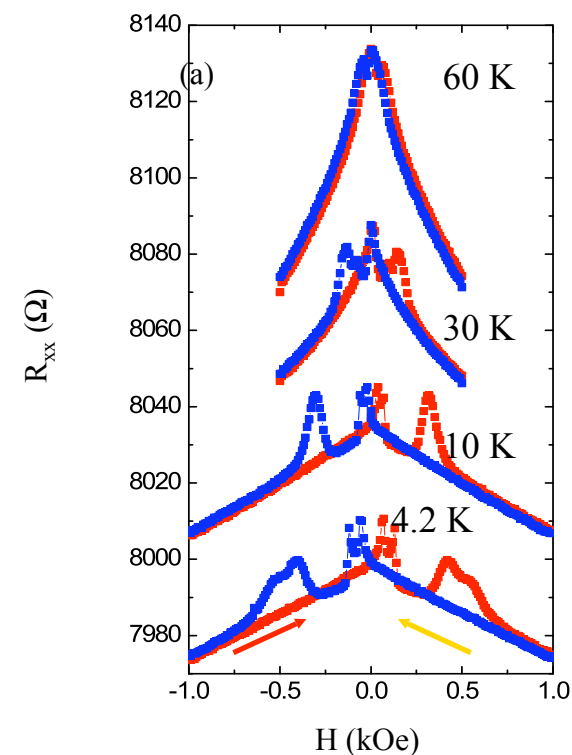
- **Planar Hall effect** tracks magnetization orientation: magnetization of each layer switches by  $90^\circ$
- Comparison between PHE and MR shows non-collinear spin valve effect: enhanced resistance for orthogonal magnetization orientation
- Unusual "structure" due to interplay between AMR and spin valve effect



GaMnAs  
GaAs:Be  
GaMnAs

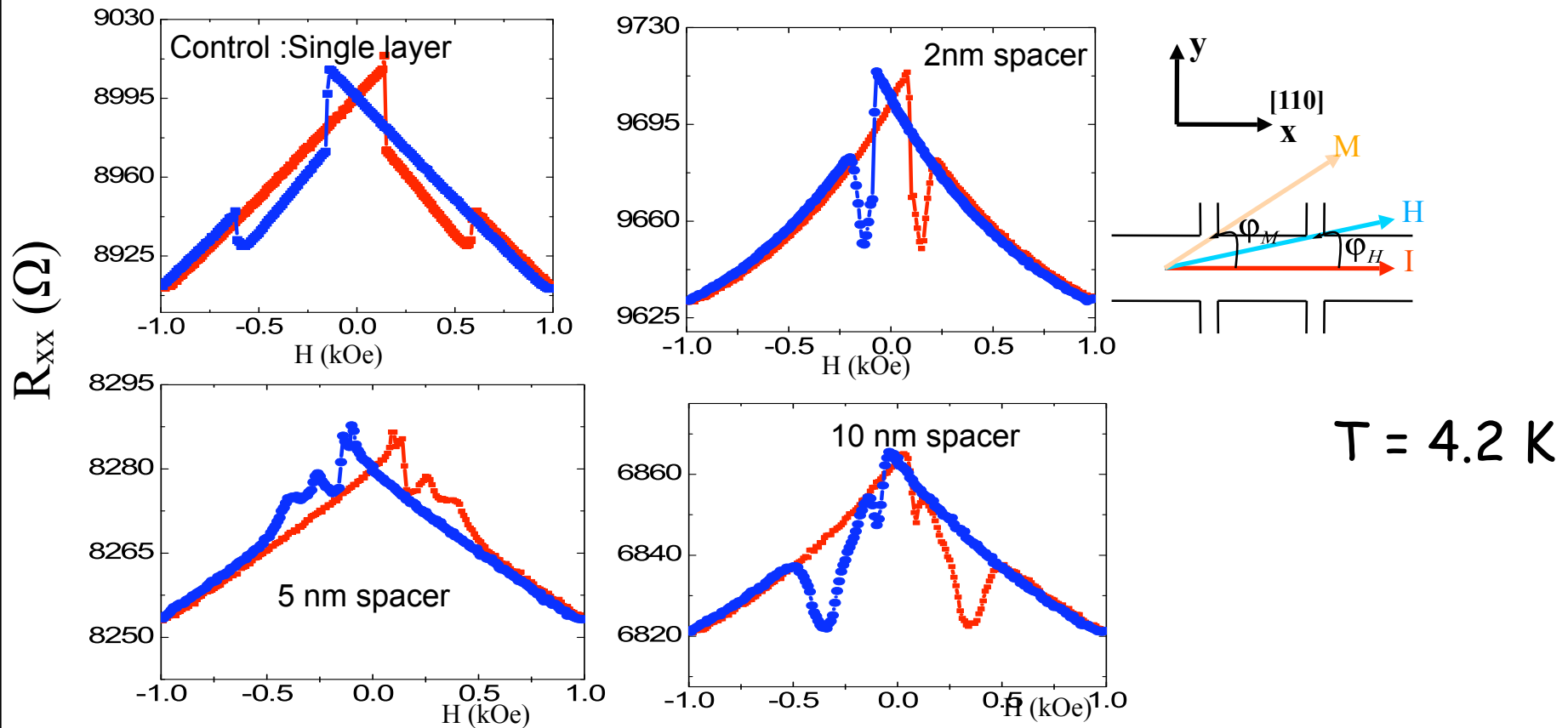


(Xiang *et al.*, in preparation)



# Magnetoresistance in trilayers: spacer thickness dependence

All devices annealed at 190°C for 1 hr



- Sample A (2nm spacer) MR similar to single FM layer's
- Sample B (5nm spacer) MR  $\rightarrow$  positive  $\Delta R$
- Sample C (10nm spacer) MR looks like addition of two FMs

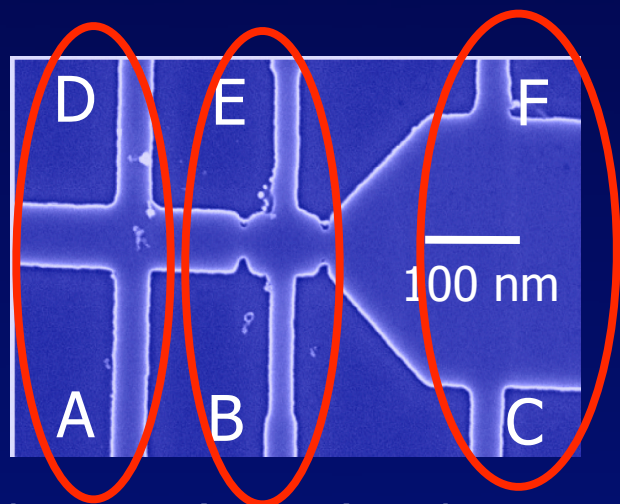
## II. Pinning, measuring and controlling domain walls in patterned (Ga,Mn)As devices

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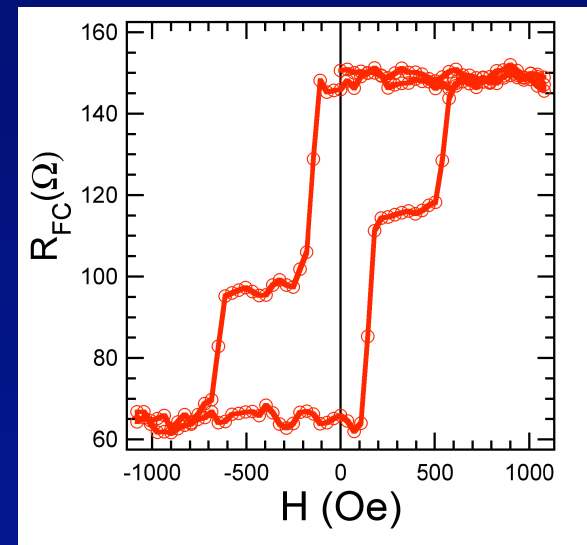
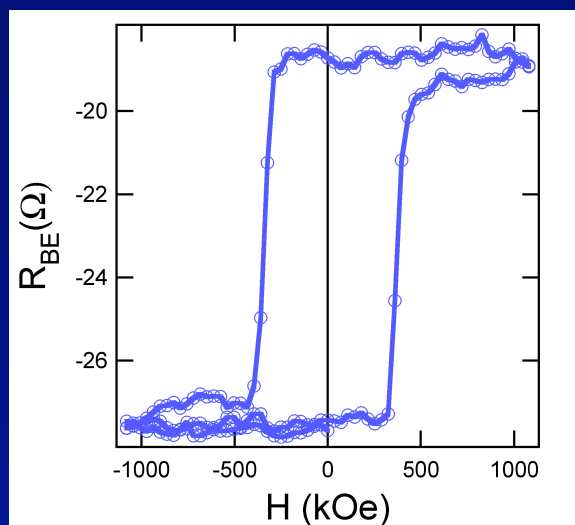
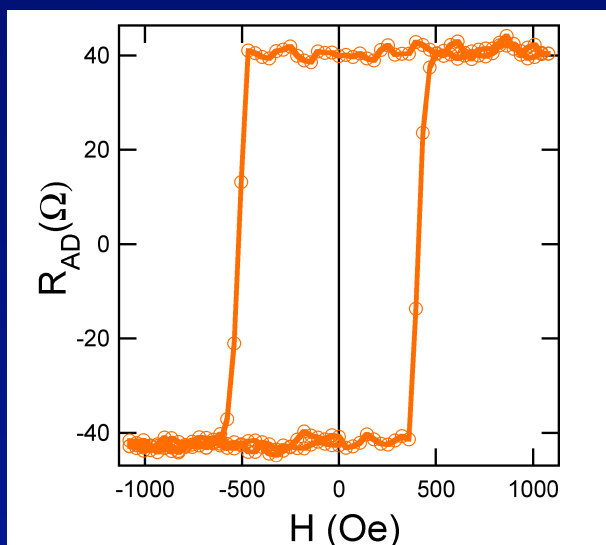
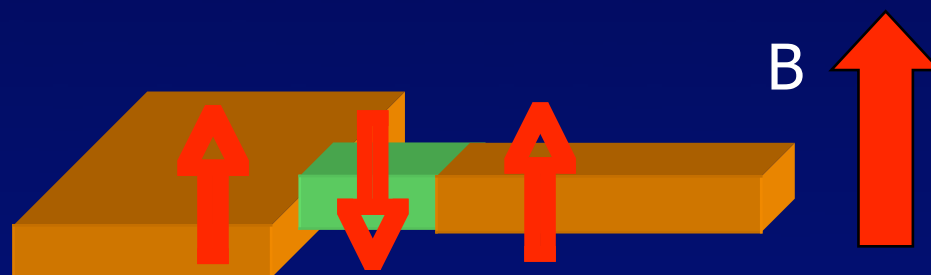
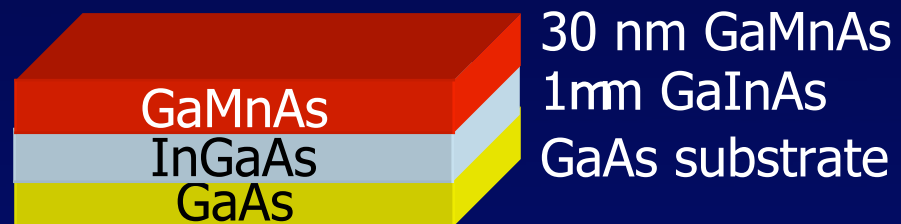
With **K. F. Eid**, G. Xiang, A. Balk, B. L. Sheu, O. Maksimov, & P. Schiffer  
(in preparation)



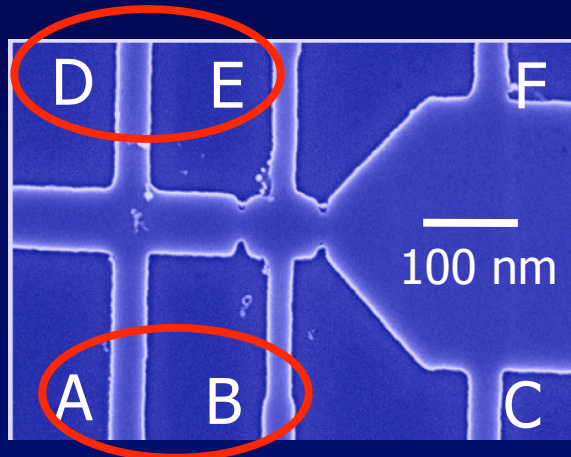
# Pinning & detecting domain walls in (Ga,Mn)As



Submicron (Ga,Mn)As device  
for domain wall pinning

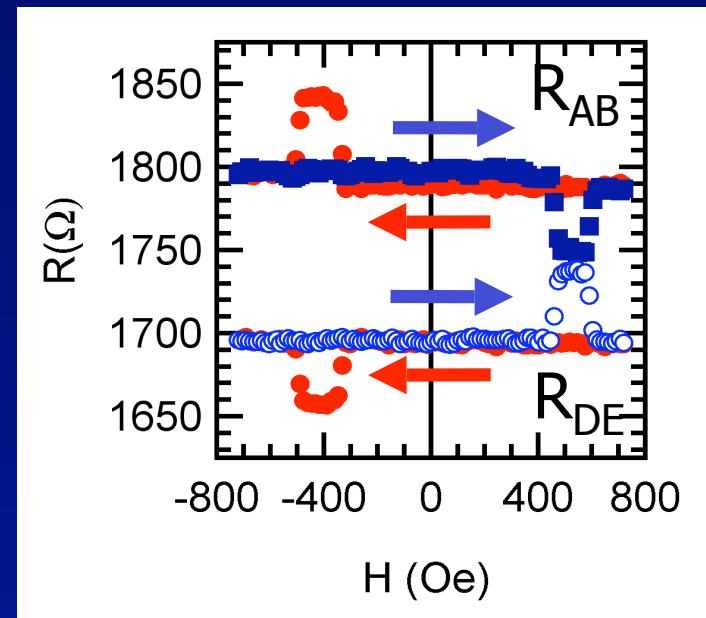
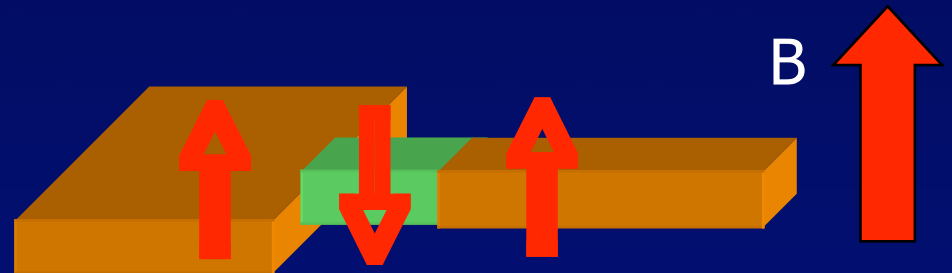
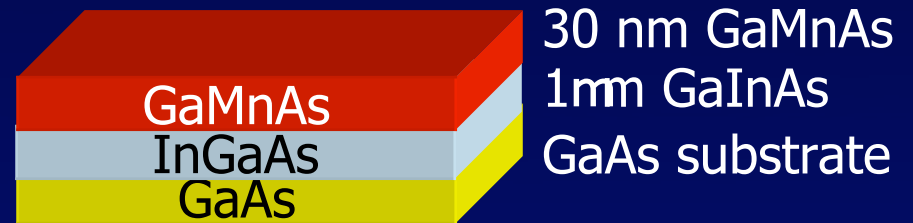


# Pinning & detecting domain walls in (Ga,Mn)As

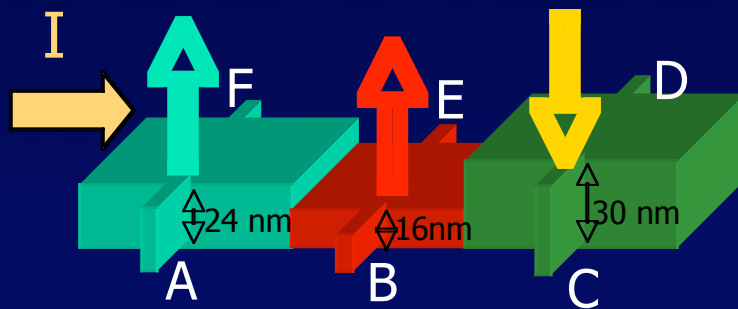


Submicron (Ga,Mn)As device for domain wall pinning

- Circulating currents in vicinity of pinned DW
- Longitudinal MR is antisymmetric in magnetic field due to  $R_{xy}$  (anomalous Hall effect) contribution to  $R_{xx}$
- Note: intrinsic DW resistance negligible (Chiba et al 2006, Tang et al 2004)

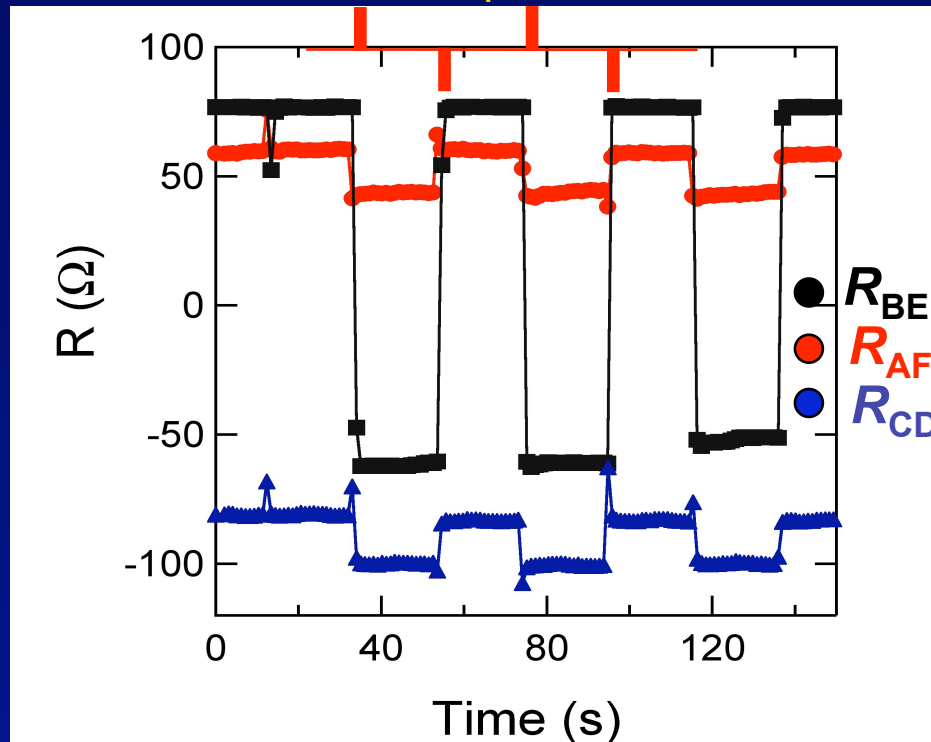


# Electrically manipulating domain walls in (Ga,Mn)As devices



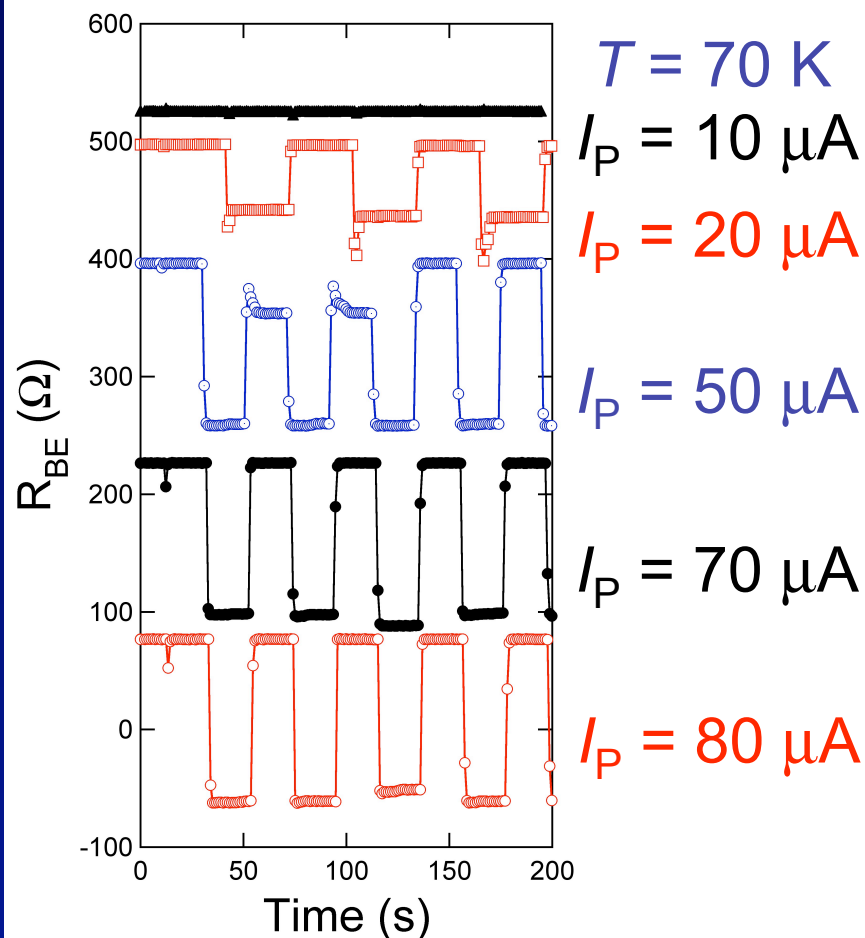
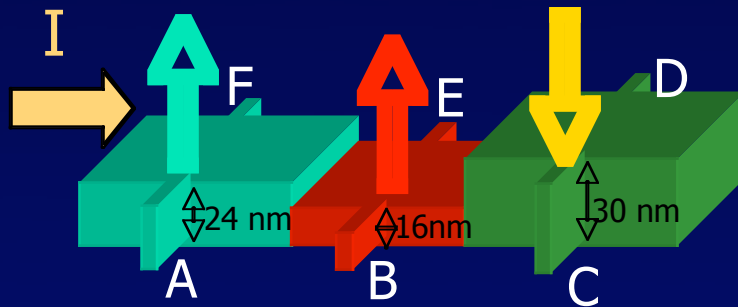
30 nm GaMnAs  
1mm GaInAs  
GaAs substrate

$T = 70 \text{ K}; I_p = 80 \mu\text{A}$

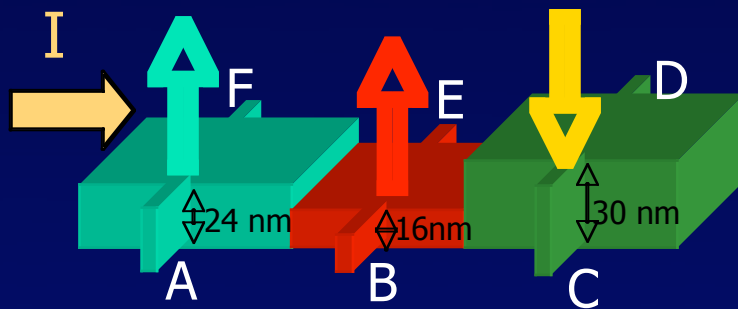


- Lateral channel size:  $20 \mu\text{m}$ ; middle element:  $40 \mu\text{m}$  long
- Vertical etching steps define (Ga,Mn)As elements with different coercivity (Yamounichi et al, Science 2004)
- Current pulses of 100 ms width sent in alternating "up" & "down" sequence at 20 s intervals
- Monitor switching of middle element via anomalous Hall effect
- Note: DW motion is opposite to direction of current pulse

# Electrically manipulating domain walls in (Ga,Mn)As devices

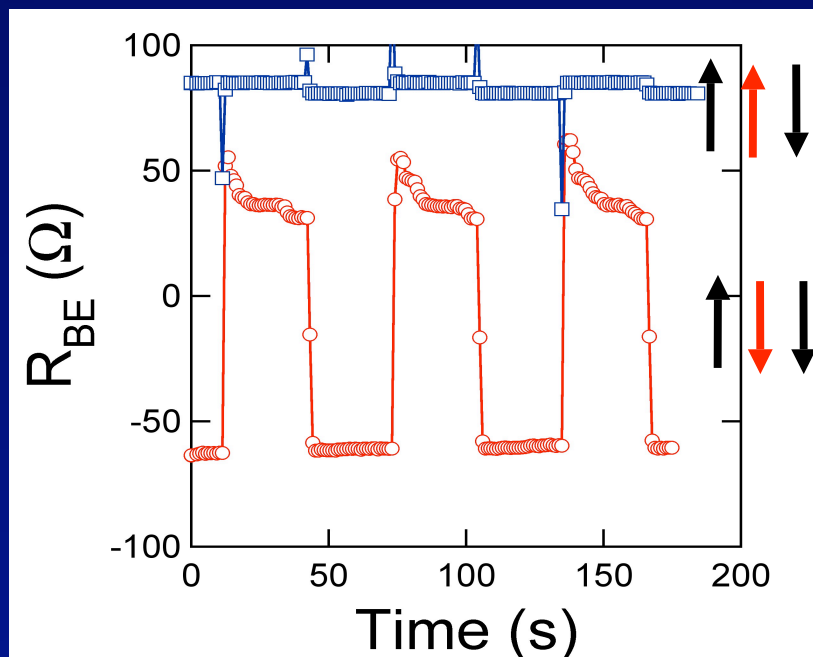


- Typical current density required to completely switch middle element (at 70 K)  $\sim 2 \times 10^4 \text{ A/cm}^2$
- Caveat: details of switching threshold very dependent on device processing, thermal history, etc.



30 nm GaMnAs  
1 mm GaInAs  
GaAs substrate

$T = 65 \text{ K}; I_p = 150 \mu\text{A}$



- Often find different threshold current density depending on initial state
- Caveat: details of switching threshold very dependent on device processing, thermal history, etc.



### III. Exchange biasing of (Ga,Mn)As

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With K. F. Eid, M. B. Stone, K. C. Ku, O. Maksimov

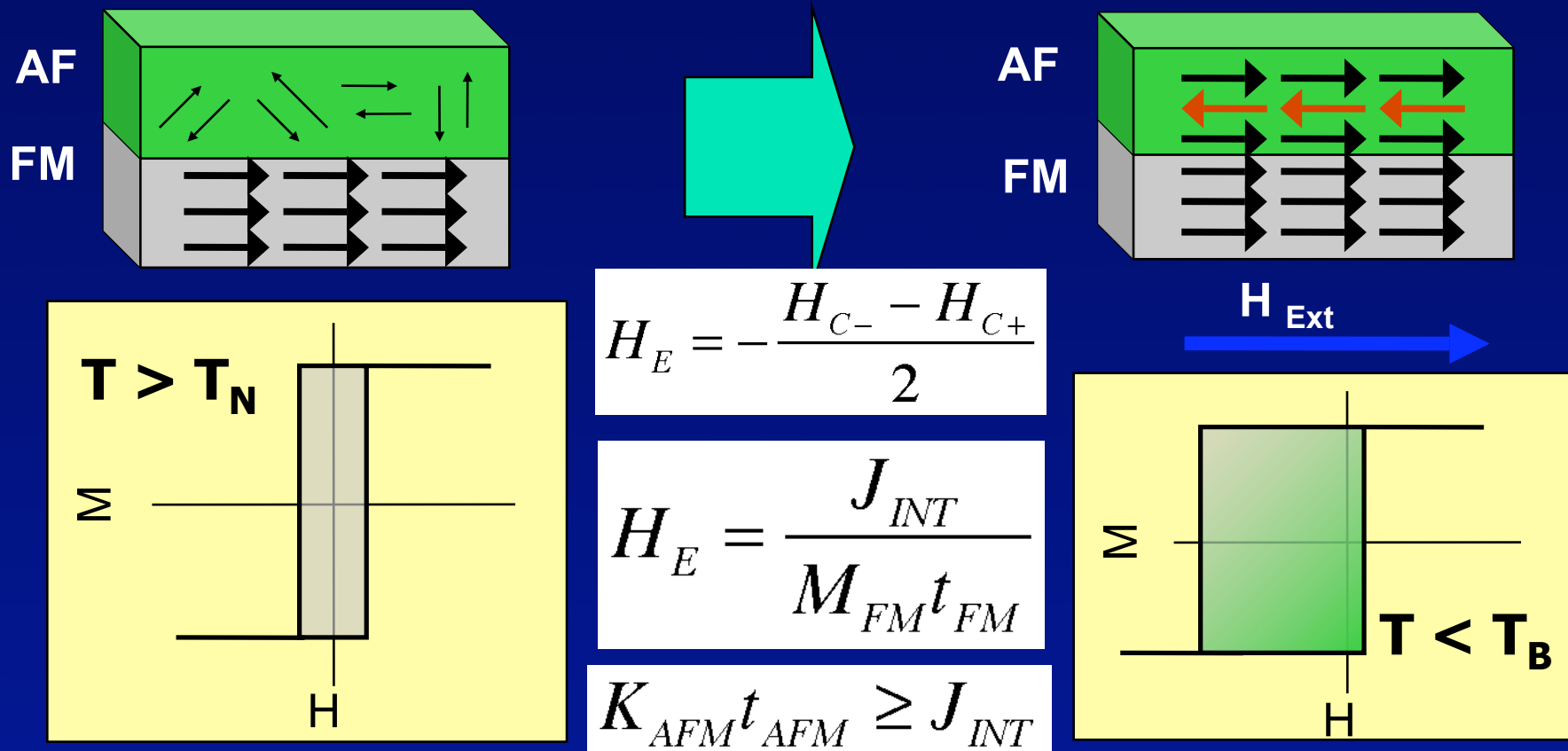
T. Shih, C. Palmstrom & P. Schiffer

Appl. Phys. Lett. **85**, 1556 [2004]

J. Appl. Phys. **97**, 10D304 [2005]

# Exchange Biasing

- Meikeljohn (1956): unidirectional anisotropy when FM/AF bilayer ( $T_N < T_C$ ) is field cooled from  $T < T_C$  to  $T < T_B < T_N$
- Magnetization hysteresis loop shifts in direction opposite to cooling field & widens below "blocking temperature" ( $T_B$ )
- Critical for spintronic devices e.g. spin valves & magnetic tunnel junctions



# Exchange biasing of (Ga,Mn)As by MnO

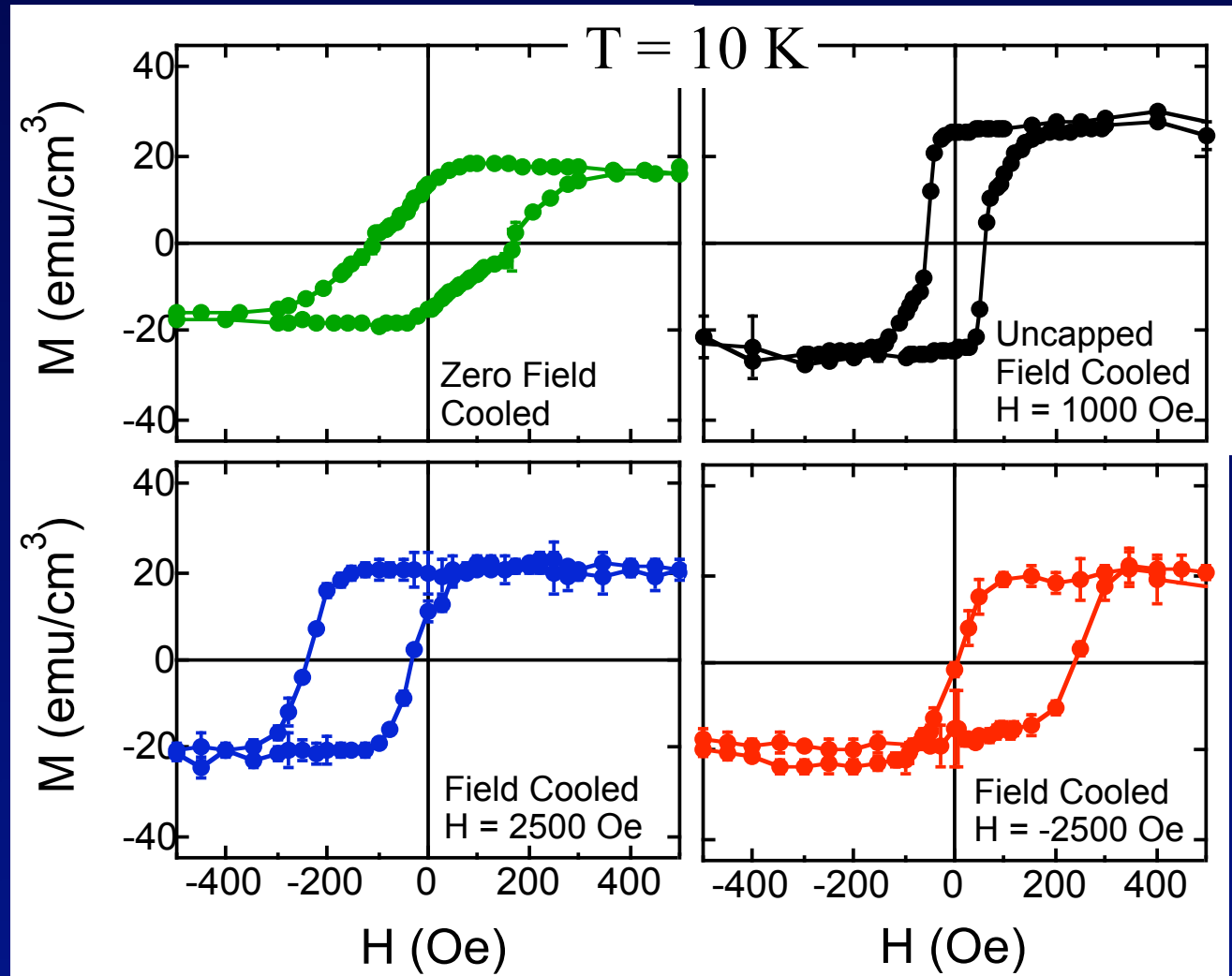
4 nm MnO

10 nm GaMnAs

$$H_E = -\frac{H_{C-} - H_{C+}}{2}$$

$$\Delta E = H_E t_{FM} M_{FM}$$

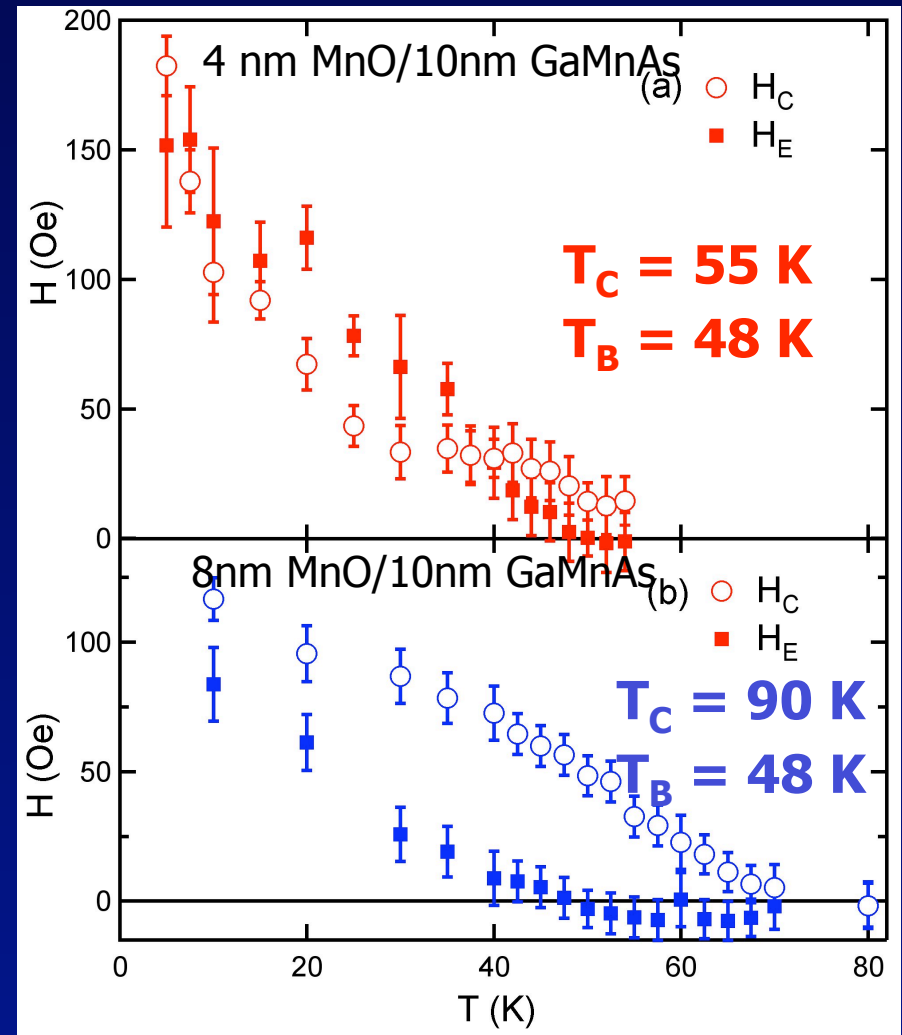
$$\approx 3 \times 10^{-3} \text{ erg/cm}^2$$



- Exchange coupling of (Ga,Mn)As with MnO: bias in field cooled hysteresis loop + increased coercivity; no effect when zero field cooled.
- Sign of bias is reversed when cooling field is reversed

# Temperature Dependence of Coercive & Exchange Fields

- **Curie temperature** ( $T_C$ )  
determined from temperature variation of remanent magnetization  $M(T)$
- **Blocking temperature** ( $T_B$ )  
determined from temperature variation of exchange bias field  $H_E(T)$
- Both cases are unconventional examples of exchange bias:  $T_C < T_N$
- Note comparison with blocking temperature:
  - Upper panel:  $T_C \sim T_B$
  - Lower panel:  $T_C \gg T_B$



## IV. Nanoengineered Curie Temperature in Laterally Patterned (Ga,Mn)As Heterostructures

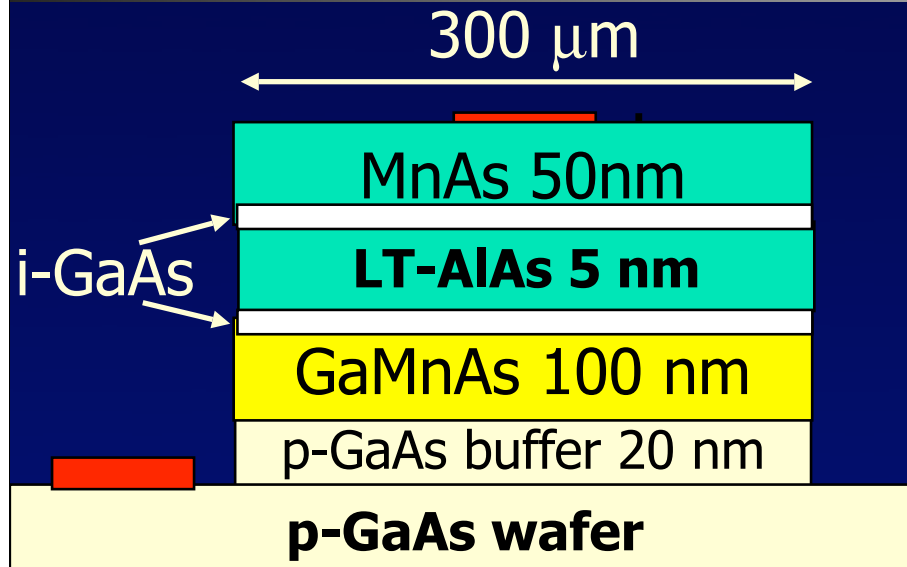
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With K. F. Eid, B. L. Sheu, O. Maksimov, M. B. Stone &  
P. Schiffer

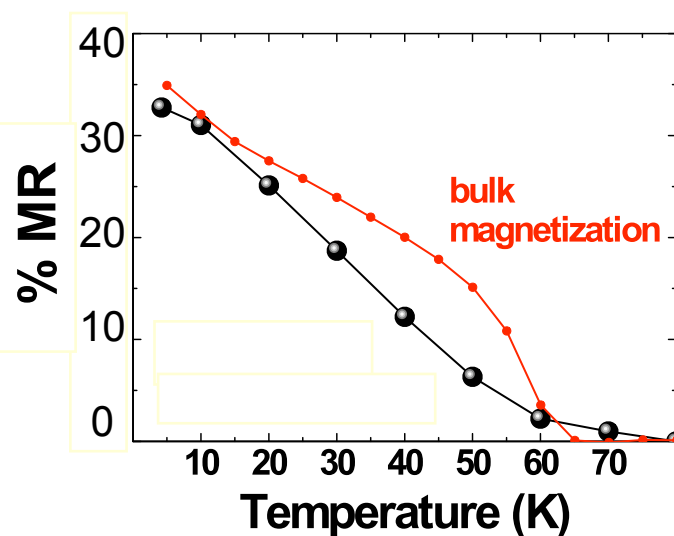
Appl. Phys. Lett. **86**, 152505 [2005]



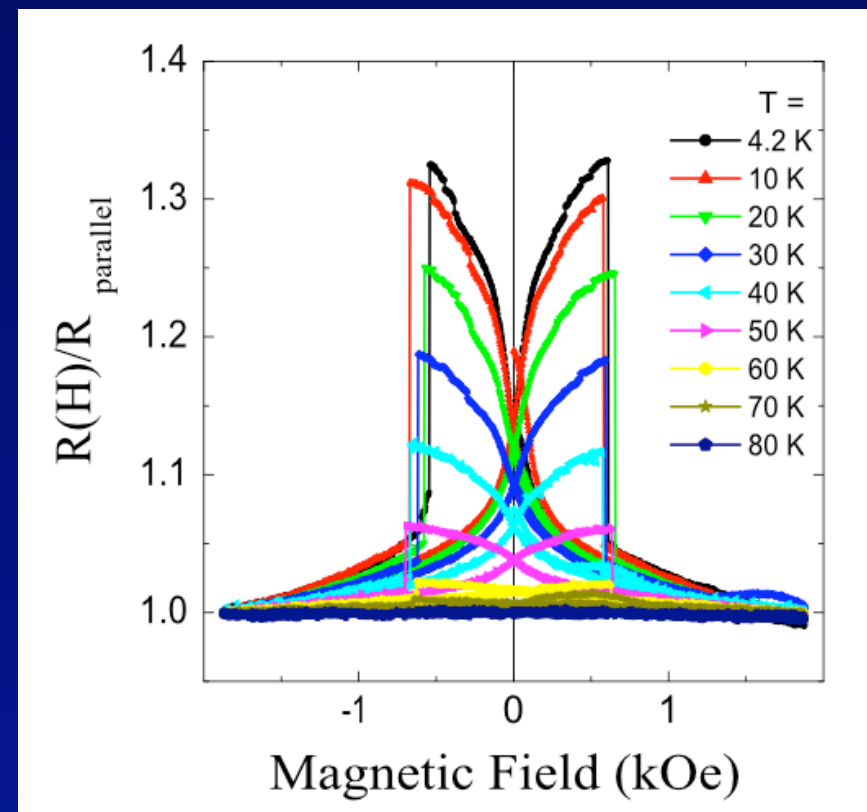
# Can annealing yield (Ga,Mn)As devices operating at $T \gg 77$ K?



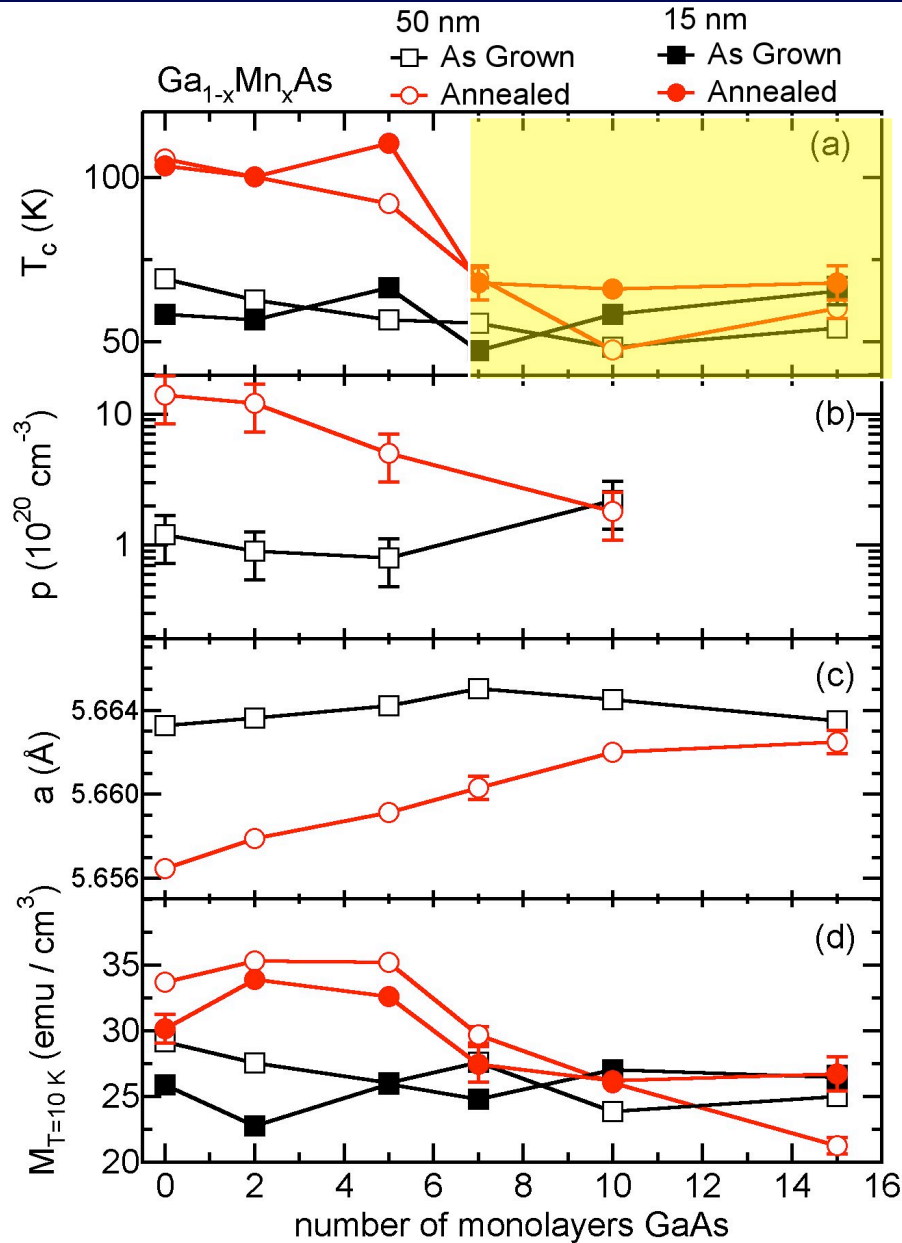
- Devices typically involve “buried” layers of (Ga,Mn)As [e.g. magnetic tunnel junctions]
- Early studies of annealing such devices showed no improvements in  $T_C$



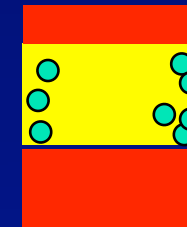
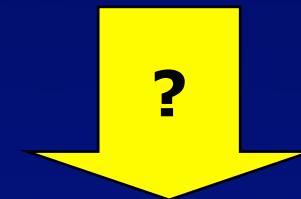
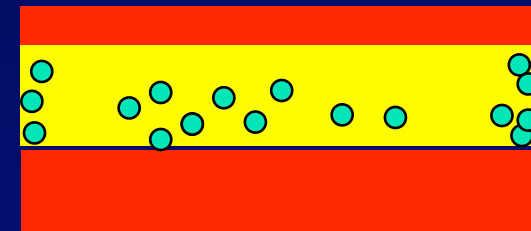
Chun et al, PRB **66**, 100408 (2002)



# Nanoengineering of Defect Diffusion Pathways

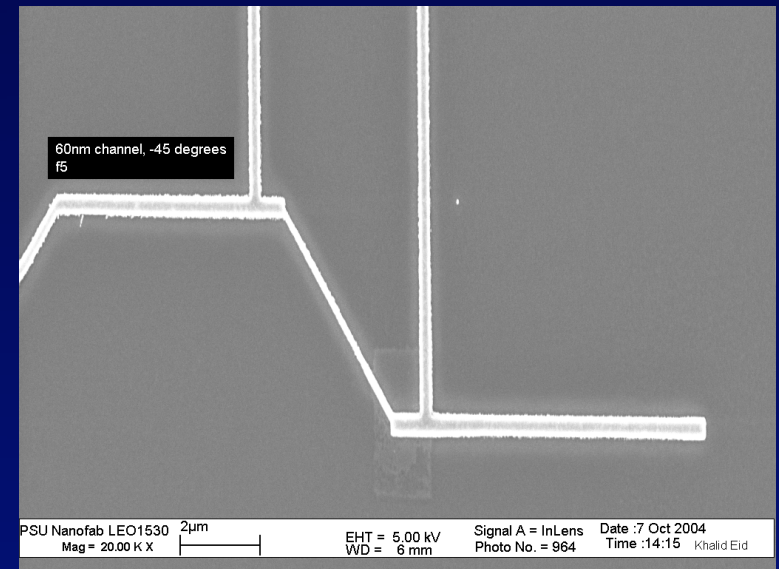
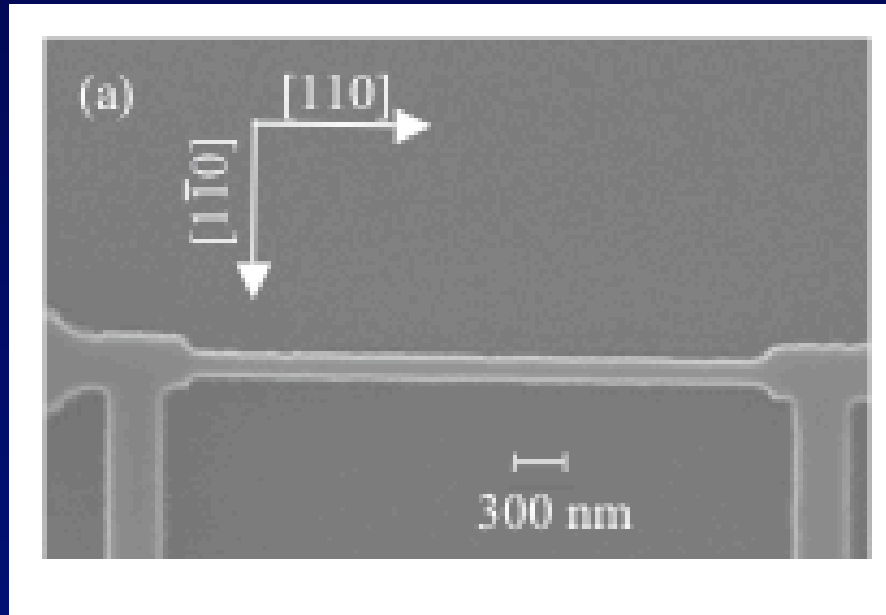


Cap layer thicknesses as small as 7 monolayers completely suppress annealing enhanced  $T_c$ ....

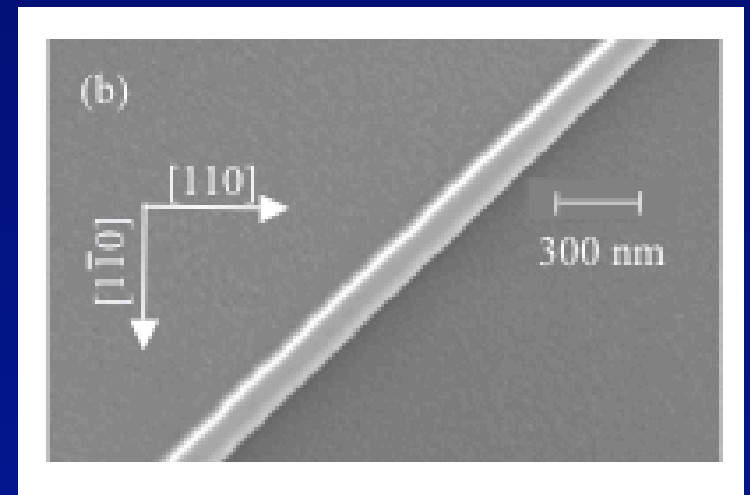


50 - 100 nm

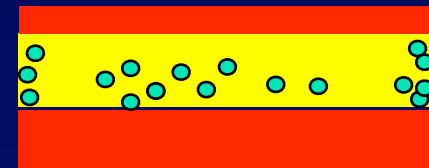
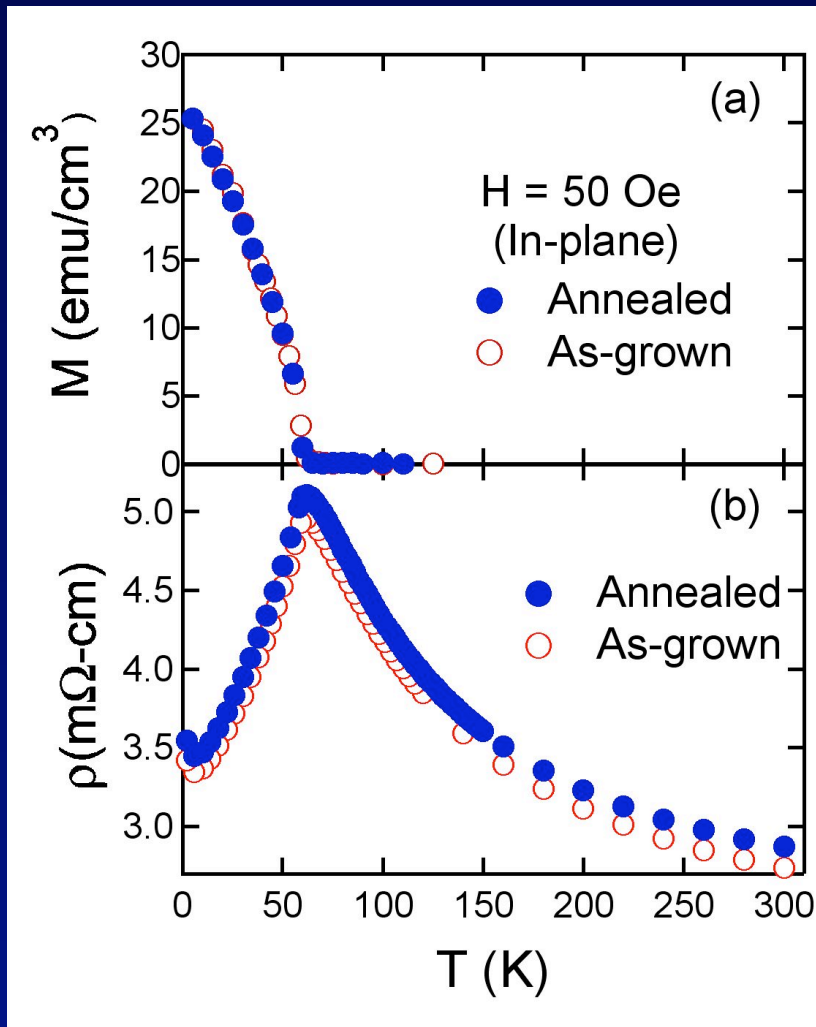
# Lithography of (Ga,Mn)As Nanowires



- Fabrication of (Ga,Mn)As nanowires using e-beam lithography & dry etching.
- Wire length 5 - 10 μm, widths 70 nm - 1 μm
- Wire orientations along different principal crystalline axes: test for possible anisotropy
- Measure Curie temperature in single wire before and after annealing (at 180°C) using temperature dependent resistivity [peak in resistivity close to  $T_c$ ]



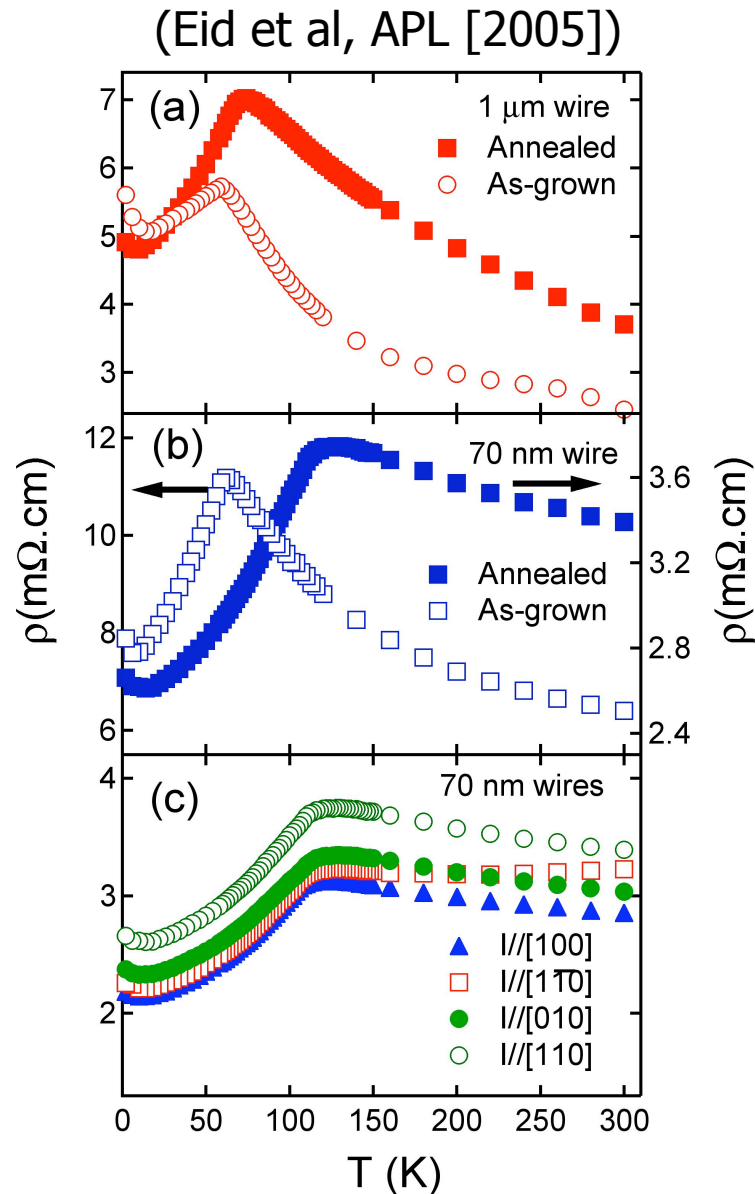
## Control measurements: unprocessed samples



Aside: origin of resistivity peak at  $T_C$  still subject of debate (see e.g. Timm, Raikh, Oppen PRL 94, 036602 [2005])

- Peak in  $\rho(T)$  closely correlated with  $T_C$  as measured by  $M(T)$ .
- Unprocessed sample:  $T_C \sim 60$  K both before and after annealing as expected from earlier studies.

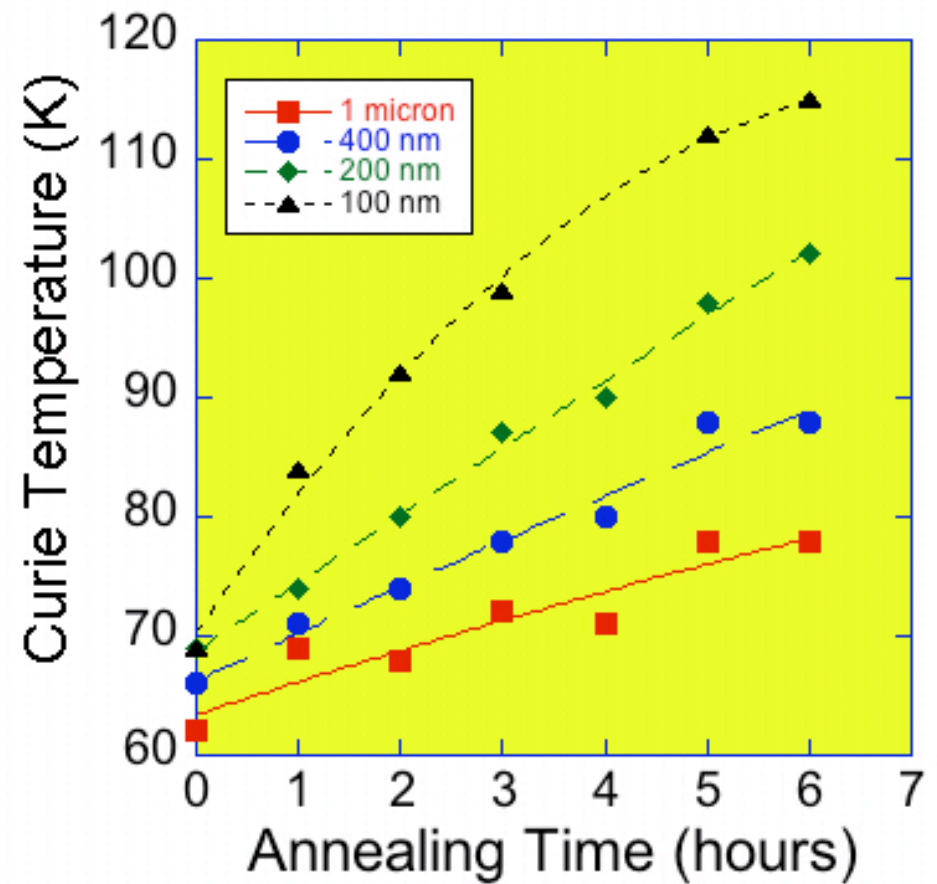
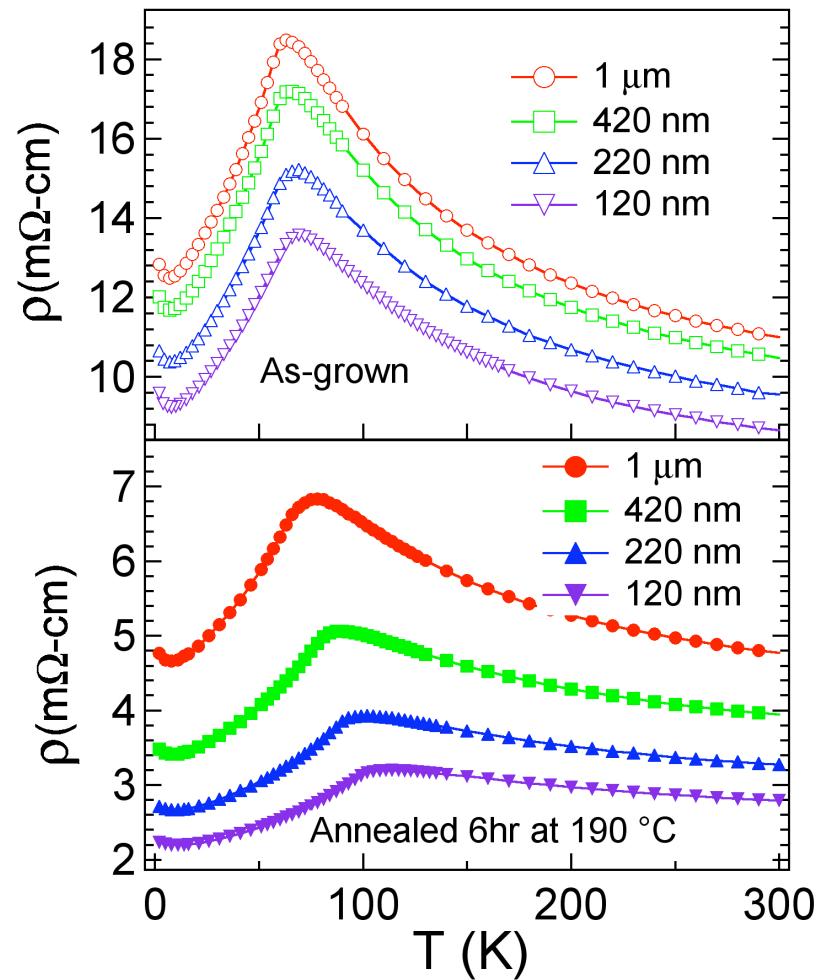
# Enhancement of $T_C$ in Nanowires



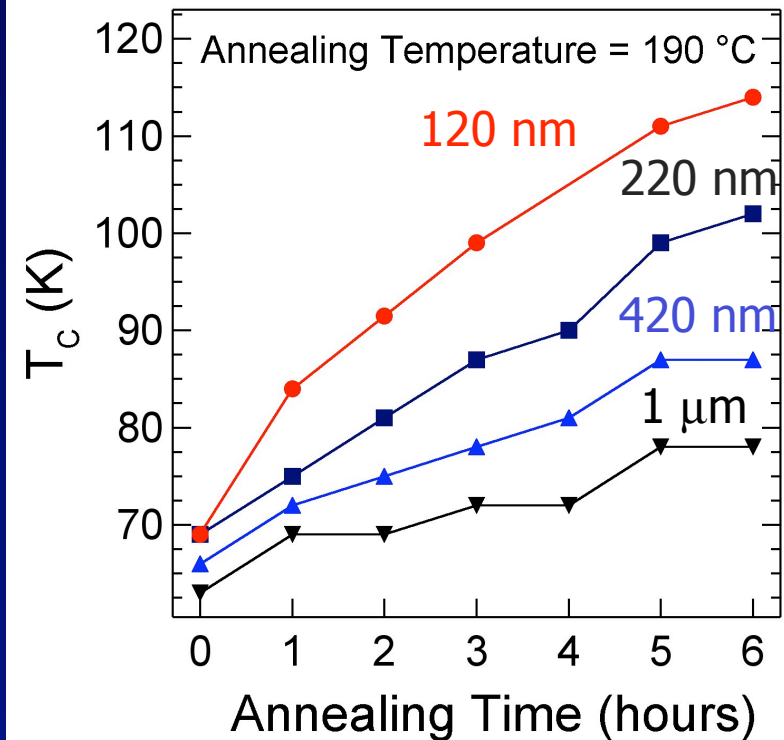
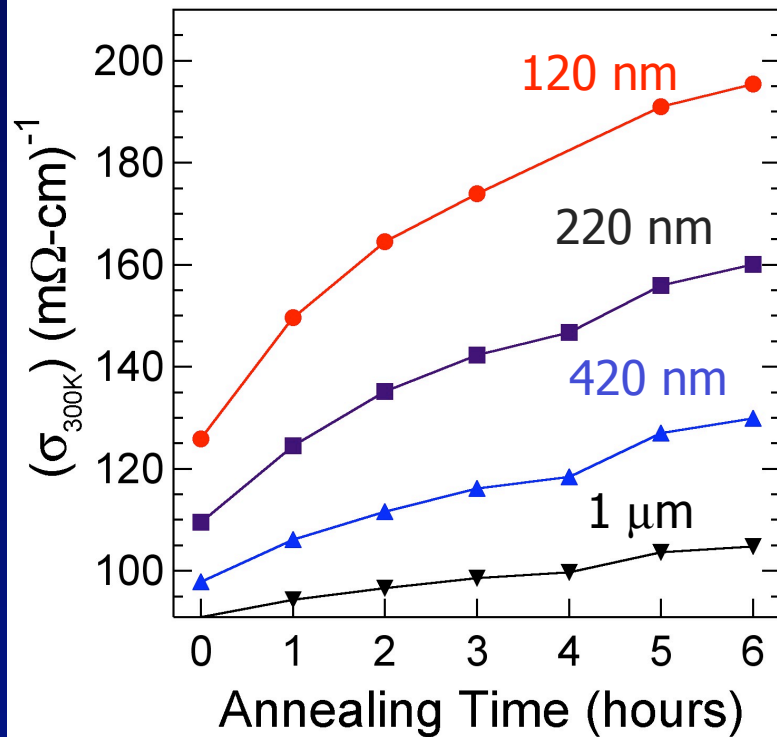
- 1  $\mu\text{m}$  width wire shows very slight increase ( $\Delta T_C \sim 5 - 10$  K)
- 70 nm wires show large increase ( $\Delta T_C \sim 40 - 60$  K)
- No observable dependence of defect diffusion on crystalline direction -- wires patterned along different crystalline axes show similar enhancements of  $T_C$ .

# Dependence of $T_C$ on annealing time and on wire width

(B. L. Sheu et al, J. Appl. Phys. [2006])



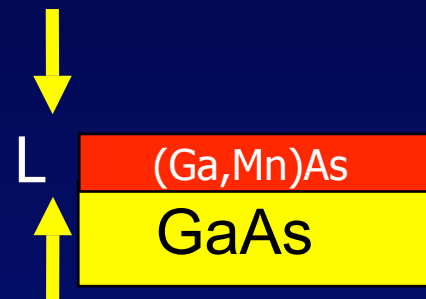
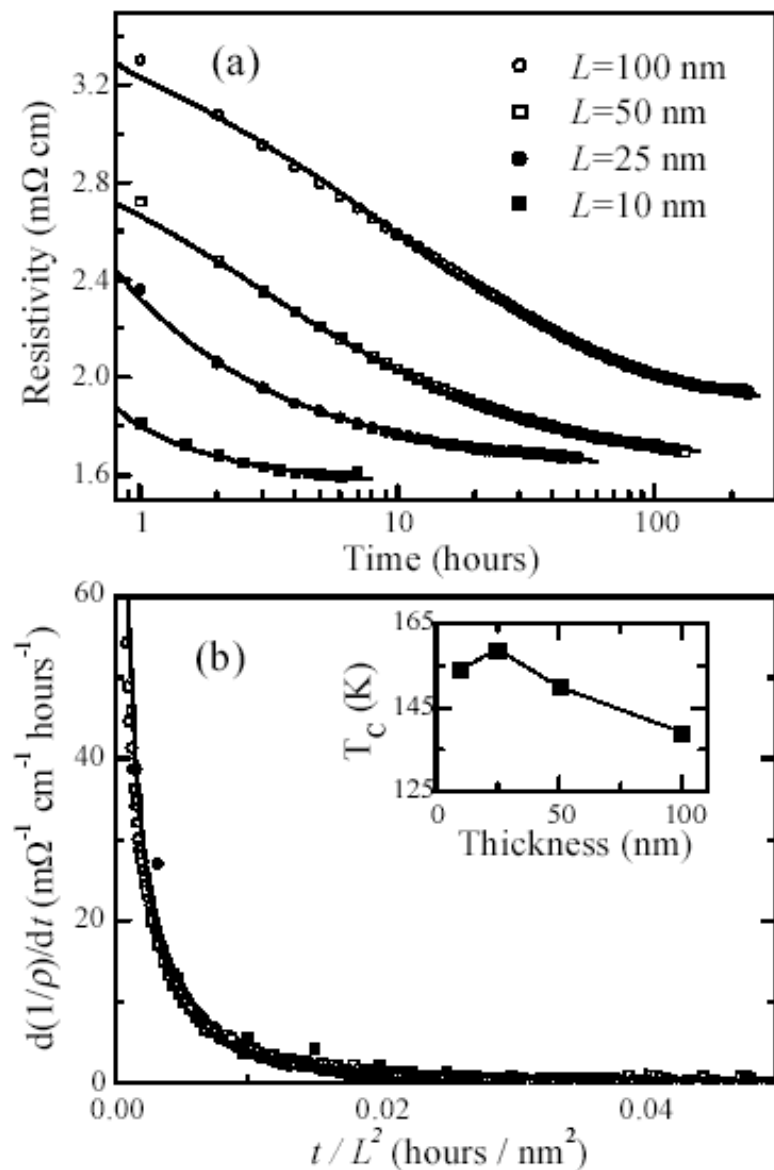
# Dependence of conductivity on annealing time and on wire width



- Variation of Curie temperature with annealing time & wire width correlates well with variation of conductivity.
- Can this be interpreted using simple diffusion models?

# 1D model for vertical outdiffusion of defects (e.g. Tuck (1974))

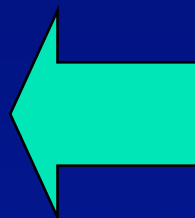
Edmonds et al, PRL (2004)



Assumption: change of conductivity entirely caused by increase in hole density

$$\sigma(t) = \sigma_0 - \sigma_1 n(t)$$

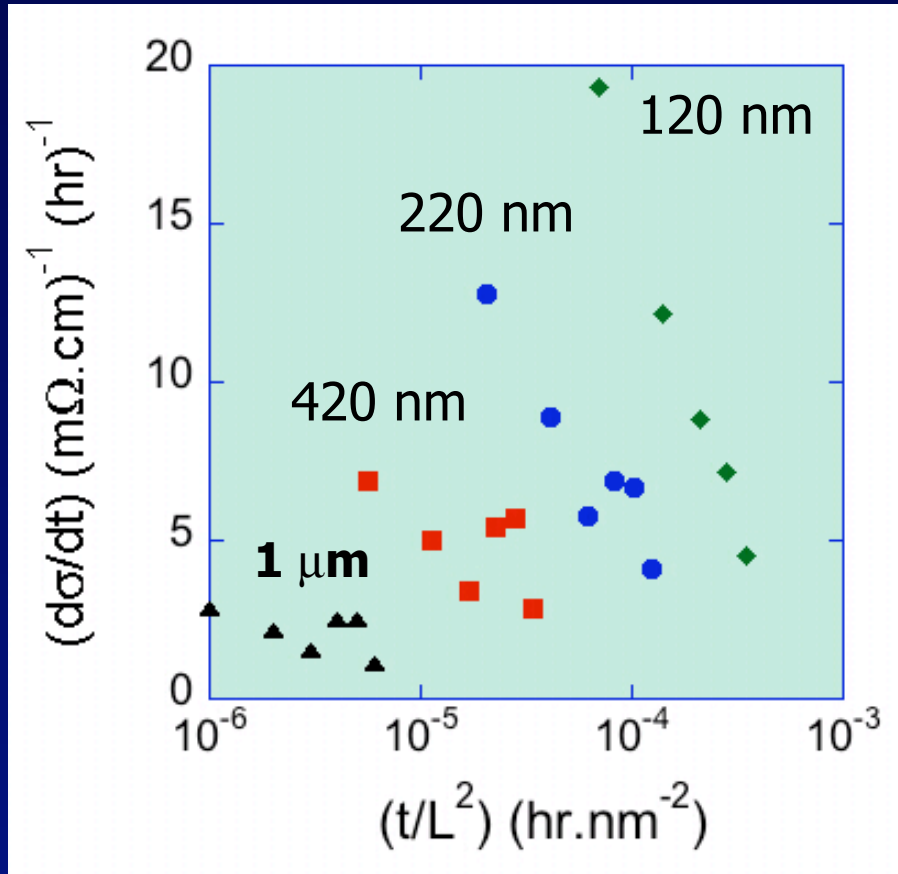
$$n(t) = \frac{N}{L} \frac{1}{\sqrt{4\pi Dt}} \iint \exp\left[-\frac{(x-x')^2}{4Dt}\right] dx dx'$$



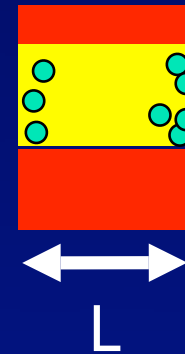
$$\frac{d\sigma}{dt} = f\left(\frac{t}{L^2}\right)$$



# Model for lateral interstitial diffusion?



$$\frac{d\sigma}{dt} = \cancel{f} \left( \frac{t}{L^2} \right) ???$$



- Data for lateral diffusion is inconsistent with naïve diffusion model
- Implies that diffusion constant decreases dramatically with wire width
- Caveat: conductivity determined by both hole density & mobility -- depletion effects + surface properties can modify mobility drastically

## Outline

- Controlling spins in semiconductor heterostructures: overview
- Spin transport & scattering in (Ga,Mn)As devices:
  - Non-collinear spin valve effect in trilayer devices
  - Pinning and controlling domain walls at constrictions & interfaces
- Interfacial control of ferromagnetism in (Ga,Mn)As
  - Exchange biasing of (Ga,Mn)As by MnO (Eid et al., APL **85**, 1556 [2004], Eid et al. J. Appl. Phys. **97**, 10D304 [2005])
  - Nanoengineered  $T_C$  in submicron (Ga,Mn)As wires (Eid et al, APL **86**, 152505 [2005])