

Do we understand (Ga,Mn)As?: prospects for high temperature ferromagnetism in (Ga,Mn)As semiconductors

Jairo Sinova
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References: Jungwirth et al Phys. Rev. B 72, 165204 (2005) and Jungwirth et al, *Theory of ferromagnetic (III,Mn)V semiconductors*, to appear in Rev. of Mod. Phys. (2006).

KITP, May 25th 2005

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OUTLINE

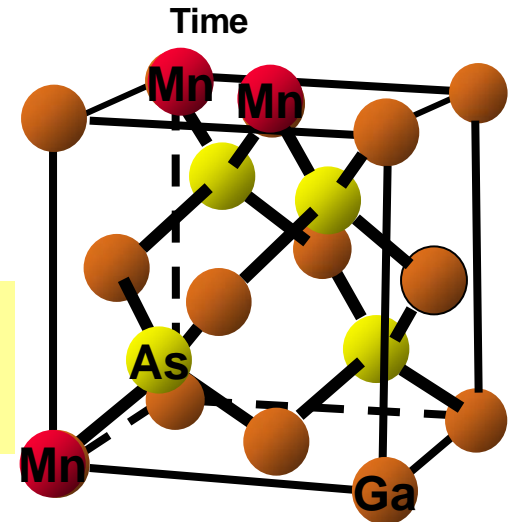
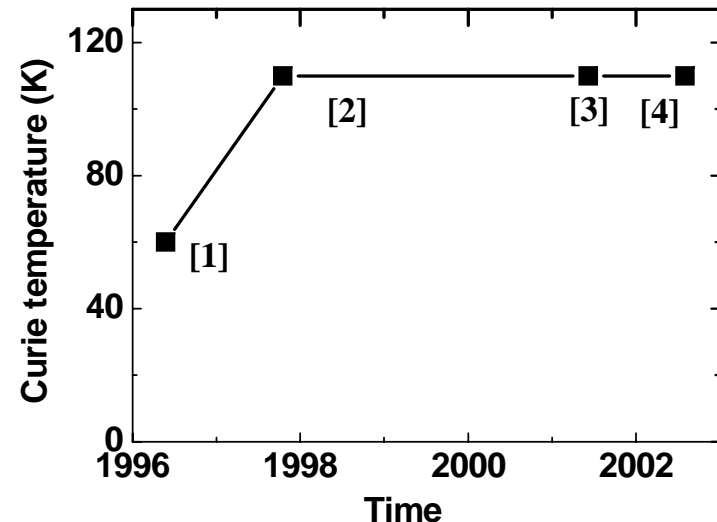
- DMS: intro to the phenomenology
 - Possible stumbling blocks to high T_c
 - Theoretical approaches to DMSs
- What is theory telling us about T_c trends
 - Is there an intrinsic limitation
 - Extrinsic limitations
- What is the data telling us: thumbs up or down?
- Other successful descriptions of system properties
 - Magnetic anisotropy
 - Temperature dependence of transport in metallic samples
 - Magnetization dynamics
 - Domain wall dynamics and resistances
 - Anisotropic magnetoresistance
 - TAMR
 - Anomalous Hall effect
- Remaining challenges:
 - Red shift in IR absorption peak
 - Seemingly large effective masses

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Problems for GaMnAs (late 2002)

- Curie temperature limited to ~110K.
- Only metallic for ~3% to 6% Mn
- High degree of compensation
- Unusual magnetization (temperature dep.)
- Significant magnetization deficit

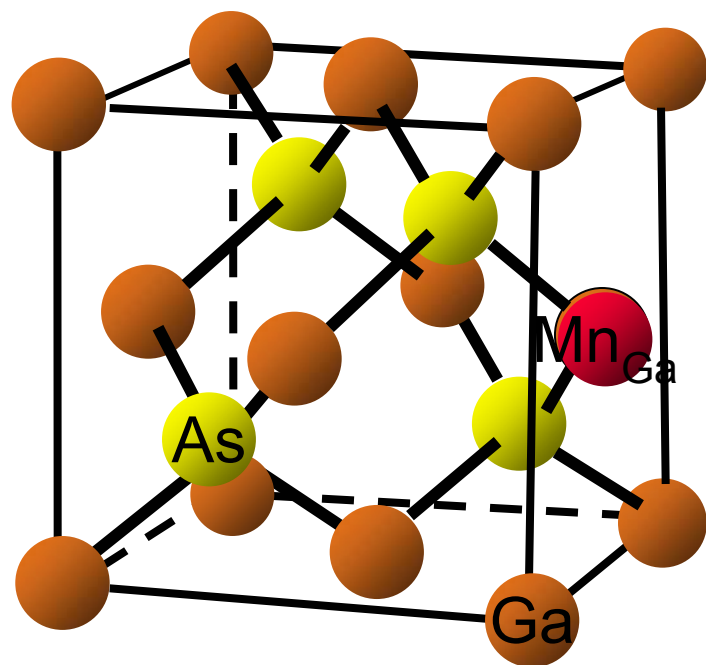
“110K could be a *fundamental limit* on T_c ”



But are these intrinsic properties of GaMnAs ??

(Ga,Mn)As diluted magnetic semiconductor

Low-T MBE - random but nearly uniform Mn distribution
up to ~ 10% doping



GALLIUM 69.72

5.91 Ga 31

[Ar] $3d^{10}4s^2$ $3p^1$

4.51 ORC 1.695
1.001

303 240

5 d-electrons with
 $L=0$, $S=5/2$

moderately shallow
acceptor (110 meV)

MANGANESE 54.938

7.43 Mn 25

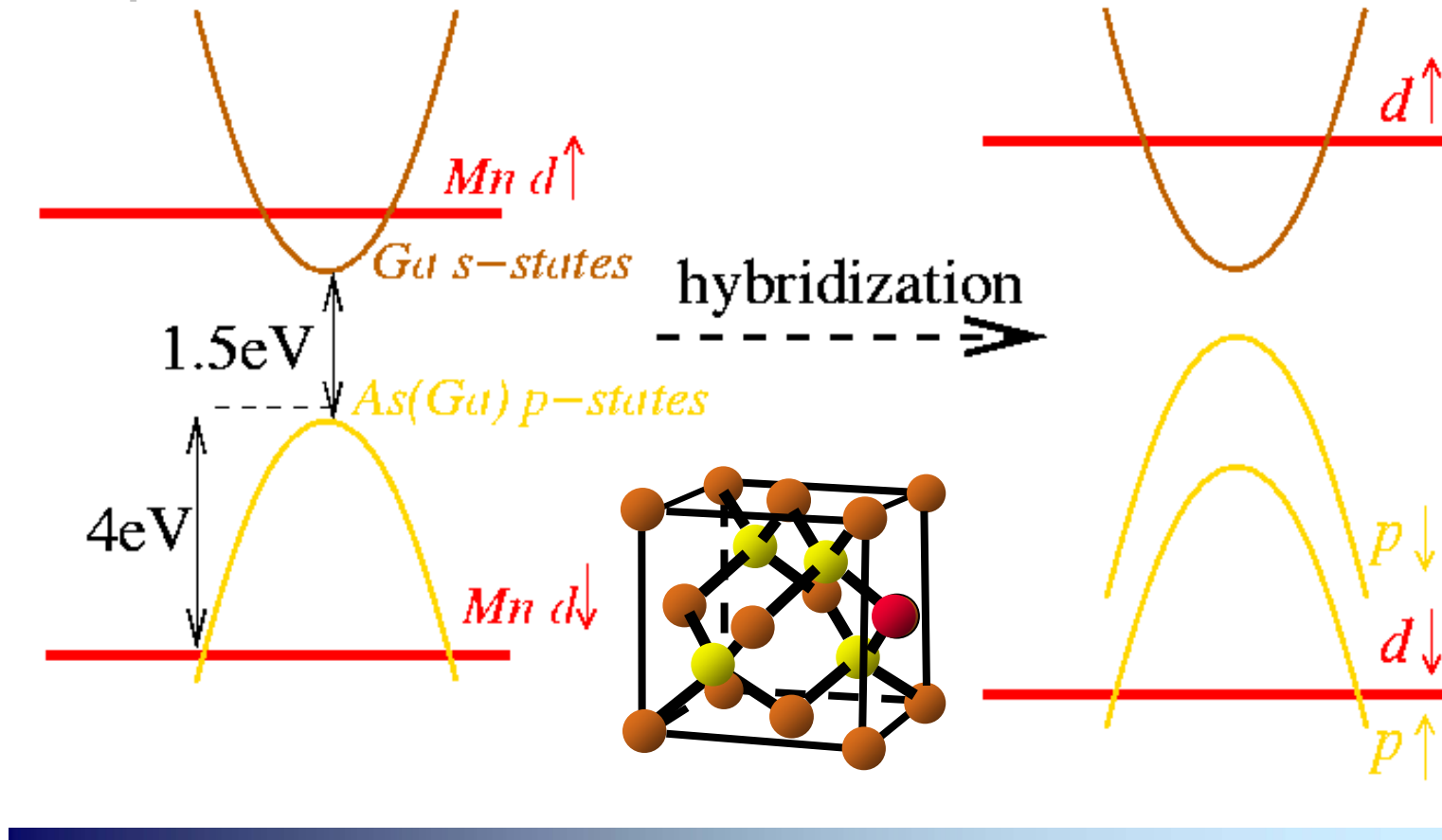
[Ar] $3d^54s^2$

8.89 CUB

1518 400

Theoretical descriptions

Microscopic: atomic orbitals & Coulomb correlation of d -electrons & hopping



Effective magnetic coupling:

$$J_{pd} \mathbf{S}_{\text{Mn}} \cdot \mathbf{s}_{\text{hole}}$$

Coulomb correlation of d -electrons & hopping \rightarrow AF kinetic-exchange coupling

Which theory is right? High noon at KITP:

Impurity bandit vs Valence Joe



KP Eastwood

Fast principles Jack



Theoretical Approaches to DMSs

- First Principles LSDA

PROS: No initial assumptions, effective Heisenberg model can be extracted, good for determining chemical trends

CONS: Size limitation, difficulty dealing with long range interactions, lack of quantitative predictability, neglects SO coupling (usually)

- Microscopic TB models

PROS: “Unbiased” microscopic approach, correct capture of band structure and hybridization, treats disorder microscopically (combined with CPA), very good agreement with LDA+U calculations

CONS: neglects coulomb interaction effects, difficult to capture non-tabulated chemical trends, hard to reach large system sizes

- $k.p \oplus$ Local Moment

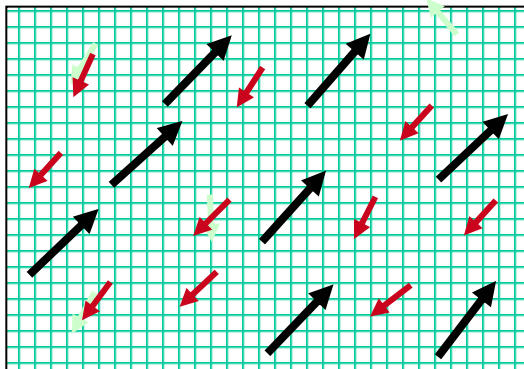
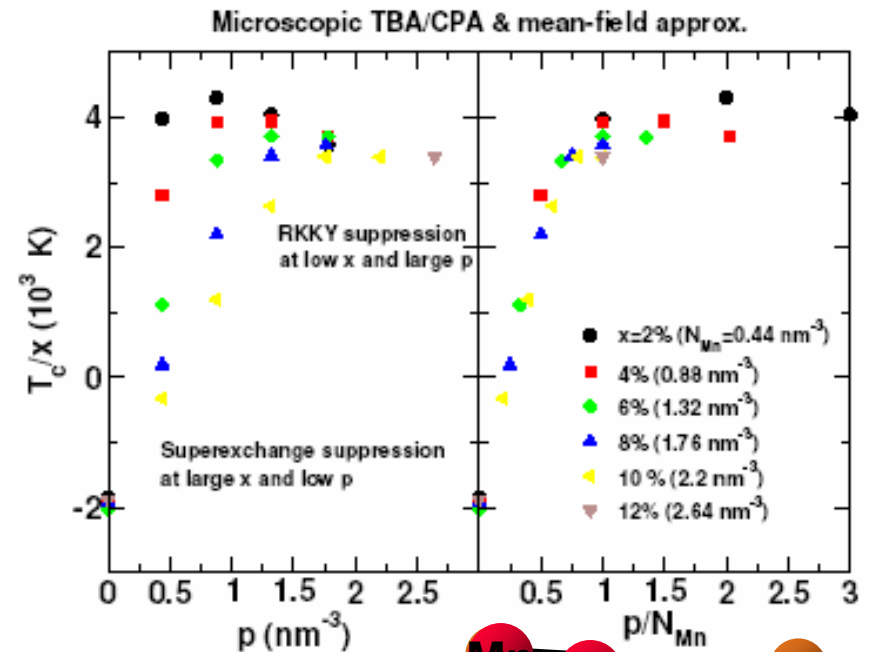
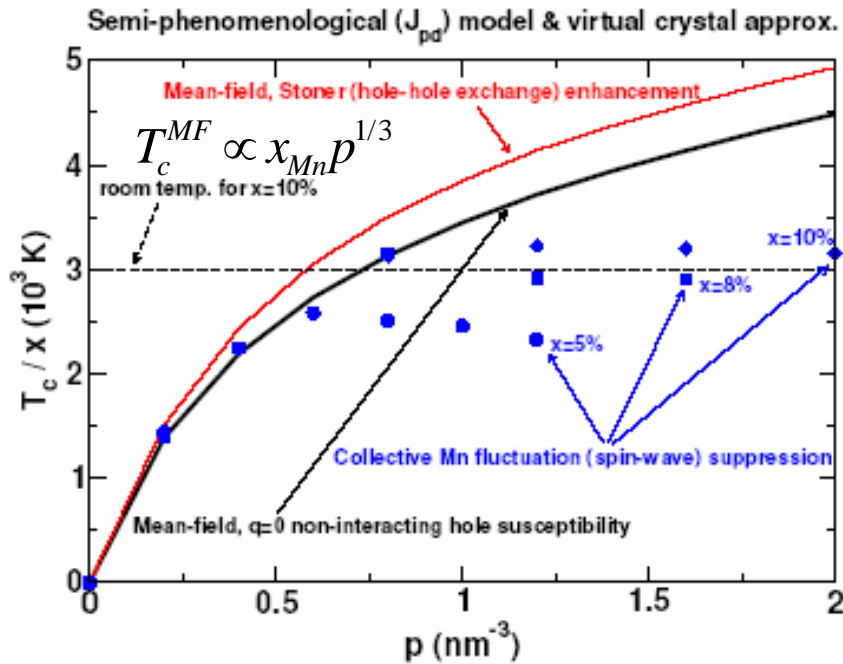
PROS: simplicity of description, lots of computational ability, SO coupling can be incorporated,

CONS: applicable only for metallic weakly hybridized systems (e.g. optimally doped GaMnAs), over simplicity (e.g. constant J_{pd}), no good for deep impurity levels (e.g. GaMnN)

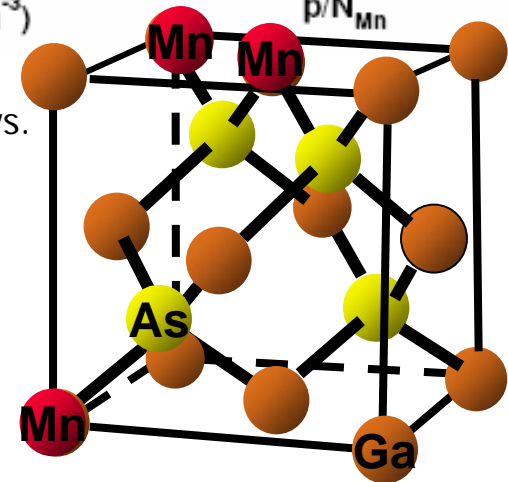
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Jungwirth, Wang, et al. Phys. Rev. B 72, 165204 (2005)



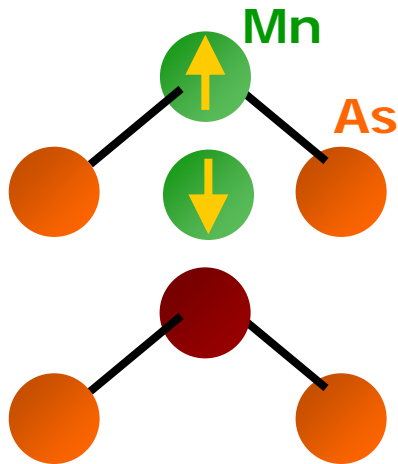
Intrinsic properties of (Ga,Mn)As: T_c linear in Mn_{Ga} local moment concentration; falls rapidly with decreasing hole density in more than 50% compensated samples; nearly independent of hole density for compensation < 50%.

Extrinsic effects: Interstitial Mn - a magnetism killer

Interstitial Mn is detrimental to magnetic order:

- compensating double-donor – reduces carrier density
- couples antiferromagnetically to substitutional Mn even in low compensation samples

Blinowski PRB '03, Mašek, Máca PRB '03

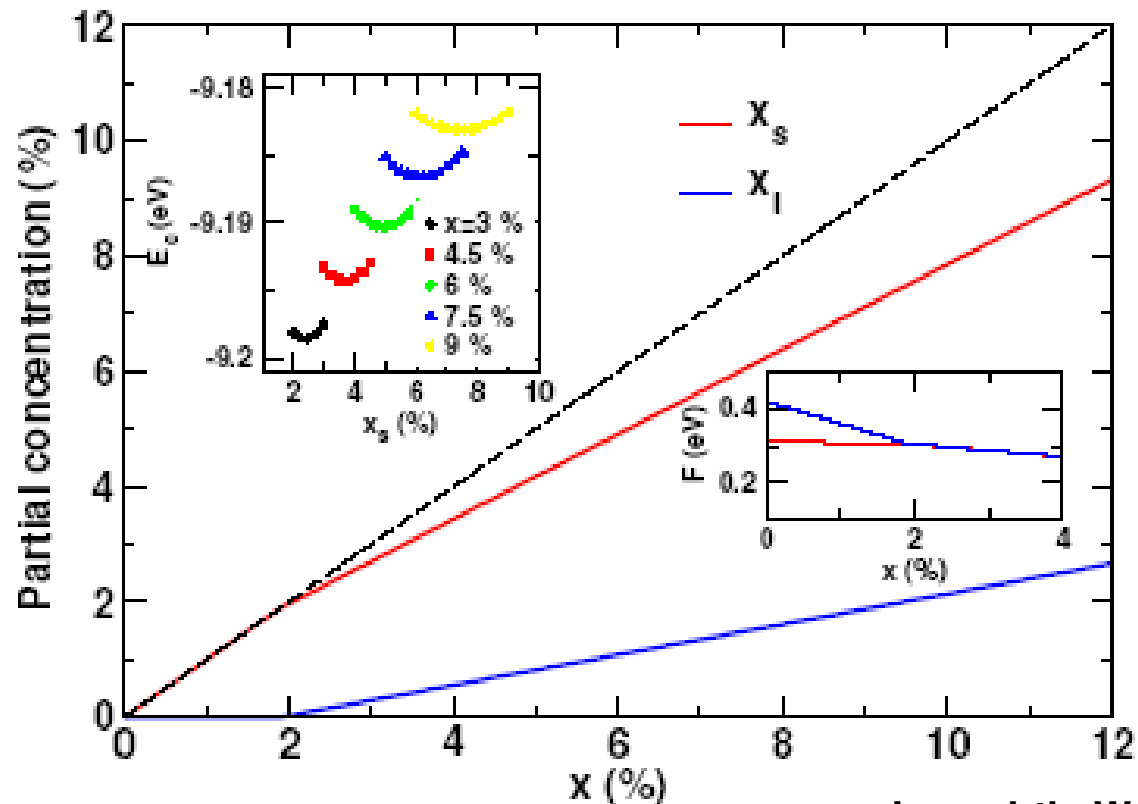


Yu et al., PRB '02:

~10-20% of total Mn concentration is incorporated as interstitials

Increased T_C on annealing corresponds to removal of these defects.

Mn_{Ga} and Mn_{I} partial concentrations



As grown
Materials
calculation

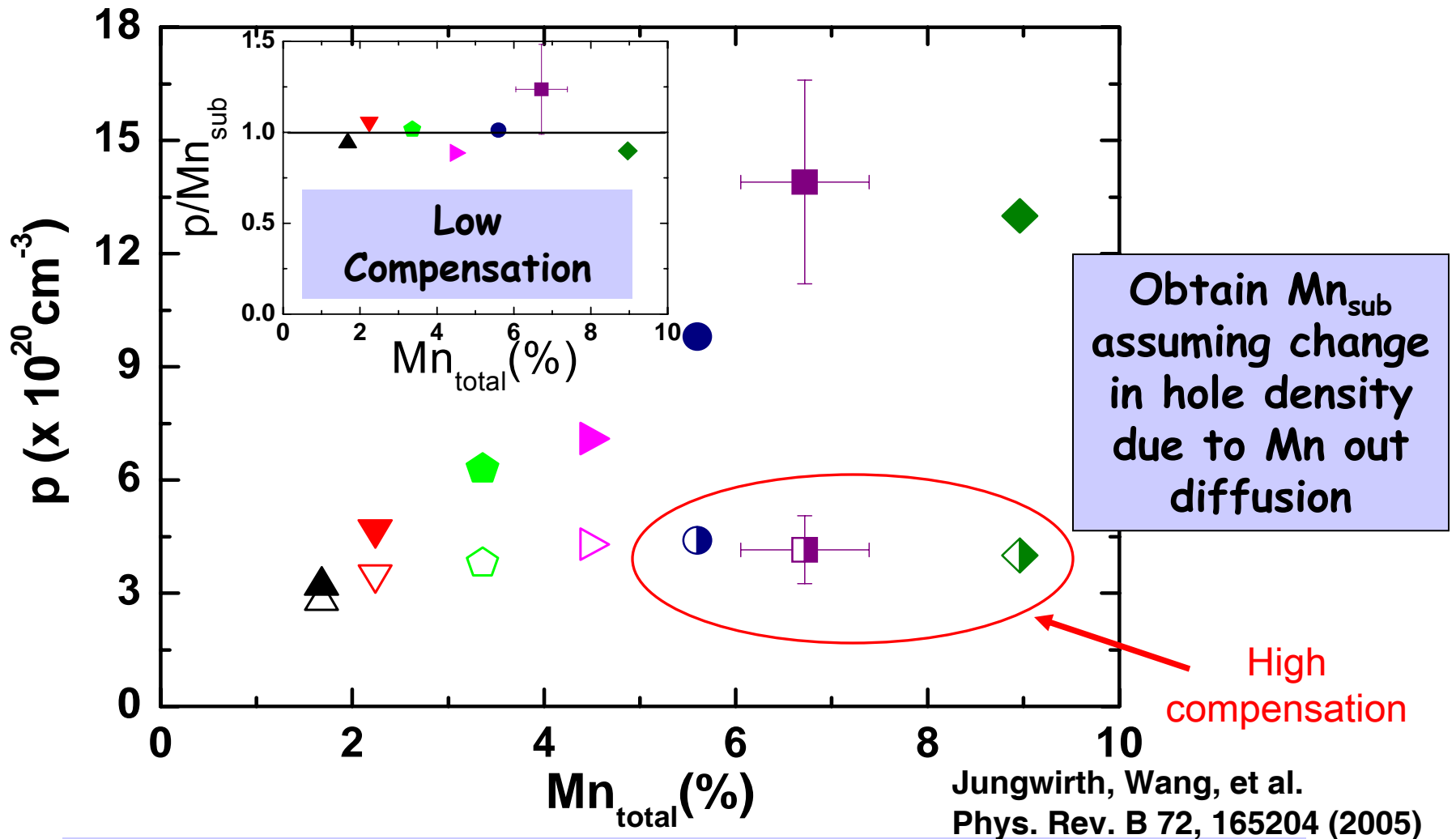
Jungwirth, Wang, et al.
Phys. Rev. B 72, 165204 (2005)

Microscopic defect formation energy calculations:

No signs of saturation in the dependence of Mn_{Ga} concentration on total Mn doping

Experimental hole densities: measured by ordinary Hall effect

Open symbols & half closed as grown. Closed symbols annealed

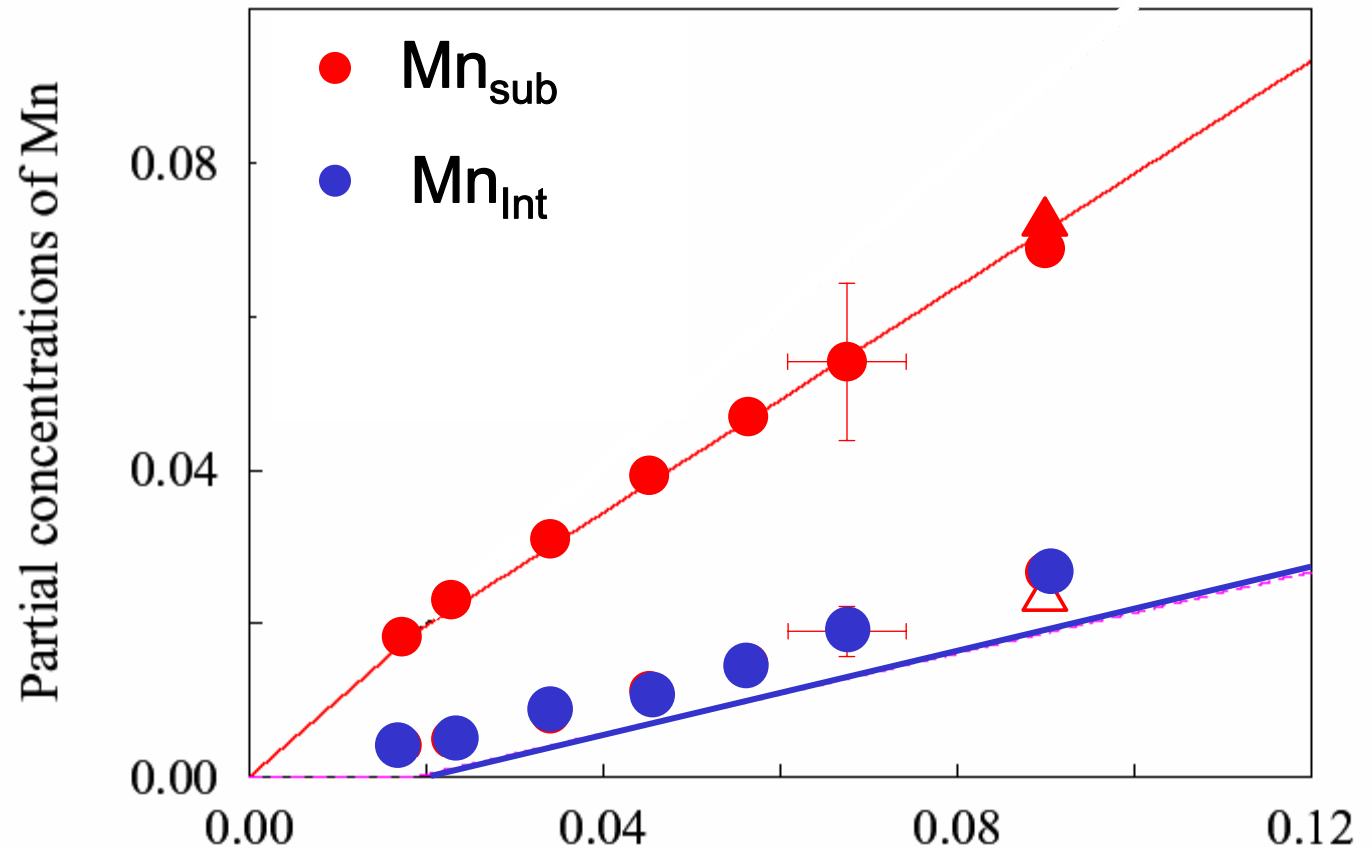


Annealing can very significantly increases hole densities.

Experimental partial concentrations of MnGa and MnI in as grown samples

Theoretical linear dependence of Mn_{sub} on total Mn confirmed experimentally

Obtain Mn_{sub} & Mn_{Int} assuming change in hole density due to Mn out diffusion



Jungwirth, Wang, et al.
Phys. Rev. B 72, 165204 (2005)

SIMS: measures total Mn concentration.
Interstitials only compensation assumed

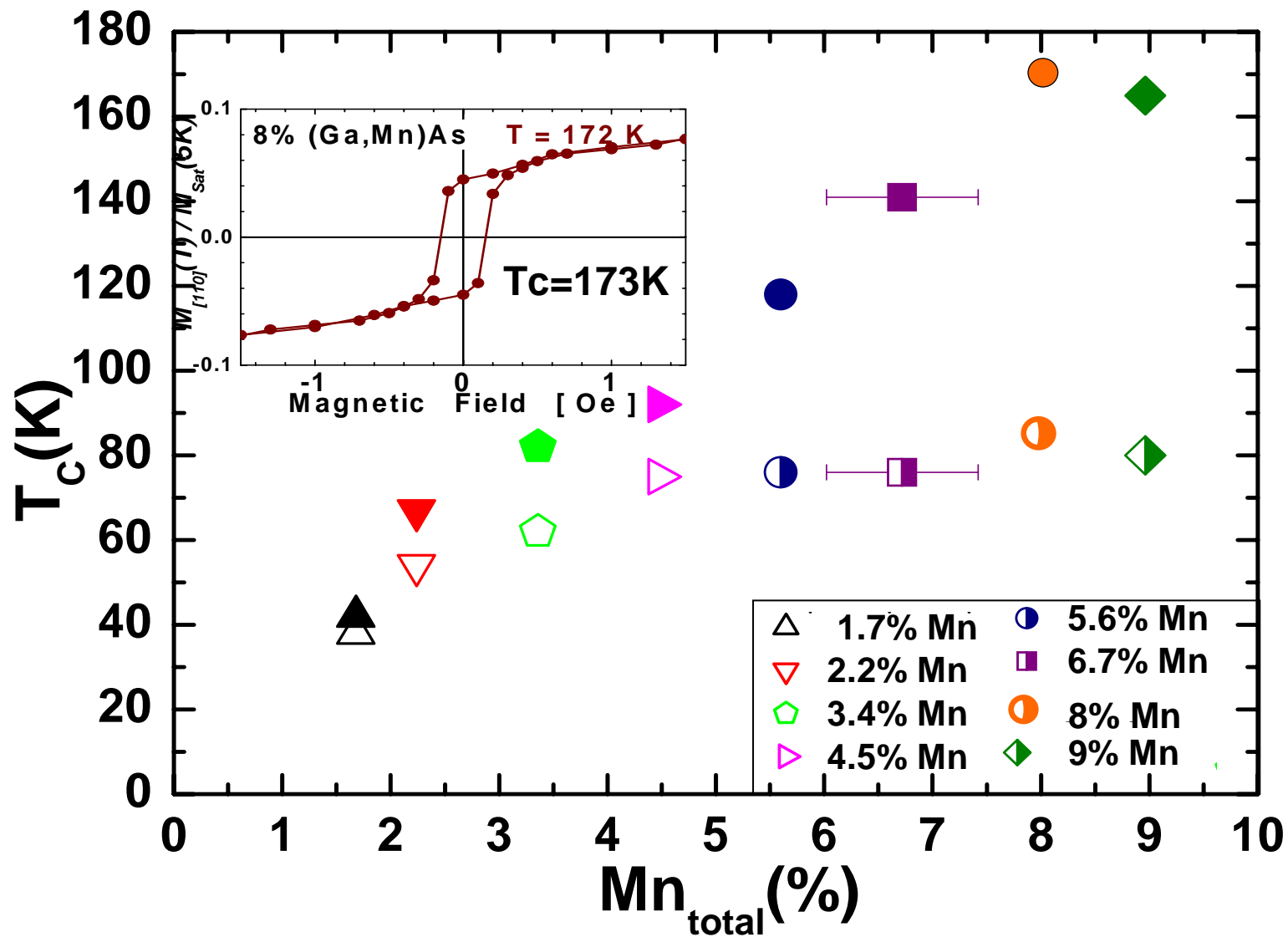
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T_c as grown and annealed samples

Open symbols as grown. Closed symbols annealed

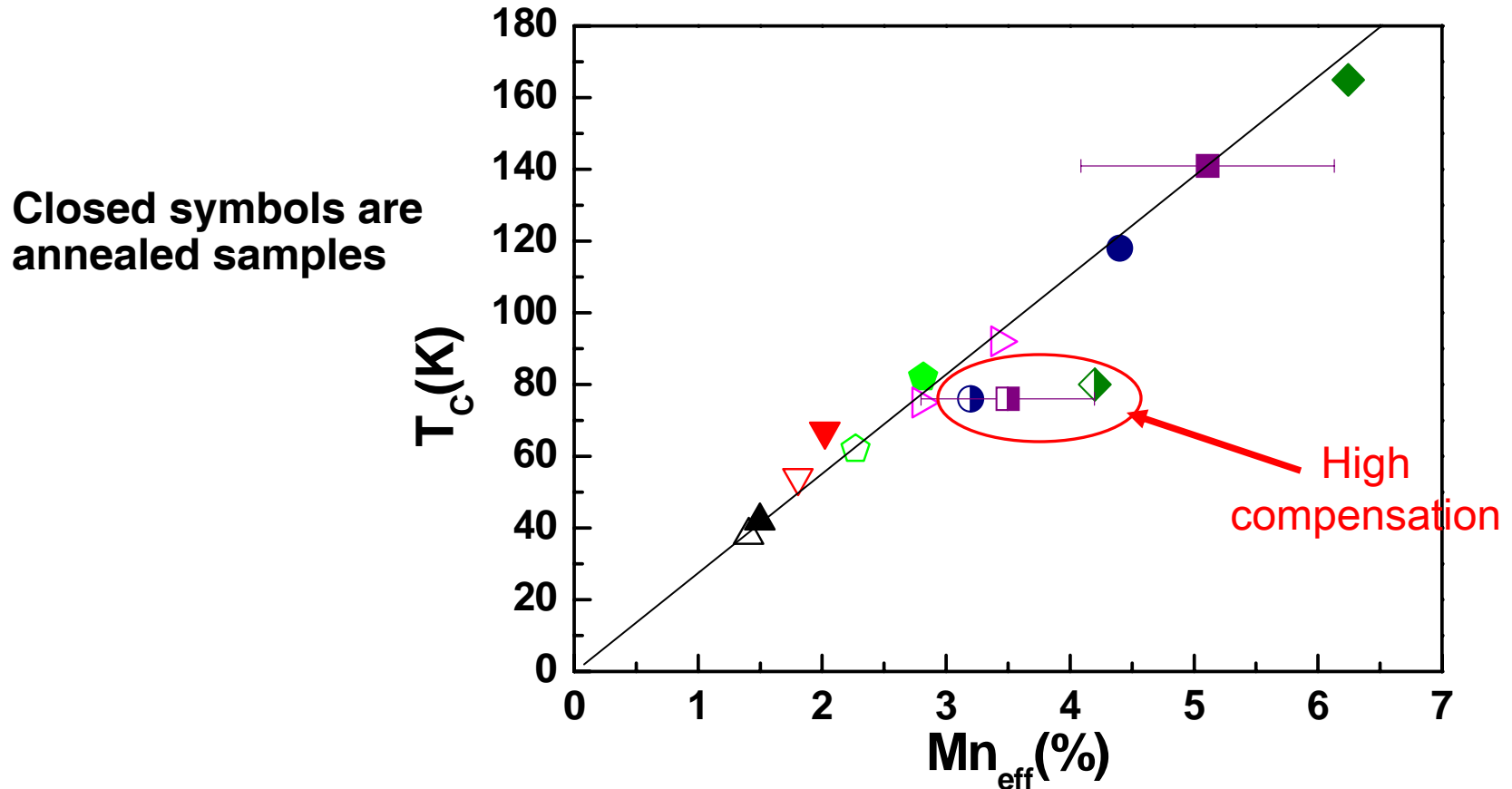


Linear increase of T_c with effective Mn

Effective Moment density, $Mn_{\text{eff}} = Mn_{\text{sub}} - Mn_{\text{Int}}$ due to AF $Mn_{\text{sub}} - Mn_{\text{Int}}$ pairs.

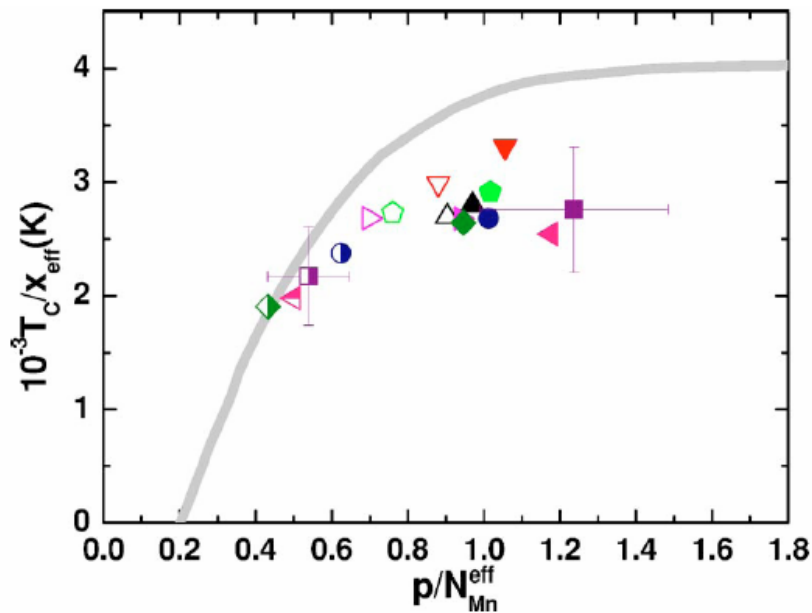
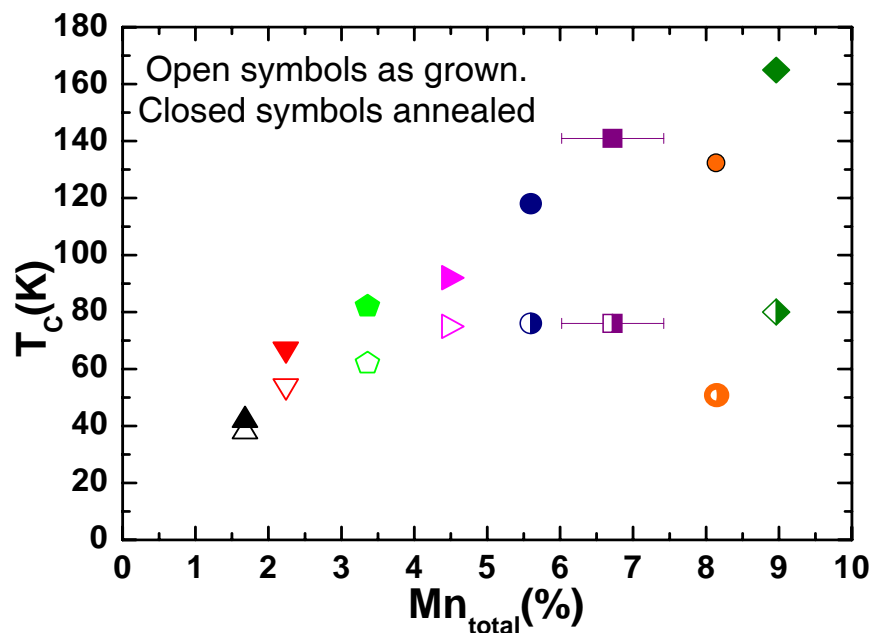
T_c increases with Mn_{eff} when compensation is less than ~40%.

No saturation of T_c at high Mn concentrations

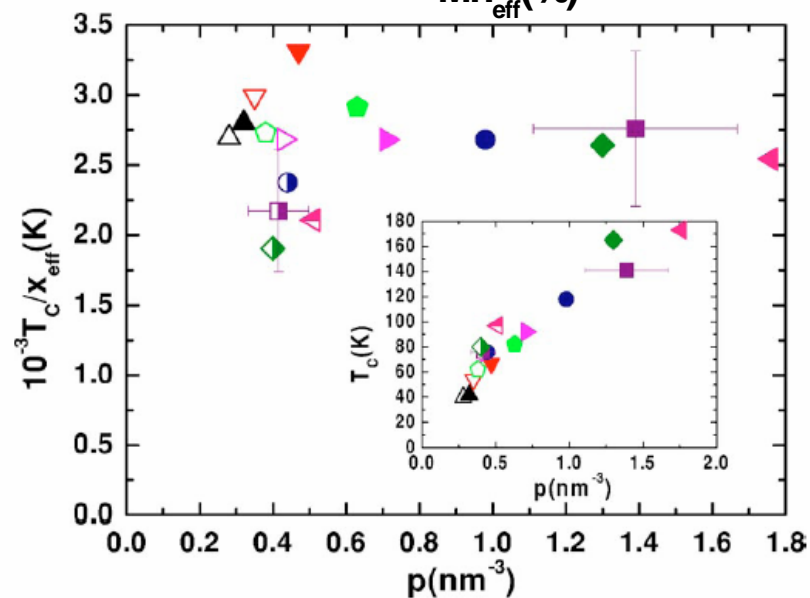
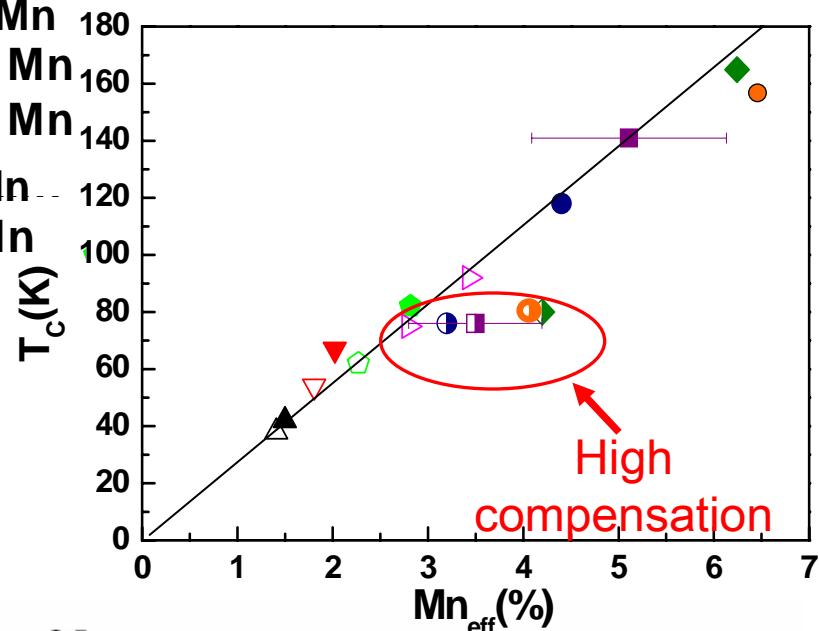


T_c as grown and annealed samples

- \triangle 1.7% Mn
- ∇ 2.2% Mn
- \triangleleft 3.4% Mn
- \triangleleft 4.5% Mn
- \bullet 5.6% Mn
- \blacksquare 6.7% Mn
- \circ 8% Mn
- \blacklozenge 9% Mn

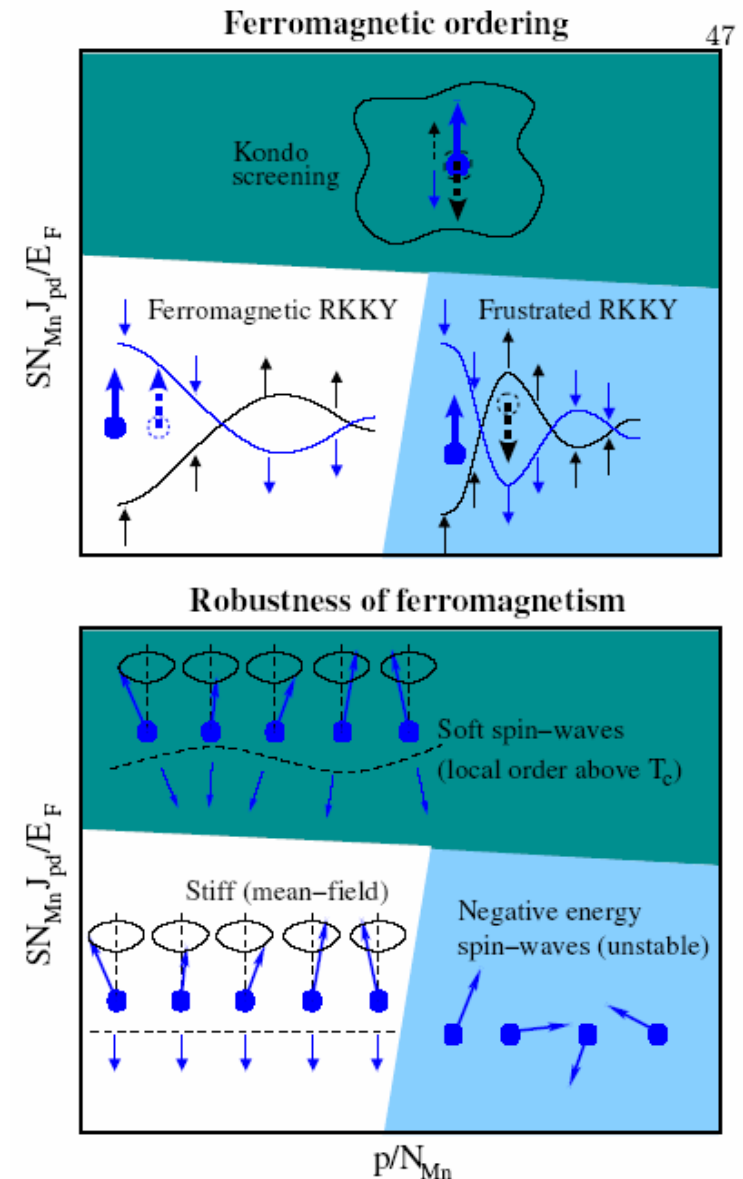


Linear increase of T_c with
 $Mn_{eff} = Mn_{sub} - Mn_{Int}$



Prospects of high T_c in DMSs

- Concentration of uncompensated Mn_{Ga} moments has to reach $\sim 10\%$. Only 6.2% in the current record $T_c = 173\text{K}$ sample
- Charge compensation not so important unless $> 40\%$
- No indication from theory or experiment that the problem is other than technological - better control of growth- T , stoichiometry
- New growth or chemical composition strategies to incorporate more MnGa local moments or enhance p-d coupling
- Window in this difficult phase space is narrow and obtaining the optimal strength of the coupling and technical difficulties for GaMnAs may make it impossible to reach room T_c
- May want to look into materials close to this material but higher coupling strength to find the optimal system

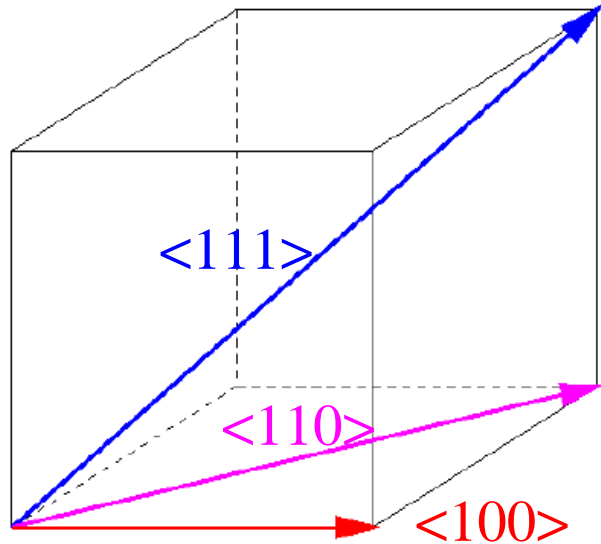


OUTLINE

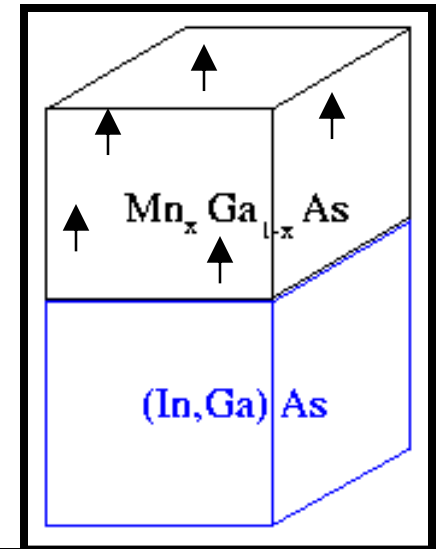
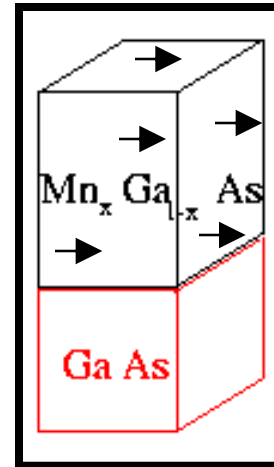
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MAGNETIC ANISOTROPY

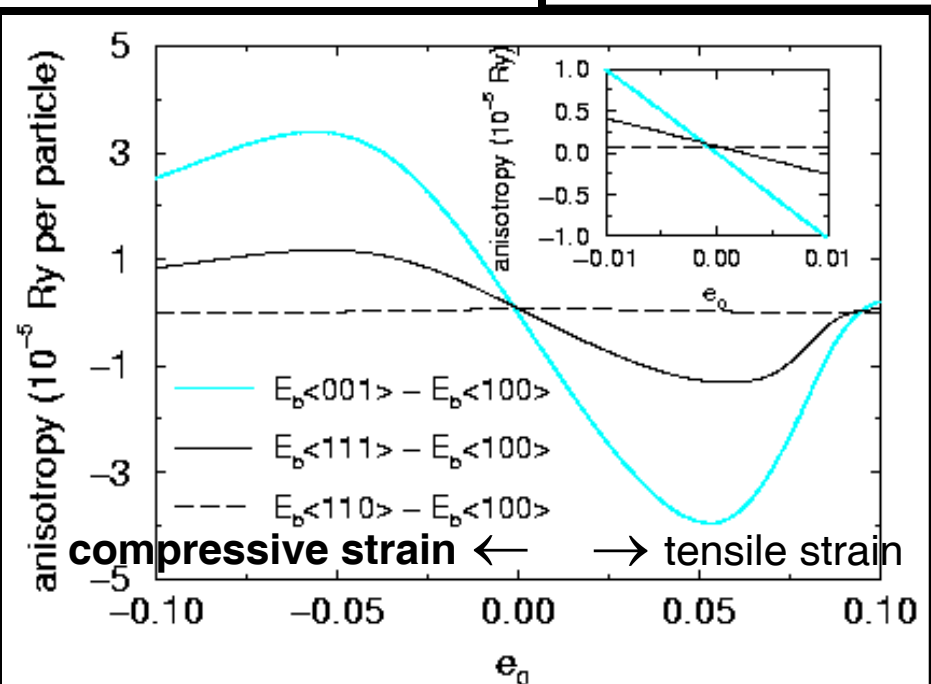
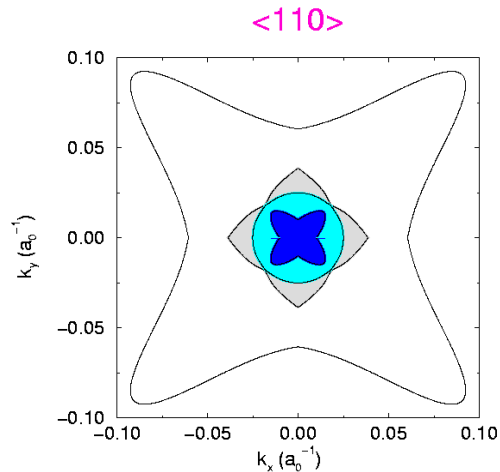


experiment:

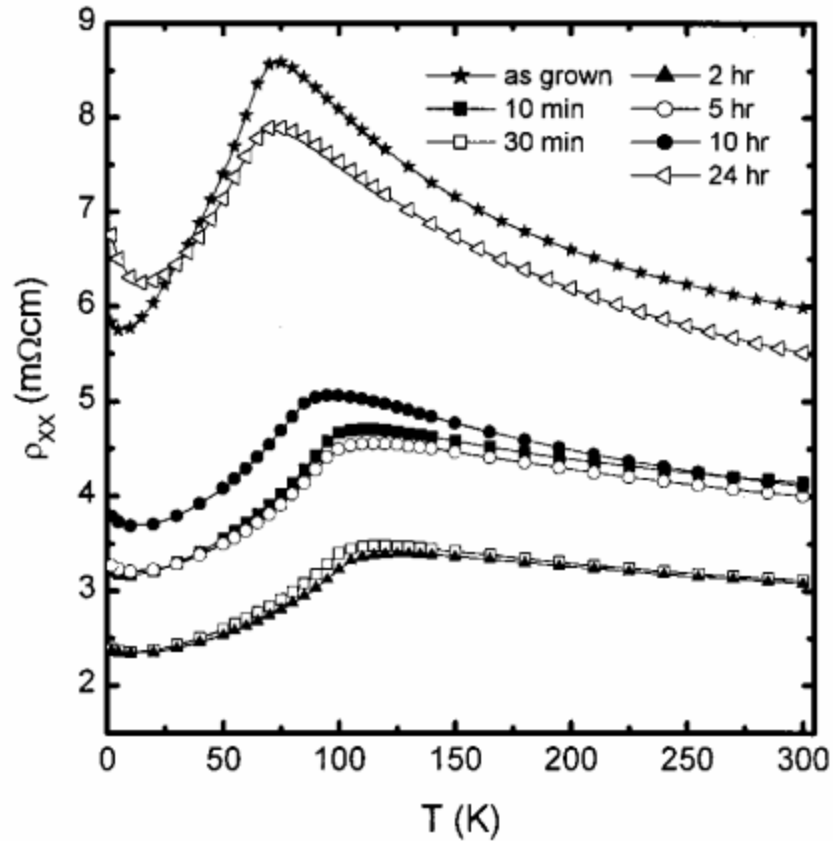


Condensation energy depends on magnetization orientation

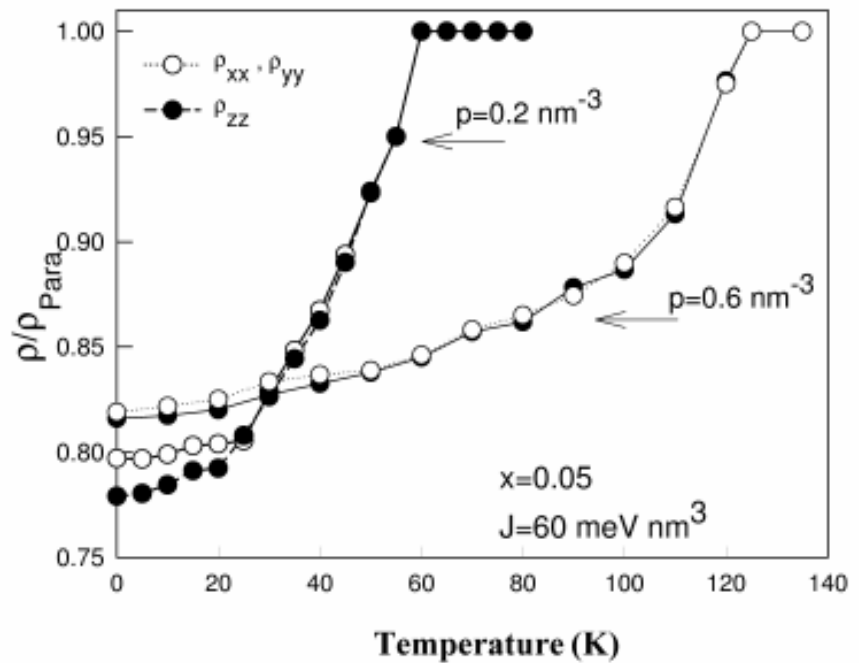
M. Abolfath, T. Jungwirth, J. Brum,
A.H. MacDonald, Phys. Rev. B 63, 035305 (2001)



Resistivity temperature dependence of metallic GaMnAs



Potashnik et al 2001

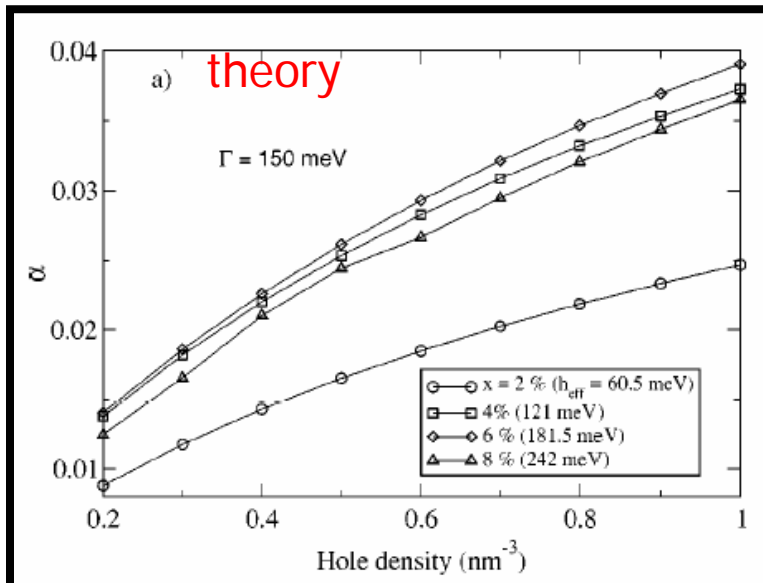


Lopez-Sanchez and Bery 2003
Hwang and Das Sarma 2005

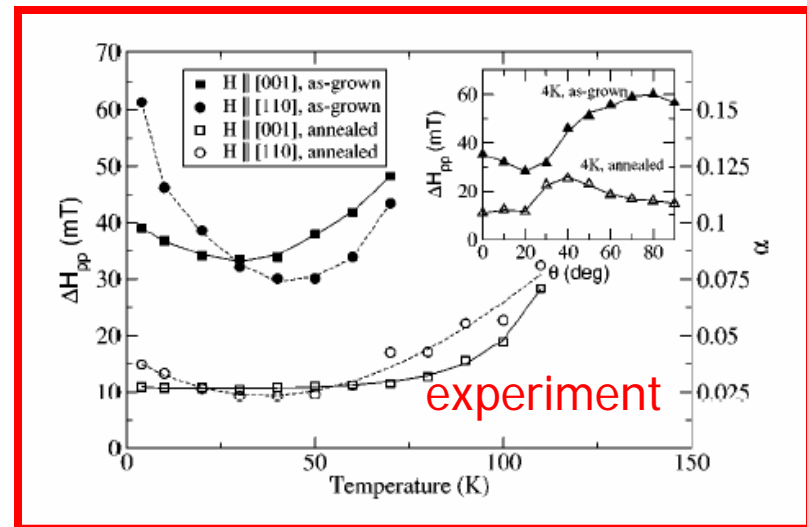
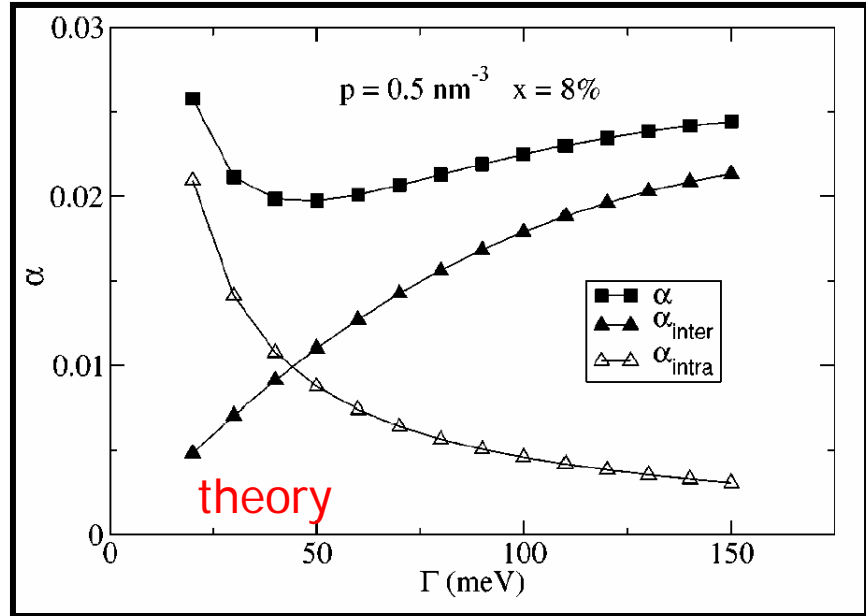
Ferromagnetic resonance: Gilbert damping

$$\alpha = \frac{J_{pd} h_{eff}}{4\hbar} \int \frac{d^3k}{(2\pi)^3} \sum_{a,b} |\langle \phi_a(\mathbf{k}) | s^+ | \phi_b(\mathbf{k}) \rangle|^2 \times A_{a,\mathbf{k}}(\epsilon_F) A_{b,\mathbf{k}}(\epsilon_F).$$

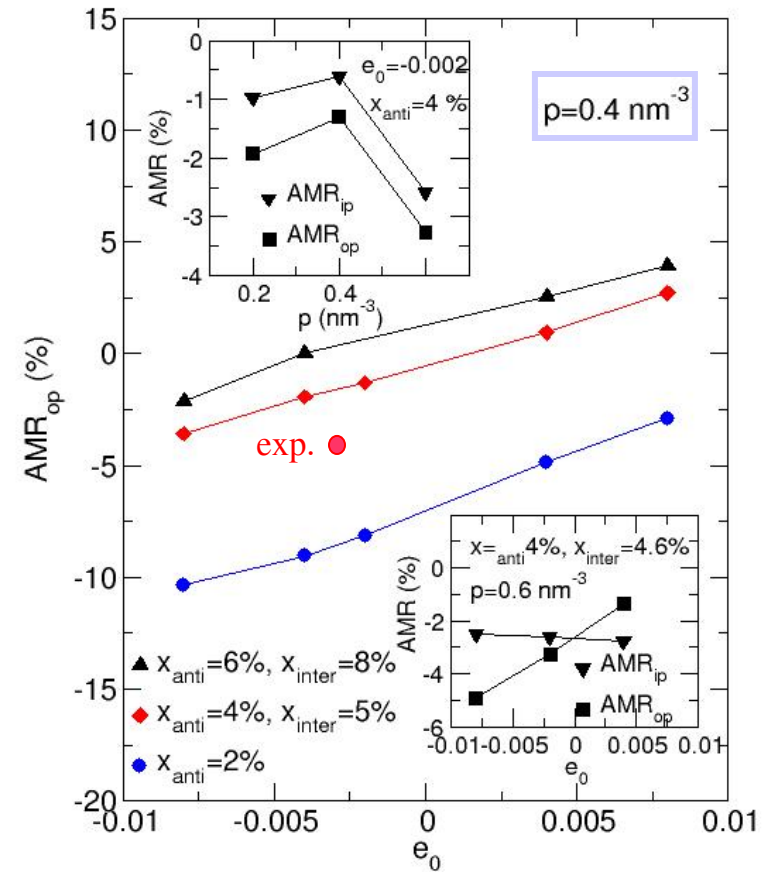
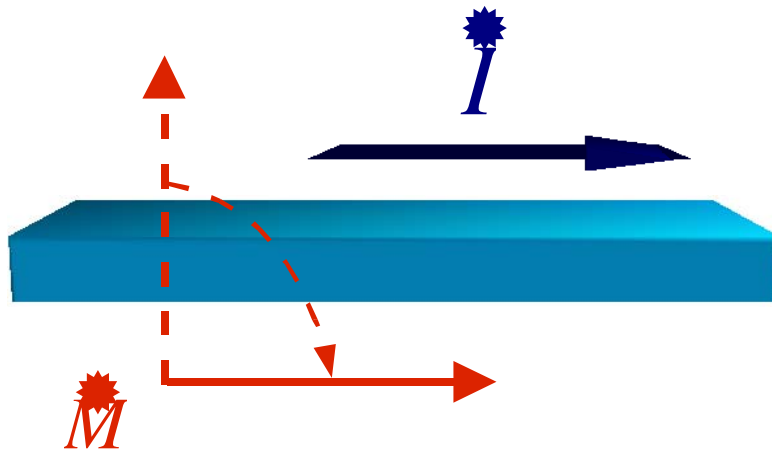
$$A_{a,\mathbf{k}}(\omega) = \Gamma / [(\epsilon - \epsilon_{a,\mathbf{k}})^2 + \Gamma^2/4].$$



$$\Delta H_{pp}(\omega) = \Delta H_{pp}(0) + \frac{2}{\sqrt{3}} \frac{\omega}{g\mu_B} \alpha.$$

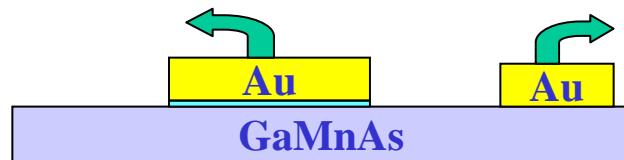


Anisotropic Magnetoresistance

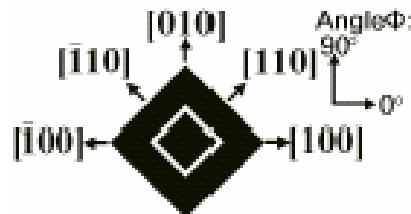
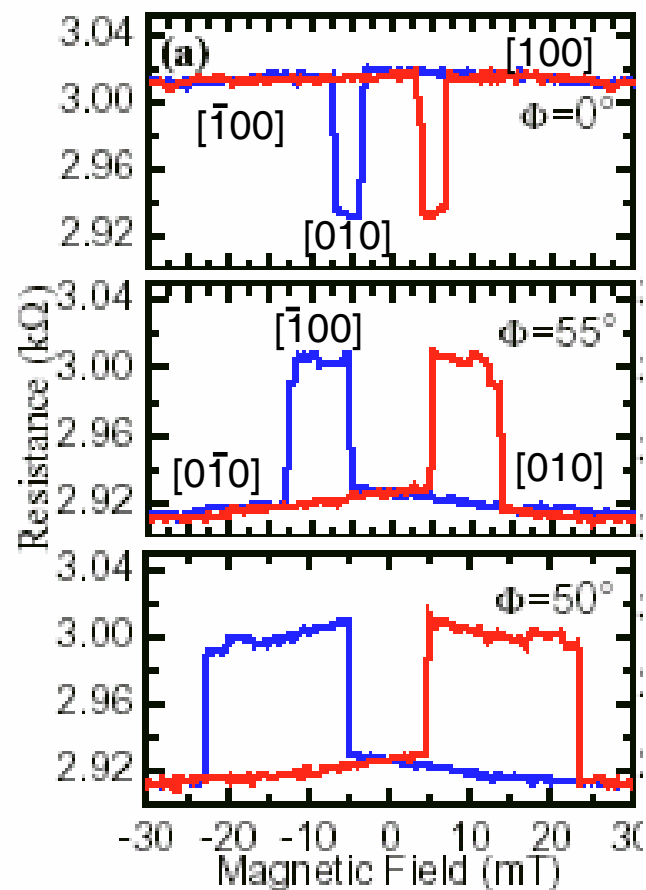


T. Jungwirth, M. Abolfath, J. Sinova, J. Kucera,
A.H. MacDonald, Appl. Phys. Lett. 2002

Tunneling anisotropic magnetoresistance (TAMR)

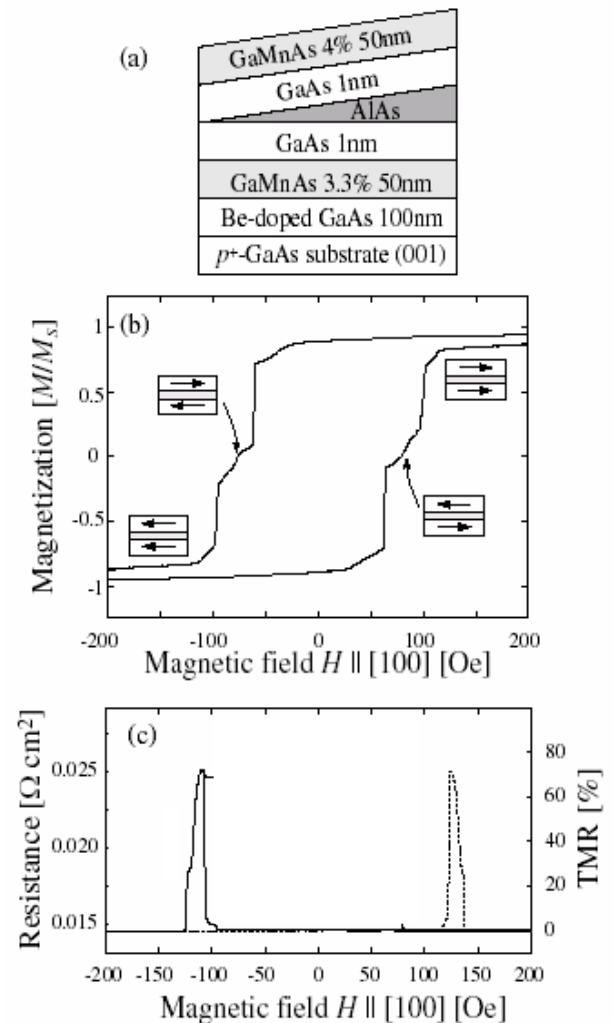


Gould, Ruster, Jungwirth, et al., PRL '04



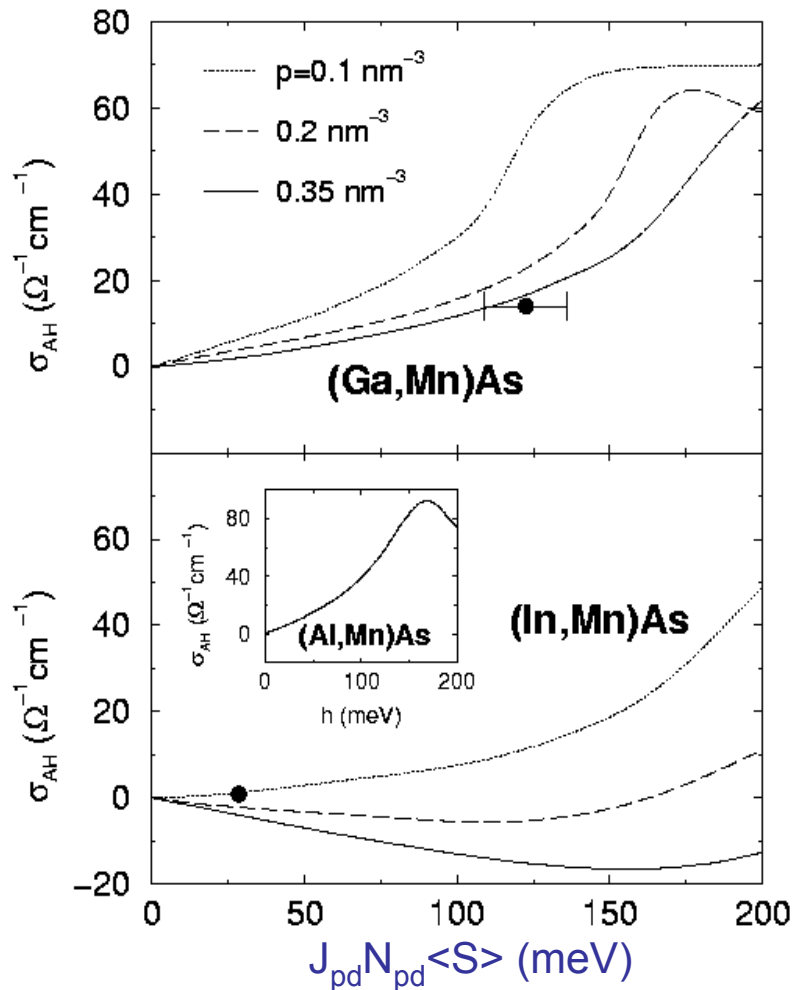
Bistable memory device with a single magnetic layer only

Giant magneto-resistance



ANOMALOUS HALL EFFECT

AHE without disorder



T. Jungwirth, Q. Niu, A.H. MacDonald,
Phys. Rev. Lett. 88, 207208 (2002)

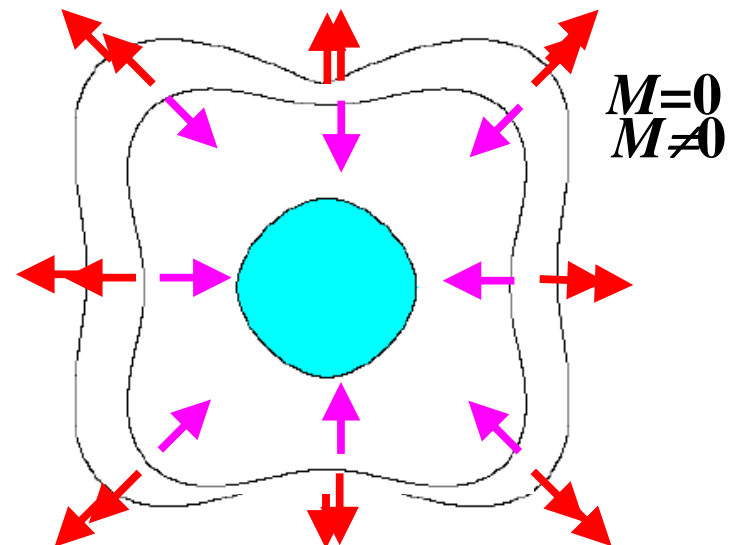
anomalous velocity:

$$\dot{x}_c = \frac{\partial \epsilon}{\hbar \partial \vec{k}} + (e/\hbar) \vec{E} \times \vec{\Omega}.$$

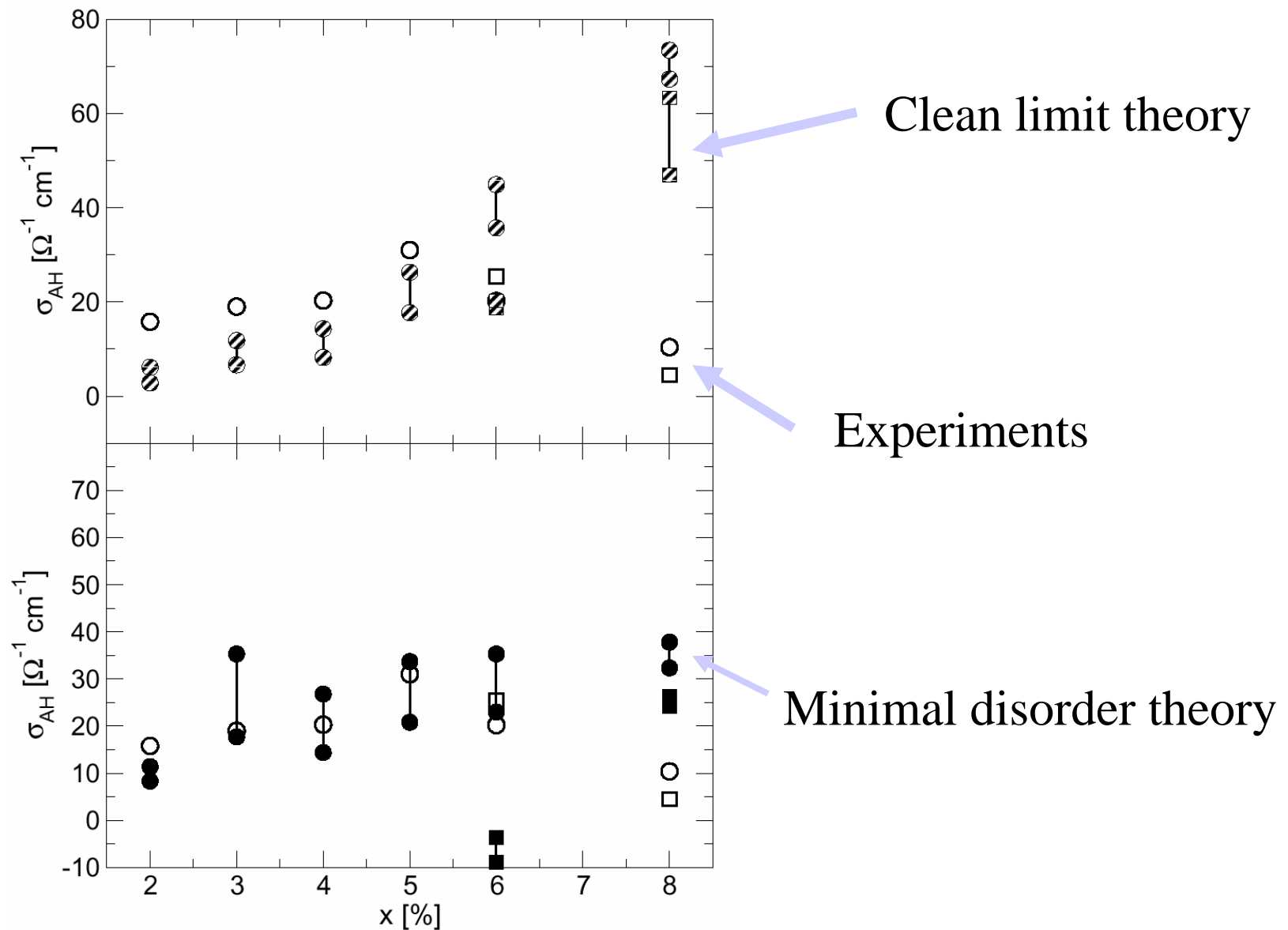
Berry curvature:

$$\Omega_z = 2\text{Im}[\langle \frac{\partial u}{\partial k_y} | \frac{\partial u}{\partial k_x} \rangle].$$

$$\sigma_{AH} = -\frac{e^2}{\hbar} \sum_n \int \frac{d\vec{k}}{(2\pi)^3} f_{n,\vec{k}} \Omega_z(n, \vec{k}),$$



ANOMALOUS HALL EFFECT IN GaMnAs



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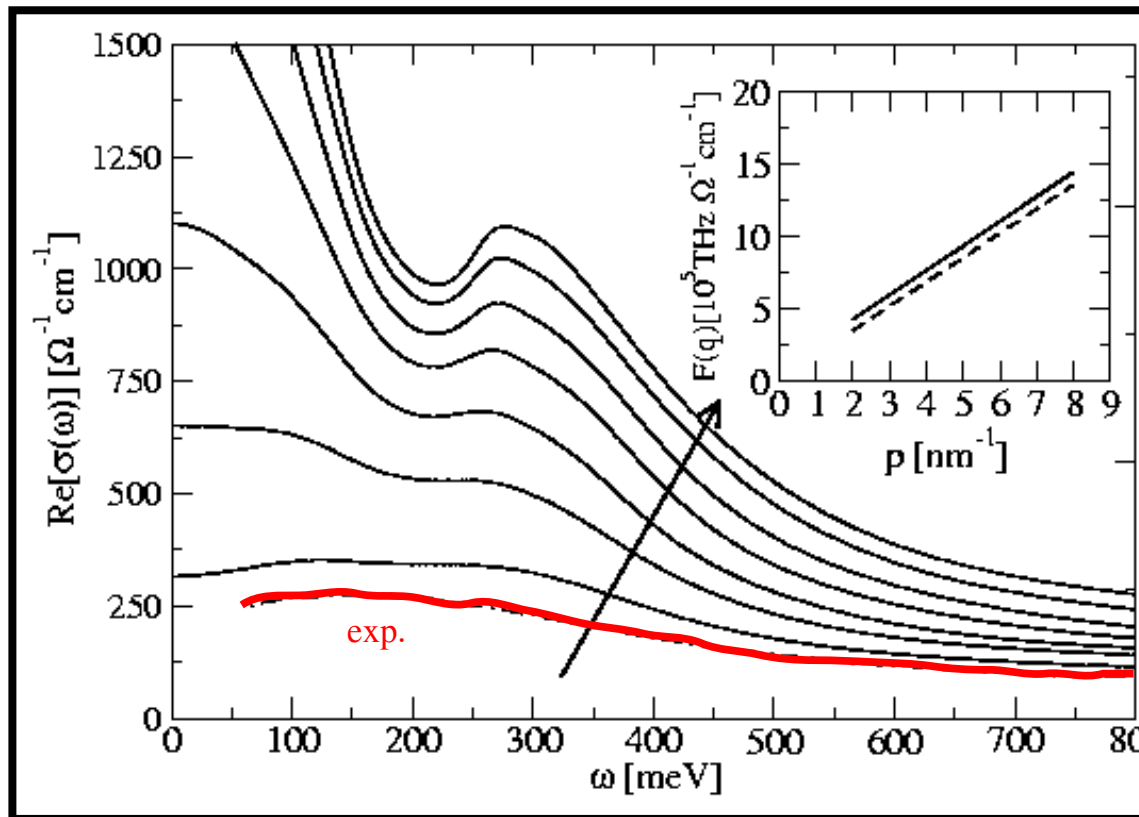
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The valence band picture of IR absorption

$$F = \int d\omega \text{Re}[\sigma(\omega)] = \pi e^2 p / 2m_{opt}$$

hole density: $p=0.2, 0.3, \dots, 0.8 \text{ nm}^{-3}$

$x=5\%$



m_{opt} independent of
(within 10%):

- density
- disorder
- magnetic state

GaAs

$$m_{op} \approx 0.24$$

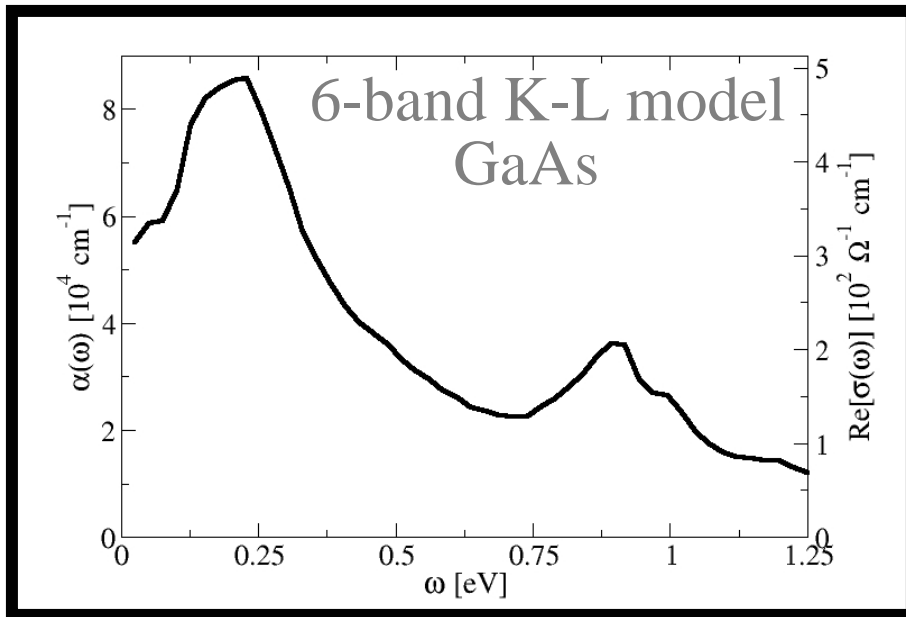
m_e

J. Sinova, et al. Phys. Rev. B
66, 041202 (2002).

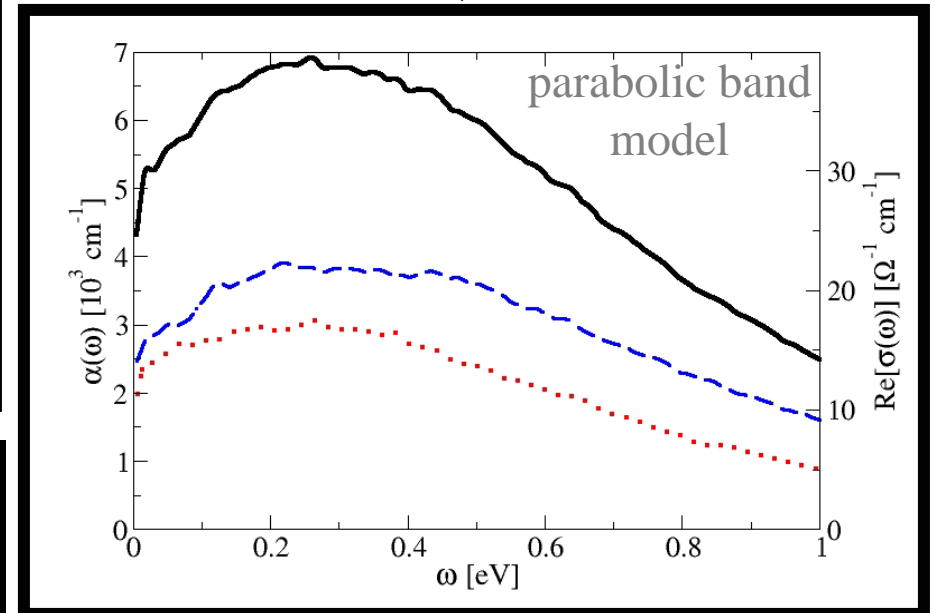
Exps: Singley et al Phys. Rev. Lett. 89, 097203 (2002)
Hirakawa, et al Phys. Rev. B 65, 193312 (2002)

infrared absorption → accurate density measurement

FINITE SIZE EXACT DIAGONALIZATION STUDIES



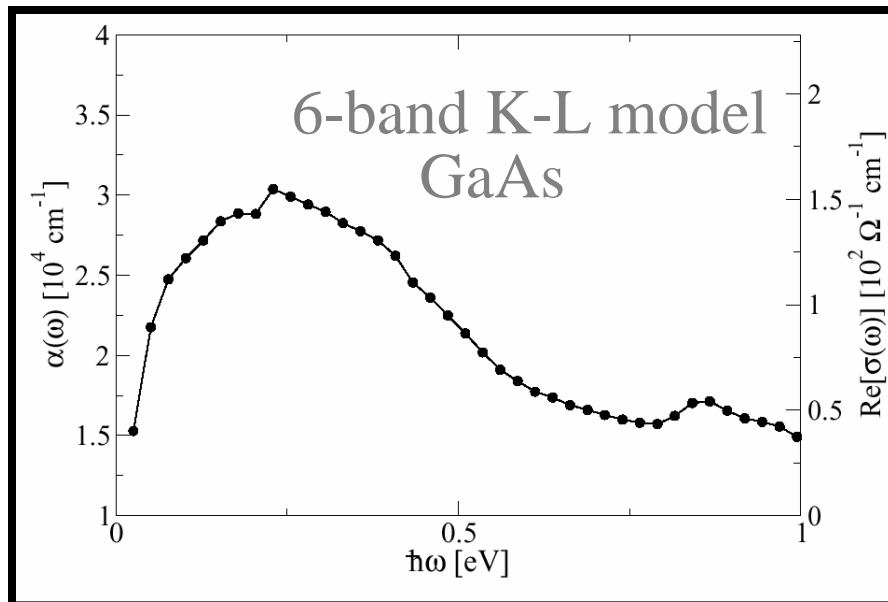
$p=0.33 \text{ nm}^{-3}$, $x=4.5\%$,
compensation from anti-sites



$p=0.2 \text{ nm}^{-3}$, $x=4.0\%$,
compensation from anti-sites



f-sum rule accurate within 10 %



S.-R. E. Yang, J. Sinova, T. Jungwirth, Y.P. Shim,
and A.H. MacDonald, PRB 67, 045205 (03)

Possible issues regarding IR absorption

- Energy dependence of J_{pd}
- Localization effects
- Contributions due to impurity states: Flatte's approach of starting from isolated impurities
- Systematic p and x_{eff} study (need more than 2 m_{eff} data points)

Keeping Score

The effective Hamiltonian (MF) and weak scattering theory (no free parameters) describe (III,Mn)V shallow acceptor metallic DMSs very well in the regime that is valid:

- Ferromagnetic transition temperatures ✓
- Magneto-crystalline anisotropy and coercivity ✓
- Domain structure ✓
- Anisotropic magneto-resistance ✓
- Anomalous Hall effect ✓
- MO in the visible range ✓
- Non-Drude peak in longitudinal ac-conductivity ✓
- Ferromagnetic resonance ✓
- Domain wall resistance ✓
- TAMR ✓

BUT it is only a piece of the theoretical mosaic with many remaining challenges!!

TB+CPA and LDA+U/SIC-LSDA calculations describe well chemical trends, impurity formation energies, lattice constant variations upon doping

Theory of ferromagnetic (III,Mn)V semiconductors,
Jungwirth, Sinova, Masek, Kucera, and MacDonald,
to appear in Rev. of Mod. Phys., in cond-mat/0603380

http://unix12.fzu.cz/ms

